

MoD-PMI
2025



From Grains to Gigabytes:

Generating Massive Virtual Specimen for Irradiation Damage Study

Younggak Shin^{a,†}, Vichhika Moul^b, Keonwook Kang^a, Byeongchan Lee^b

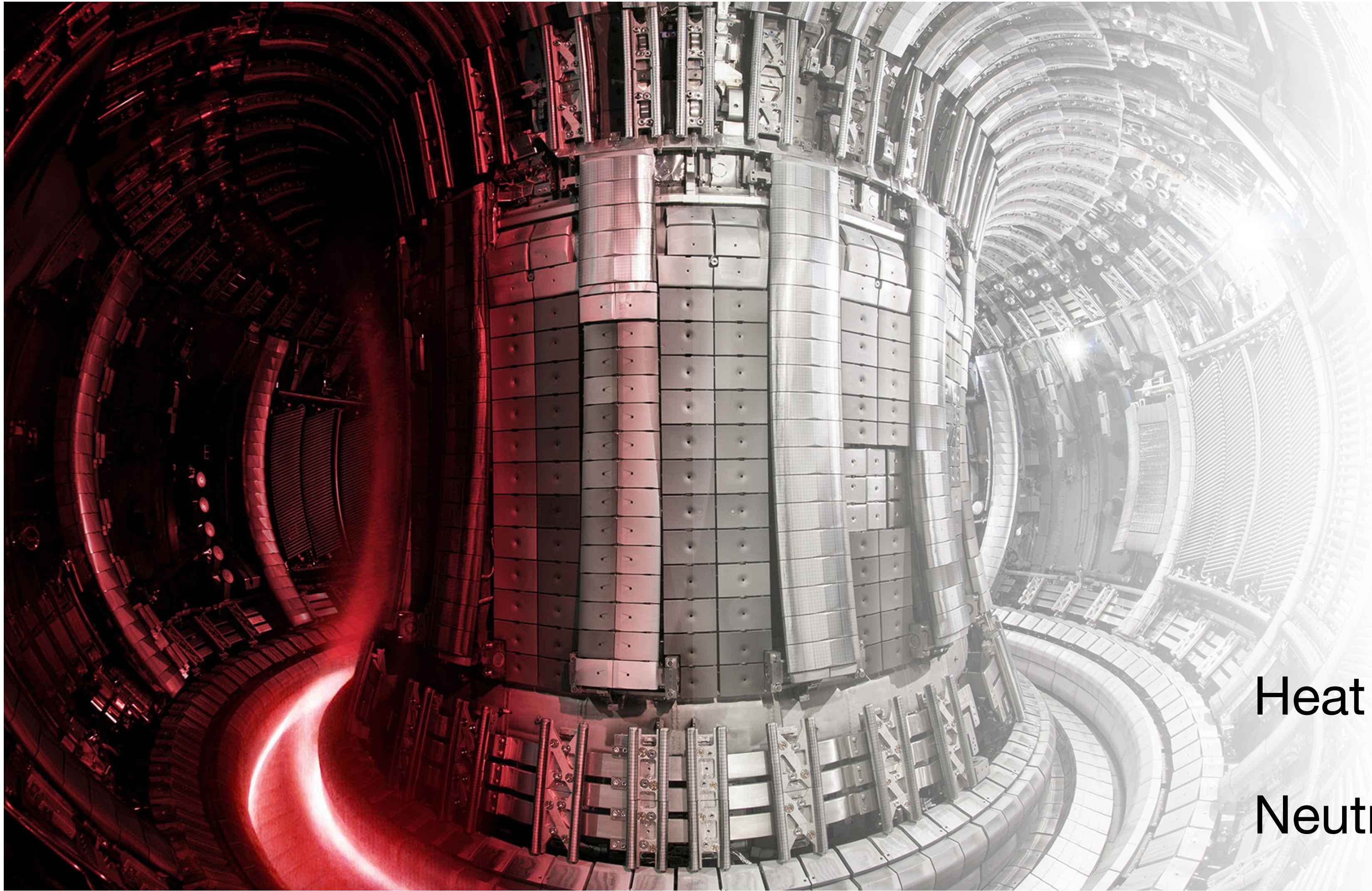
^a *Yonsei University, Seoul, Republic of Korea*

^b *Kyung Hee University, Yong-in, Republic of Korea*

[†] Presenting author: zeroangle@yonsei.ac.kr

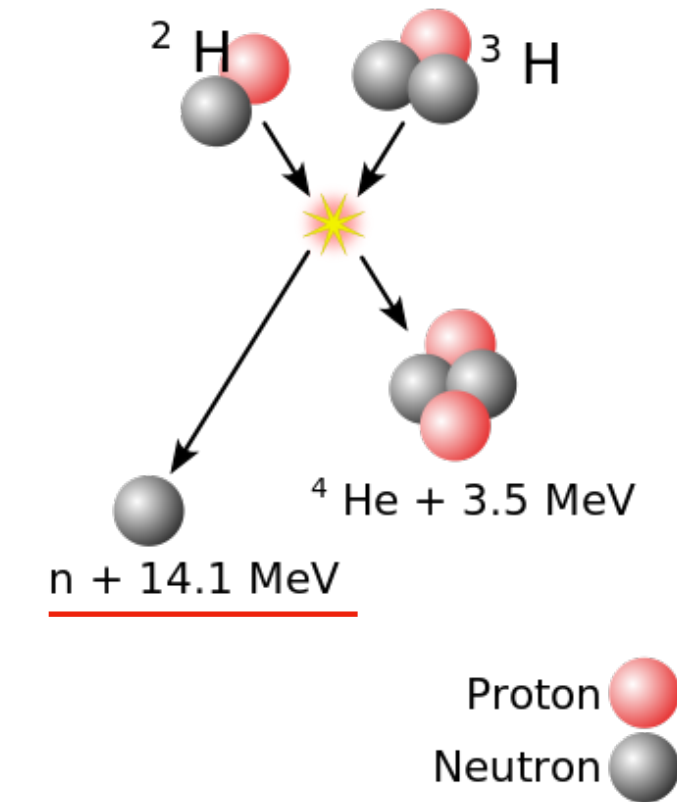
Vienna
May 26 2025

Irradiation Damage in Fusion Environment



Eurofusion - JET

D-T fusion reaction



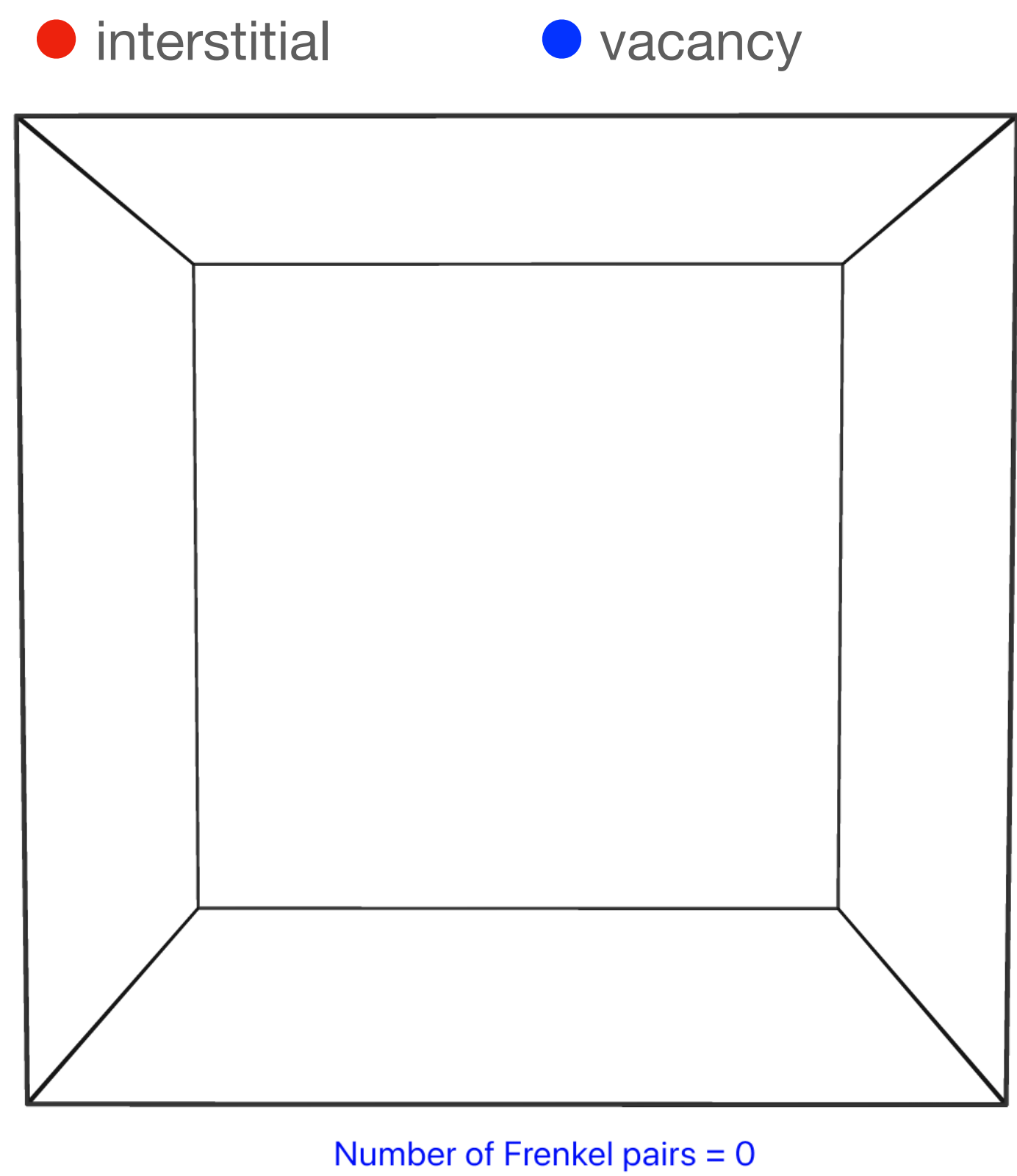
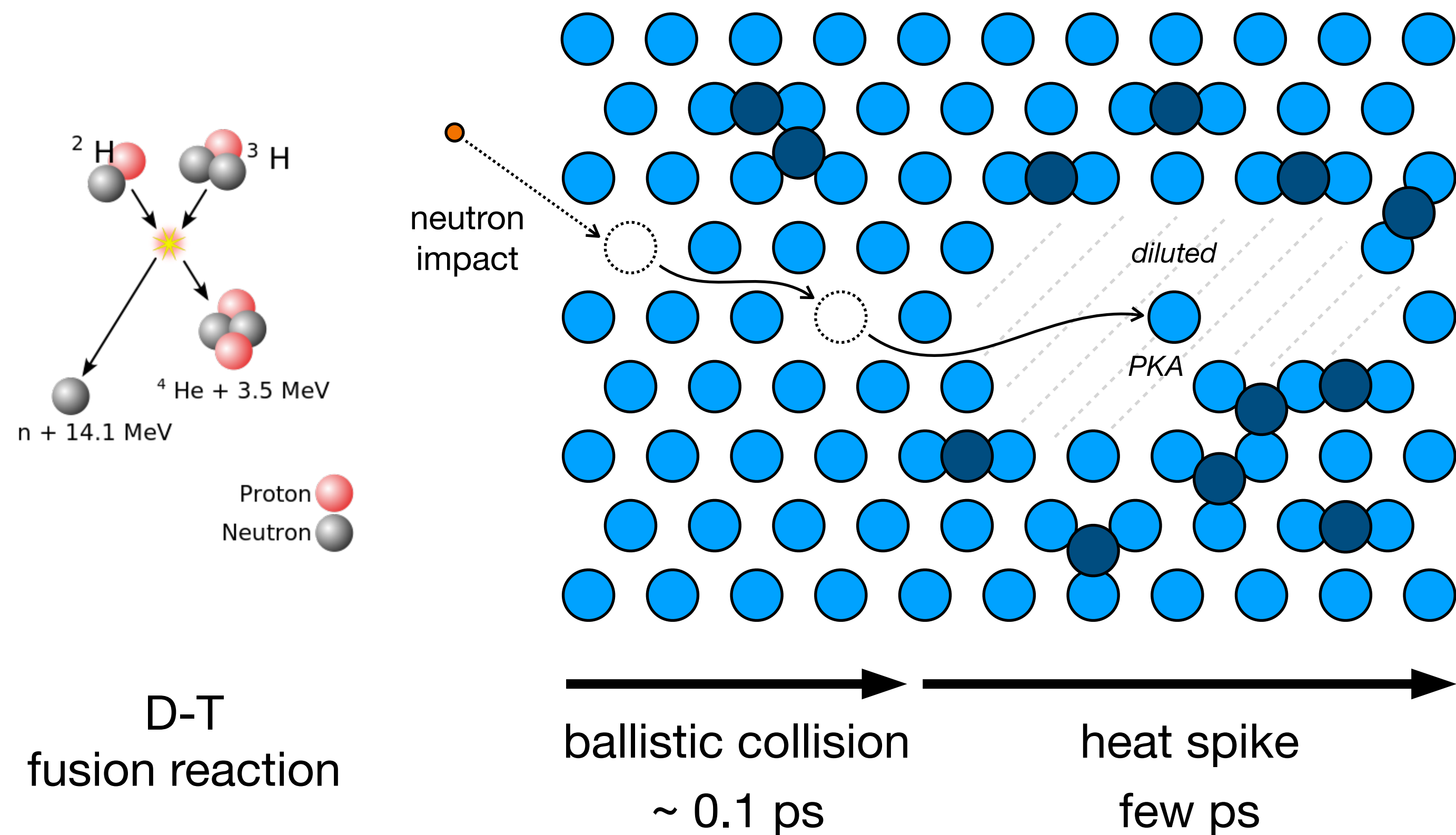
Divertor

Heat flux : $10 - 20\text{ MW/m}^2$

Neutron flux : $10^{14}\text{ n} \cdot \text{cm}^{-2} \cdot \text{s}^{-1}$

Irradiation Damage in Fusion Environment

Collision Cascades



Irradiation Damage in Fusion Environment

Collision Cascades → Volumetric Defects

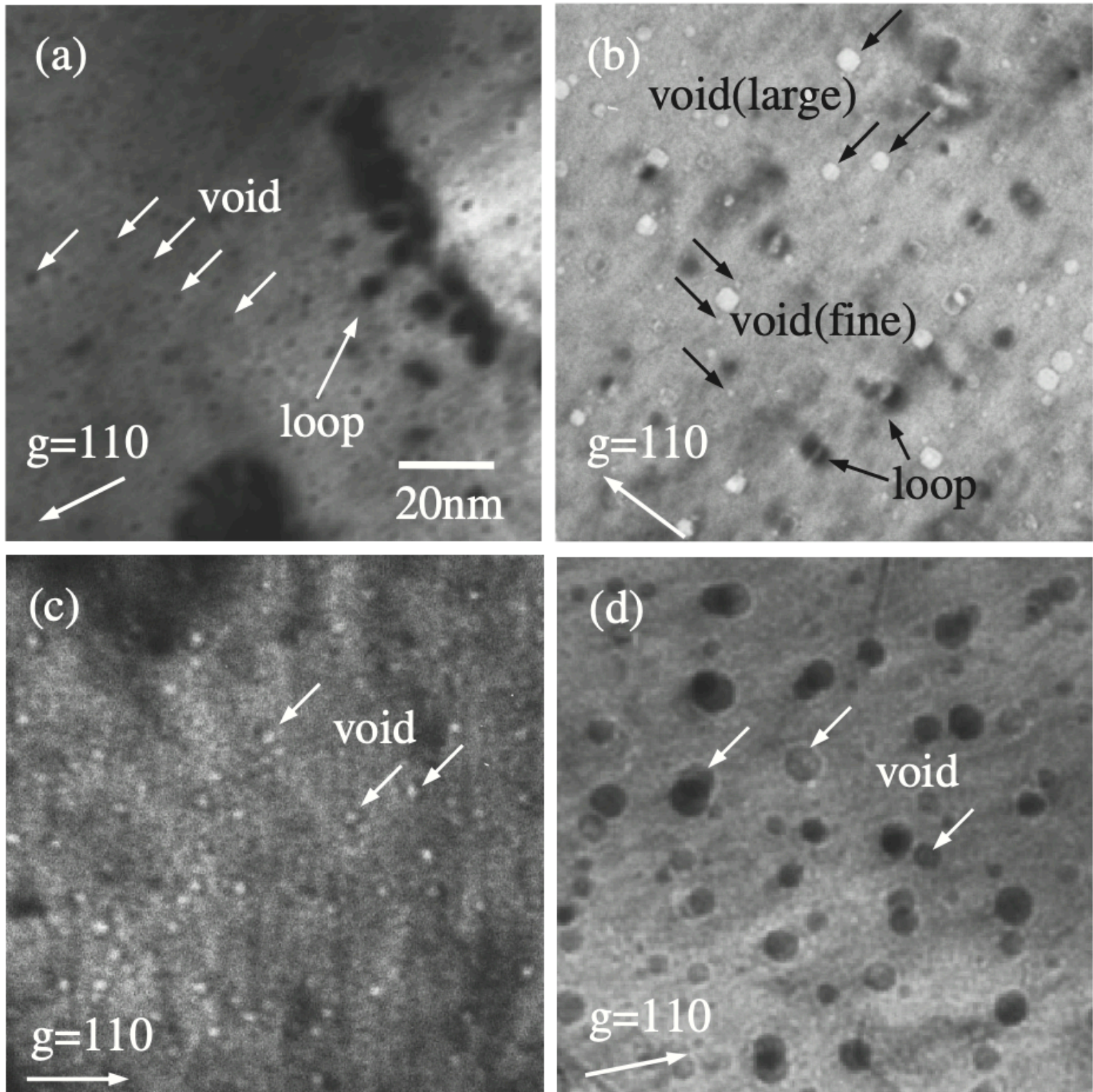


Fig. 2 Microstructural observations of pure W irradiated to (a) 0.17 dpa at 400°C, (b) 0.96 dpa at 538°C, (c) 0.40 dpa at 740°C and (d) 1.54 dpa at 750°C. The void images in (a) and (d) are black, because these were taken in an “over focused” condition.

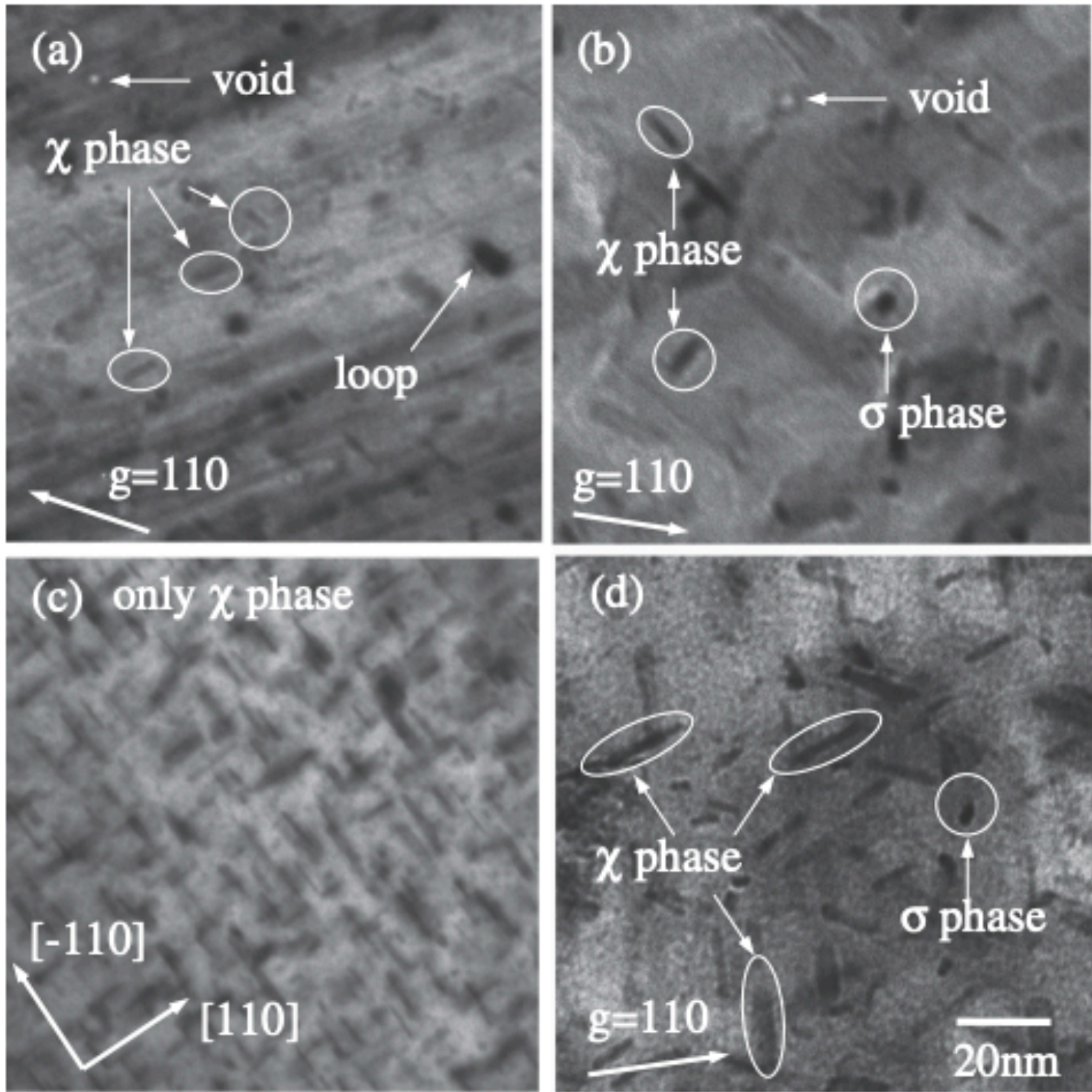


Fig. 4 Microstructural observations of (a) (b) W-5Re and (c) (d) W-10Re, irradiated to (a) (c) 0.96 dpa at 538°C and (b) (d) 1.54 dpa at 750°C. Plate-like precipitates were along {110} planes.

Irradiation Damage in Fusion Environment

Polycrystal: Grain Growth

“Recrystallisation”

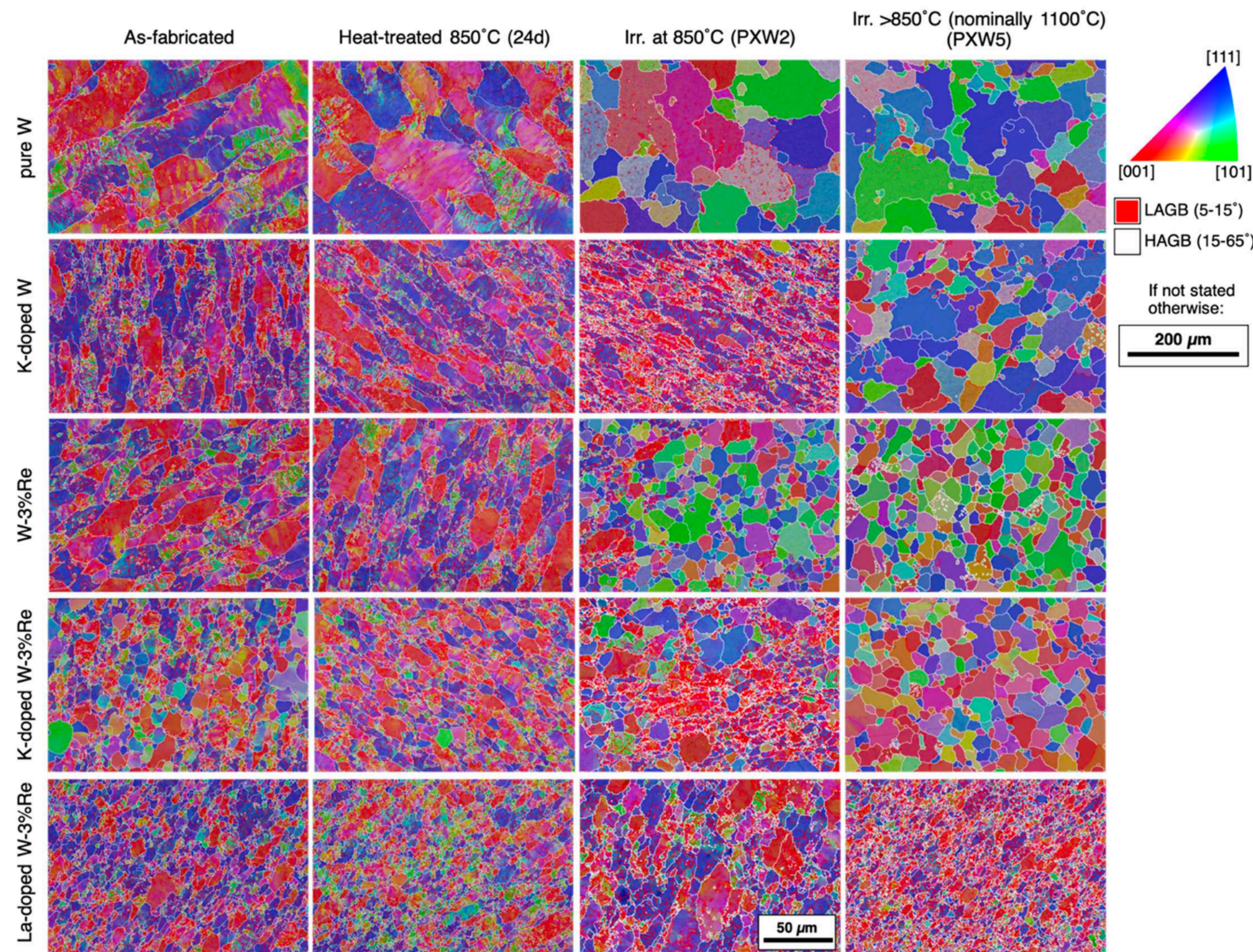
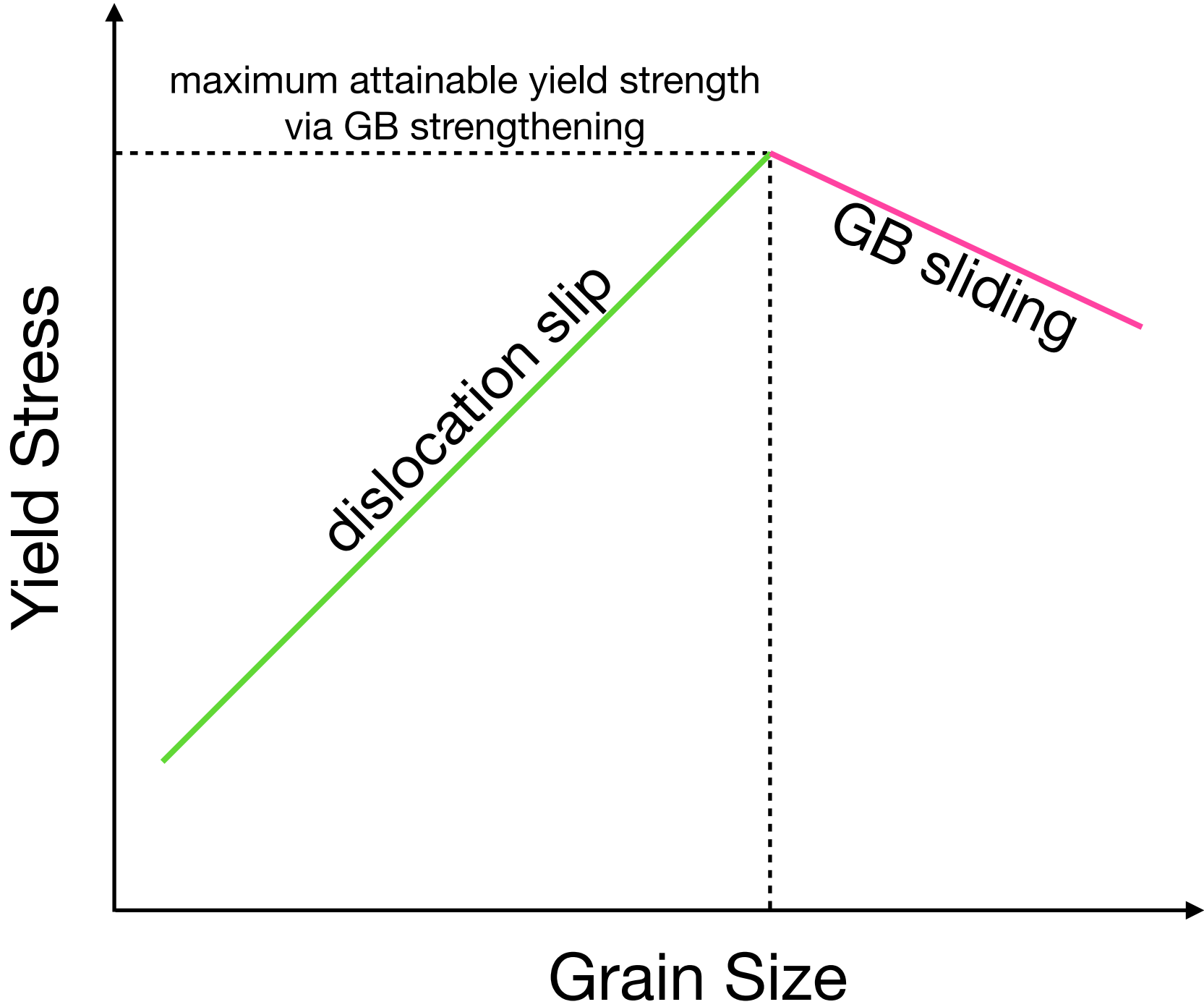


Fig. 1. EBSD orientation maps for the nonirradiated and irradiated materials after correction and spike reduction. HAGBs are marked in white, and LAGBs are marked in red.

Mechanical properties



Gietl et al., J. Alloys Compd. (2022)

Irradiation Damage in Fusion Environment

Polycrystal: Grain Growth

“Recrystallisation”

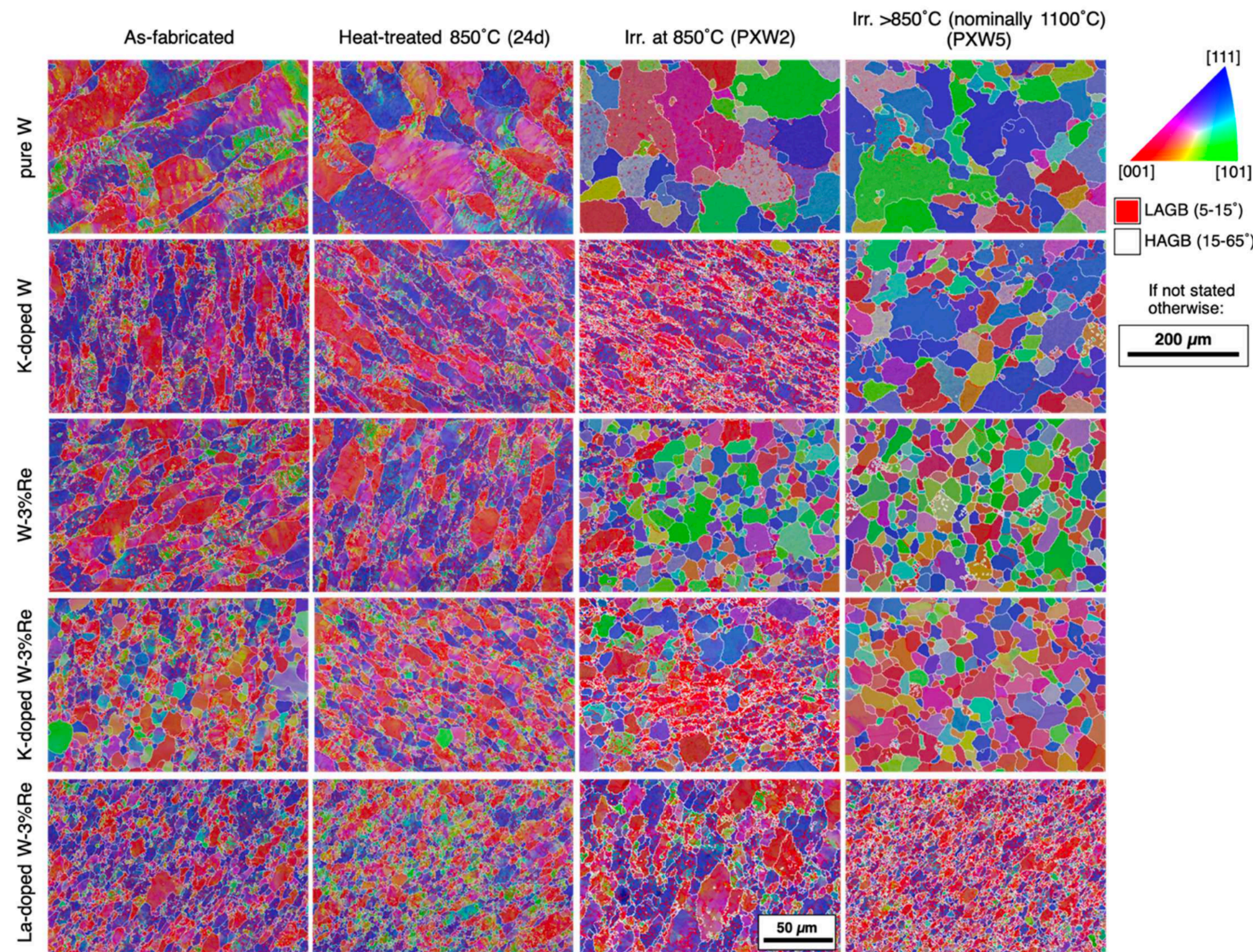
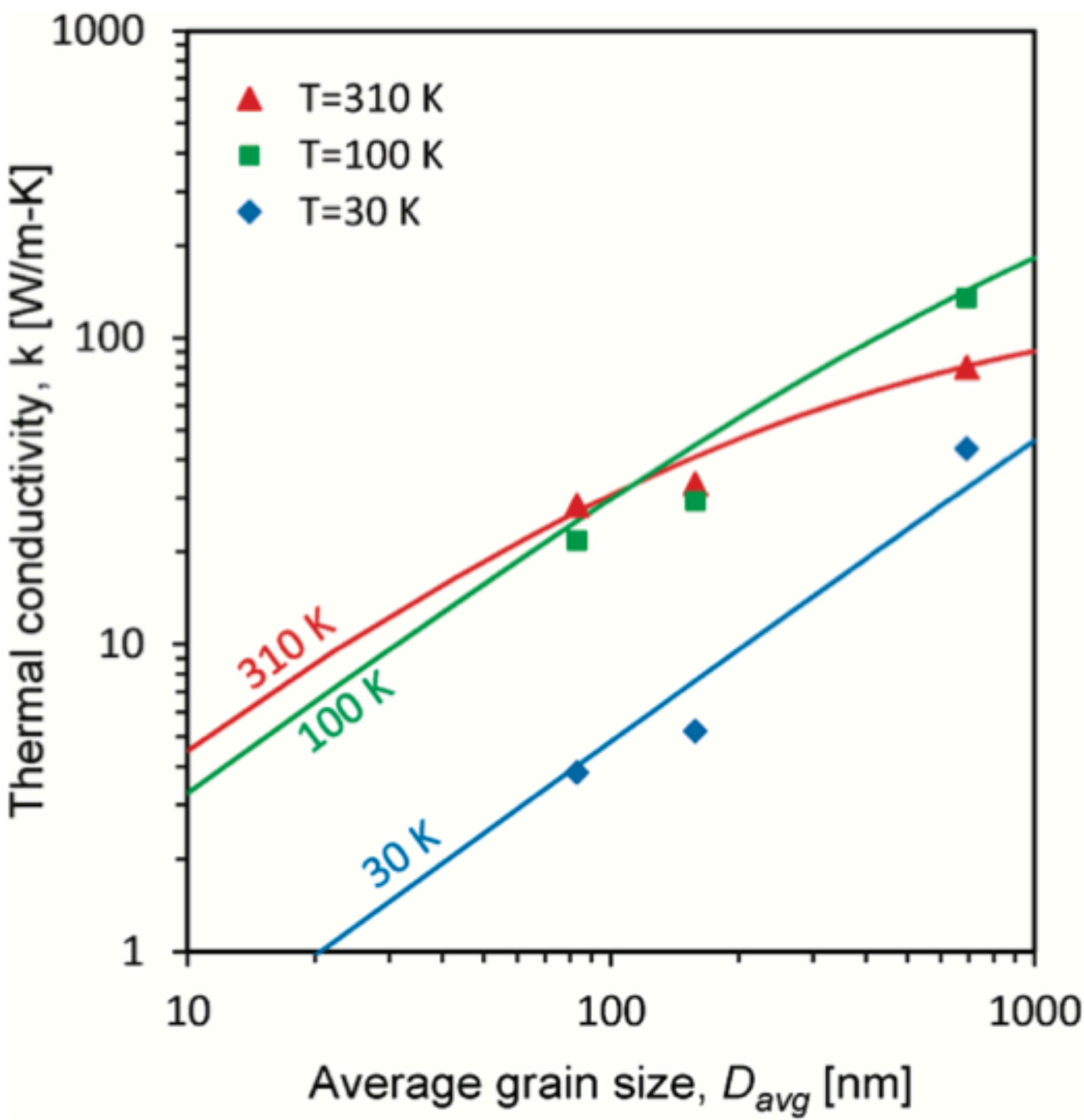


Fig. 1. EBSD orientation maps for the nonirradiated and irradiated materials after correction and spike reduction. HAGBs are marked in white, and LAGBs are marked in red.

Thermal properties



Wang et al., Nano Lett. (2011)

Gietl et al., J. Alloys Compd. (2022)

Irradiation Damage in Fusion Environment

Polycrystal: Grain Boundary as Defect Sink

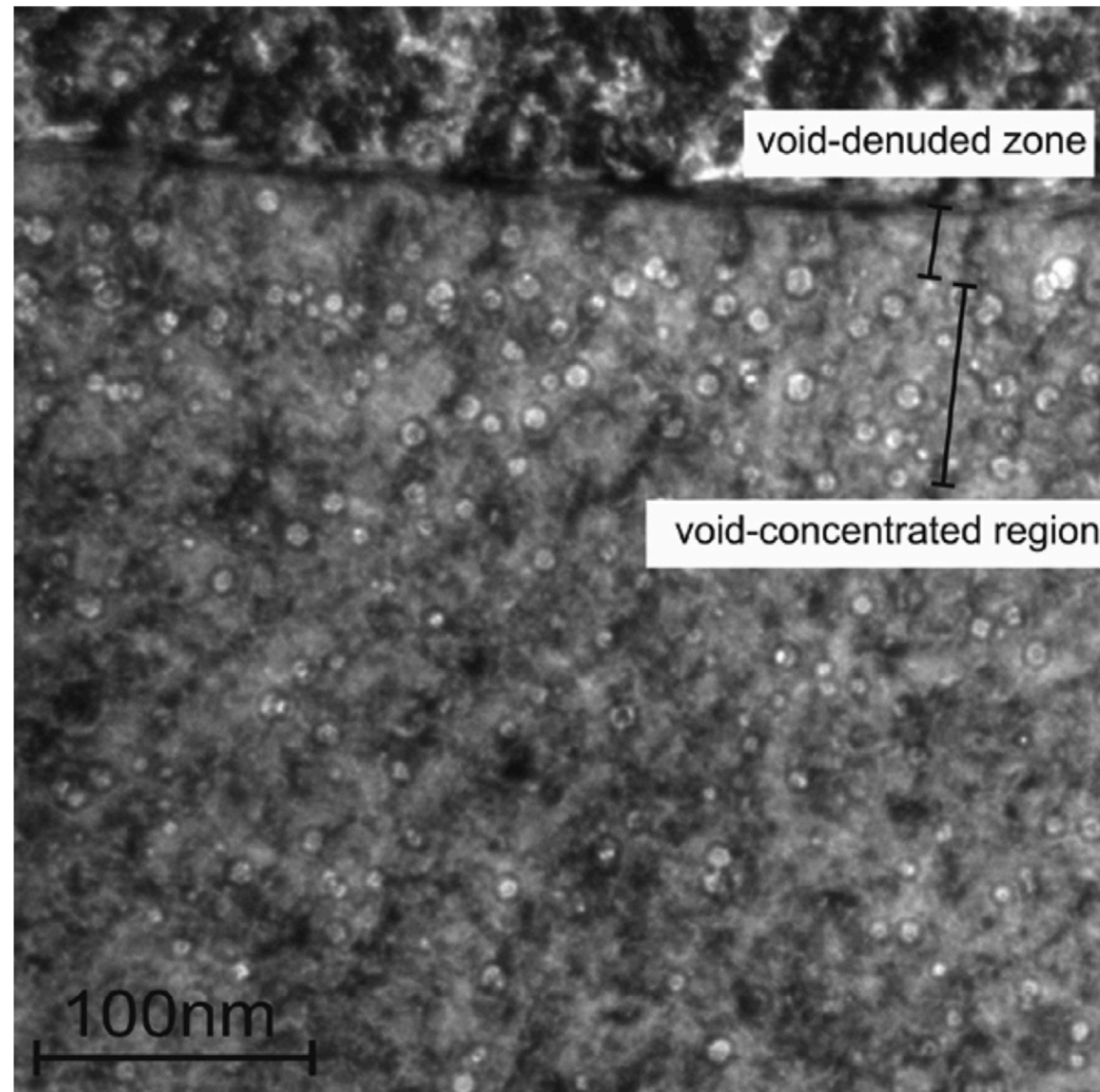


Fig. 7. TEM image of cavities in the region near a grain boundary.

Klimenkov, Nucl. Mater. Energy (2016)

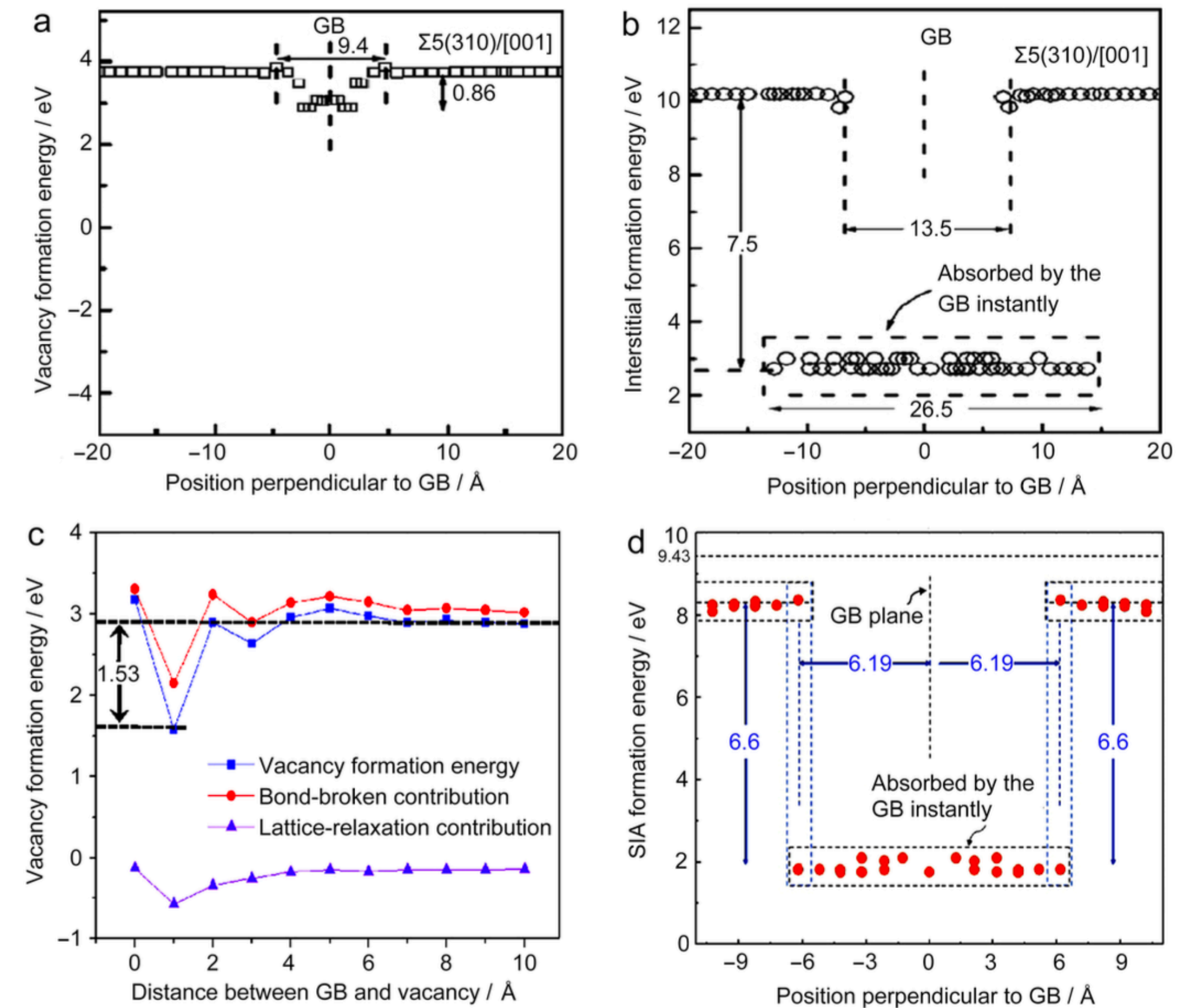


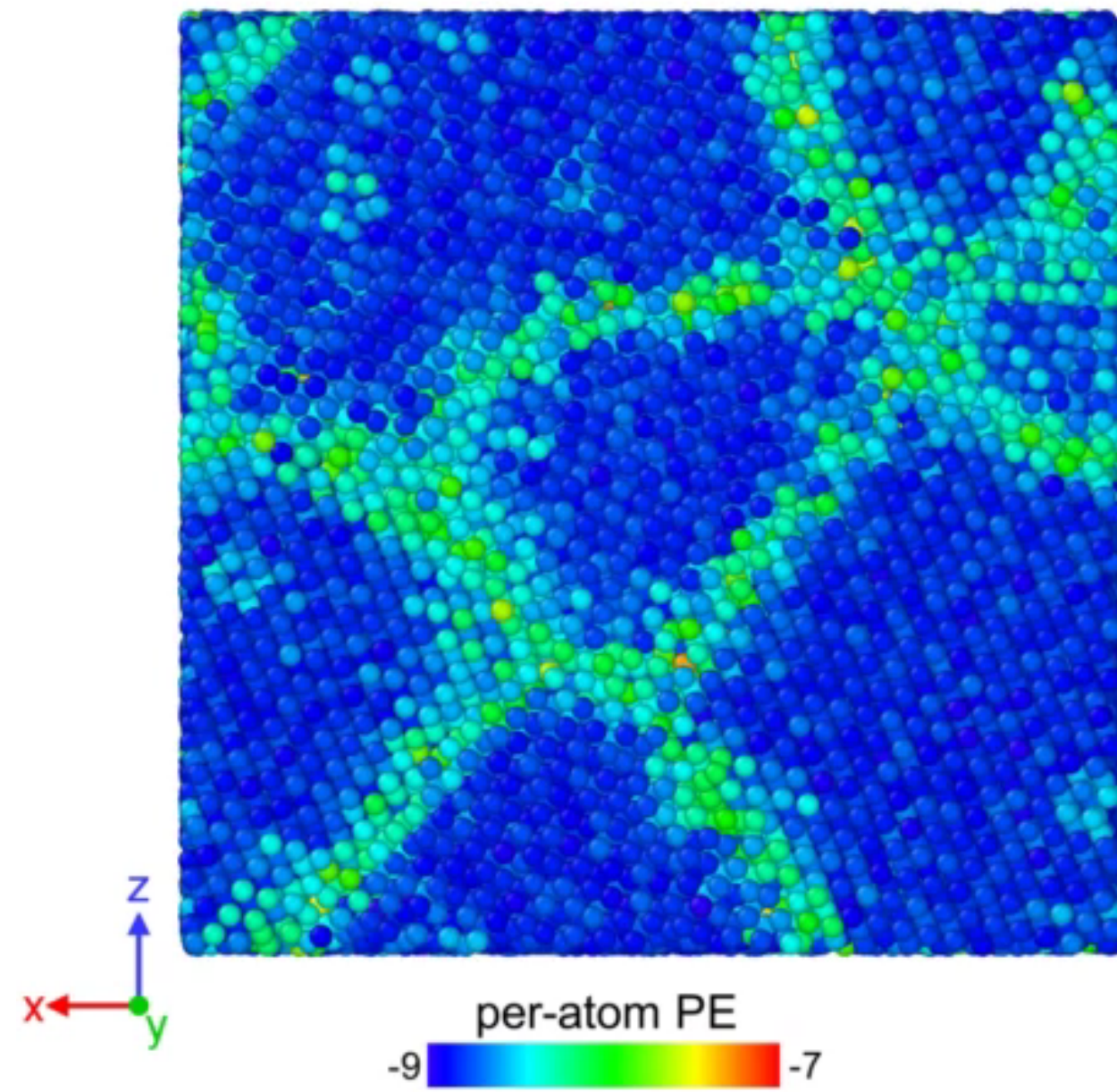
Fig. 4 Variation of the V/SIA formation energy with the initial V/SIA distance from the GB of $\Sigma 5(3\ 1\ 0)/[0\ 0\ 1]$ in W for **a** Vs and **b** SIAs, respectively. Reproduced with permission from Ref. [20]. Copyright

2013 IAEA. **c** and **d** Corresponding results calculated using a first-principles method. Reproduced with permission from Ref. [23]. Copyright 2017 Elsevier

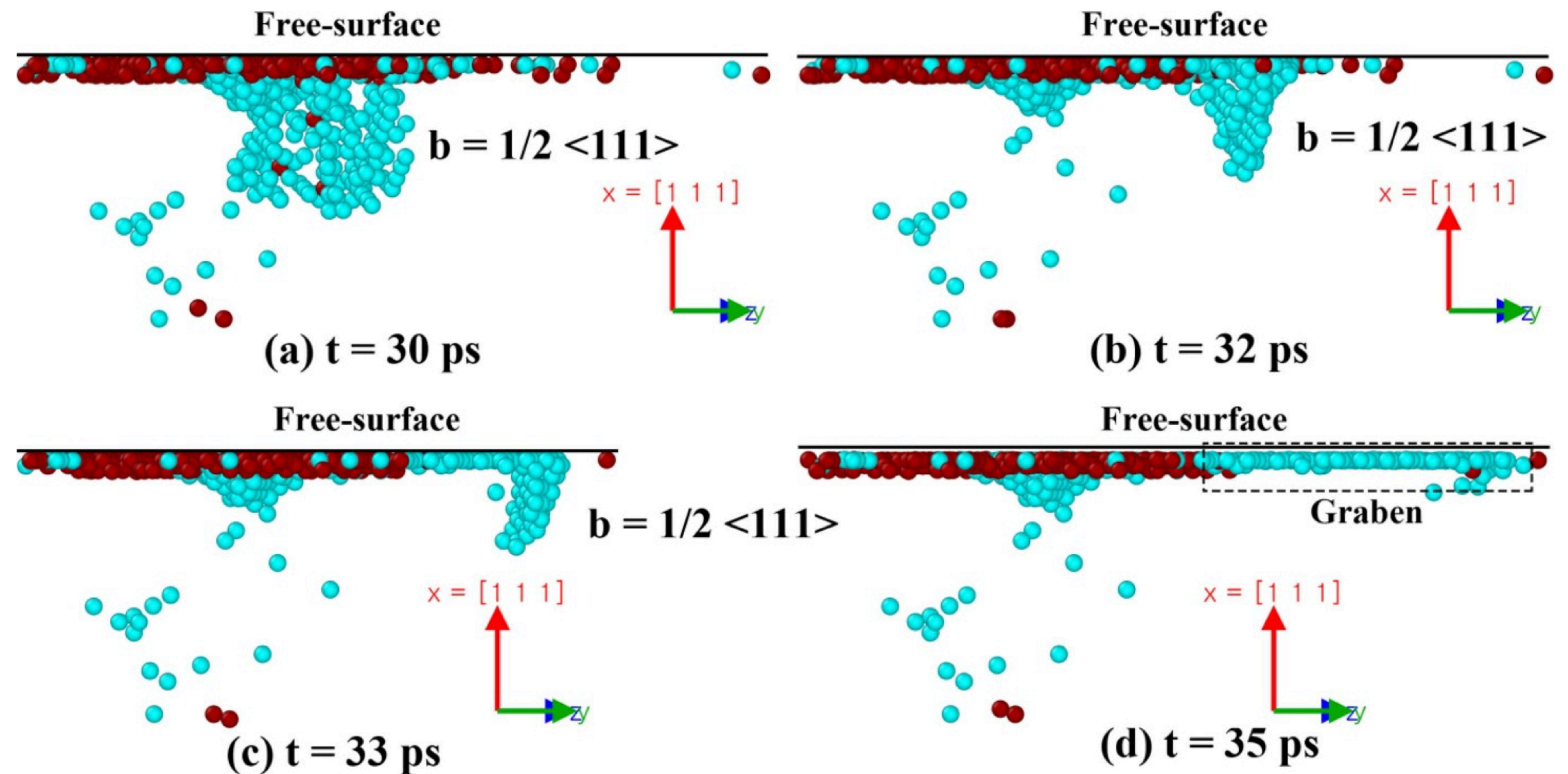
Li et al, Tungsten 2 (2020)

Why We Need Massive-Scale MD Simulation

Tiny Model



Unpublished, courtesy of Yoo



Lee et al., Nucl. Fusion (2020)

Unrealistically small grains

Severe recrystallisation

Large interfacial area

Intense absorption of self-interstitial atoms

Why We Need Massive-Scale MD Simulation

Tiny Model versus Massive Model

	Tiny (< tens of millions of atoms)	Massive (~ 10 billion atoms)
Microstructure fidelity	singlecrystal bicrystal ultrafine-grained*	Full grain-size spectrum
Phenomena captured	aggressive recrystallisation no grain boundary pinning	GB-defect interaction Hall-Petch strengthening grain boundary sliding
Parameter transferability	impractical for multiscale modelling	direct hand-off comparable with experimental data

Massive MD: **Present**

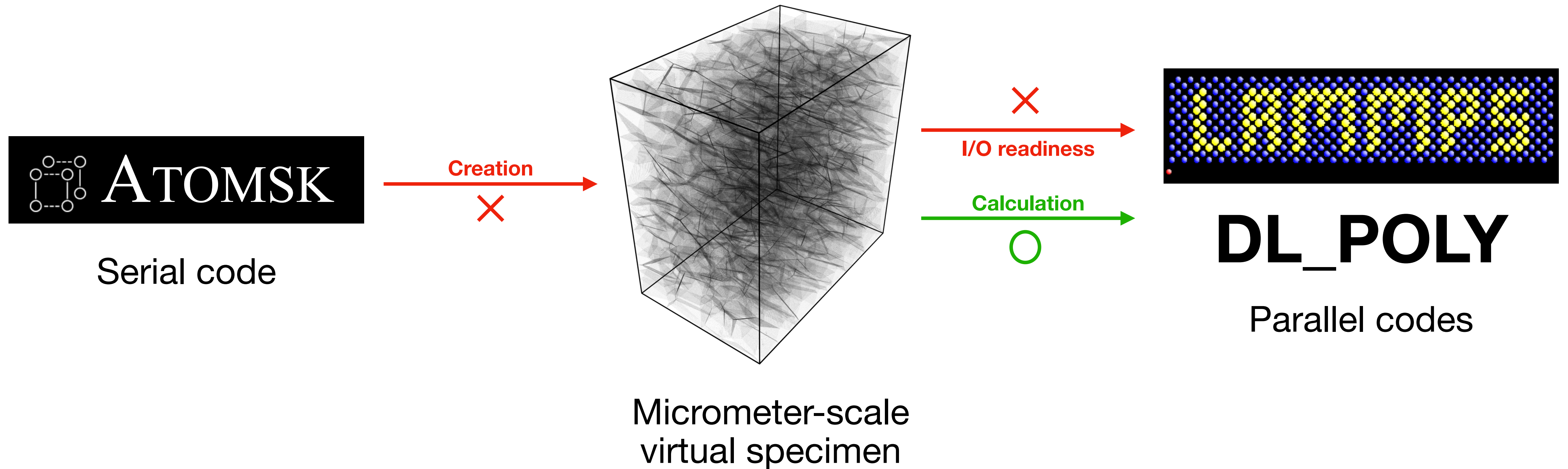


OAK RIDGE National Laboratory website

Frontier HPC achieved ExaFLOPS in 2023

Are we ready for this?

Massive MD: **Present**

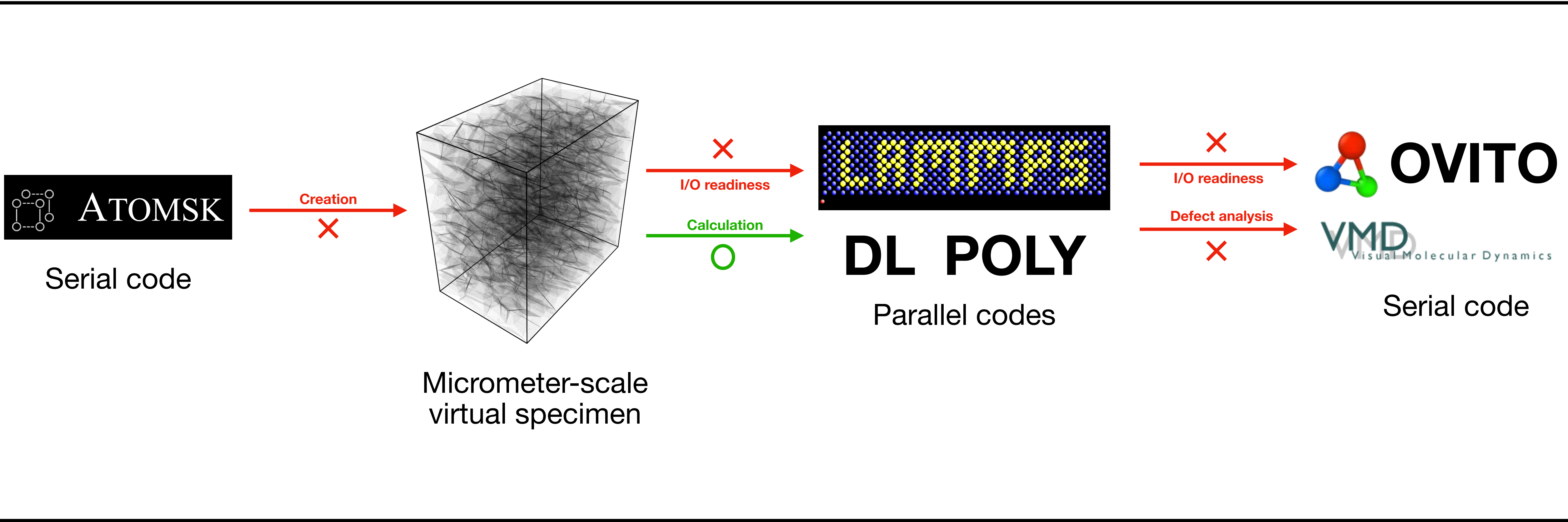


“No parallel atomic structure generation code exists.”

“No detail control of polycrystalline structure.”

“LAMMPS community discarded parallel I/O system.”

Massive MD: **Present**



“No parallel atomic structure generation code exists.”
“No detail control of polycrystalline structure.”

“LAMMPS community discarded parallel I/O system.”

“No parallel post-processing code exists.”

Massive MD: **Future**

Atomic structure
generation

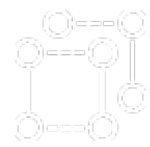
MD simulation

Post-processing

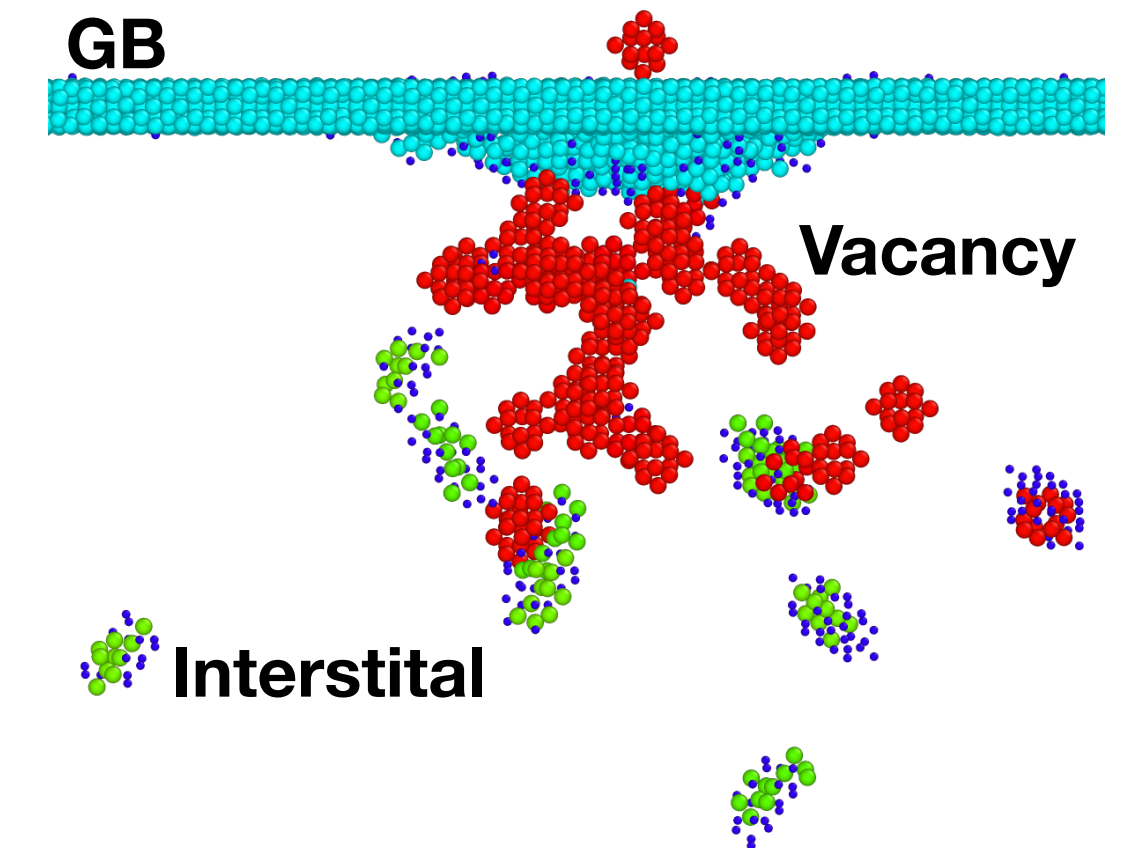
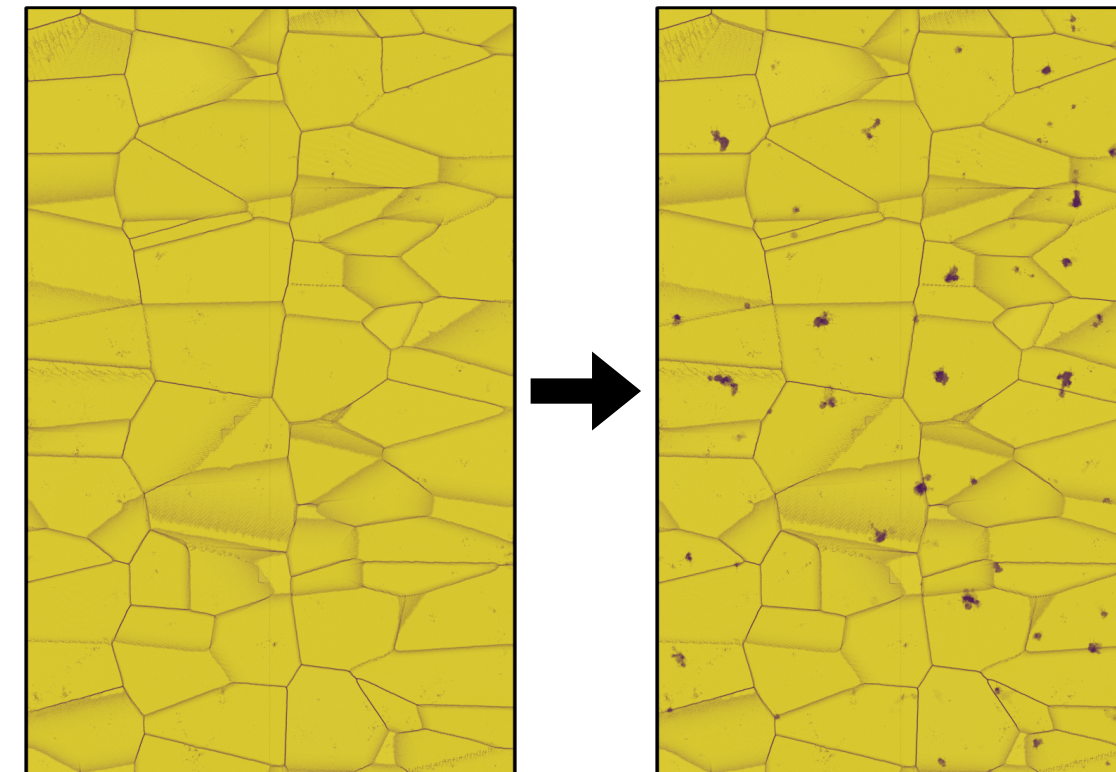
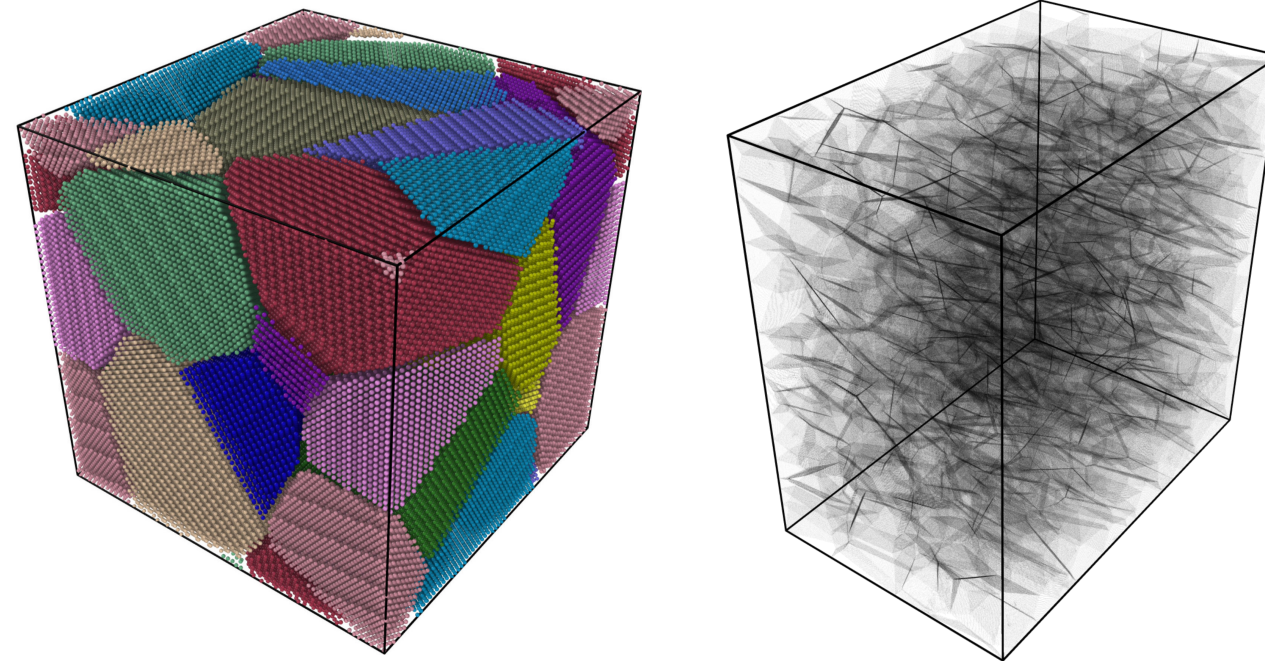
PolyPal
(published)

VITMAS
(developed)

Untitled
(in progress)



Parallel
codes



- Integrated parallel I/O framework
 - Efficient memory management
 - GB generation and appropriate defect analysis
- + Multi-PKA (Primary Knock-on Atom) simulation feature

PolyPal

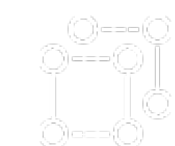
Parallel I/O

Parallelised task without
inter-core communication

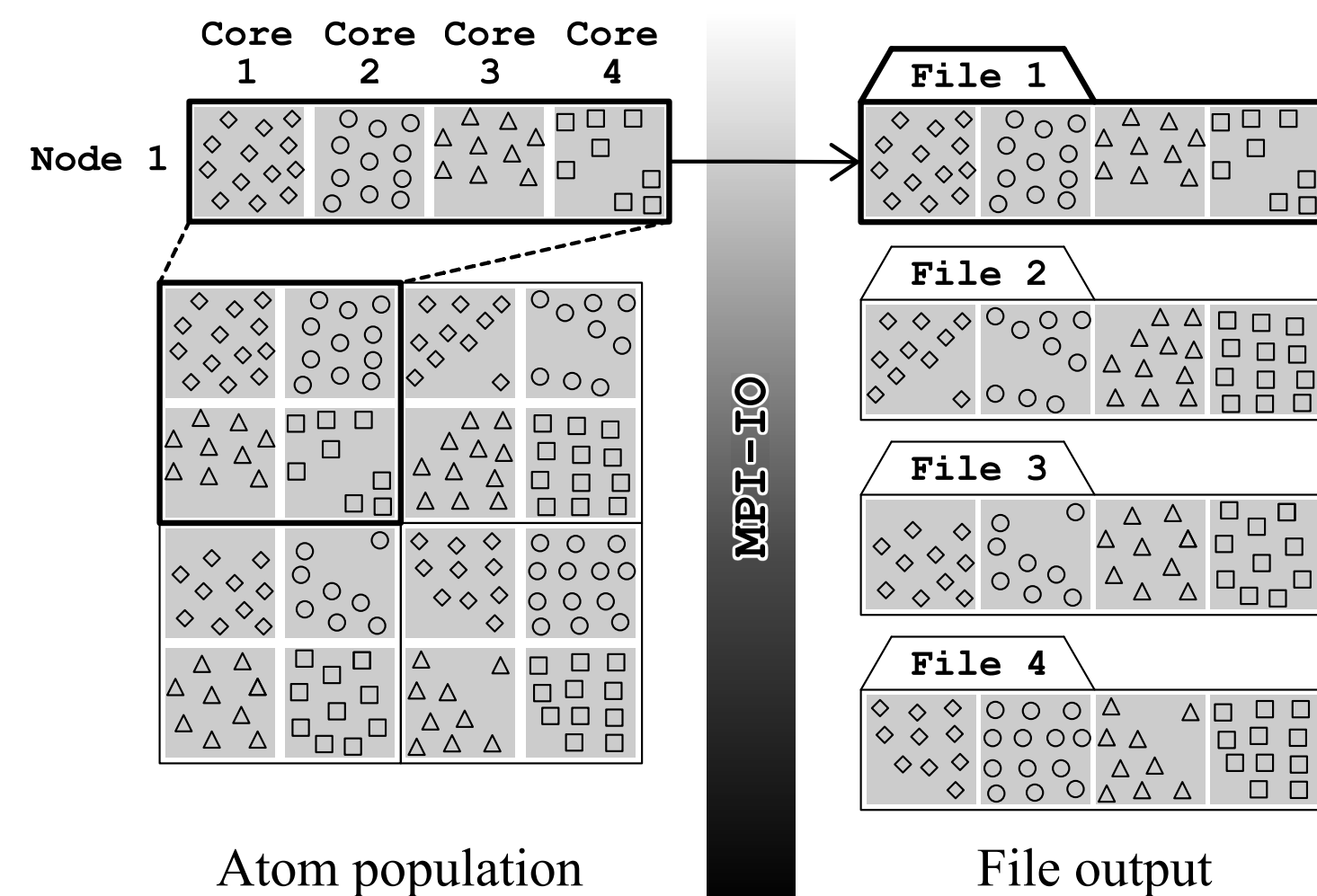
Versatile options

PolyPal: Features

Parallel I/O

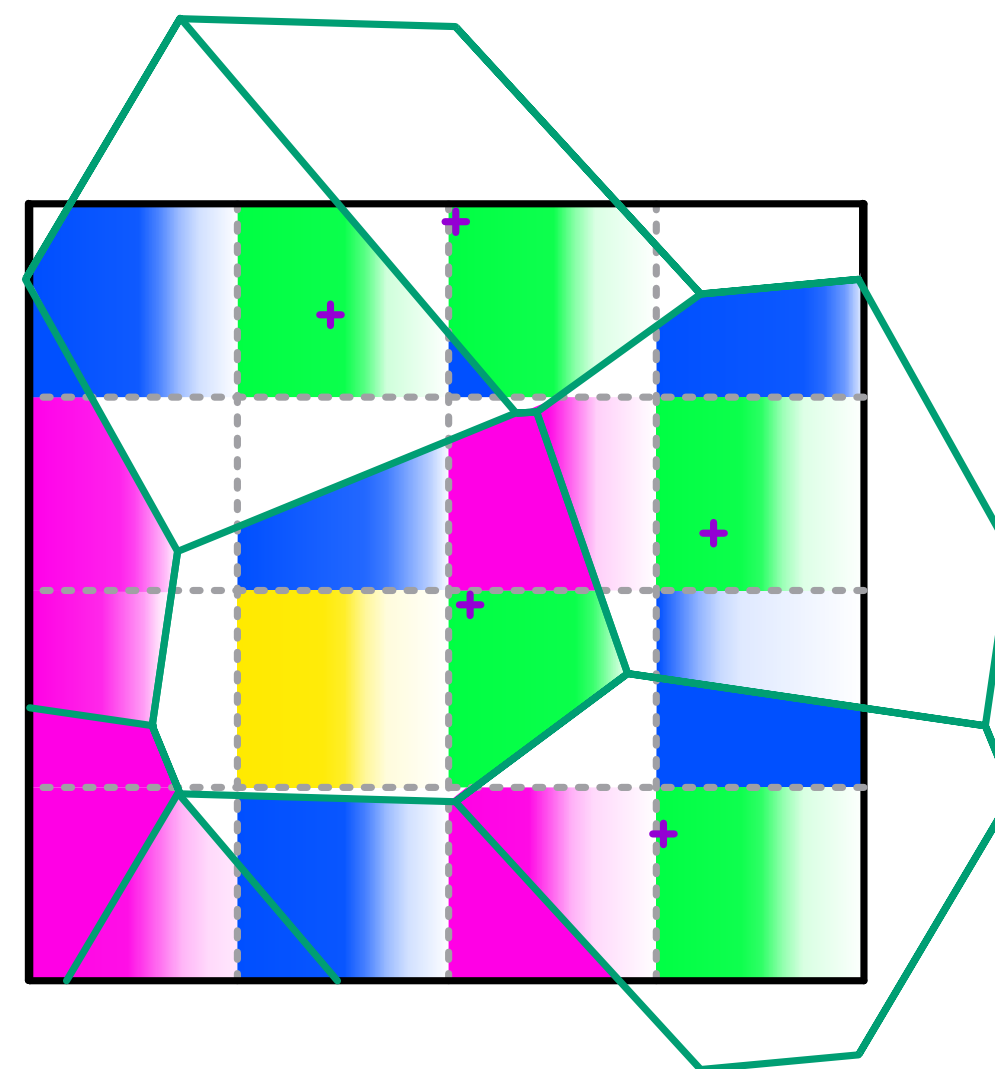


- Fast
- Memory efficient
- Domain-preserving
→ Elimination of atom sorting



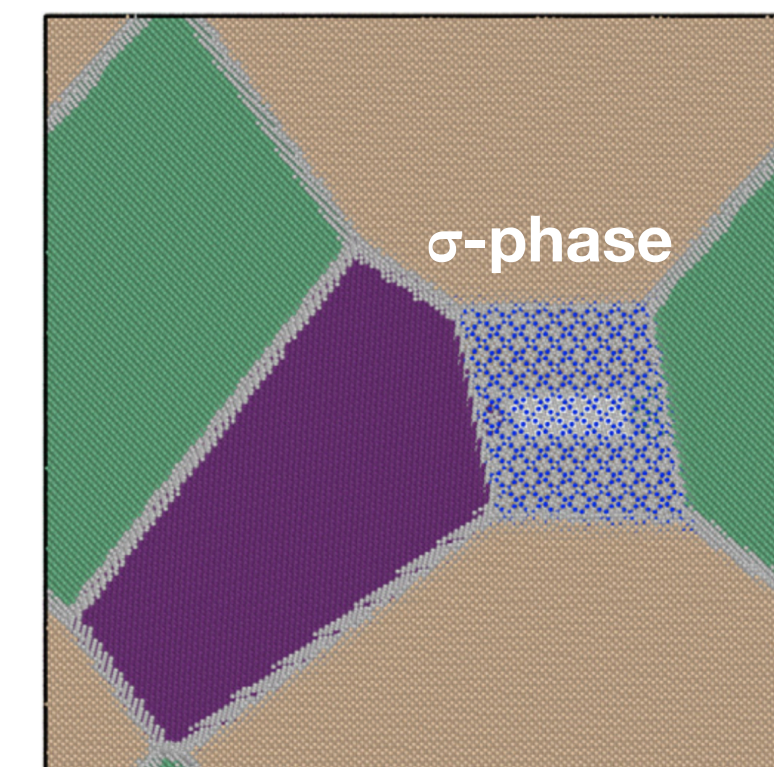
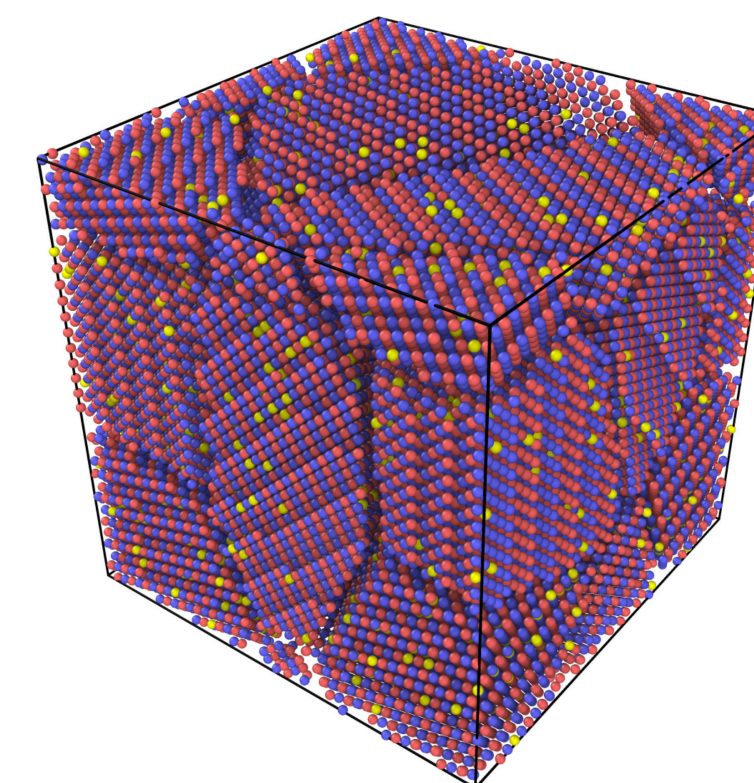
Parallelised task without inter-core communication

- Fast
- Memory efficient



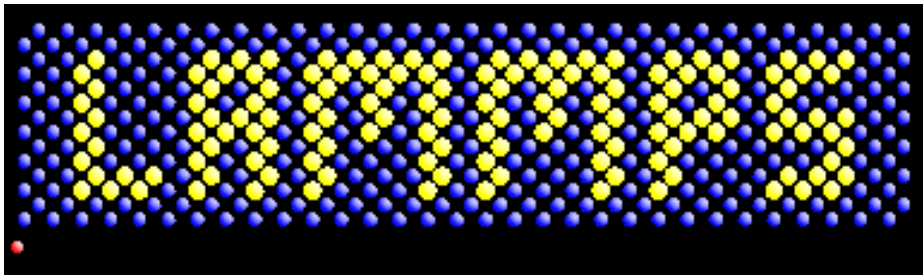
Versatile options

- Grain orientation control
- Grain morphology control
- Multi-phase
- Species substitution



PolyPal: Features

File Format



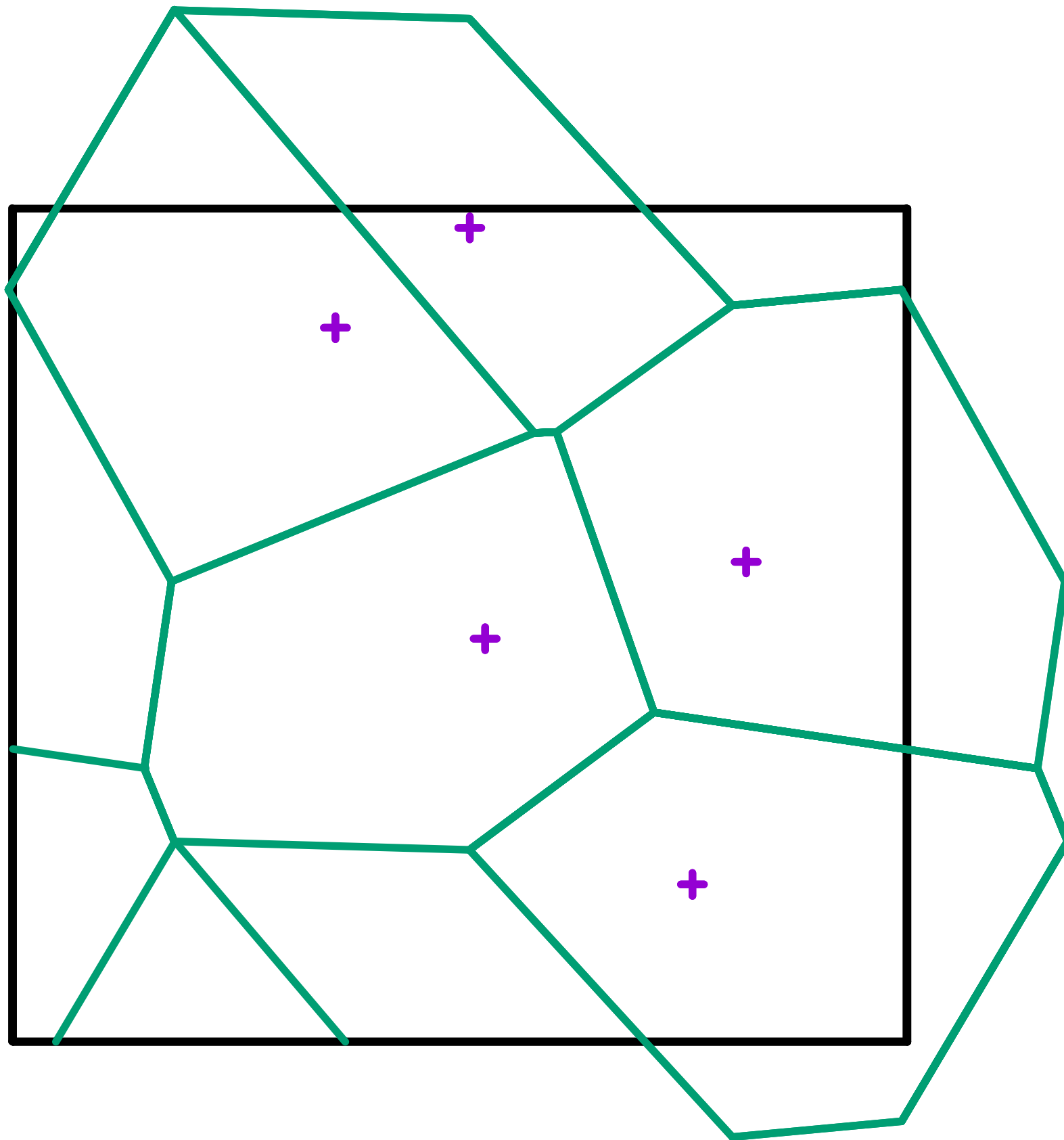
DL_POLY

VITMAS

ASCII text	✓	✓	✓
Binary	X	X	✓
Mutil-flle	✓	X	✓

PolyPal: Features

Structure Design: Grain Morphology



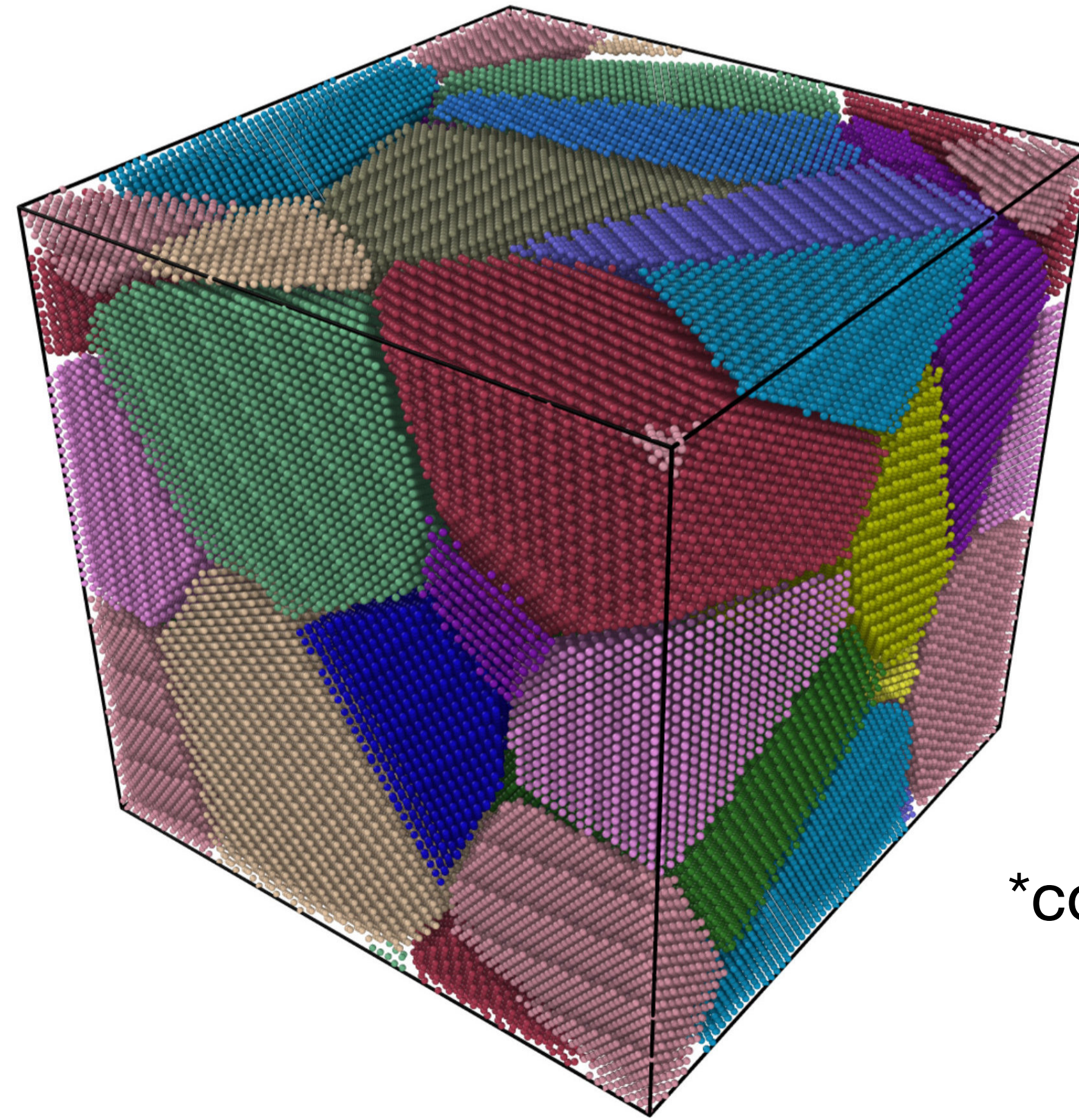
Seed distribution

=

Grain morphology

***PolyPal*: Features**

Structure Design: Grain Morphology

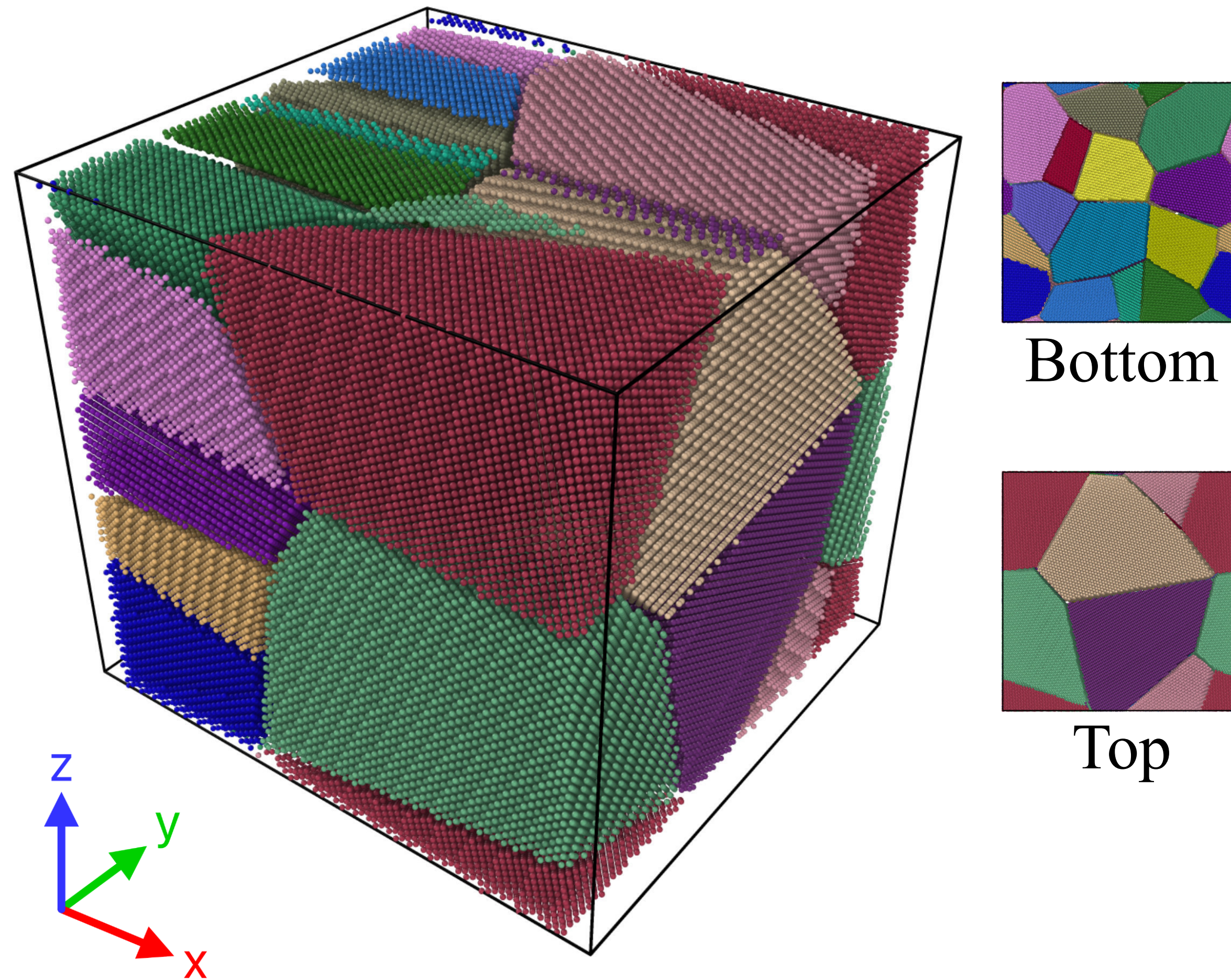


*coloured by lattice
orientation

Uniformly and randomly distributed

***PolyPal*: Features**

Structure Design: Grain Morphology

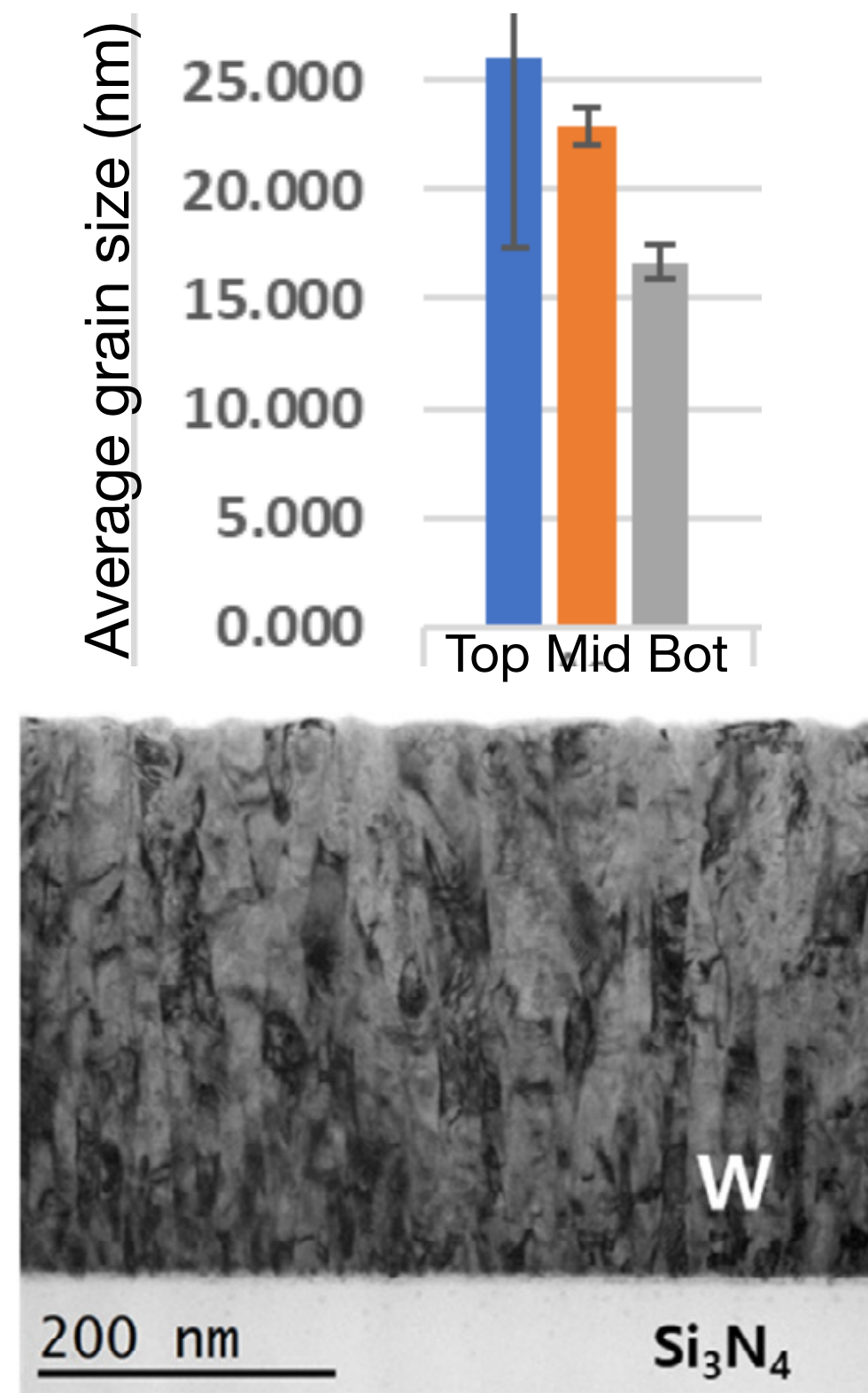


Layered structure with free surfaces

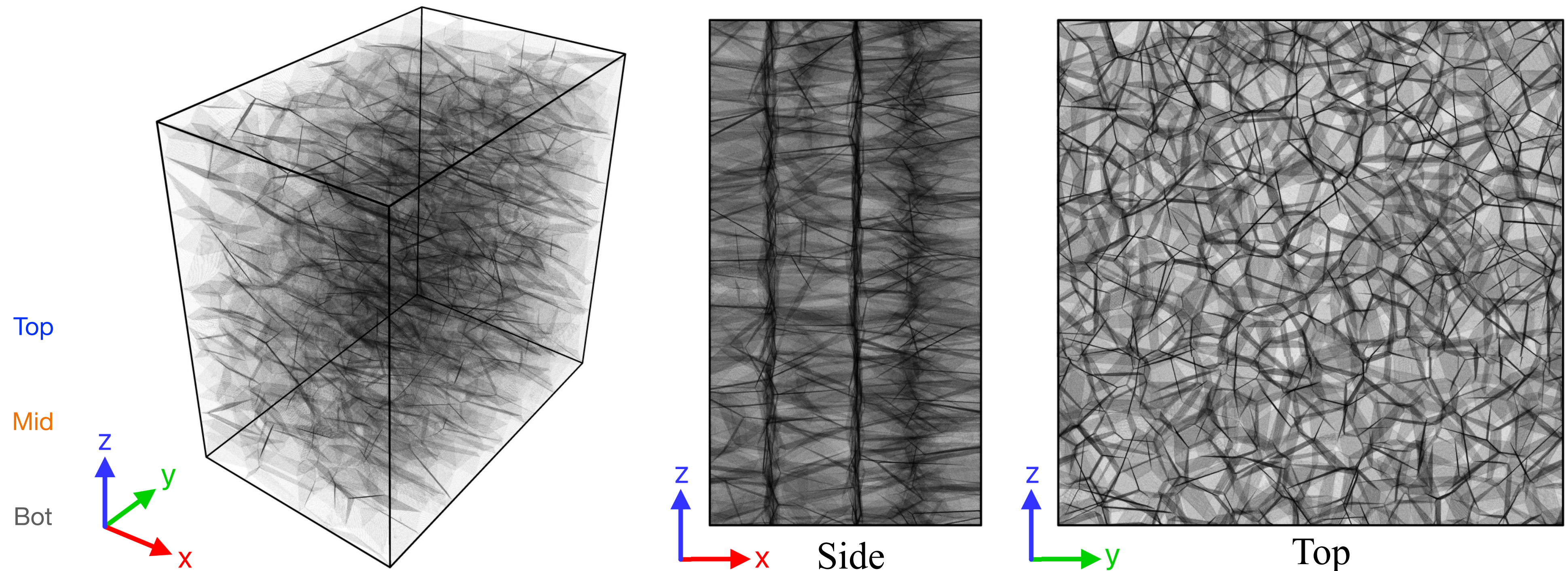
PolyPal: Features

Structure Design: Grain Morphology

*Only non-crystalline atoms are visualised.



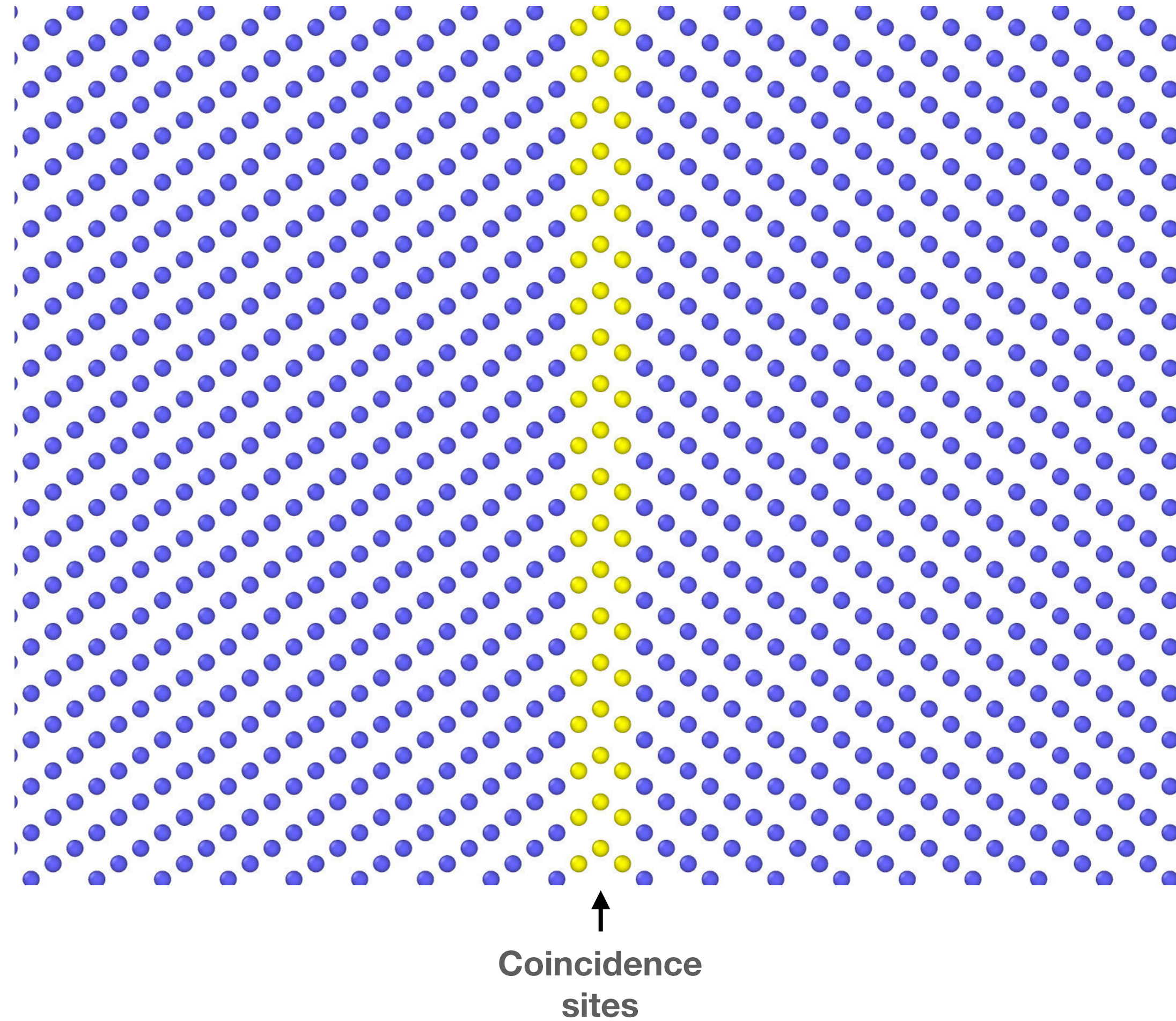
Oh et al., *Extreme Mech. Lett.* (2020)



Massive virtual test specimen (~8 billion atoms)

***PolyPal*: Features**

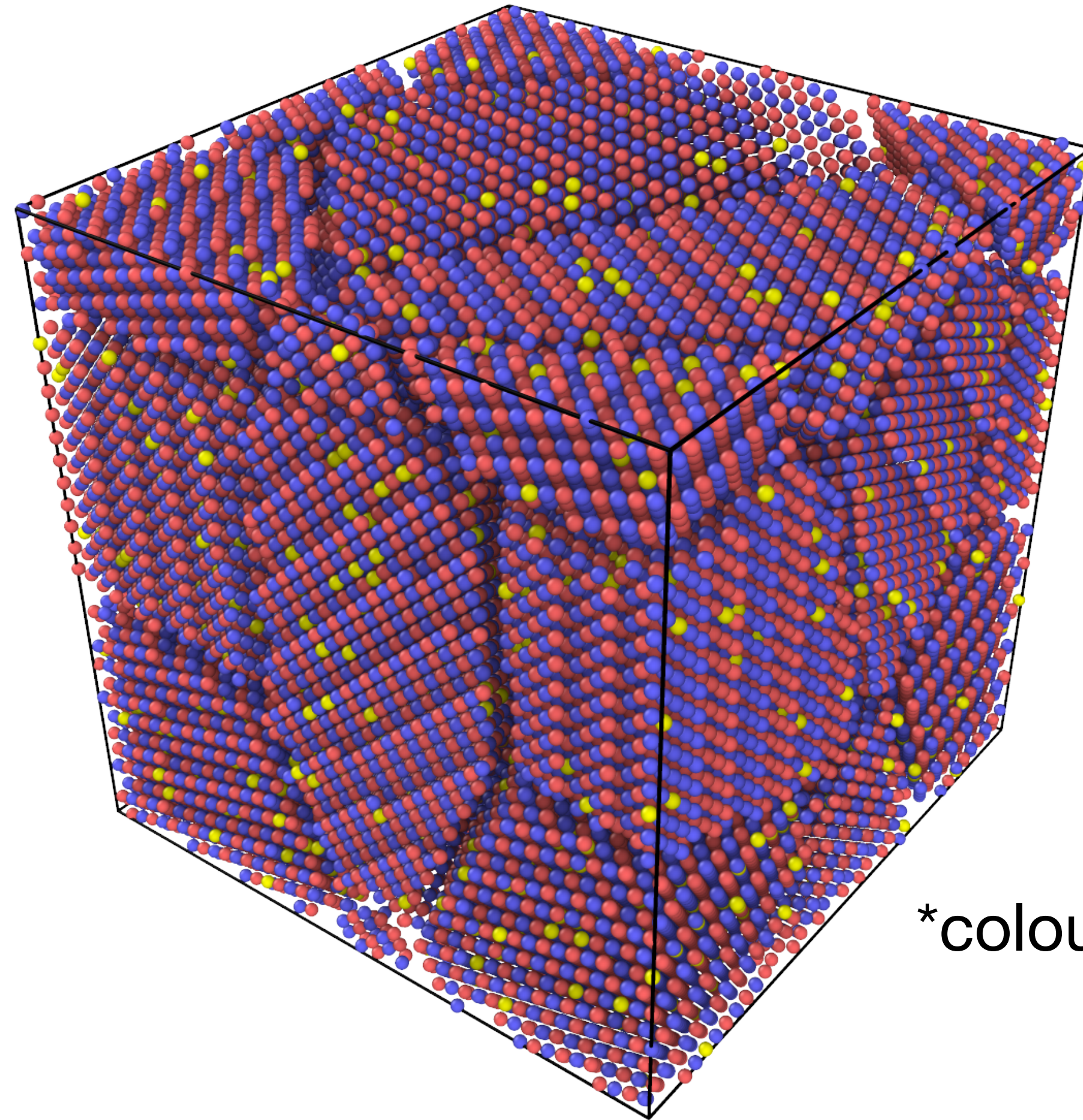
Structure Design: Grain Morphology



Bicrystal: bcc $\Sigma 3(112)$ symmetric tilt

PolyPal: Features

Structure Design: Texture

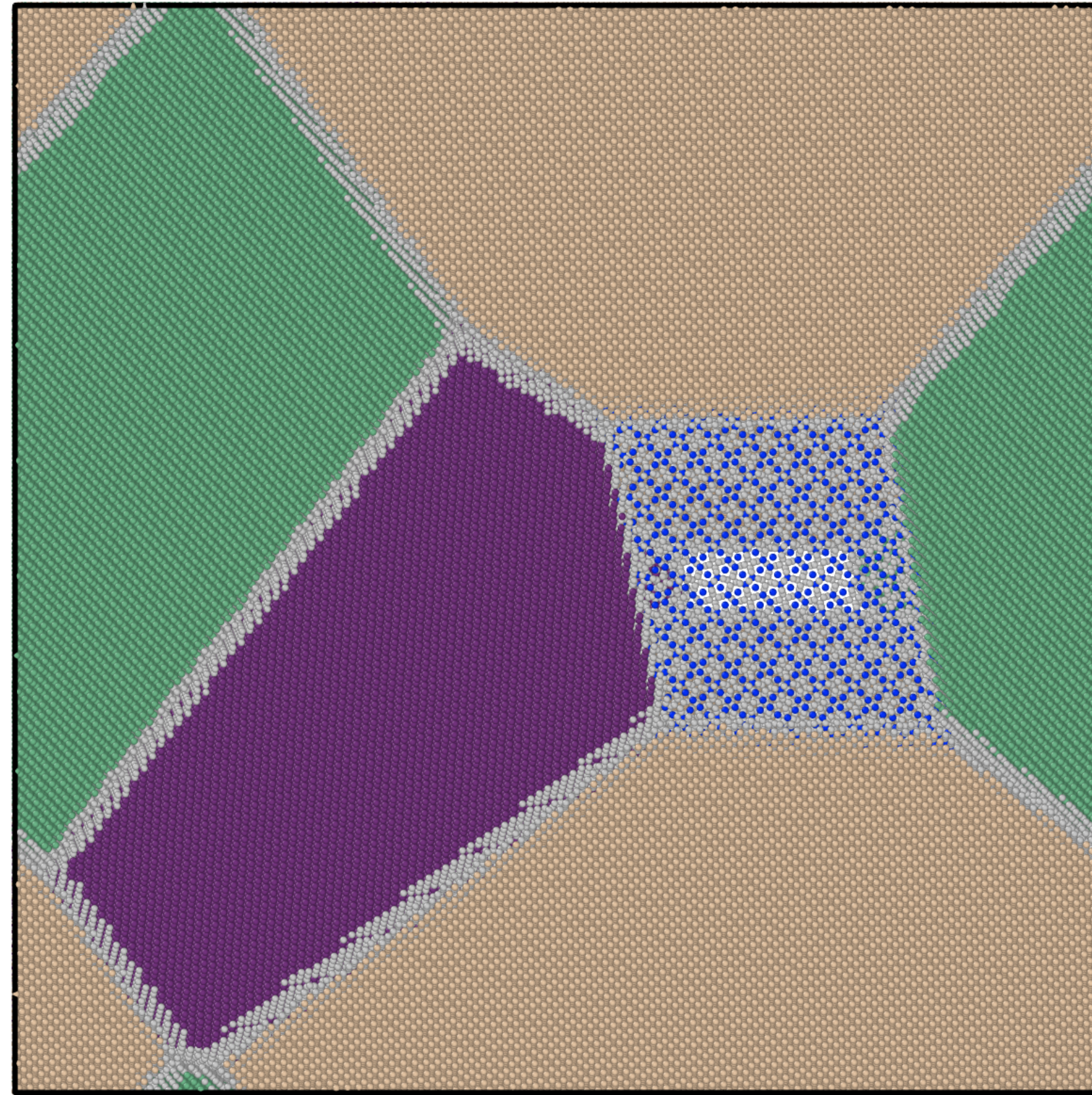


*coloured by atom species

Equiatomic fcc with solute atoms

PolyPal: Features

Structure Design: Texture

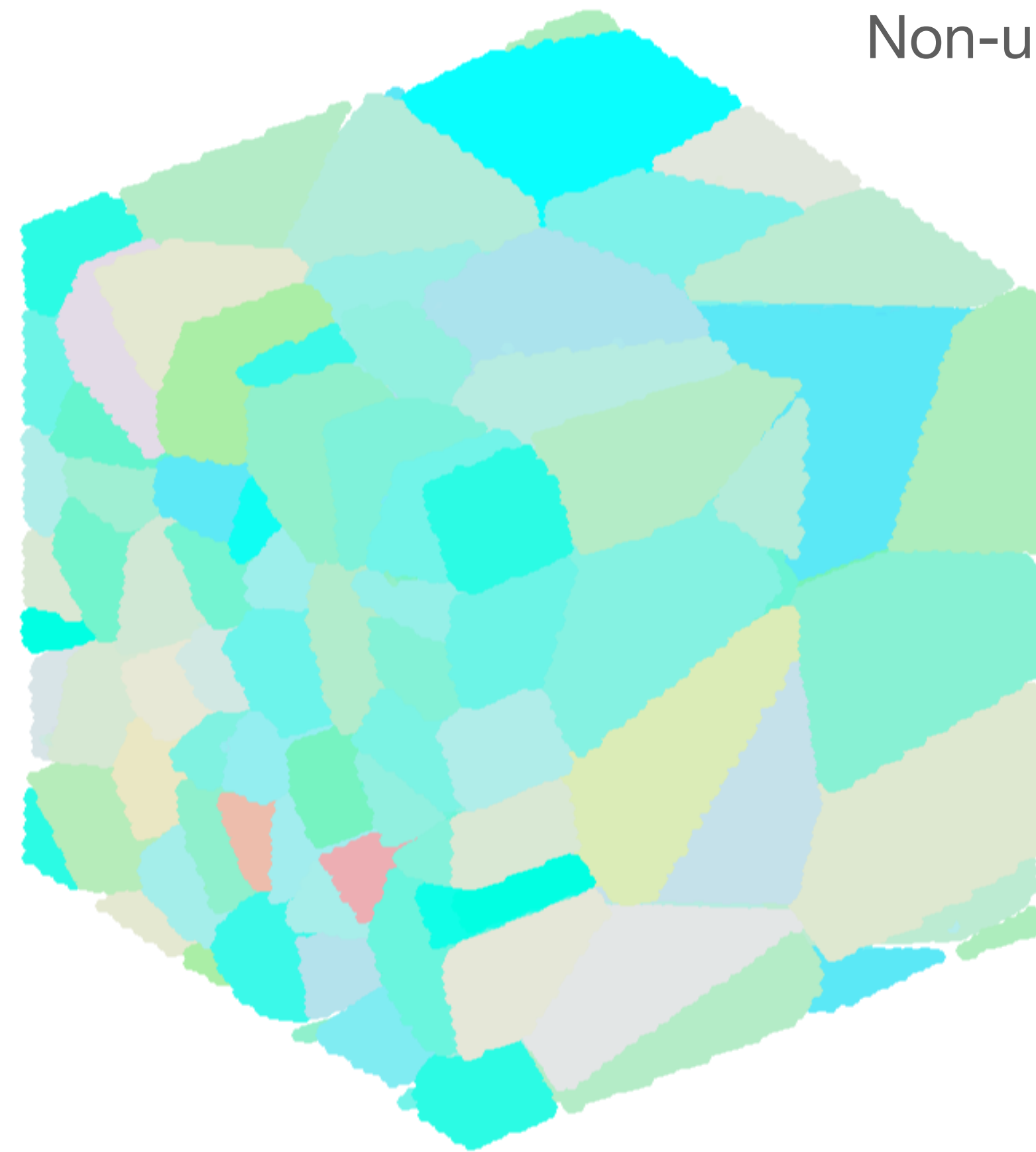
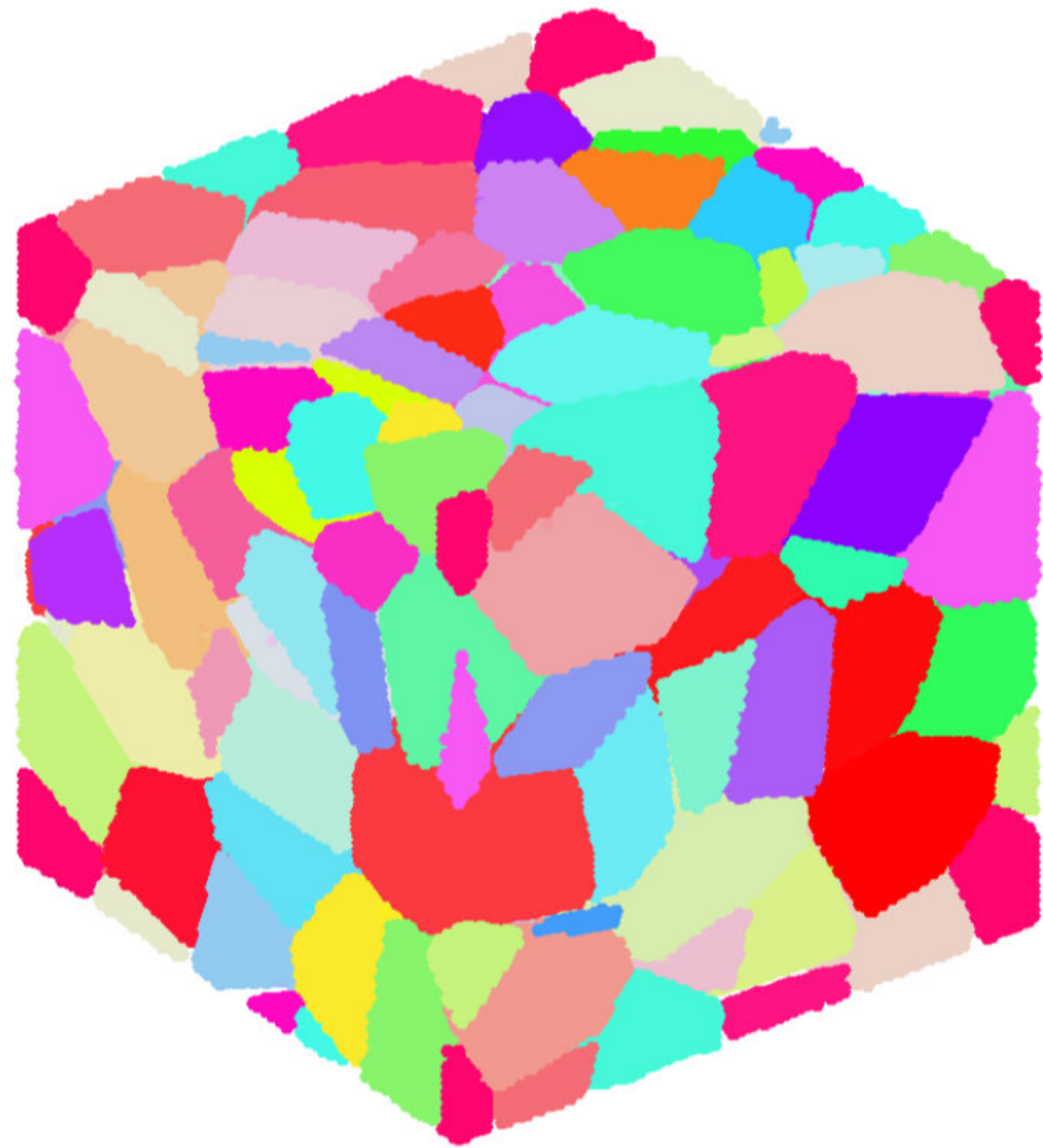


Multiple unit cells

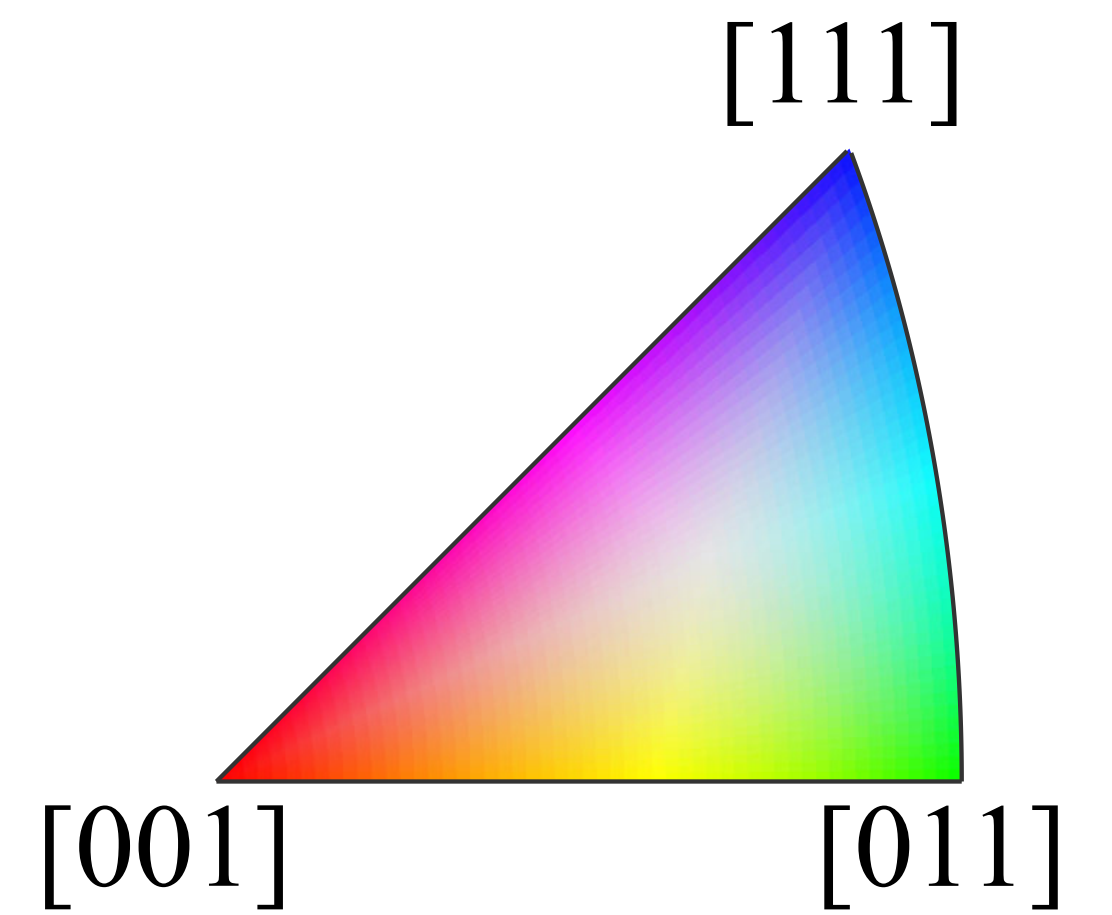
σ -phase precipitates

PolyPal: Features

Structure Design: Texture

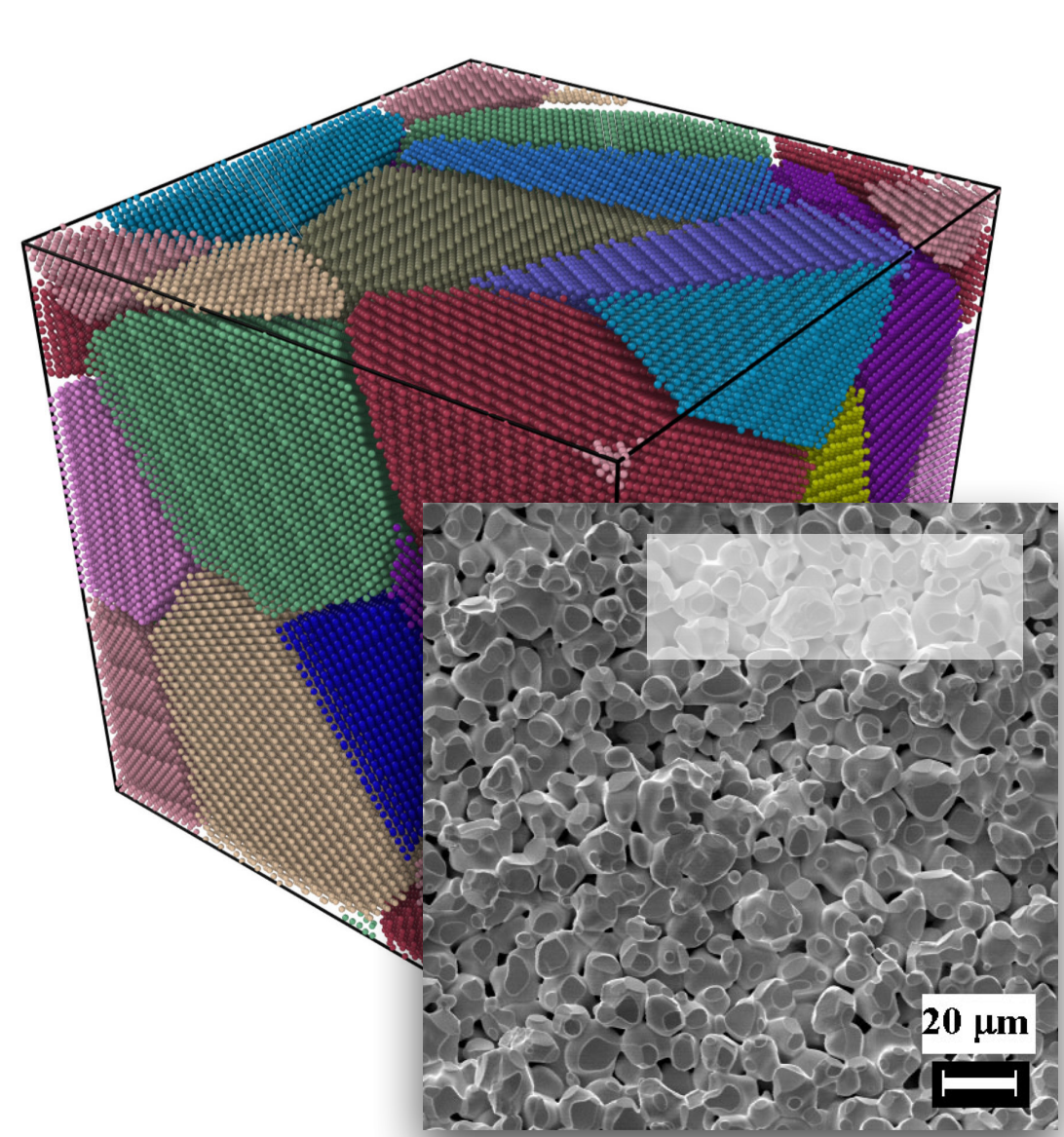


Biased crystallographic direction
Non-uniform grain size distribution

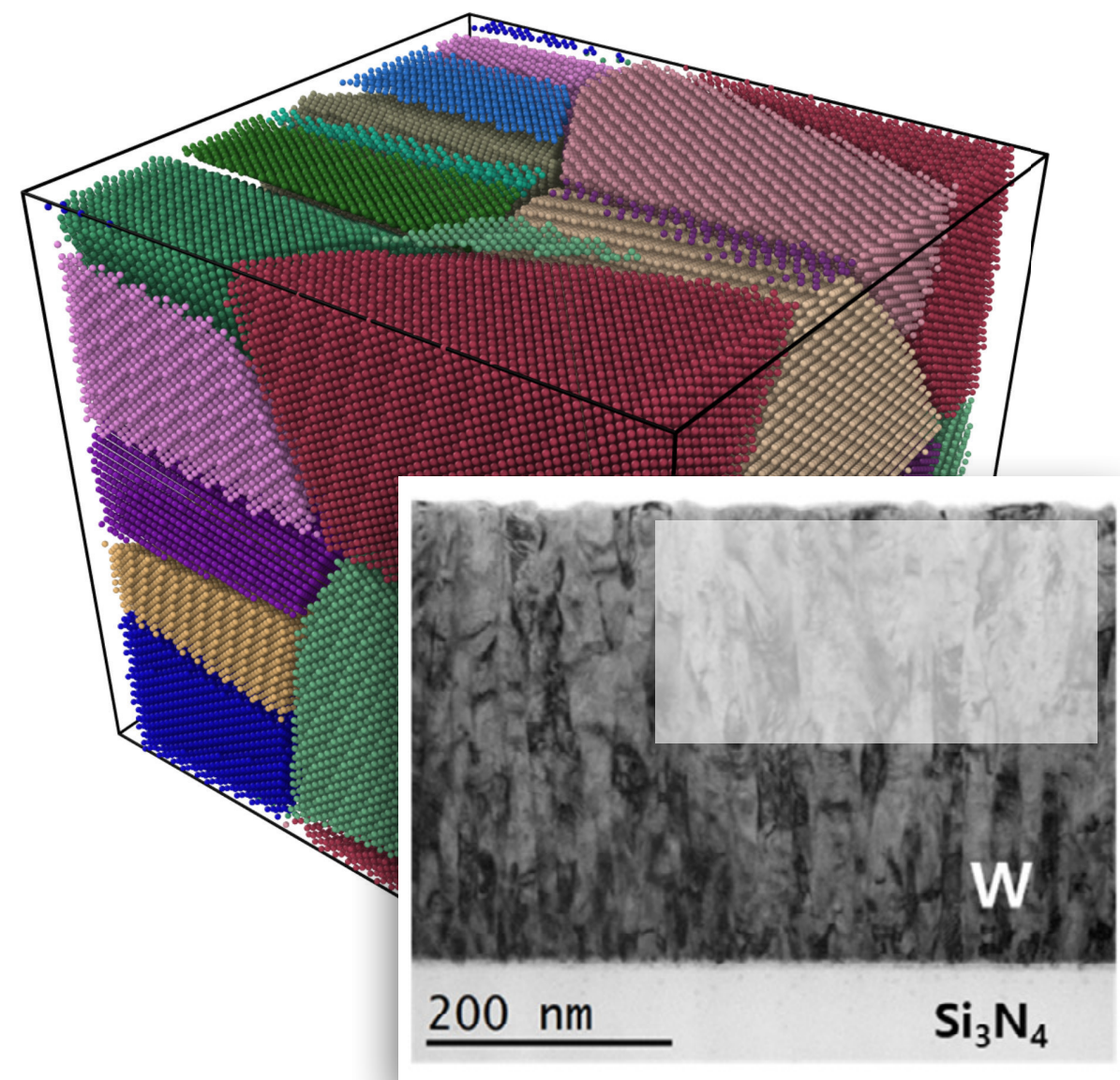


Isotropic versus anisotropic

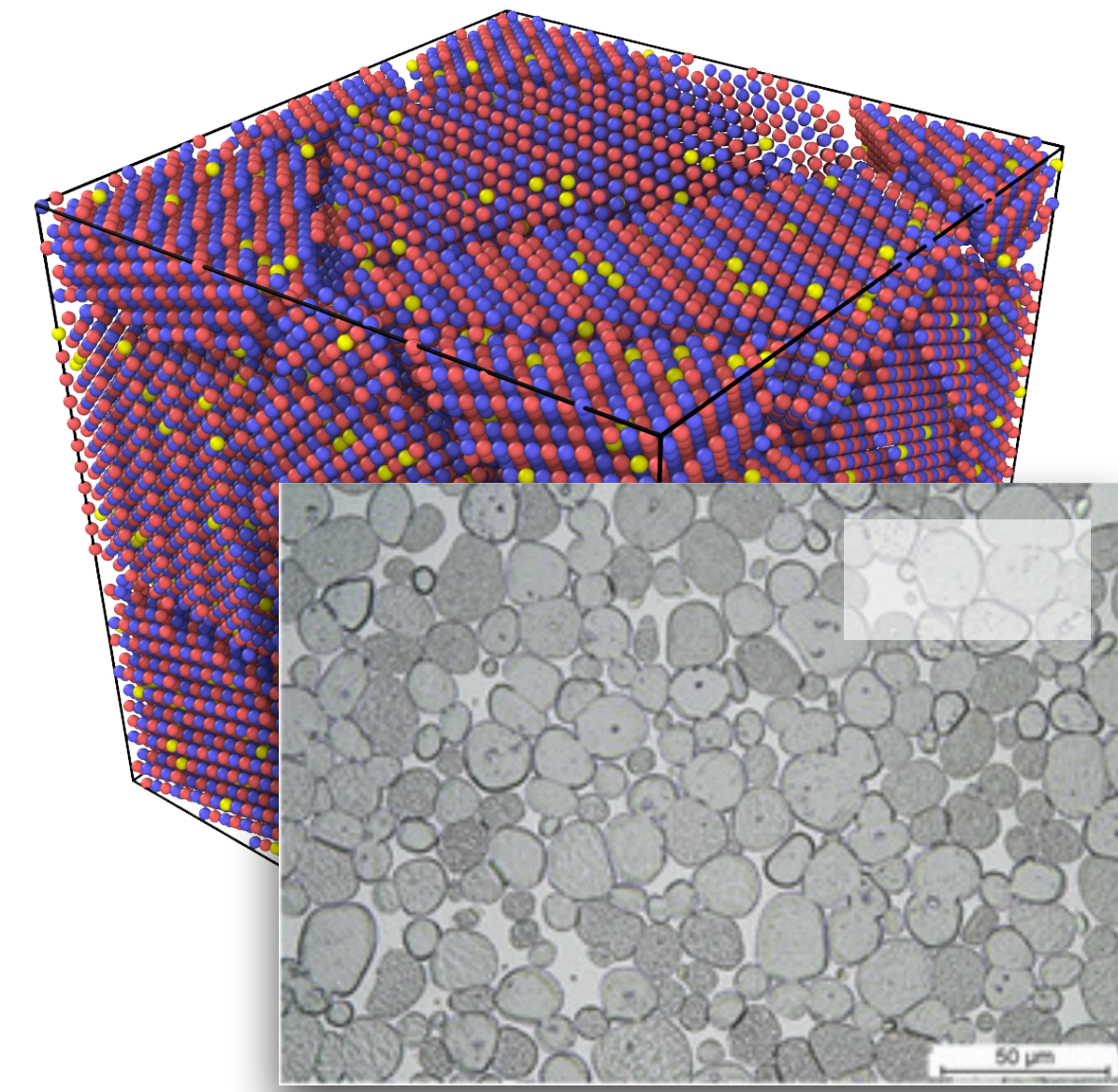
PolyPal Group Photo: Applications



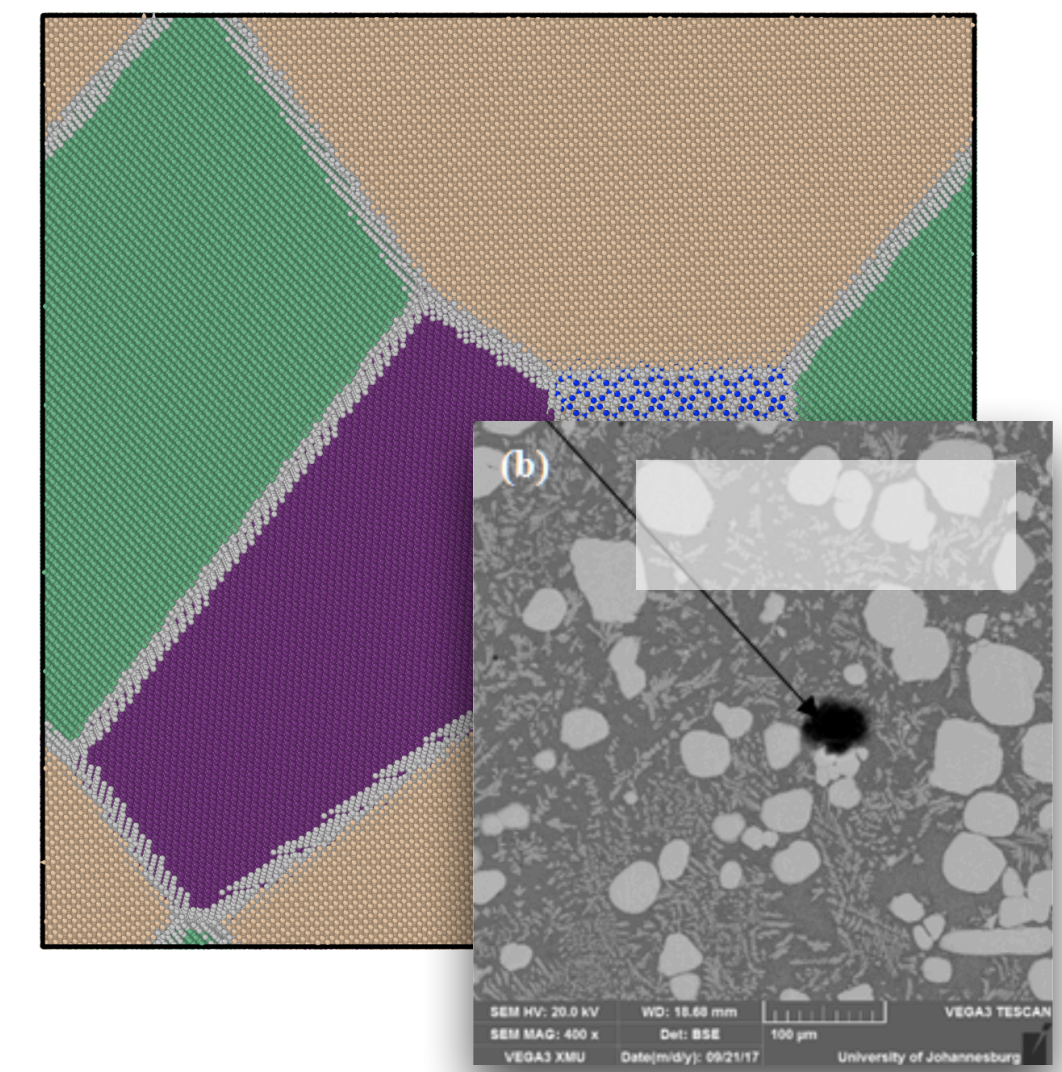
Grabis et al., Proc. Est. Acad. Sc Eng. (2004)



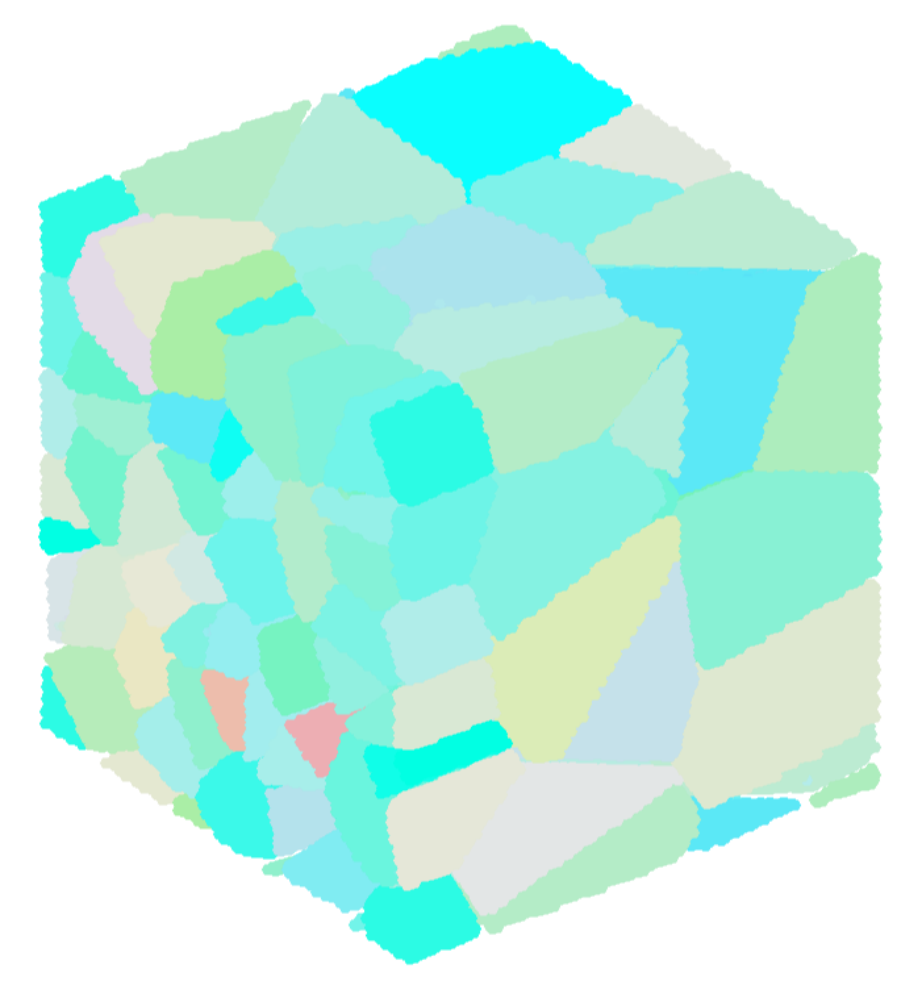
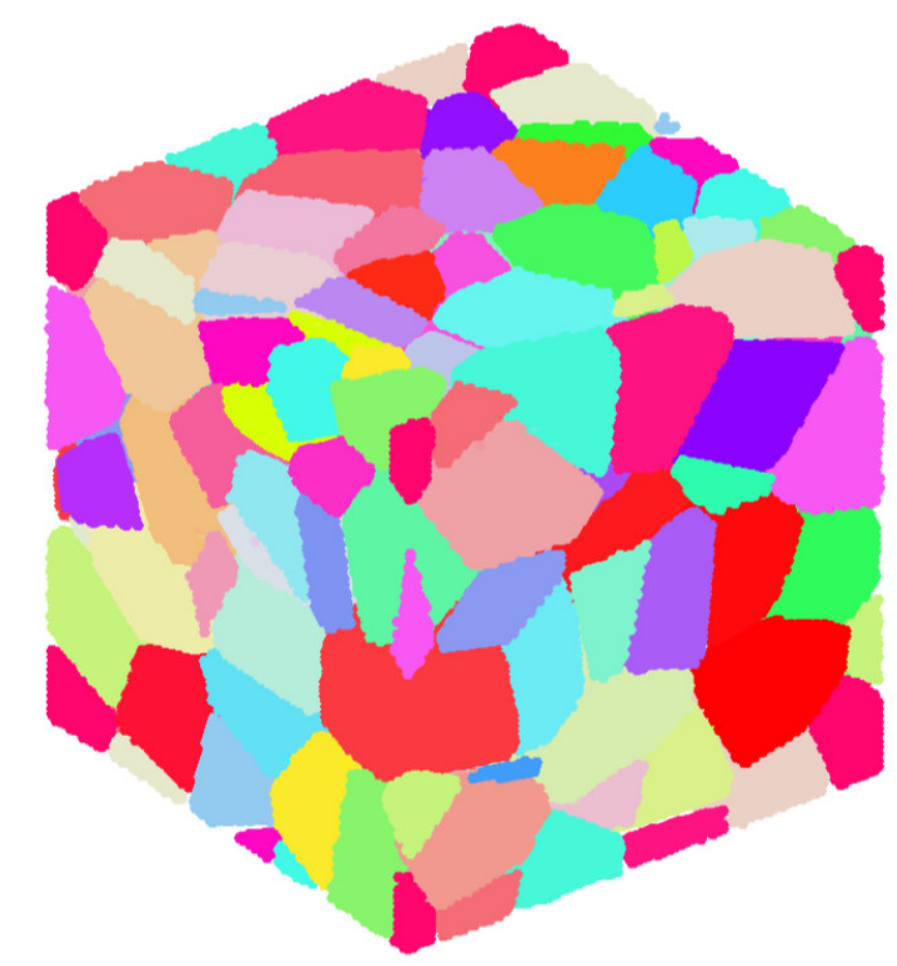
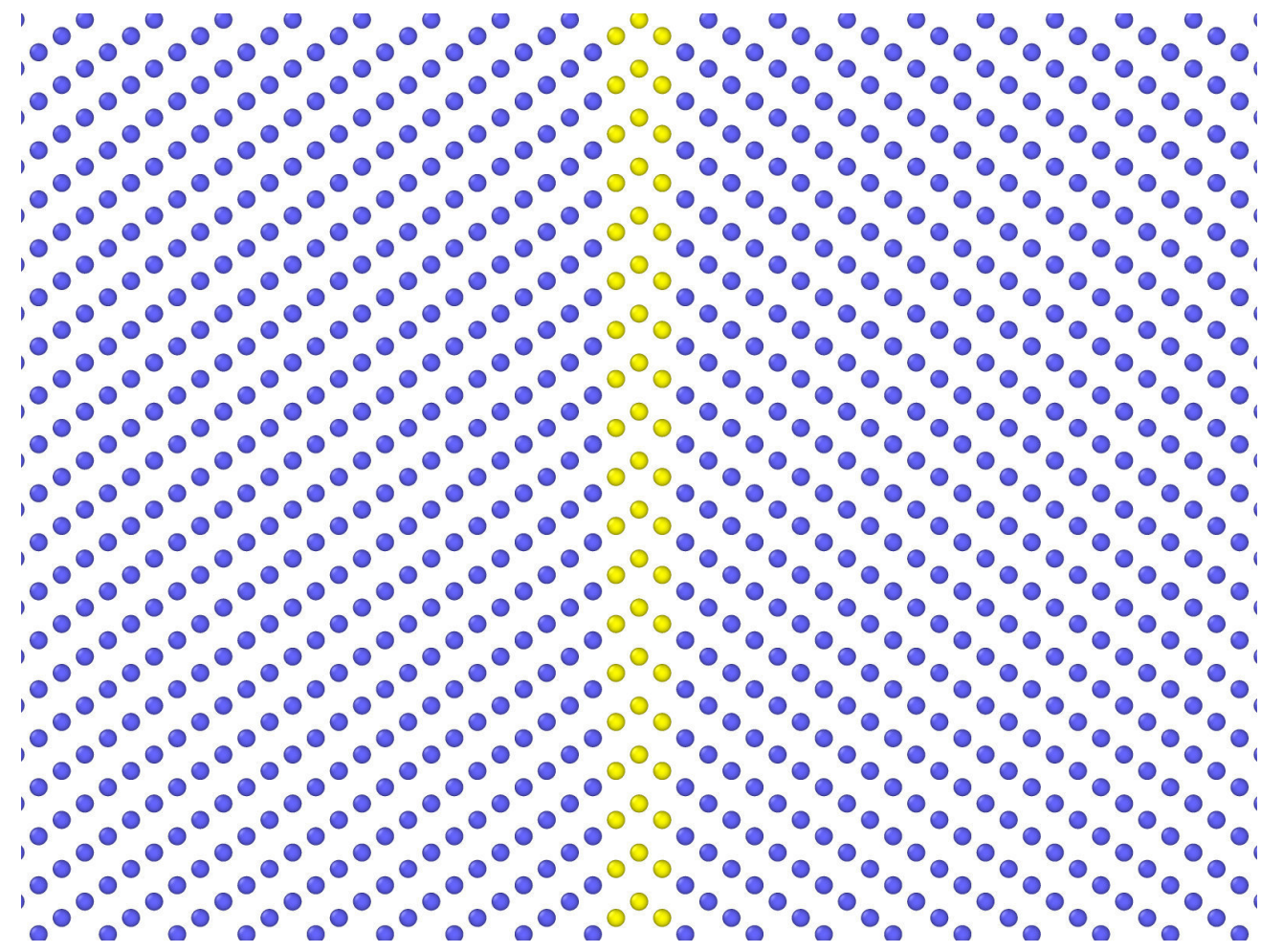
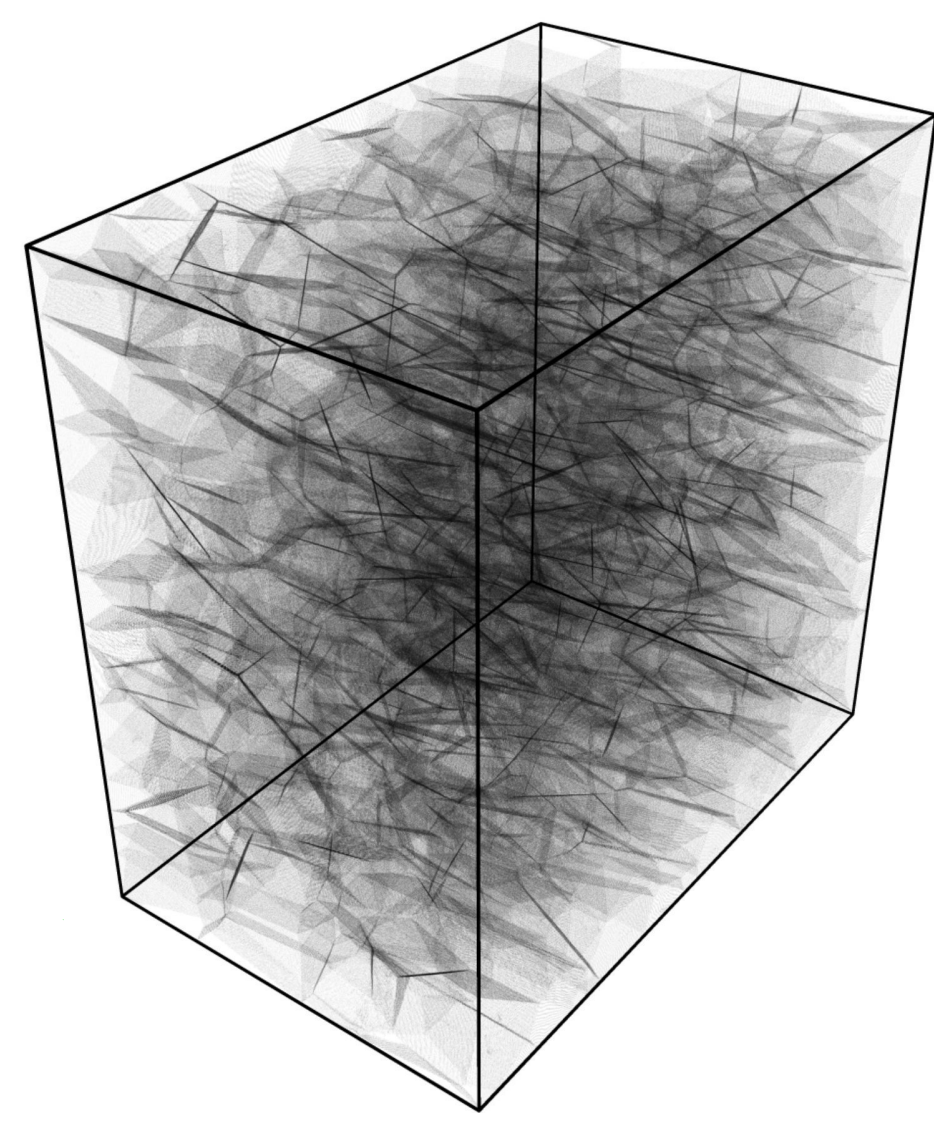
Oh et al., Extreme Mech. Lett. (2020)



Skoczylas et al., Materials (2021)

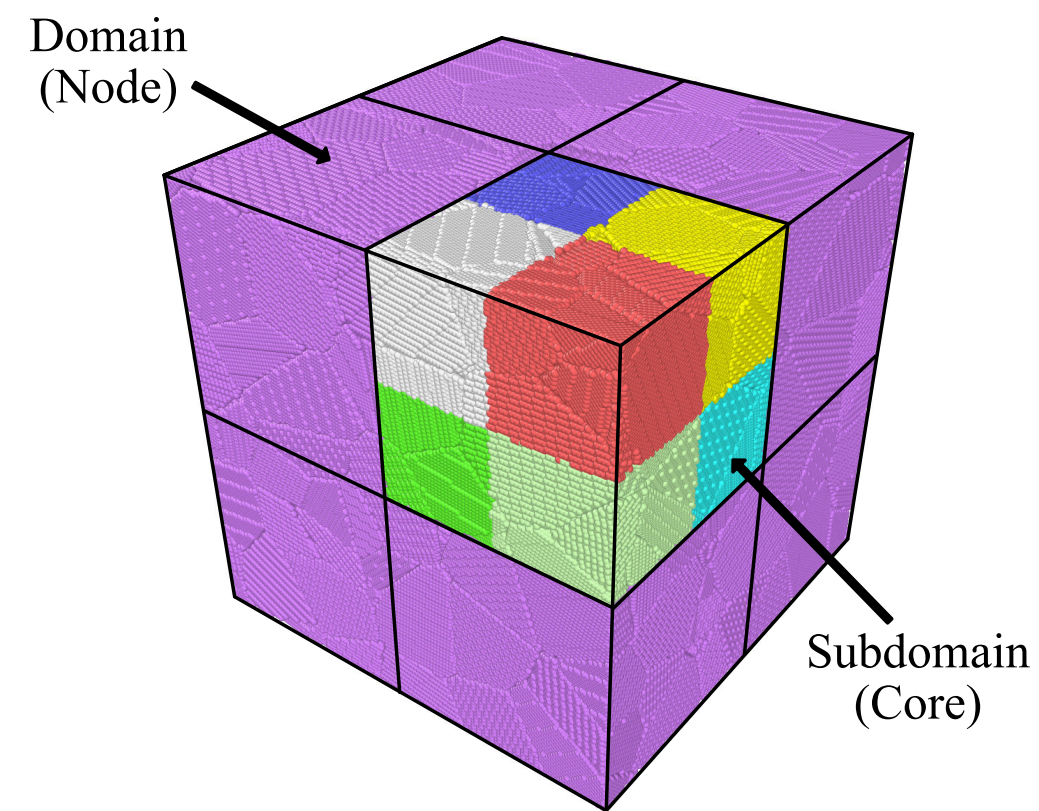


Skoczylas et al., Materials (2021)

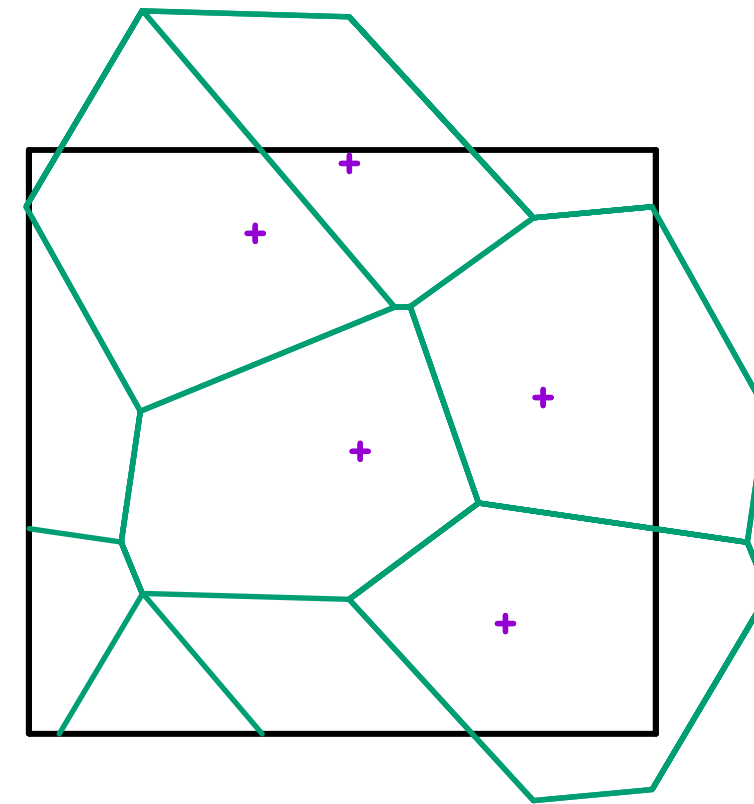


Workflow Overview

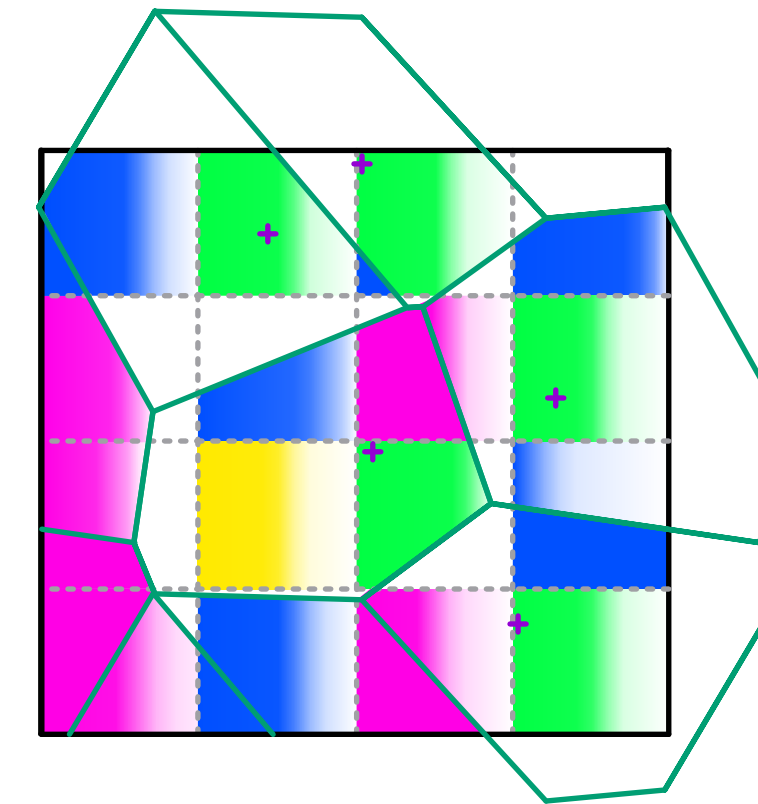
(1) Domain decomposition



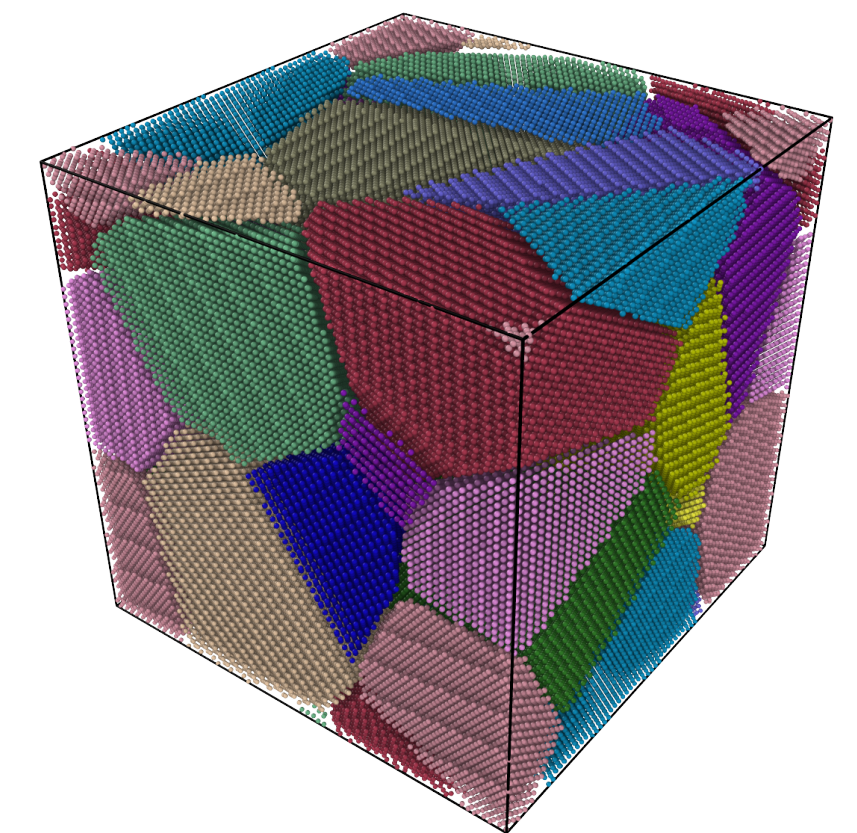
(2) Grain population



(3) Grain distribution & atom filling

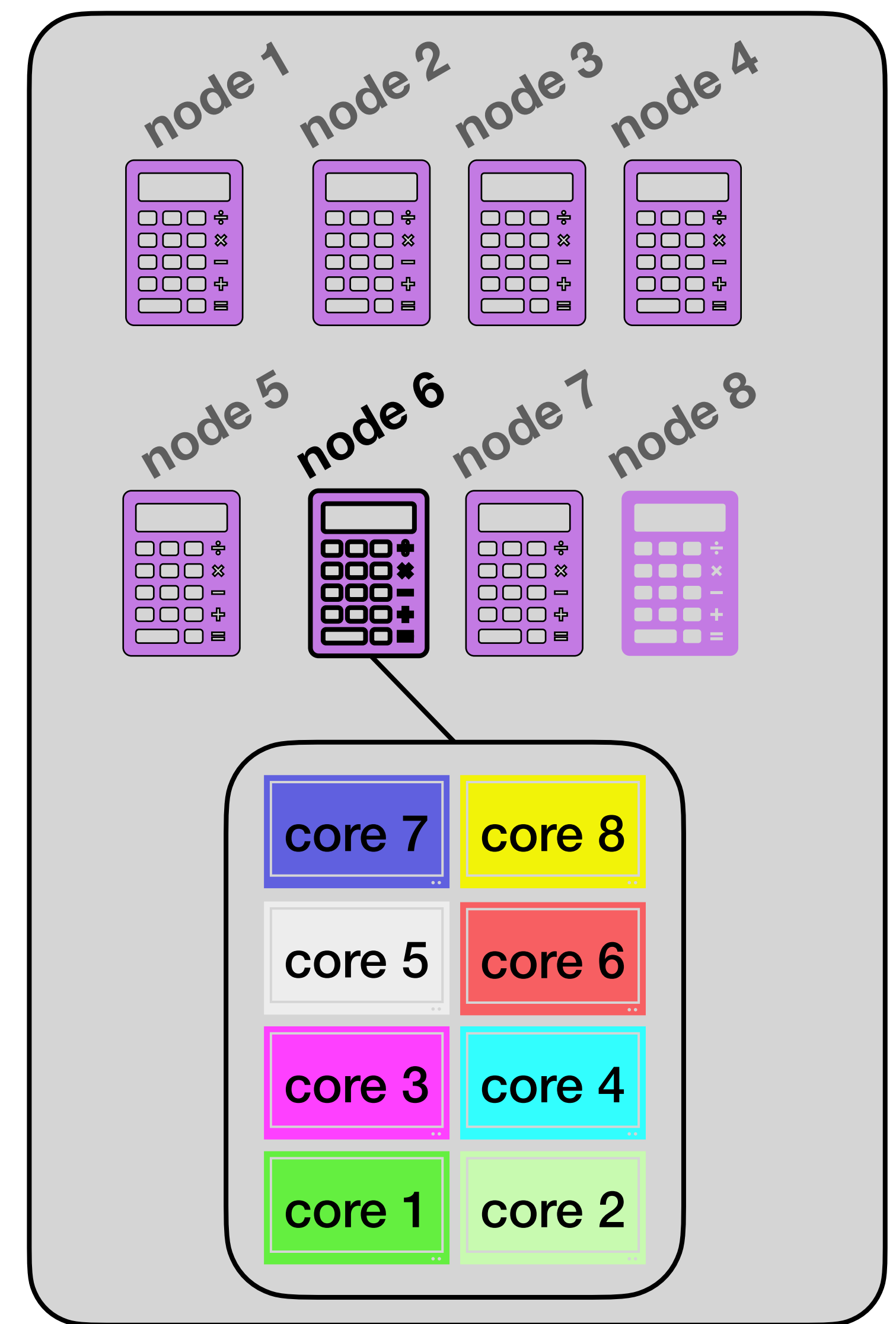
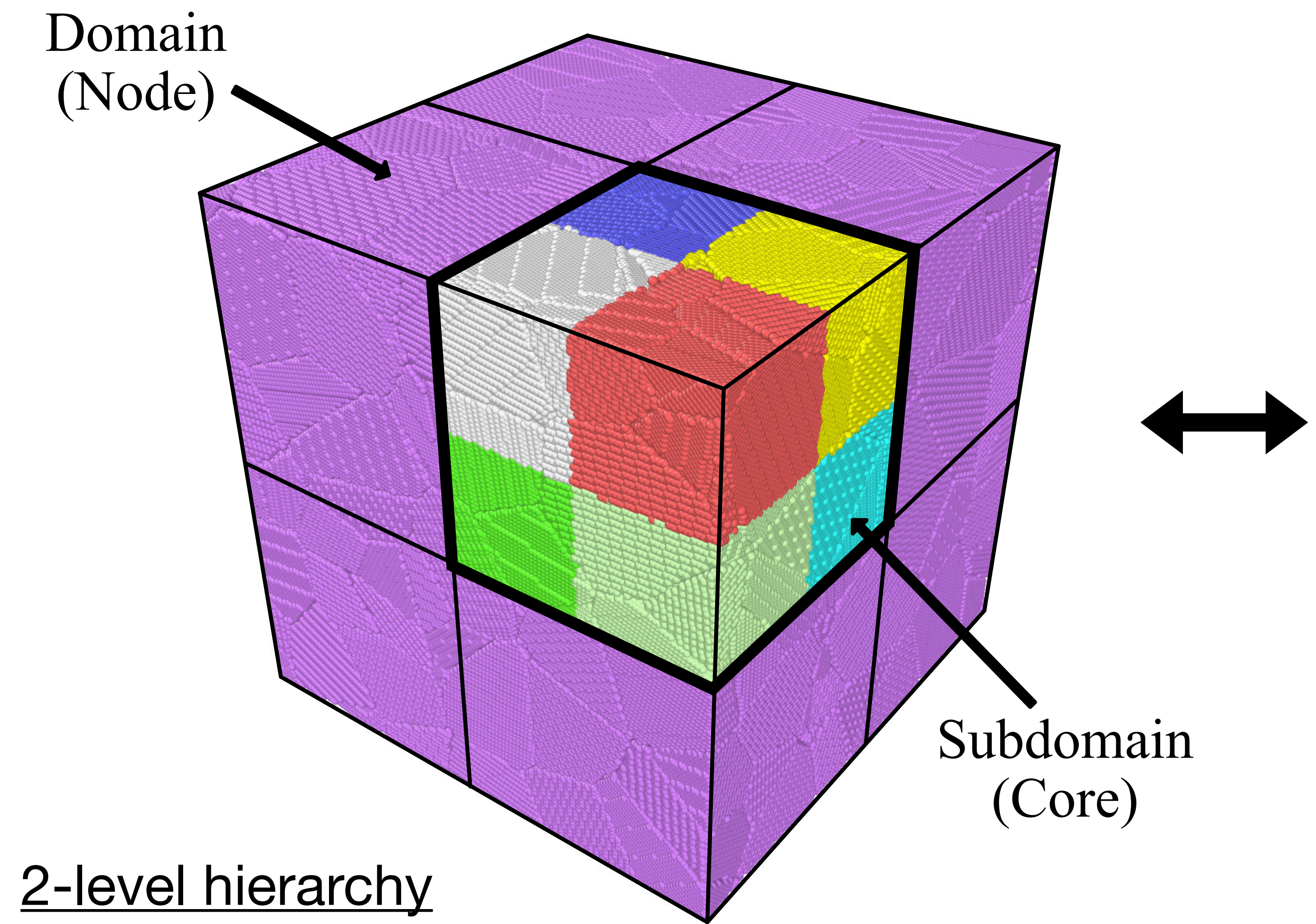


(4) File output

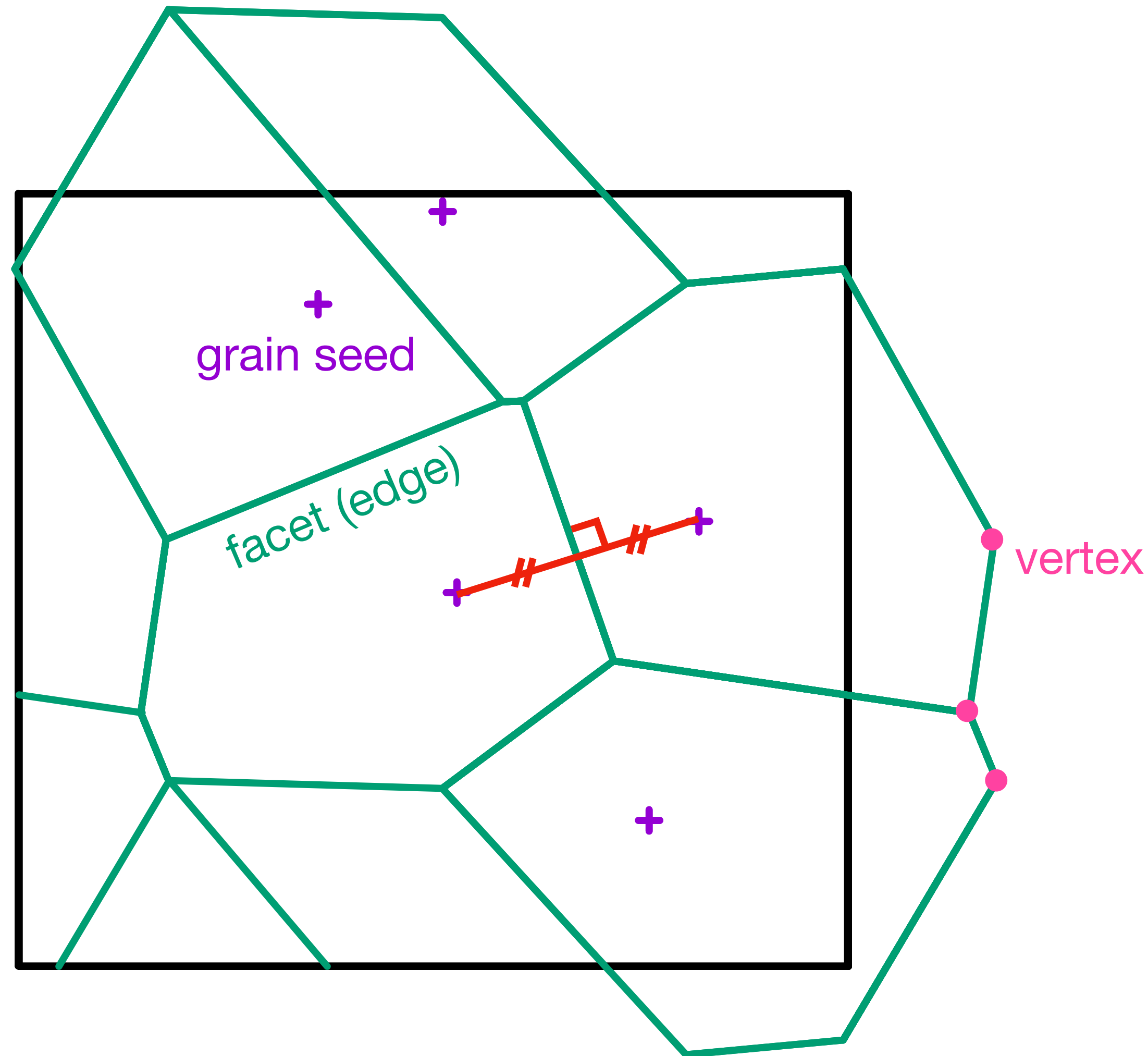


Domain Decomposition

HPC (8 nodes x 8 cores)

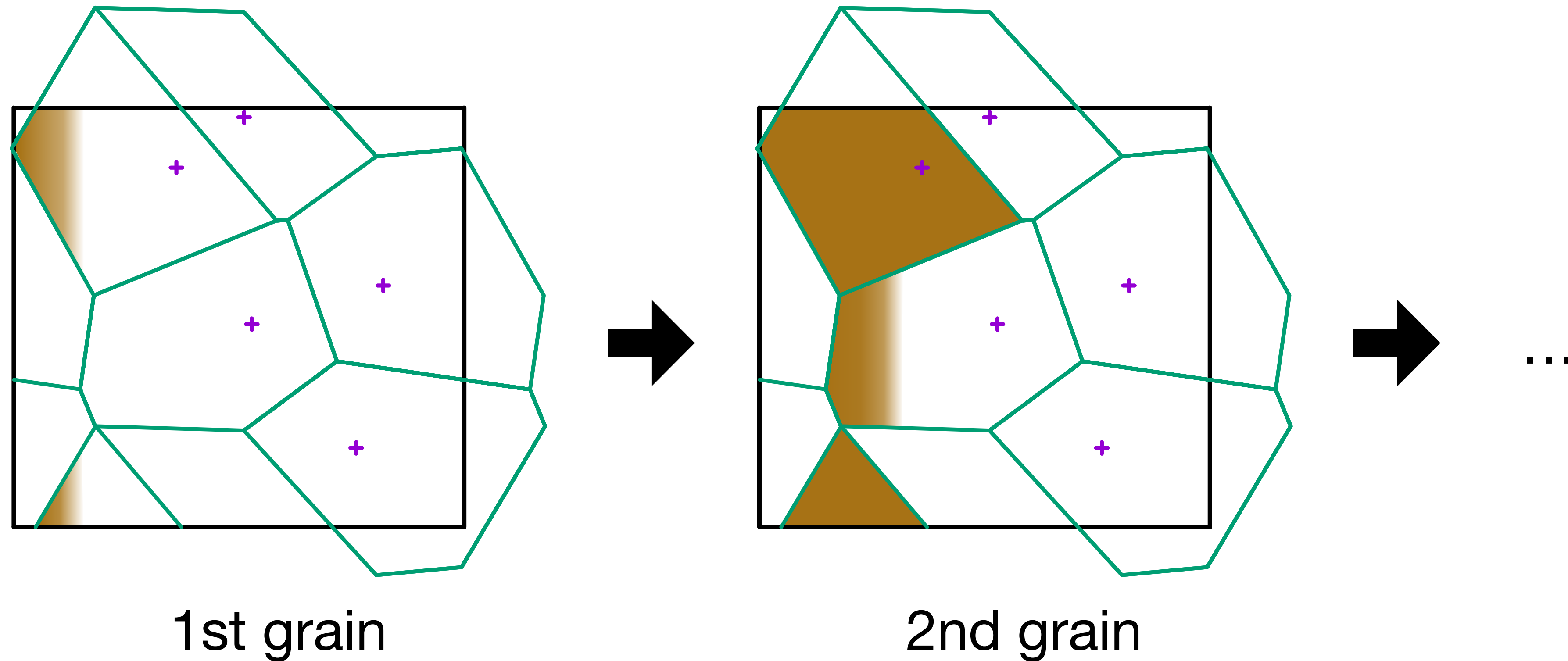


Grain Population



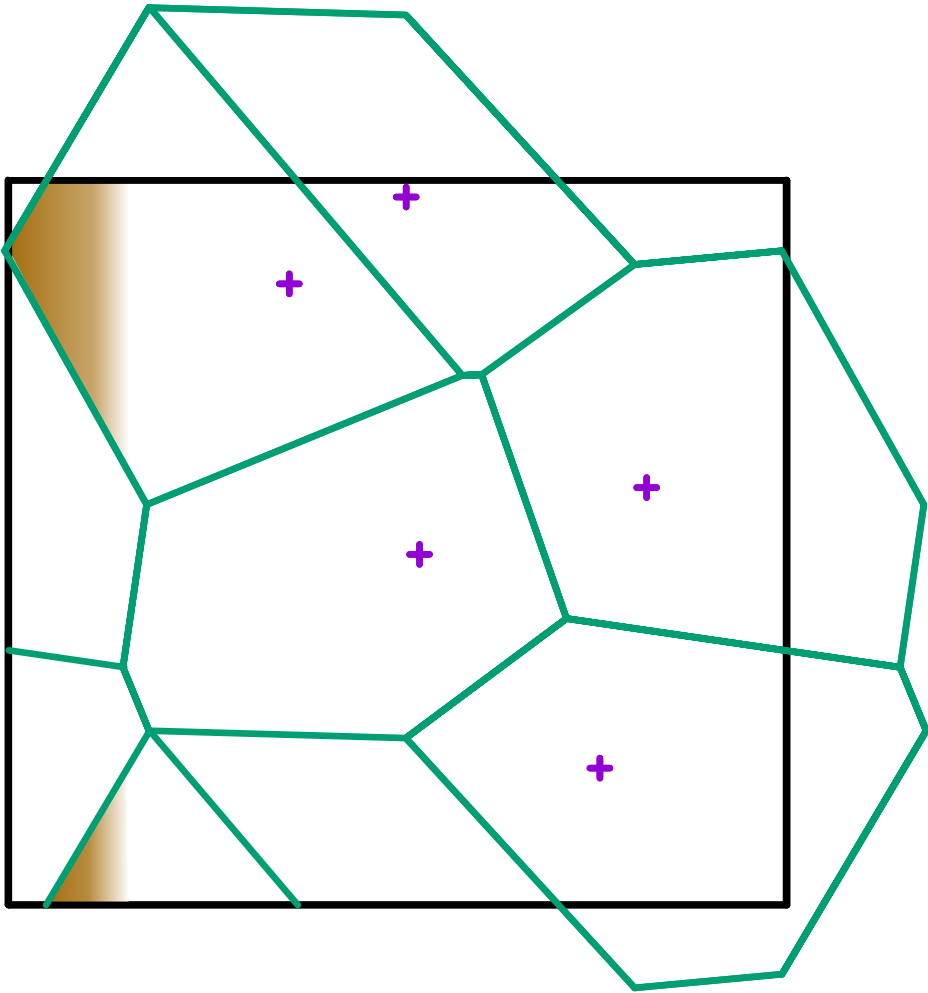
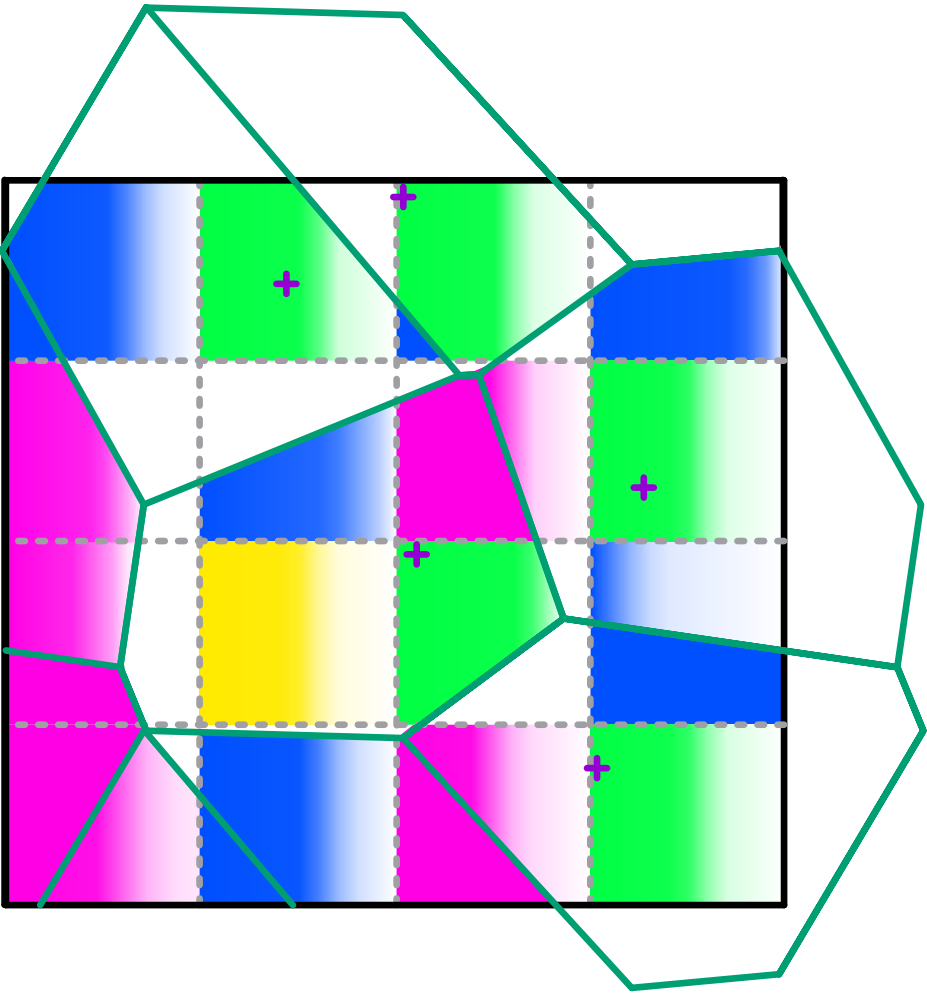
- Centroidal Voronoi tessellation
- Grains constructed from
 - seeds with crystal orientation
- Grain information
 - crystallographic orientation
 - seed points
 - vertices
 - edges
 - list of neighbour grains

Grain Distribution: Serial

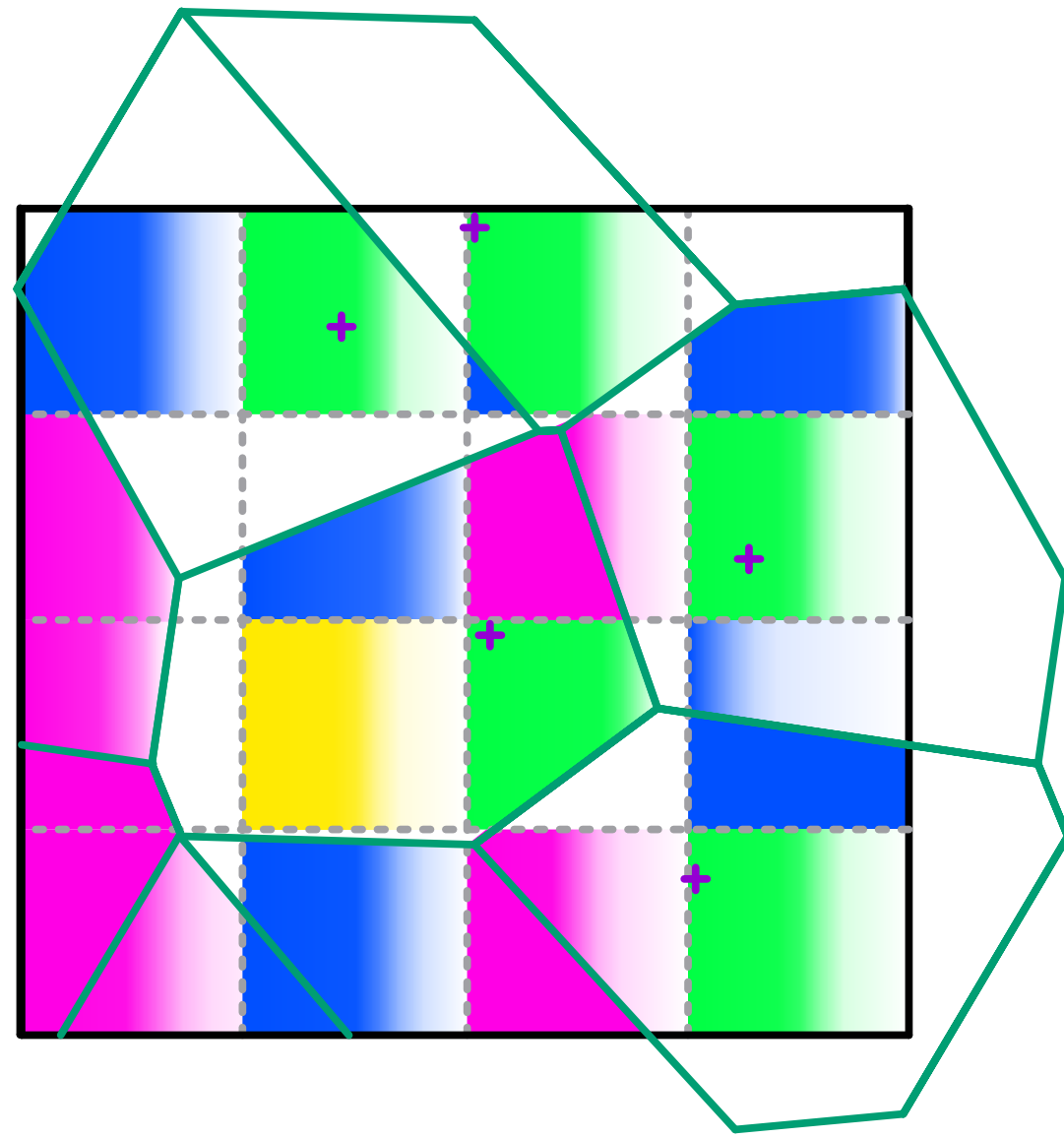


One grain at a time: no grain distribution required in serial approach

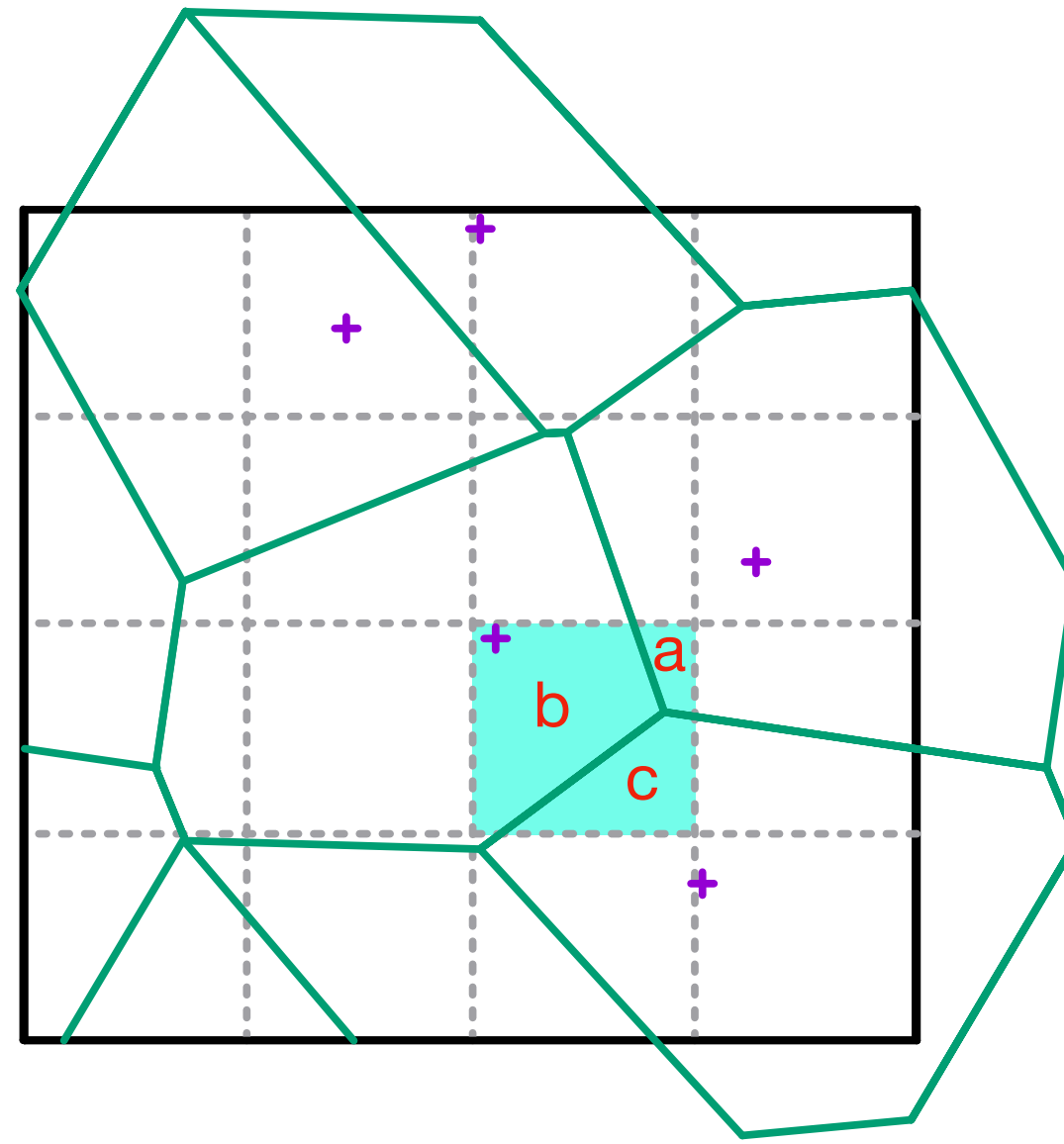
Grain Distribution: Parallel

Parallelization strategy	 w/o domain decomposition Each core assigned to a grain	 All detected grains in a subdomain partially filled by corresponding core
# working cores	restricted to number of grains	all cores
Inter-core comm.	required if parallel I/O applied	no
Memory	potential memory lackage issue	all cores require similar amount of memory

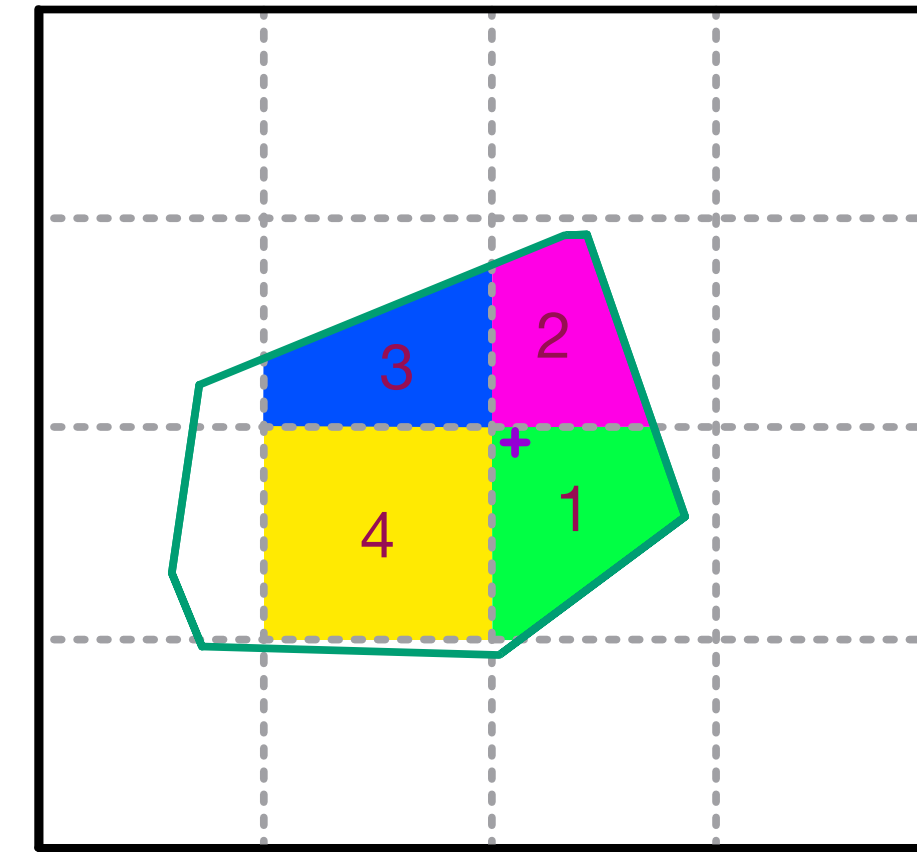
Grain Distribution and Atom Generation



- 1) Each core finds grains in its subdomain.
- 2) All cores fill individual subdomains simultaneously.
(solving plane equations)

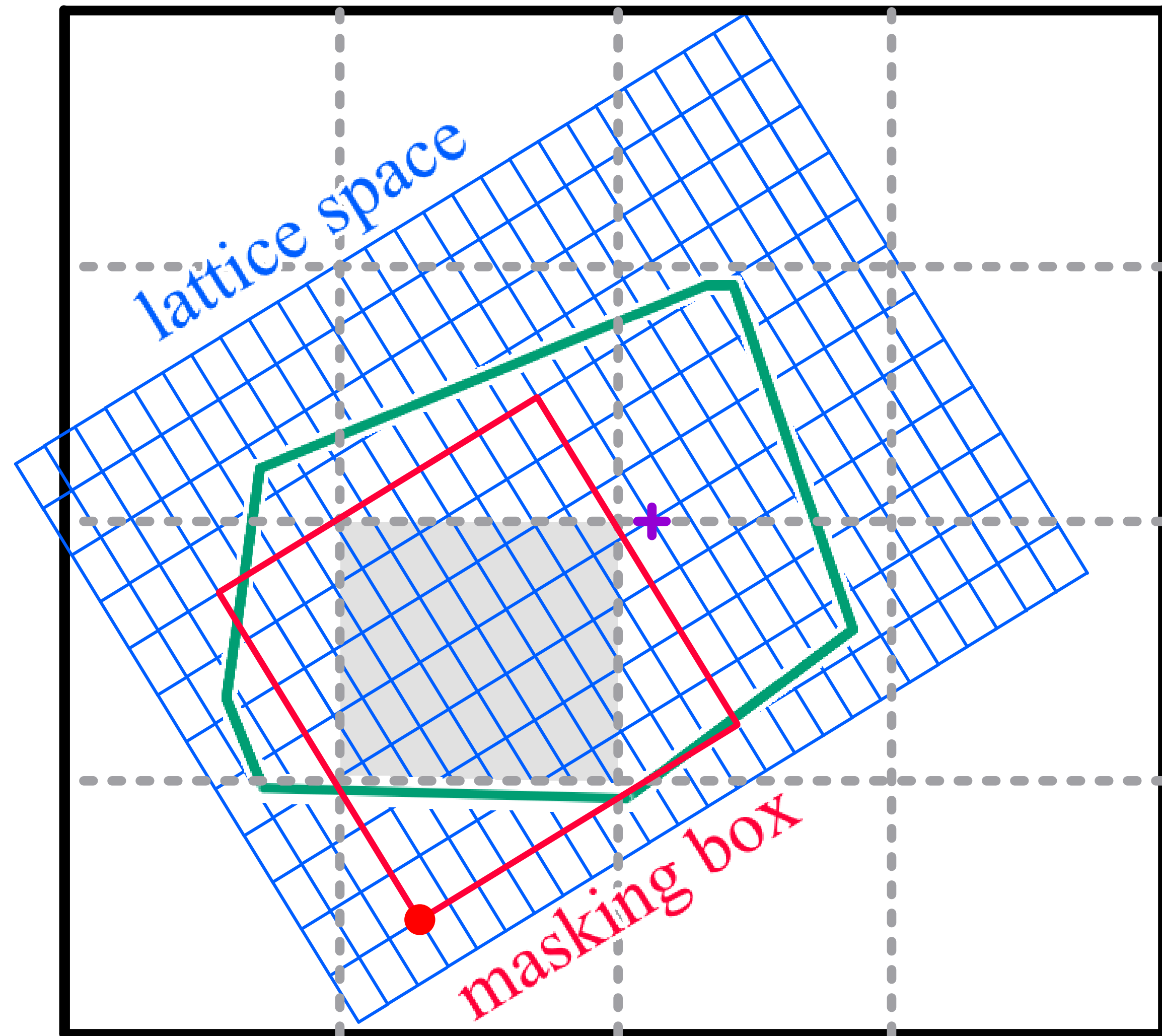


A core can have multiple grains.
A grain can span over multiple subdomains.

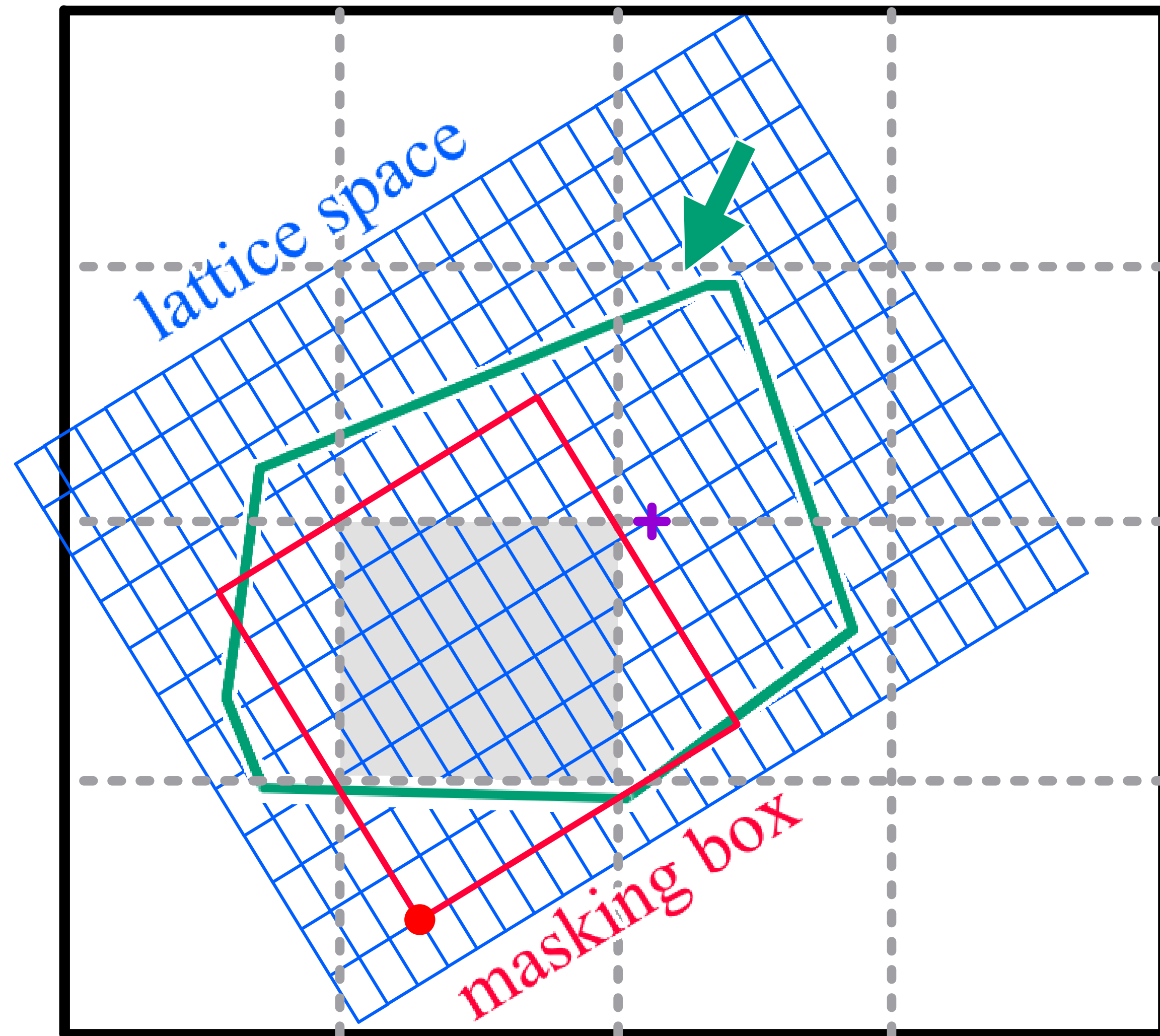


- (1-green) A seed located inside subdomain
- (2-magenta) At least one vertex inside subdomain
- (3-blue) At least one edge penetrating across the subdomain
- (4-yellow) A grain wrapping the subdomain.

Grain Distribution and Atom Generation

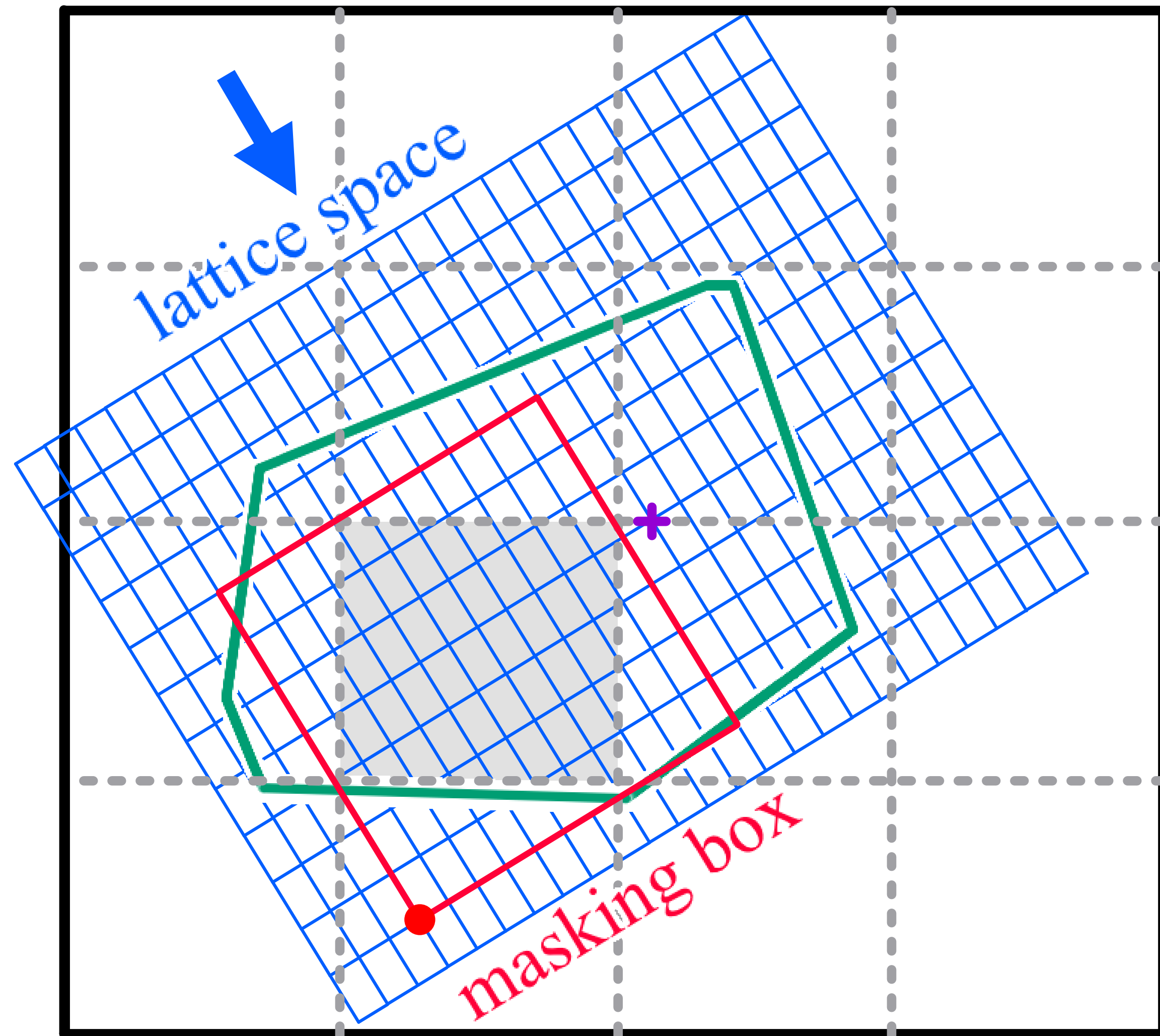


Grain Distribution and Atom Generation



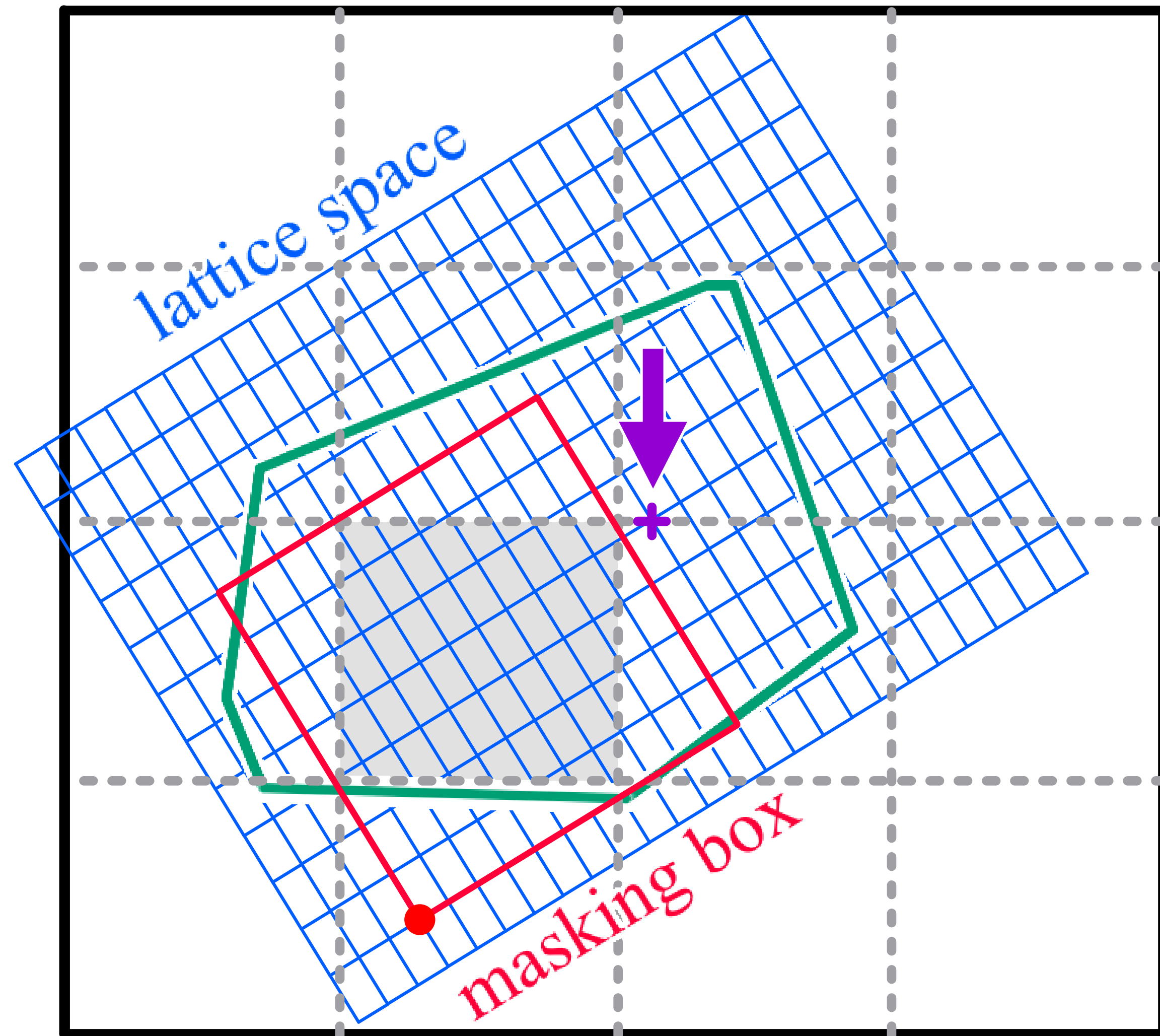
- A grain can span over multiple subdomains.

Grain Distribution and Atom Generation



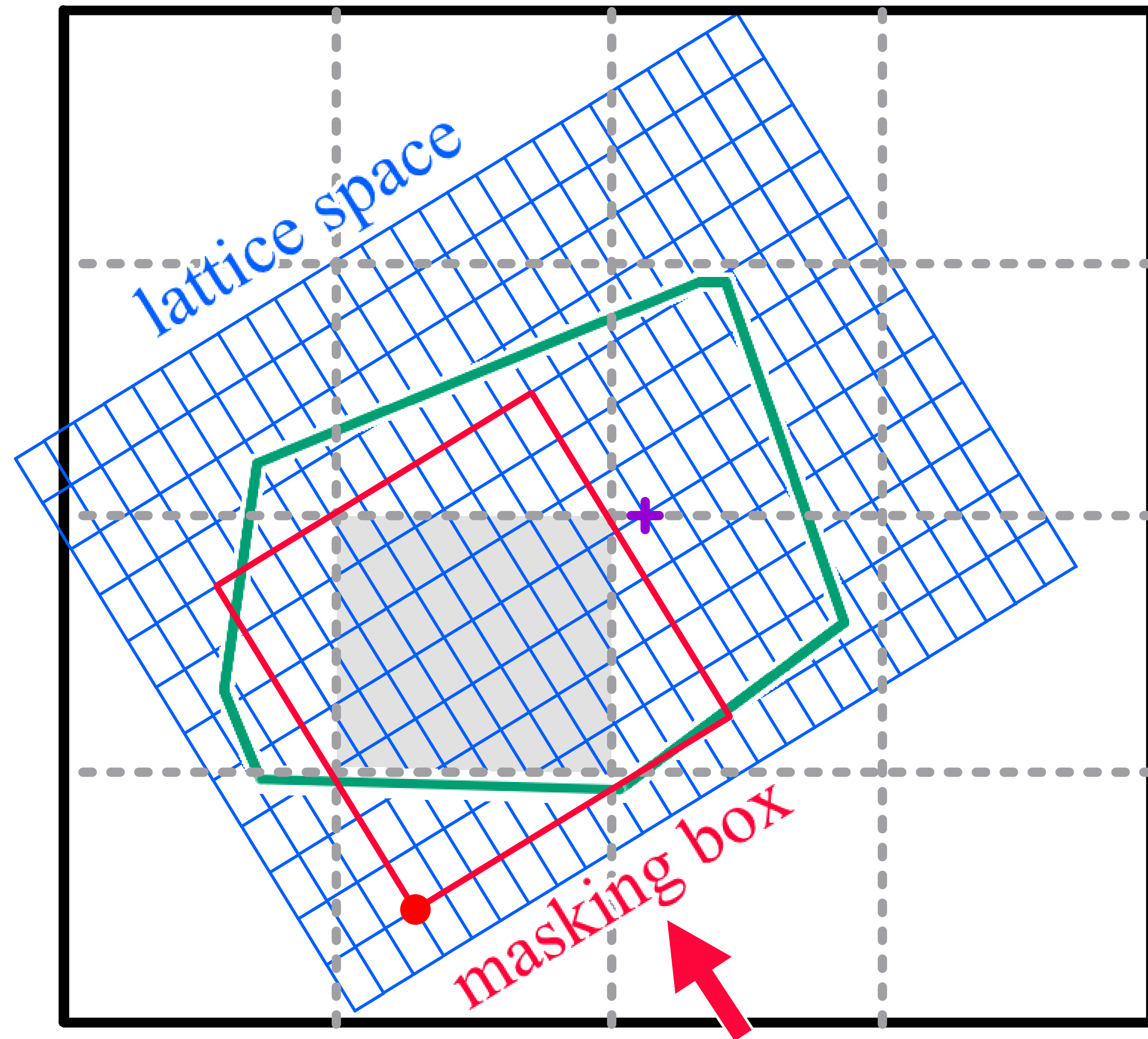
- A grain can span over multiple subdomains.
- To ensure the continuity, cores share the same lattice space.

Grain Distribution and Atom Generation



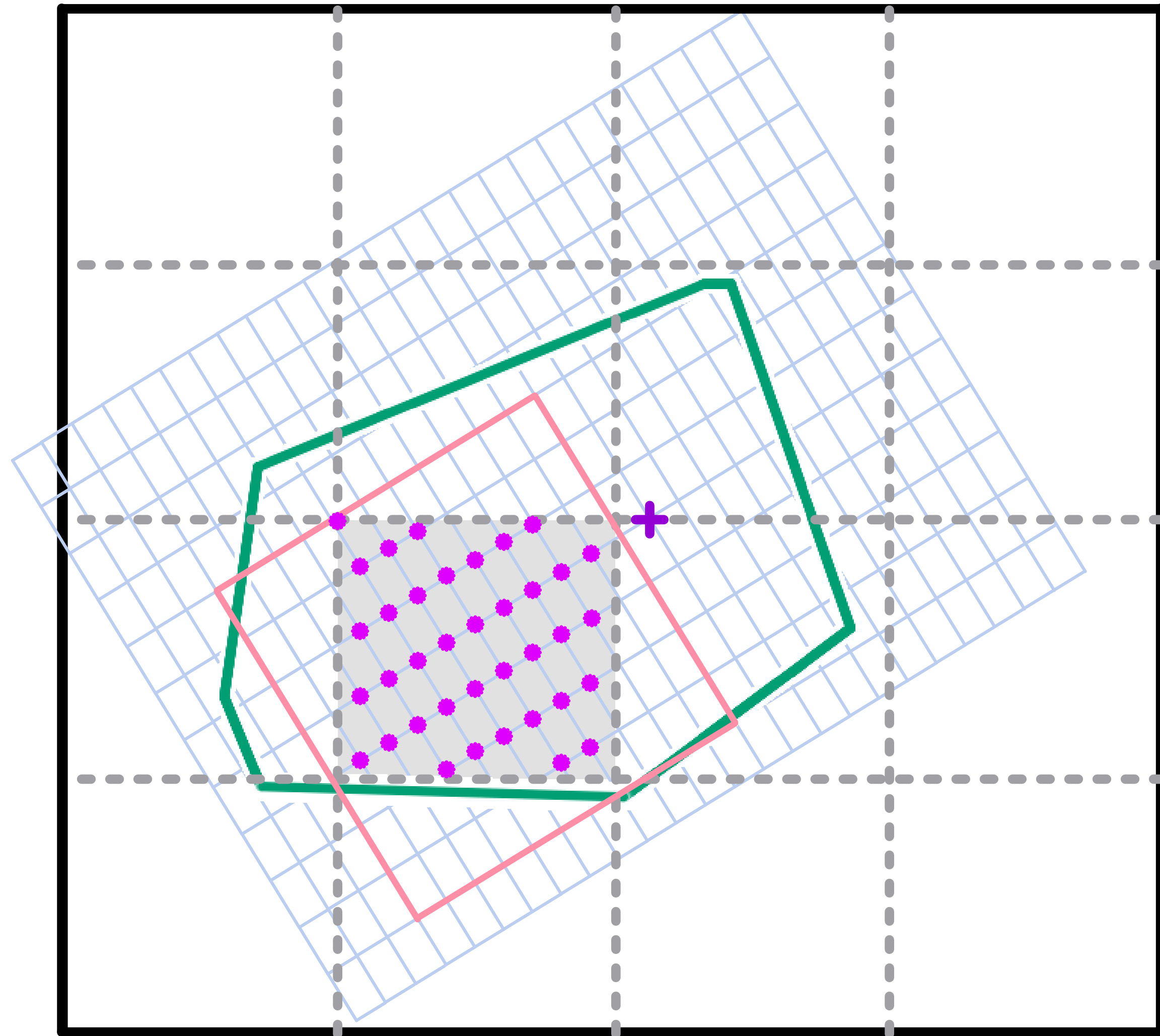
- A grain can span over multiple subdomains.
- To ensure the continuity, cores share the same lattice space.
- The grain seed point (+) is set as the zero point of the lattice space.

Grain Distribution and Atom Generation



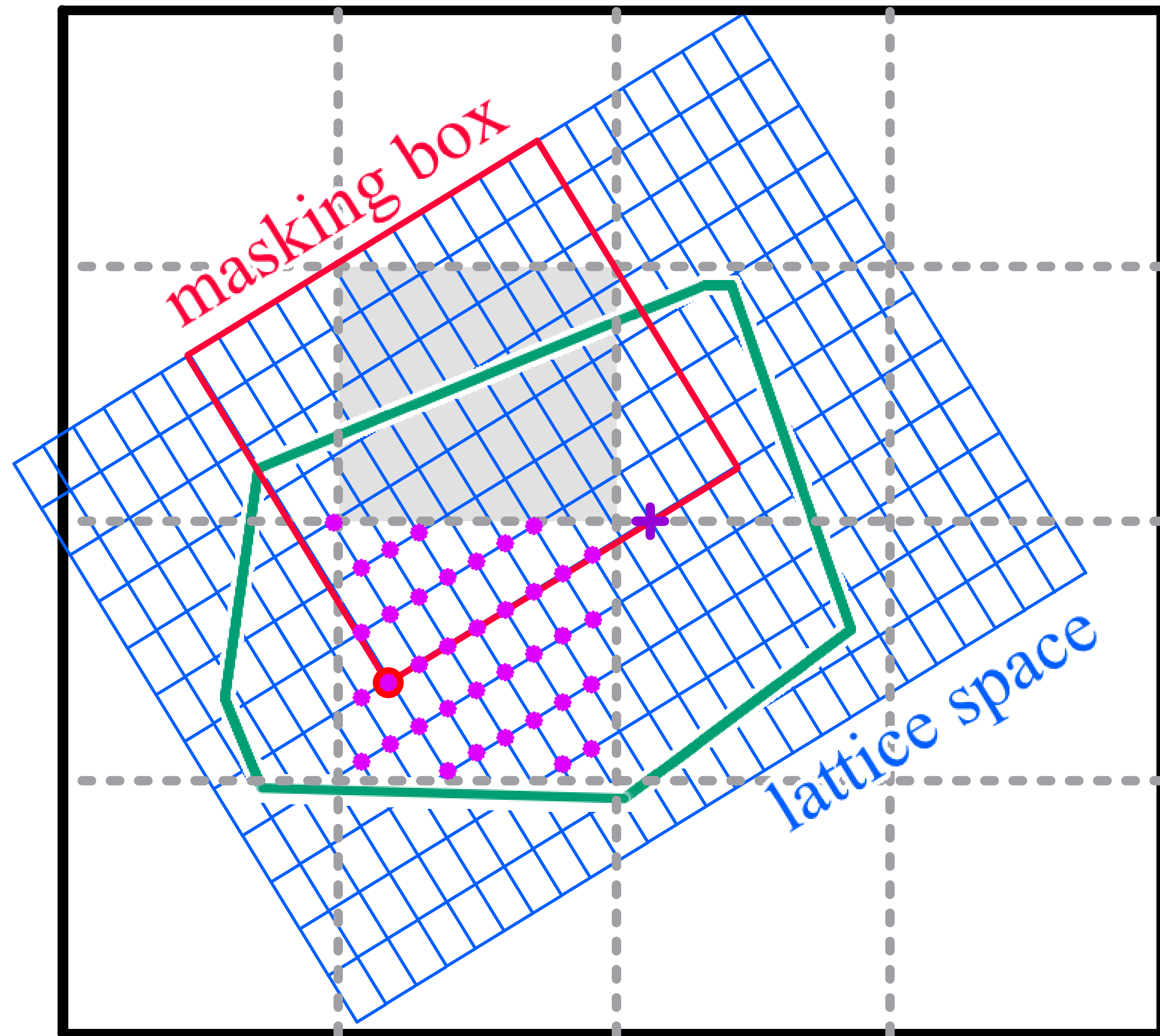
- A grain can span over multiple subdomains.
- To ensure the continuity, cores share the same lattice space.
- The grain seed point (+) is set as the zero point of the lattice space.
- A fictitious masking box, bigger enough to cover the subdomain, is created.

Grain Distribution and Atom Generation



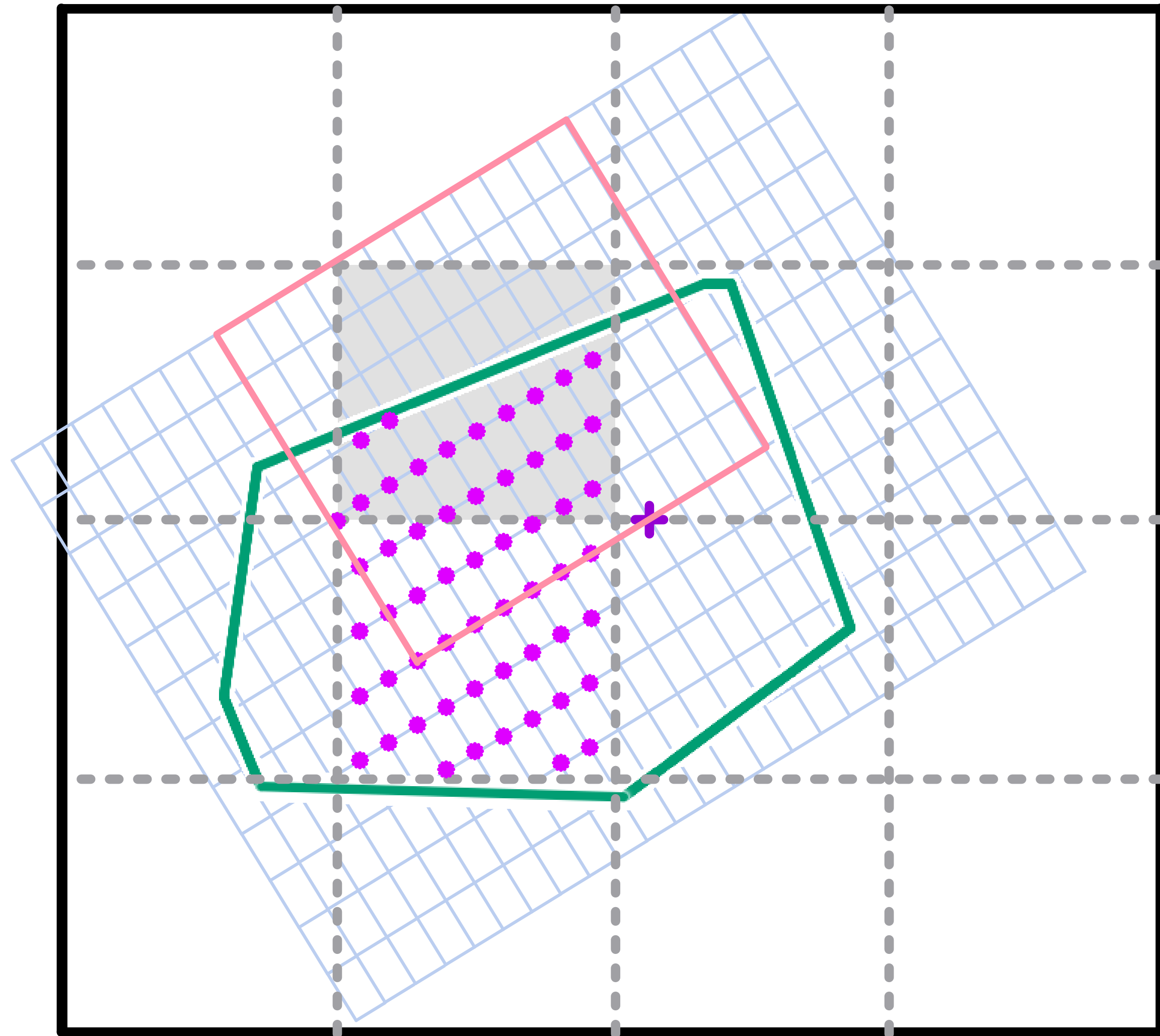
- A grain can span over multiple subdomains.
- To ensure the continuity, cores share the same lattice space.
- The grain seed point (+) is set as the zero point of the lattice space.
- A fictitious masking box, bigger enough to cover the subdomain, is created.
- Sweep lattice points inside the masking box and fill the space with atoms.

Grain Distribution and Atom Generation



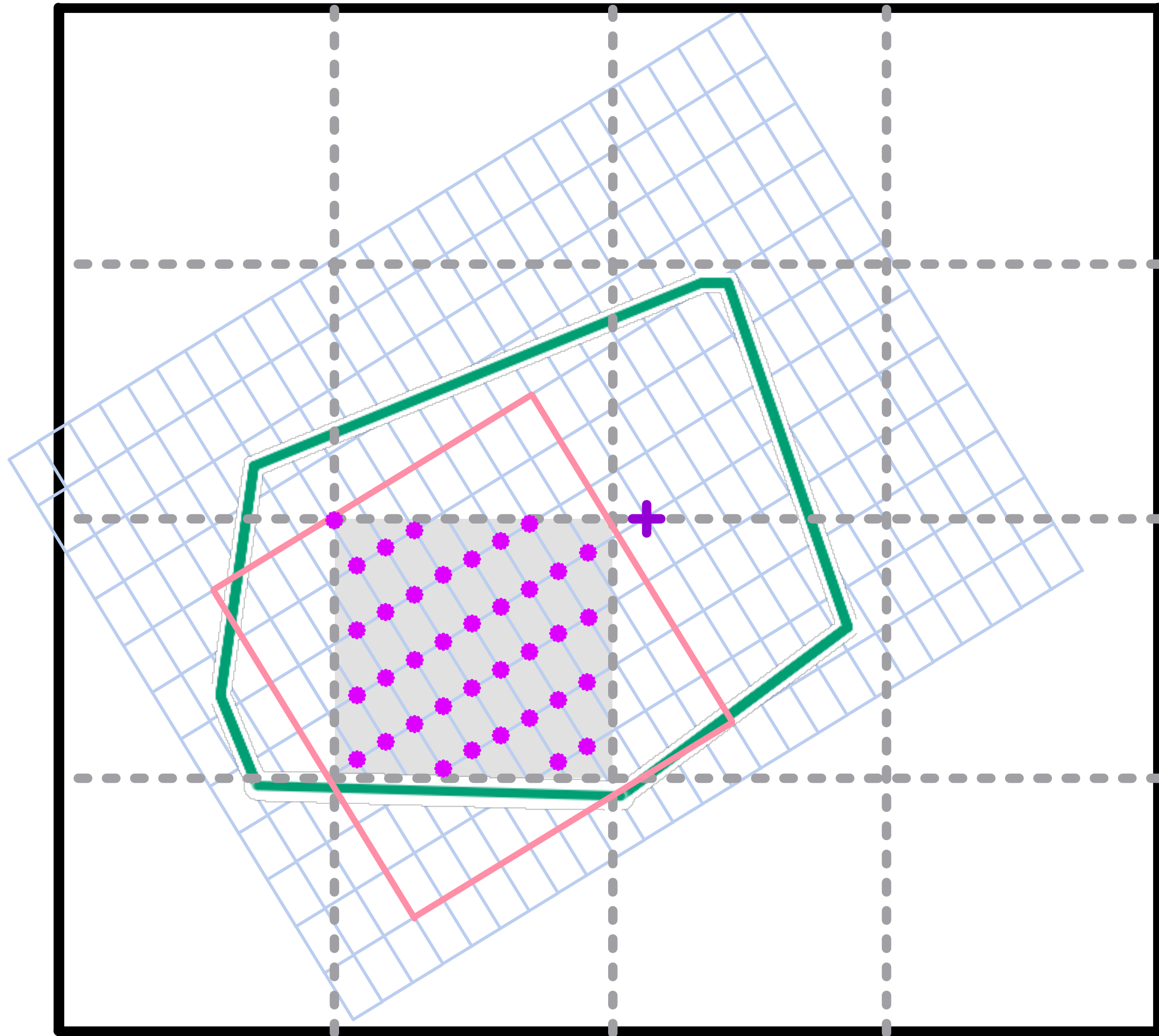
- A grain can span over multiple subdomains.
- To ensure the continuity, cores share the same lattice space.
- The grain seed point (+) is set as the zero point of the lattice space.
- A fictitious masking box, bigger enough to cover the subdomain, is created.
- Sweep lattice points inside the masking box and fill the space with atoms.

Grain Distribution and Atom Generation

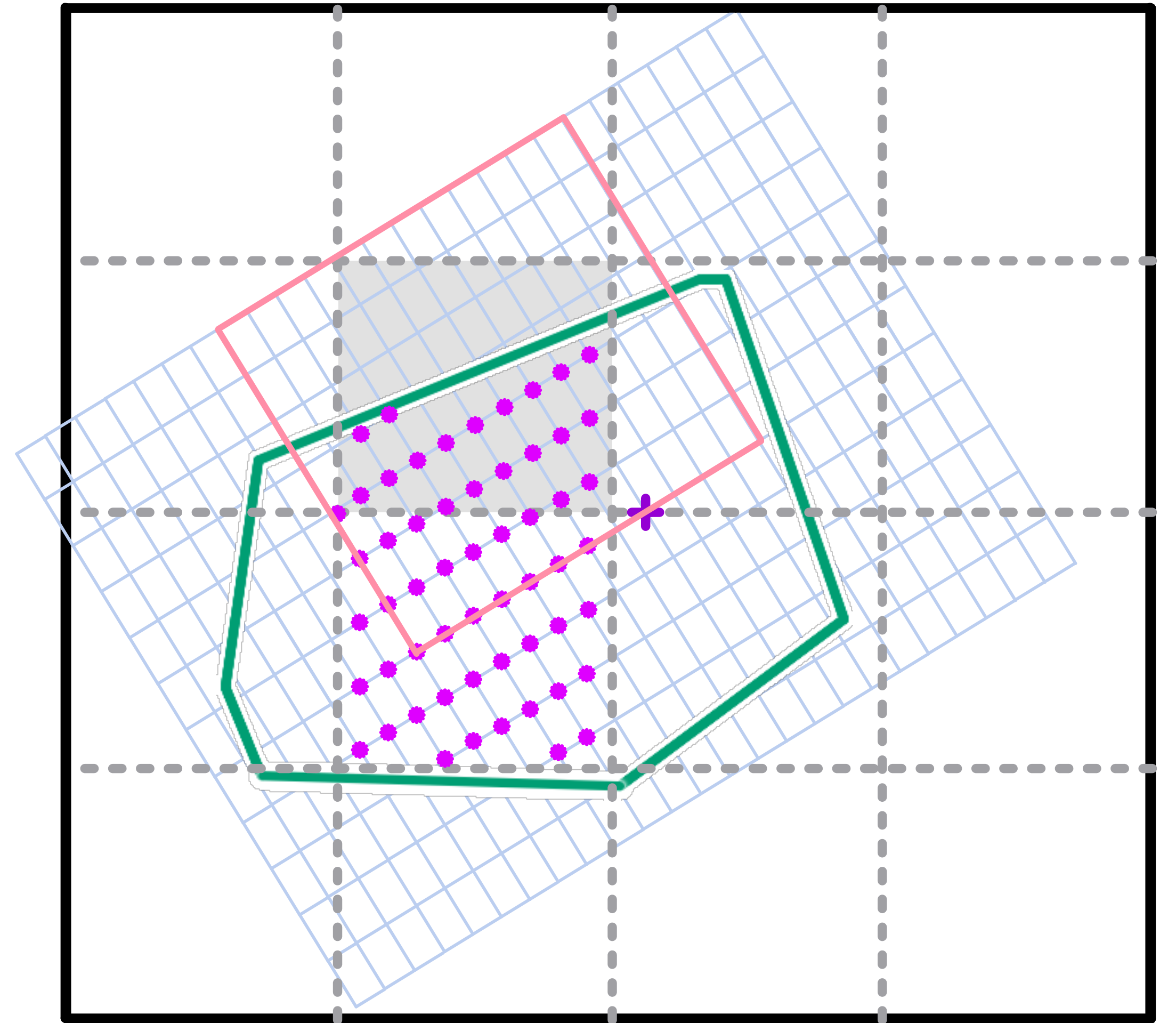


- A grain can span over multiple subdomains.
- To ensure the continuity, cores share the same lattice space.
- The grain seed point (+) is set as the zero point of the lattice space.
- A fictitious masking box, bigger enough to cover the subdomain, is created.
- Sweep lattice points inside the masking box and fill the space with atoms.

Grain Distribution and Atom Generation



1st subdomain filled

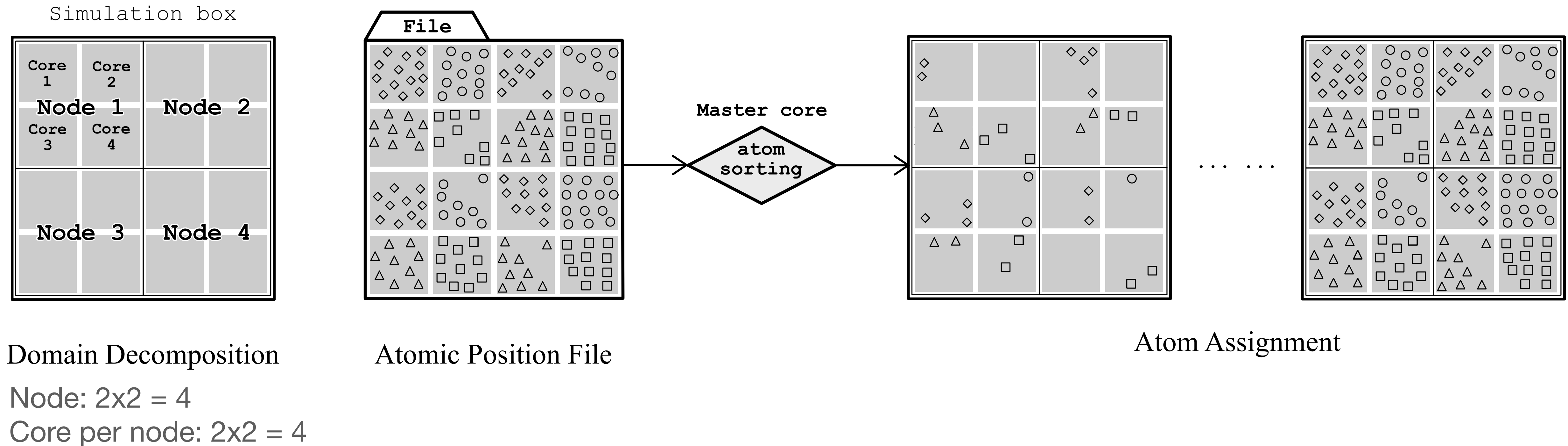


1st & 2nd subdomains filled

Parallel I/O

Parallel I/O

Position File: Conventional MD Code

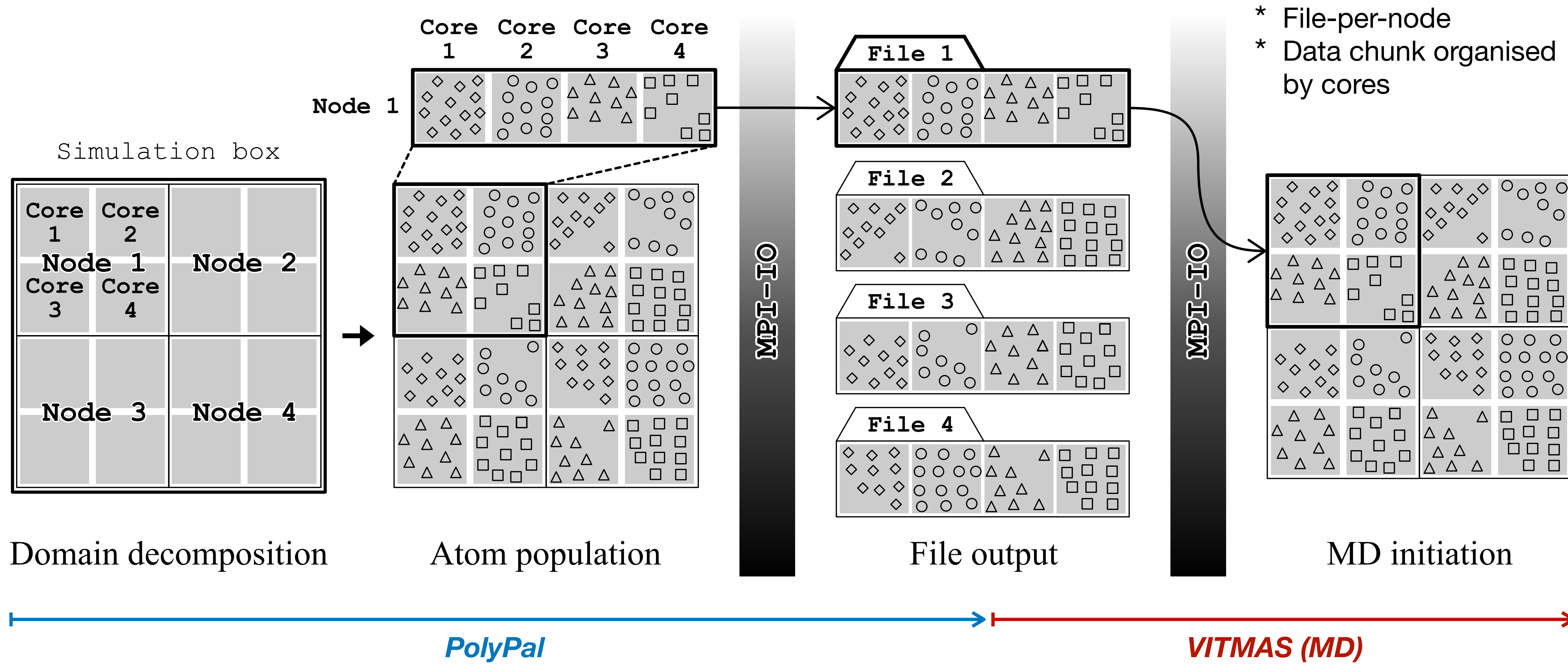


File reading and atom distribution by a master core

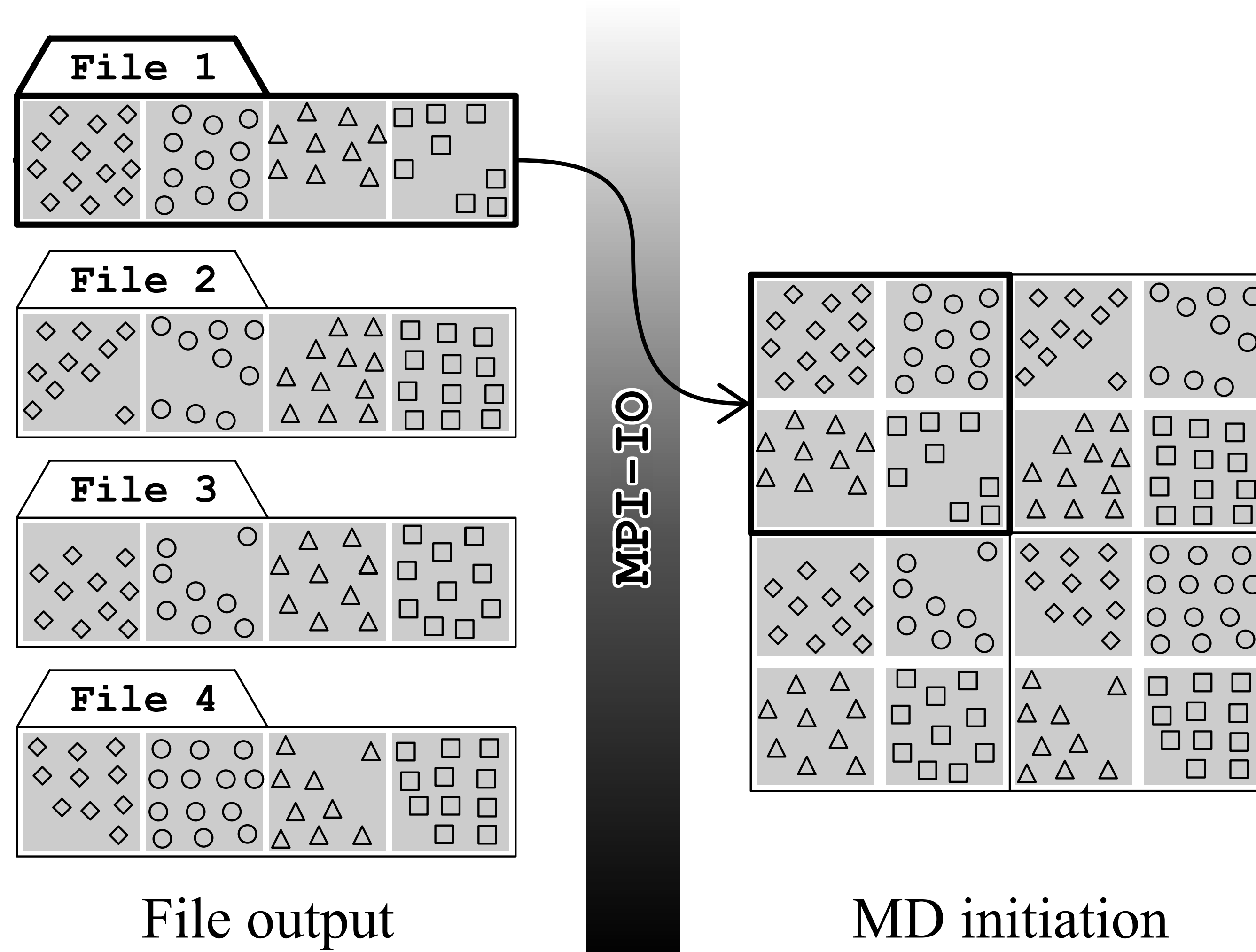
Single-core processing & extensive core-to-core communications

Position File: Domain-Preserving Parallel I/O

Shin et al., Comput. Phys. Commun. (2025)



Position File: Domain-Preserving Parallel I/O



MD init. test

1 billion atoms
100 x 64 cores

PolyPal + VITMAS

- File per node
- MPI-IO

1m 30s

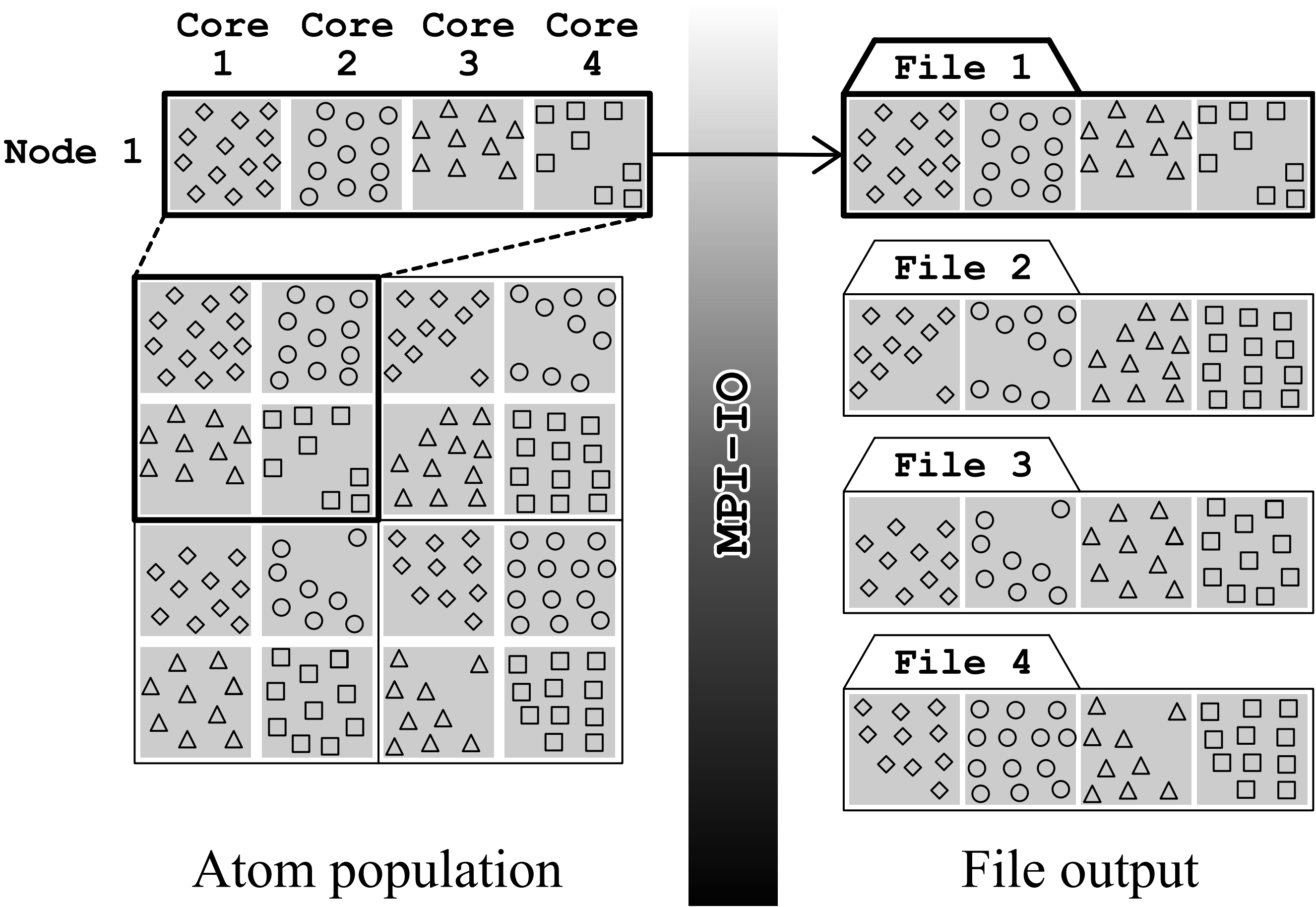
LAMMPS

- Single file
- no MPI-IO

7 hrs (x264)



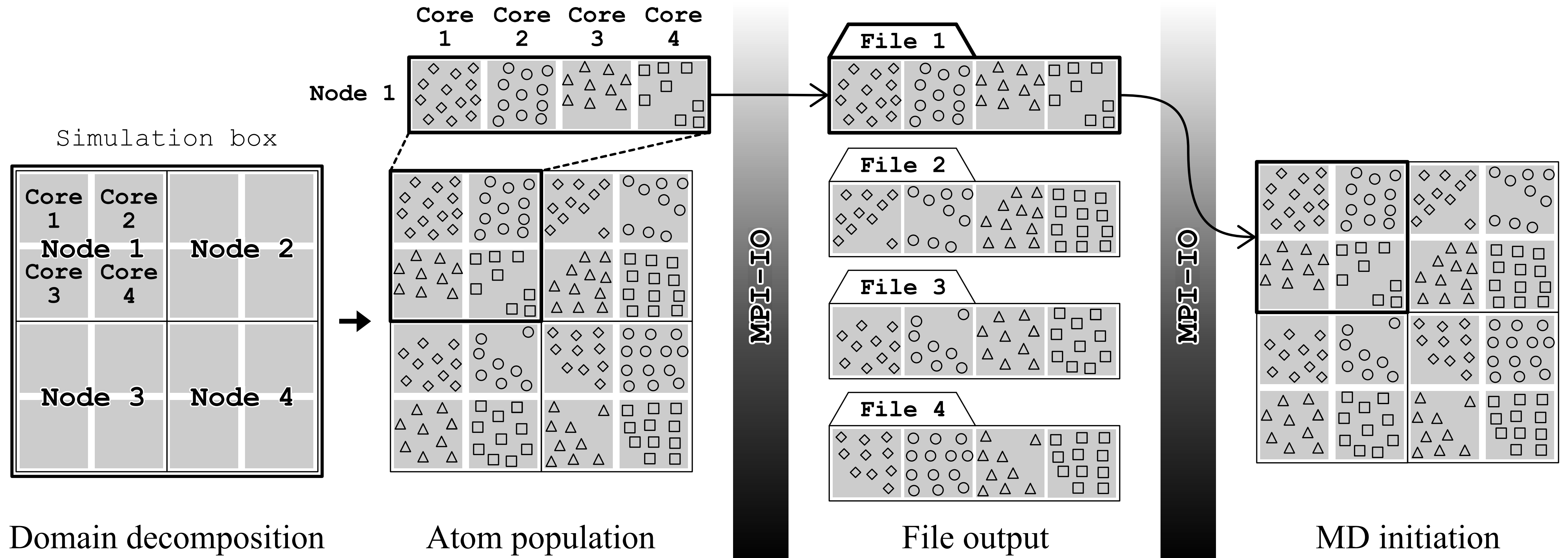
Position File: Domain-Preserving Parallel I/O



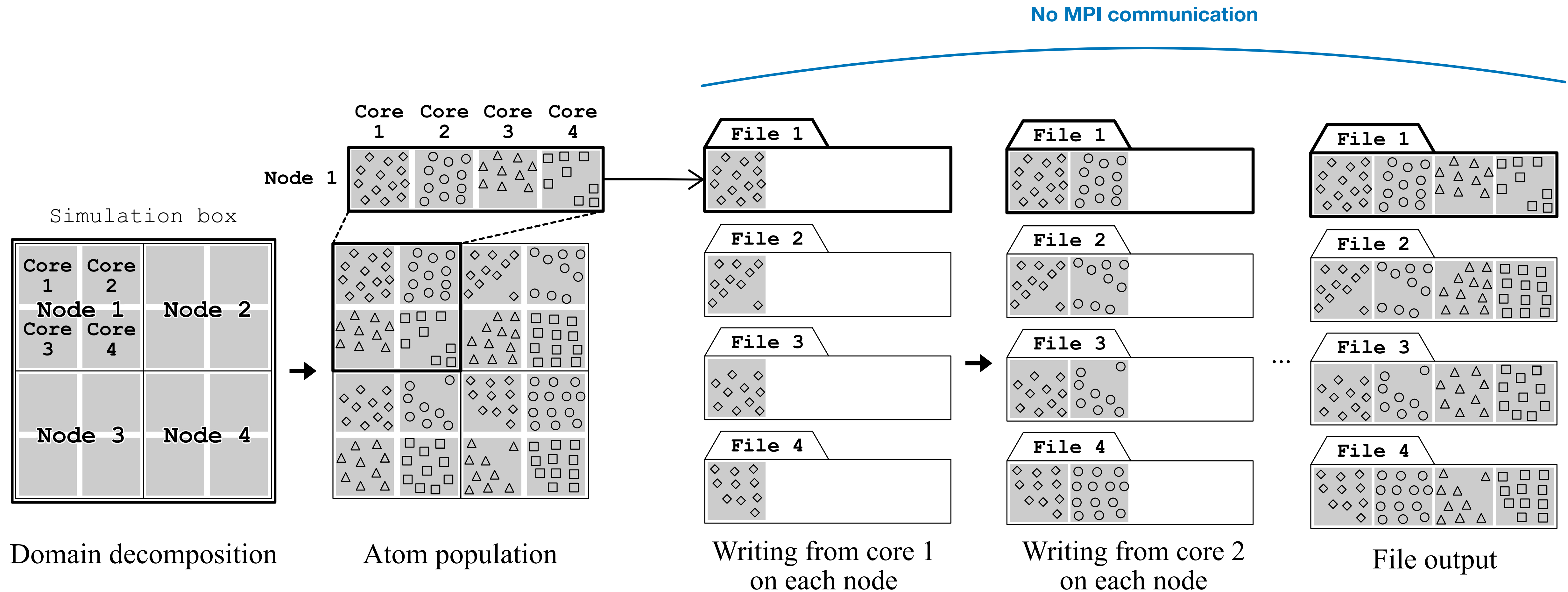
header = map

2	Number of species		
74	75	Atomic numbers (W, Re)	
15558	304	Numbers of W/Re atoms in core 1	
15479	254	Numbers of W/Re atoms in core 2	
15367	270	Numbers of W/Re atoms in core 3	
15601	311	Numbers of W/Re atoms in core 4	
316.51900	0.00028	0.11128	
316.51900	0.00028	3.27648	
...	...		

Position File: File-per-Node MPI-IO

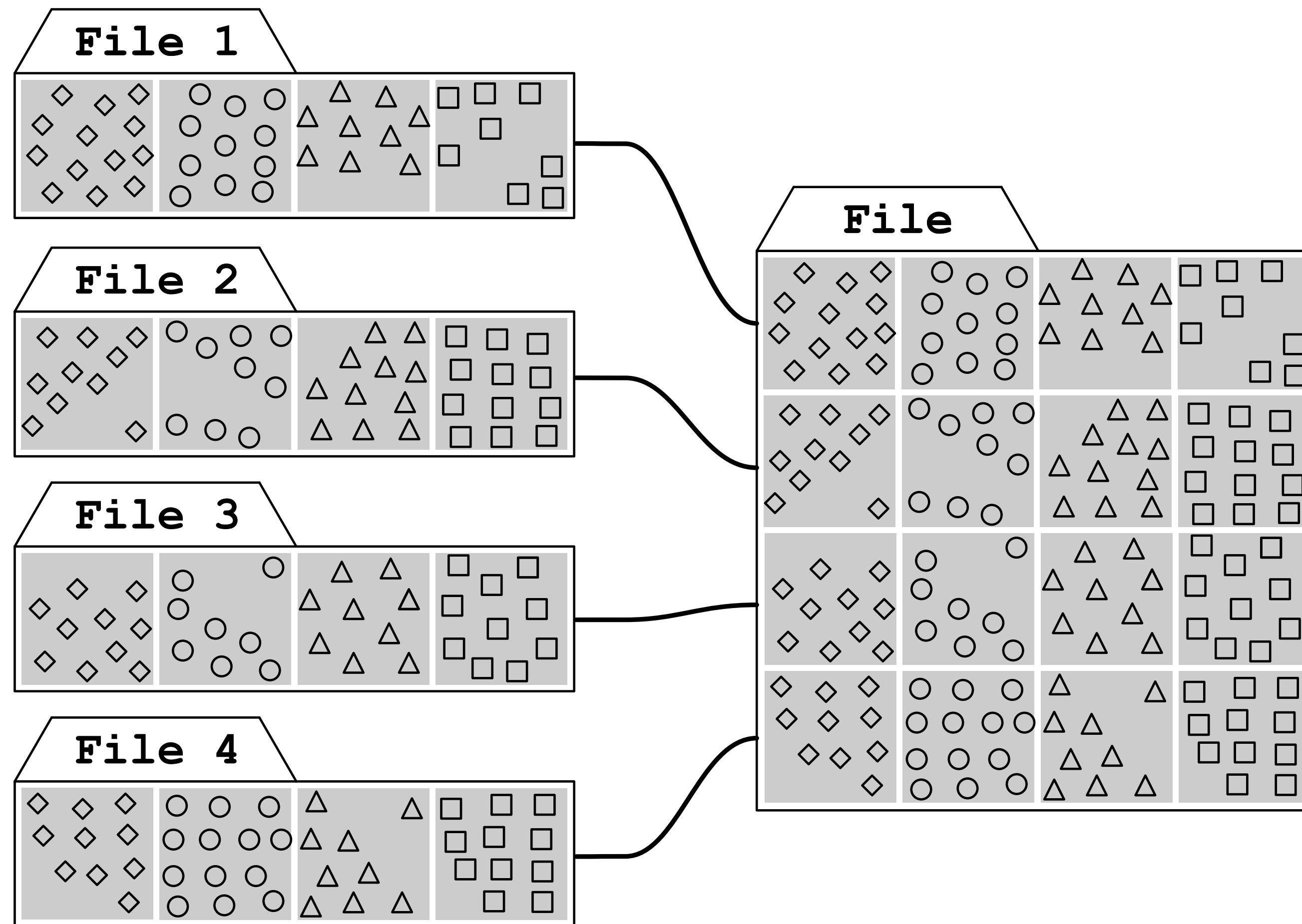


Position File: File-per-Node Non-MPI-IO



Asynchronous data dump from all cores.

Position File: Single-File Non-MPI-IO



Concatenation

No additional memory required
No additional processing required

Performance Metrics

Output				Wall time (s)	
Format	File policy	MPI-IO	Writer	File output	Others
Binary	Single	×	Global master	1099	151
Binary	Single	×	All cores*	10	151
Binary	File/node	×	All cores*	8	151
Binary	File/node	○	All cores	3	151
ASCII text	Single	×	Global master	4423	151
ASCII text	Single	×	All cores*	100	150
ASCII text	File/node	×	All cores*	54	151
ASCII text	File/node	○	All cores	45	151

HPC system: 6400 cores (64 cores/node) from Nurion-5 at KISTI

- Intel Xeon Phi 7250 1.4 GHz CPU
- 96 GB/node memory with Lustre file system

Atomic structure: total 10,030,152,440 atoms in 300 grains

Shin et al., Comput. Phys. Commun. (2025)

Performance Metrics

Output				Wall time (s)	
Format	File policy	MPI-IO	Writer	File output	Others
Binary	Single	×	Global master	1099	151
Binary	Single	×	All cores*	10	151
Binary	File/node	×	All cores*	8	151
Binary	File/node	○	All cores	3	151
ASCII text	Single	×	Global master	4423	151
ASCII text	Single	×	All cores*	100	150
ASCII text	File/node	×	All cores*	54	151
ASCII text	File/node	○	All cores	45	151

file output: 2%
others: 98%

HPC system: 6400 cores (64 cores/node) from Nurion-5 at KISTI
- Intel Xeon Phi 7250 1.4 GHz CPU
- 96 GB/node memory with Lustre file system
Atomic structure: total 10,030,152,440 atoms in 300 grains

Shin et al., Comput. Phys. Commun. (2025)

Only takes **2m 30s** to generate a **10-billion**-atom system.

Performance Metrics

Output				Wall time (s)	
Format	File policy	MPI-IO	Writer	File output	Others
Binary	Single	×	Global master	1099	151
Binary	Single	×	All cores*	10	151
Binary	File/node	×	All cores*	8	151
Binary	File/node	○	All cores	3	151
ASCII text	Single	×	Global master	4423	151
ASCII text	Single	×	All cores*	100	150
ASCII text	File/node	×	All cores*	54	151
ASCII text	File/node	○	All cores	45	151

HPC system: 6400 cores (64 cores/node) from Nurion-5 at KISTI

- Intel Xeon Phi 7250 1.4 GHz CPU
- 96 GB/node memory with Lustre file system

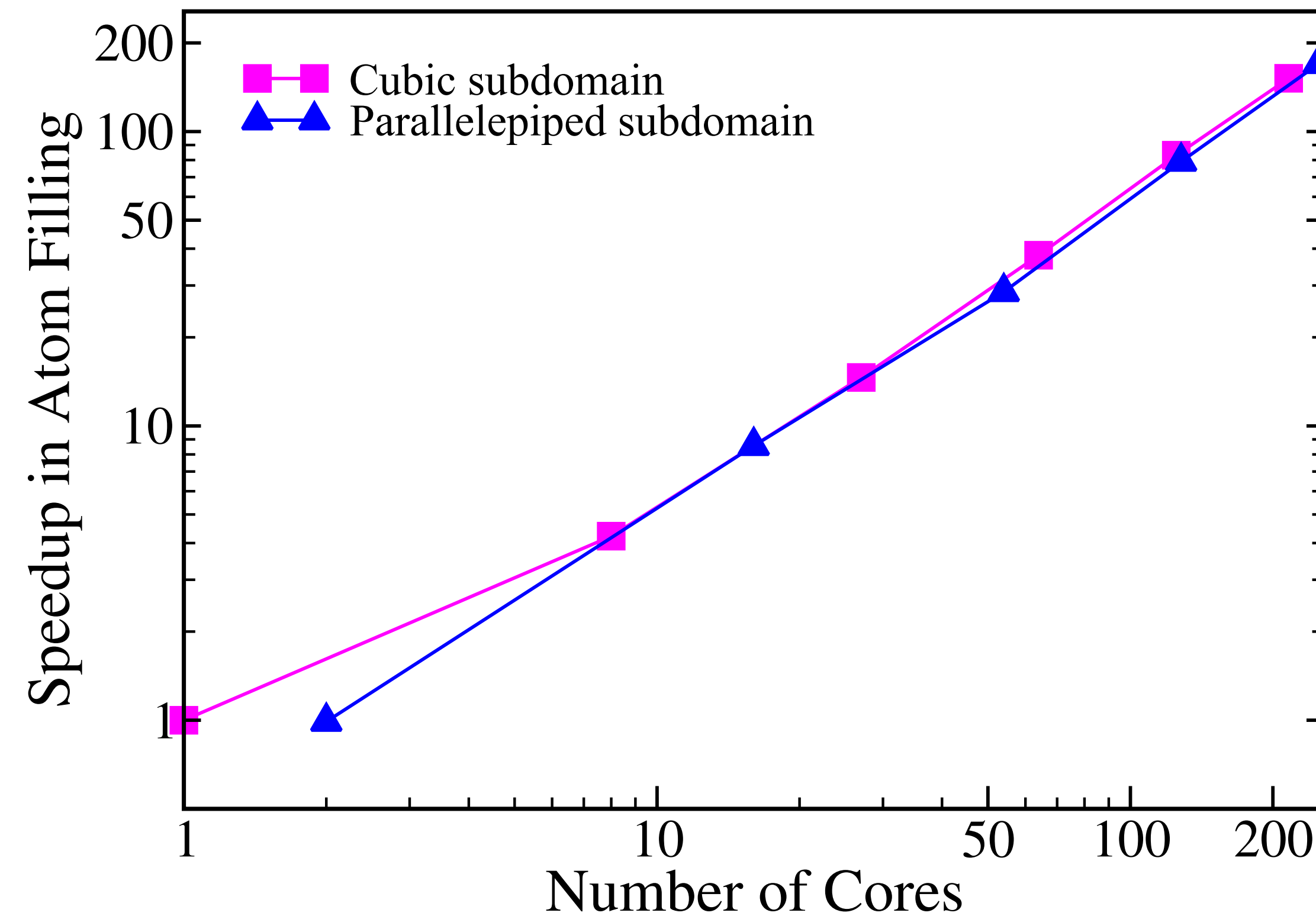
Atomic structure: total 10,030,152,440 atoms in 300 grains

Shin et al., Comput. Phys. Commun. (2025)

*binary ~225gb / ASCII ~534gb

Binary format output: faster and light-weighted*

Performance Metrics: Strong Scaling Test



Shin et al., Comput. Phys. Commun. (2025)

Near-perfect load balancing + no inter-core comm. = **excellent scalability**

Takeaway

- Massive-scale polycrystalline structures → realistic simulations
- File I/O: a huge bottleneck for massive-scale simulation
- PolyPal & VITMAS domain-preserving file I/O strategy
- Excellent scalability due to near-perfect load balancing and no inter-core communication
- Versatile options for tailoring desired polycrystalline structure

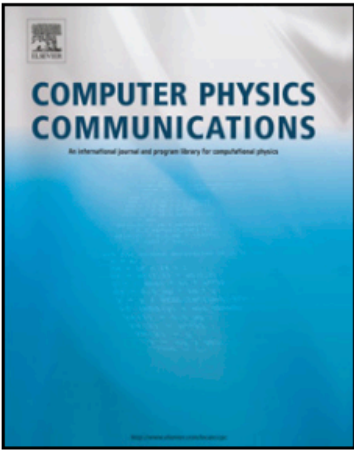
“We need appropriate tools for massive-scale MD simulations.”



Contents lists available at ScienceDirect

Computer Physics Communications

journal homepage: www.elsevier.com/locate/cpc



Computer Programs in Physics

PolyPal: A parallel microscale virtual specimen generator

Youngguk Shin ^{a,b}, Vichhika Moul ^a, Keonwook Kang ^b, Byeongchan Lee ^{a,*}

^a Kyung Hee University, 1732 Deogyeong-daero, Yongin, Gyeonggi 17104, Republic of Korea

^b Yonsei University, 50 Yonsei-ro, Seodaemun-gu, Seoul 03722, Republic of Korea



A R T I C L E I N F O

Keywords:

Atomistic simulations

Polycrystal

Microscale

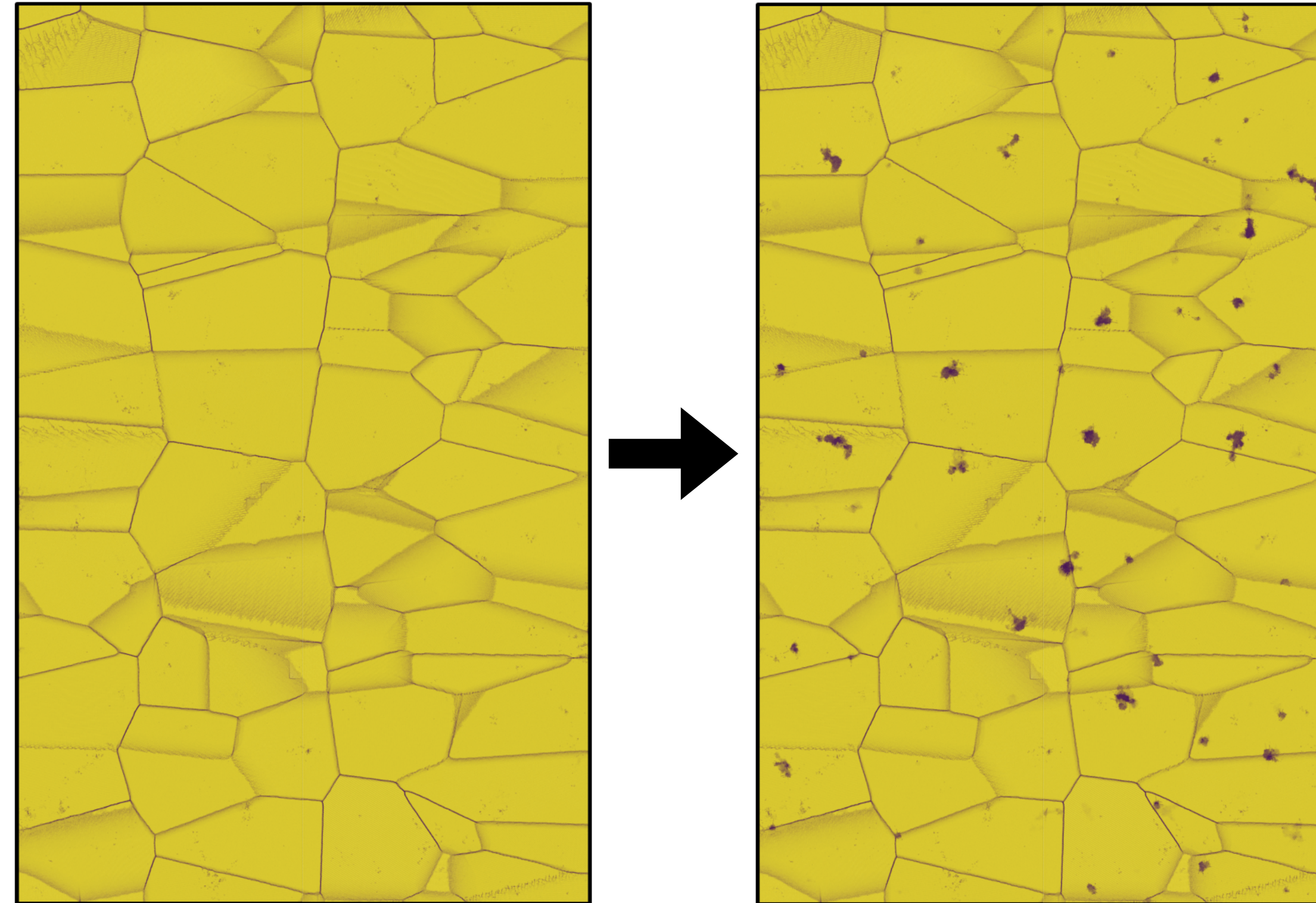
Parallel computing

A B S T R A C T

We present an open source program, PolyPal, that can generate a polycrystalline virtual specimen in the micrometer scale for atomistic calculations and visualization. Unlike regular meshes or perfect lattices, atomic positions in polycrystalline materials need to be defined before calculations, and the capability of an atom-generation code is evaluated by the maximum size of the virtual specimen it can generate as well as by the efficiency of the necessary input-output process. Present atom-generation codes are implemented in a serial fashion, and the maximum size of the virtual specimen is limited by the on-board memory. Furthermore, it is difficult to handle a single position file with billions of atoms not only because it takes a long time to read in a row but also full domain decomposition takes hours. PolyPal addresses these challenges with a fully parallelized MPI input-output scheme that supports multiple export options on a Linux cluster. It has no limit in the system size with virtually perfect scalability. Additionally by controlling the size distribution and homogeneity of grains, the program can simulate different microstructures, as typically found in the bulk system or in thin-film samples, prepared with different fabrication processes. PolyPal will harness molecular dynamics codes in the coming age of the exascale computing.

Future Plans

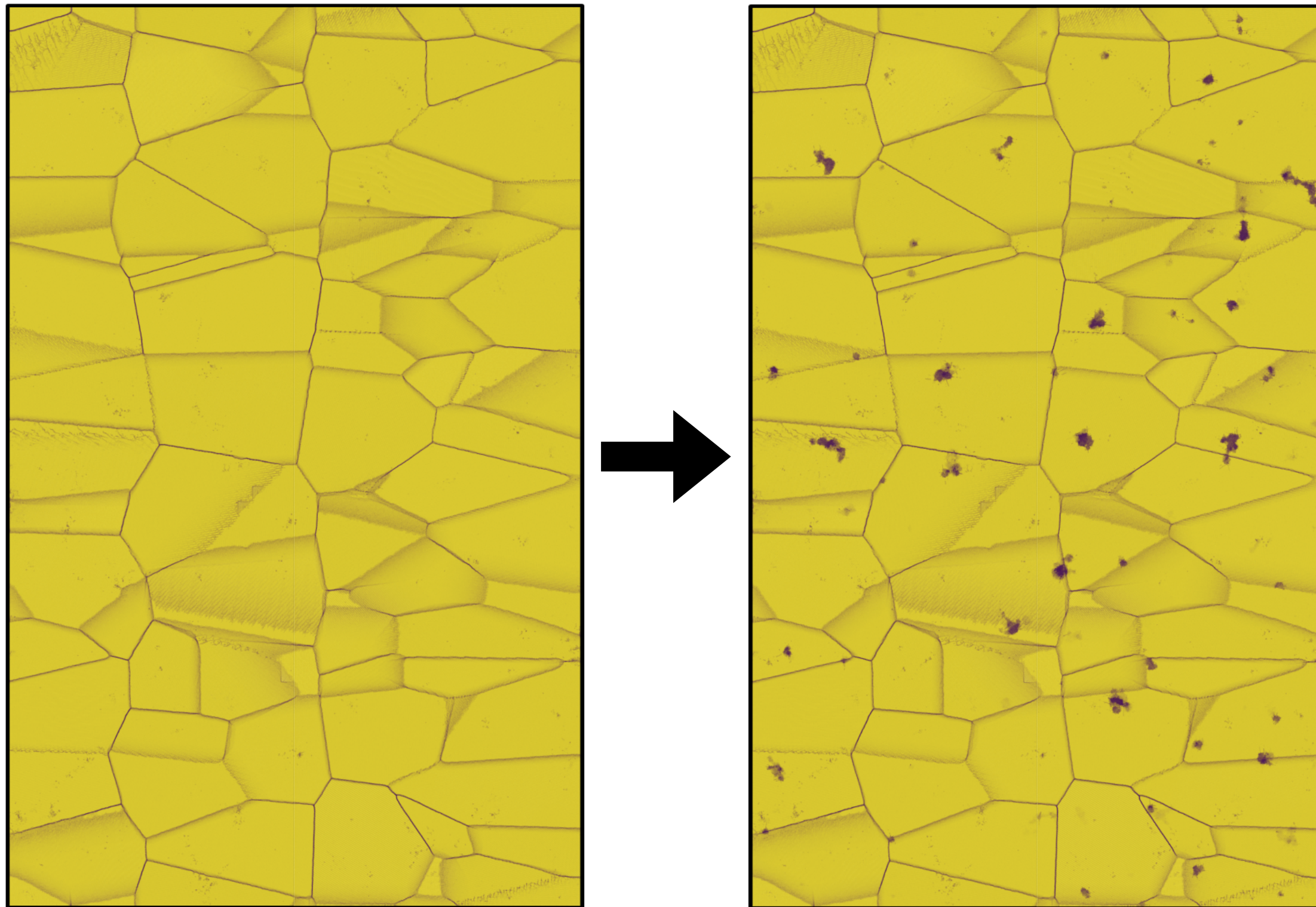
Analysis on Massive-Scale MD Simulation



- Virtual test sample containing 8-billion atoms that mimics W thin film specimen
- Multiple-PKA simulations using *VITMAS*

Future Plans

Analysis on Massive-Scale MD Simulation



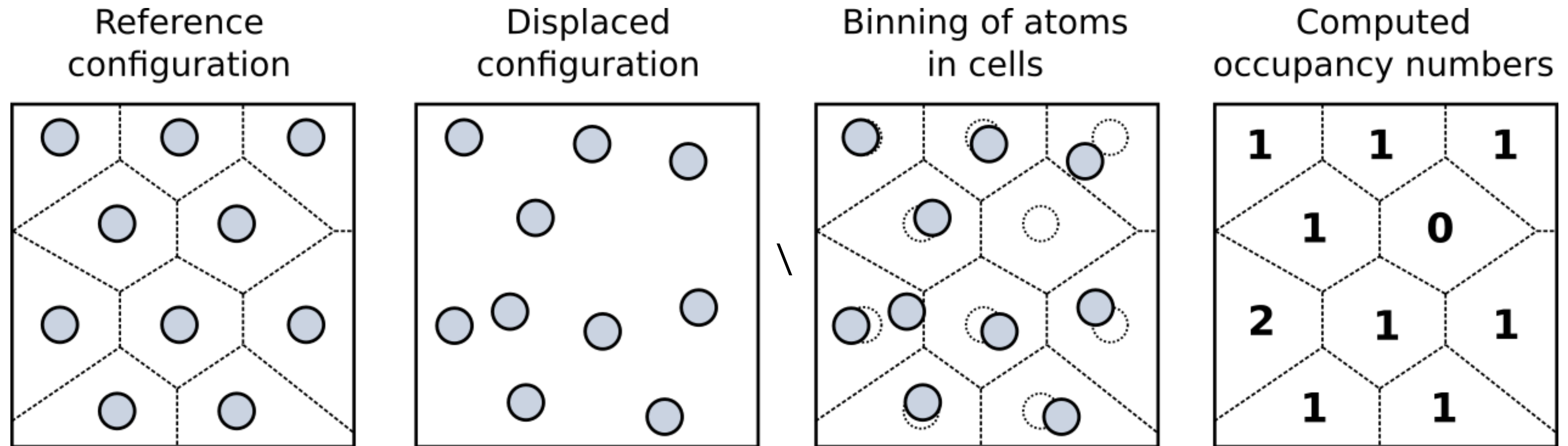
“We’ve created and simulated...!”

“Now... analysis...?”

- Virtual test sample containing 8-billion atoms that mimics W thin film specimen
- Multiple-PKA simulations using *VITMAS*

Future Plans

Wigner-Seitz Method Fails on Polycrystal

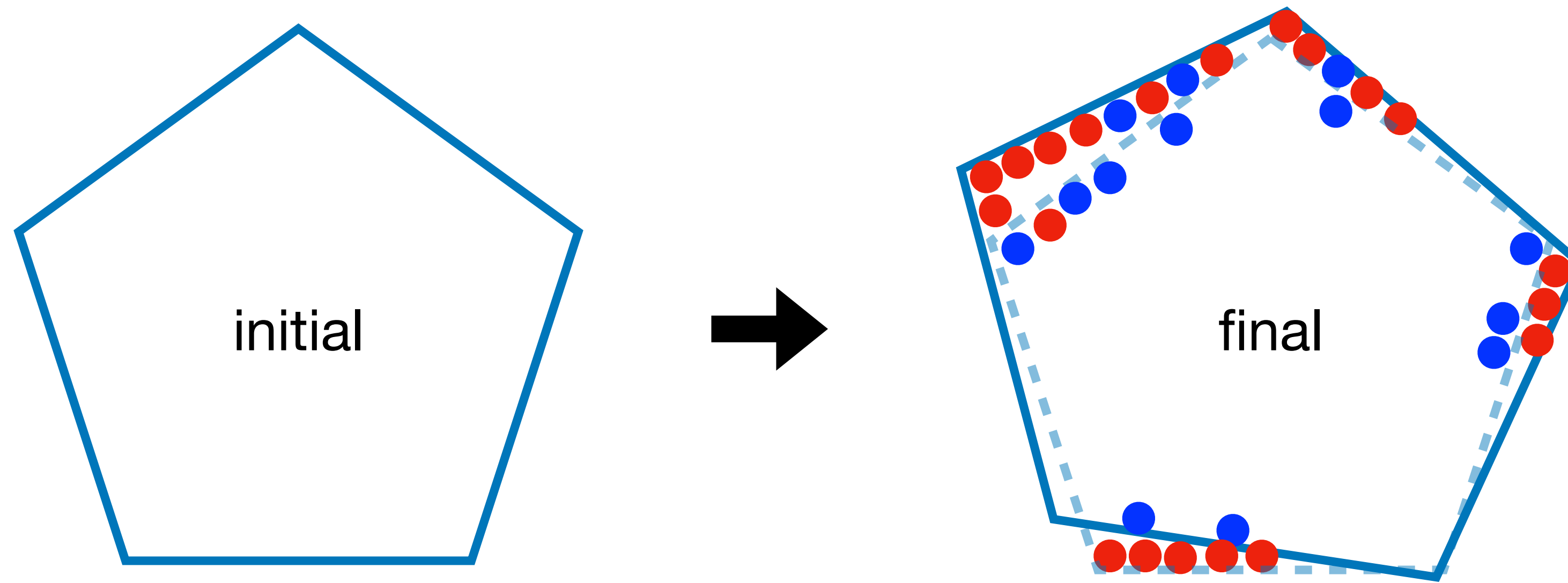


Stukowski, Modelling Simul. Mater. Sci. Eng (2010°

- WS is frequently used to locate and quantify defects in PKA simulations.
- It creates WS cells at the initial and check occupancy; 0: vacancy, >1 : interstitial.

Future Plans

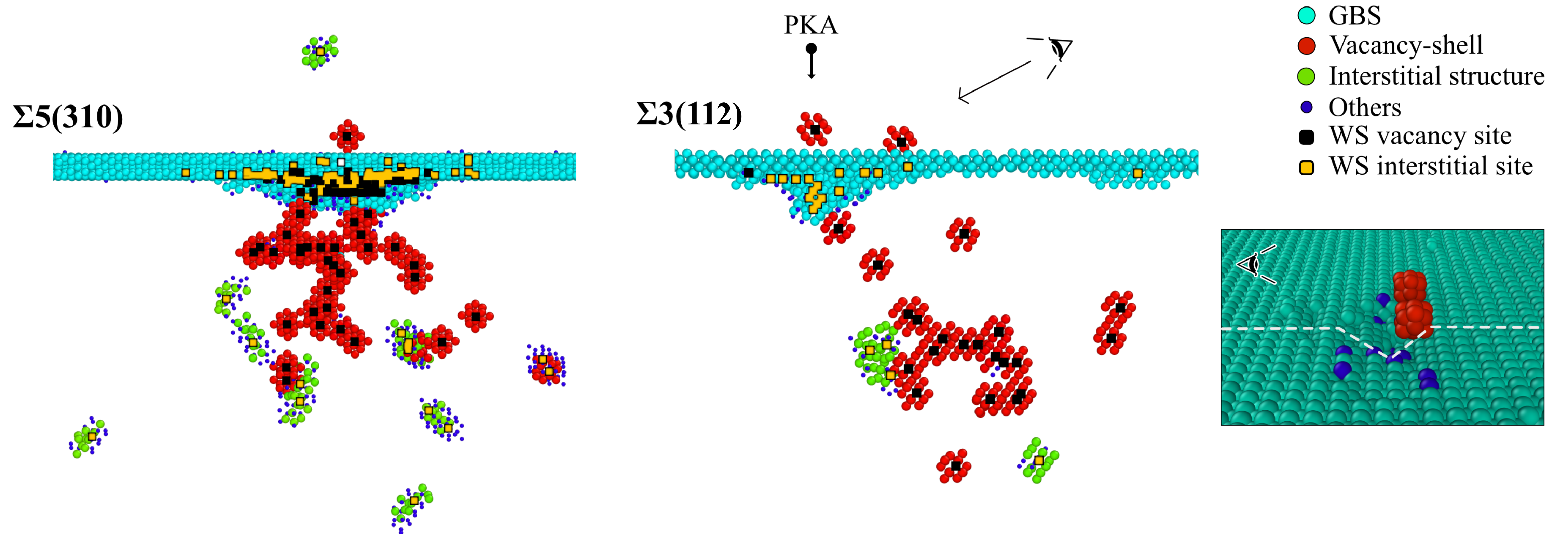
Wigner-Seitz Method Fails on Polycrystal



- Grains rotate, translate and grow.
- W-S cells at the interfaces are larger than those in bulk.
- Miscount of point defects occurs.

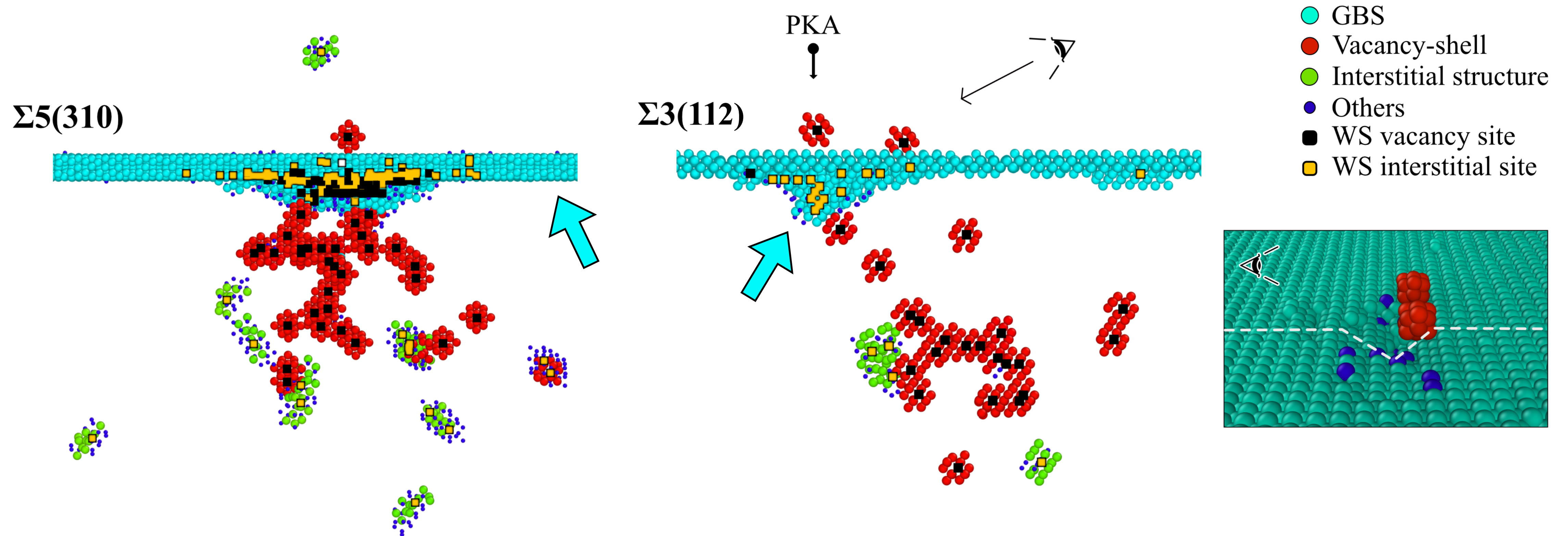
Future Plans

Alternative: CNA + Graph-Theoretical Pattern Recognition



Future Plans

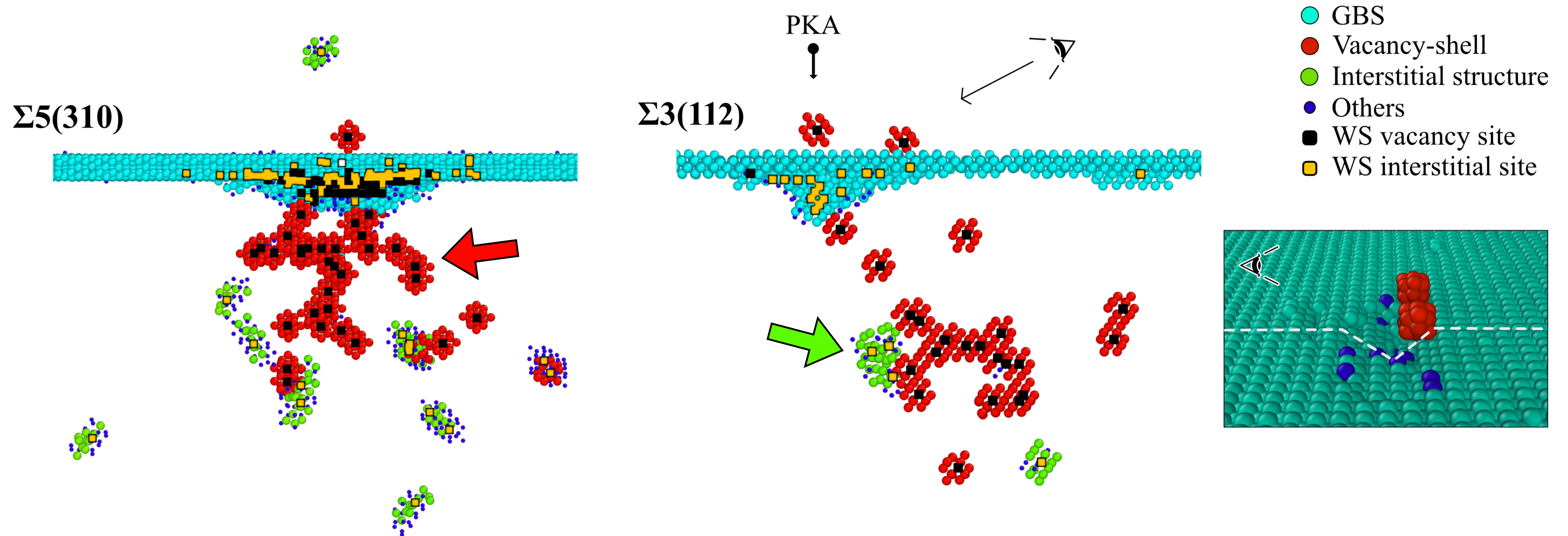
Alternative: CNA + Graph-Theoretical Pattern Recognition



Separation of grain boundary structure (GBS)

Future Plans

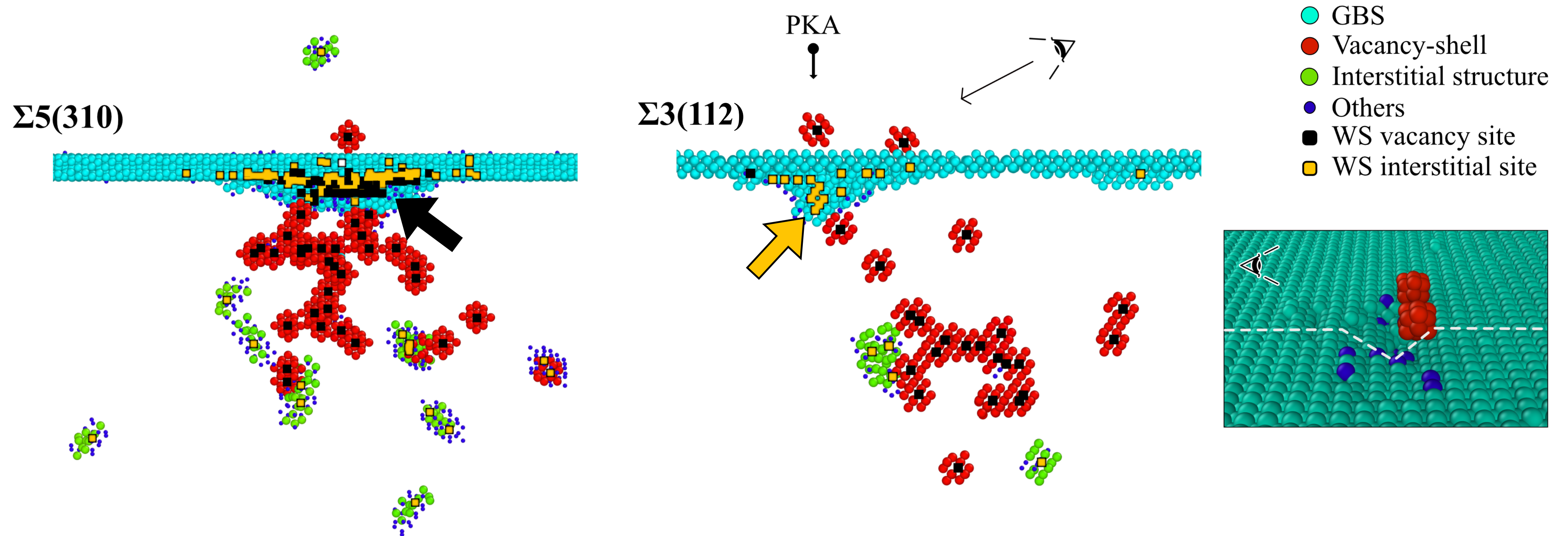
Alternative: CNA + Graph-Theoretical Pattern Recognition



Type-resolved defect identification: $\langle 110 \rangle$ / $\langle 111 \rangle$ interstitial or vacancy

Future Plans

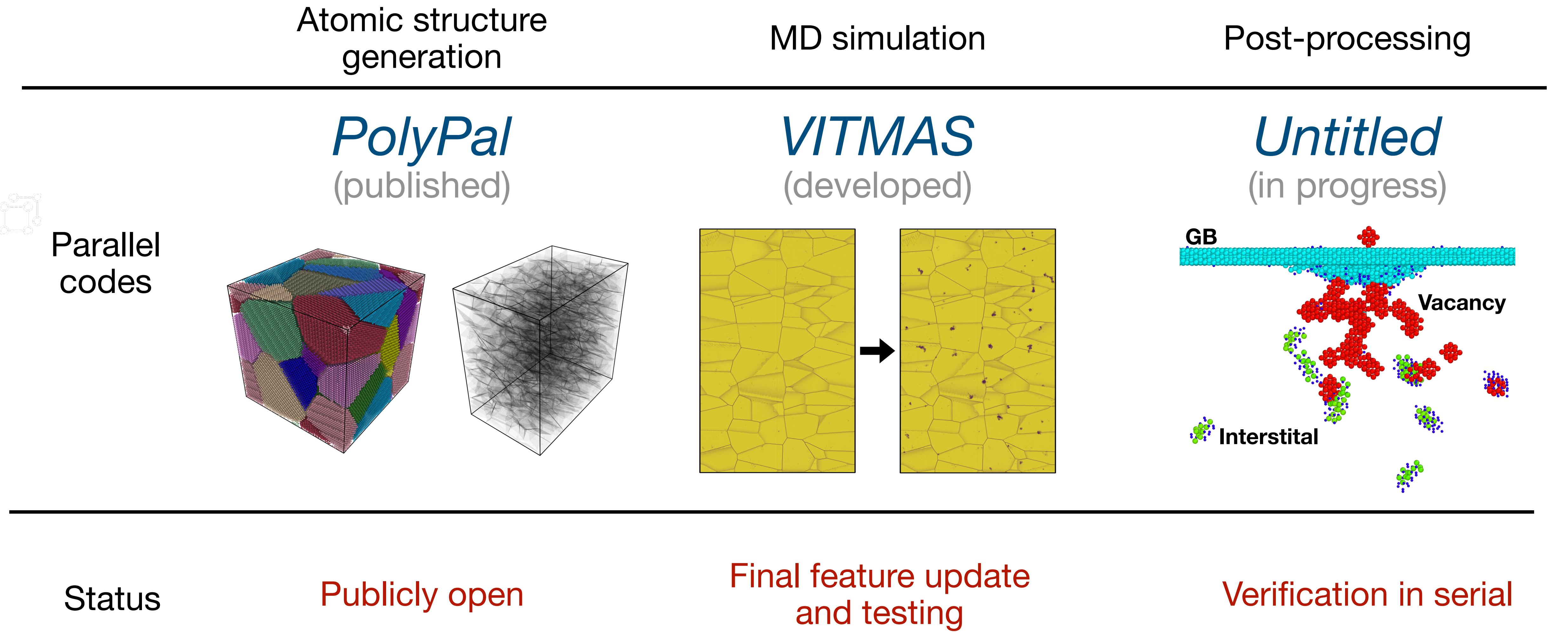
Alternative: CNA + Graph-Theoretical Pattern Recognition

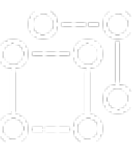


No false detection of point defect at grain boundary

Future Plans

Look into Our Masterplan Again



 Parallel codes

“What I cannot create, I do not understand.”

Richard Feynman

Thank You

This research was supported by the Nano & Material Technology Development Program through the National Research Foundation of Korea(NRF) funded by Ministry of Science and ICT(RS-2024-00445448)