

AMPMI 2024 @ Metsätalo, University of Helsinki, Finland

Bulk, surfaces, and grain boundaries

in the lifetime of cascades in W

Byeongchan Lee Kyung Hee University, Korea, July 18, 2024



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KYUNG HEE UNIVERSITY

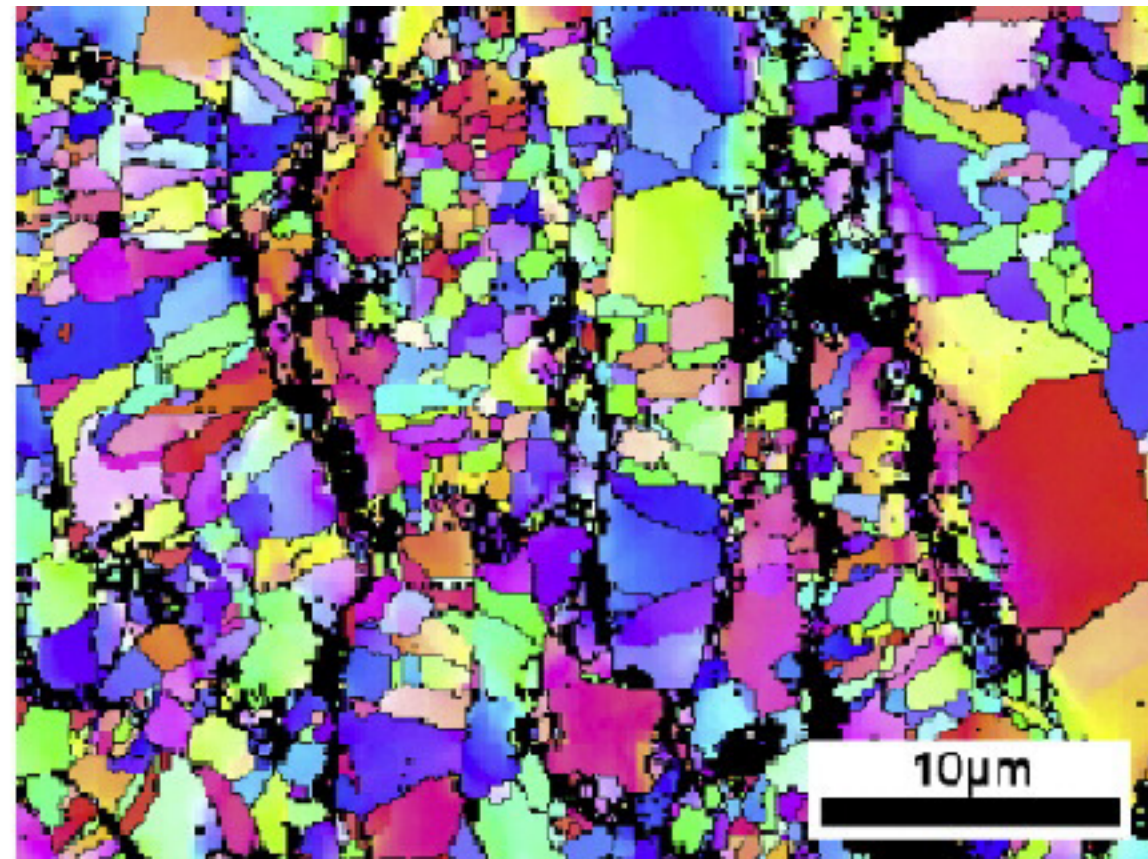
Acknowledgments

- PhD students: **Youngguk Shin, Vichhika Moul** (Kyung Hee University)
- Collaborator: **Keonwook Kang** (Yonsei University)

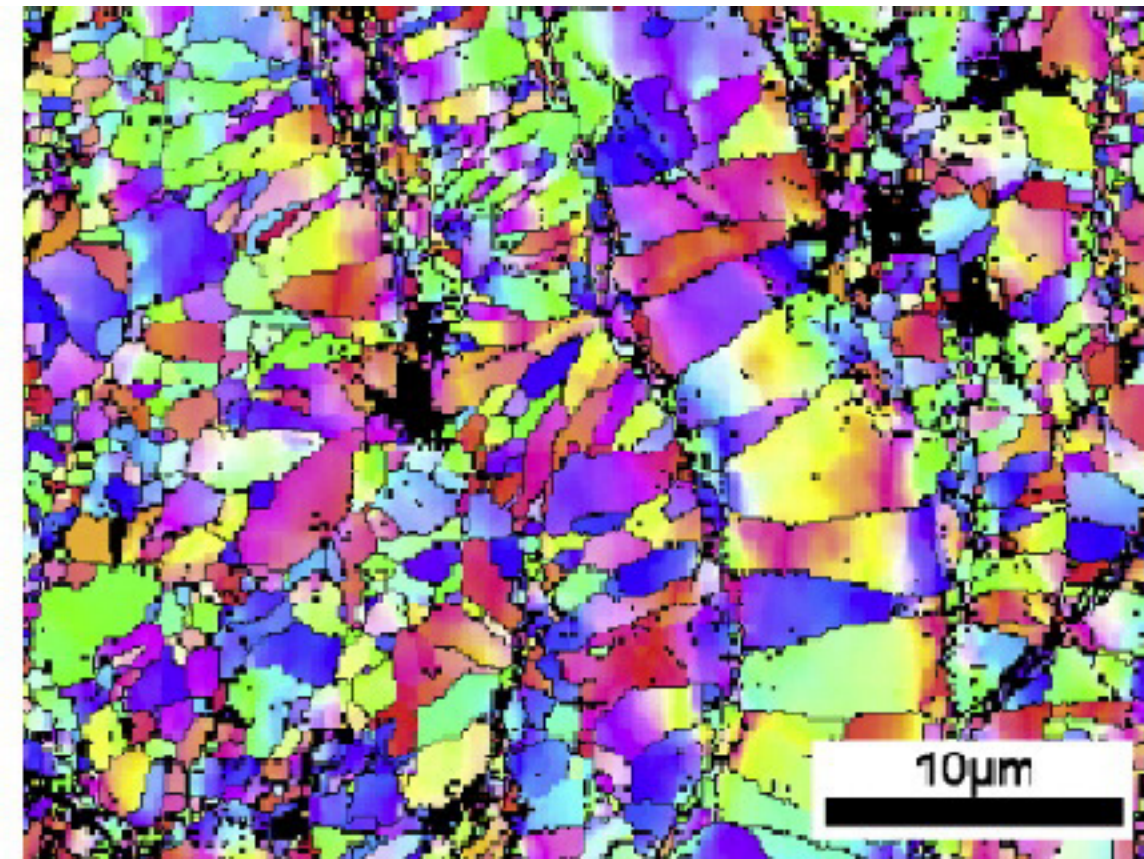
- Funding
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 - Korea Hydro & Nuclear Power Co., LTD (No. 2022-Tech-11).

Mechanical properties

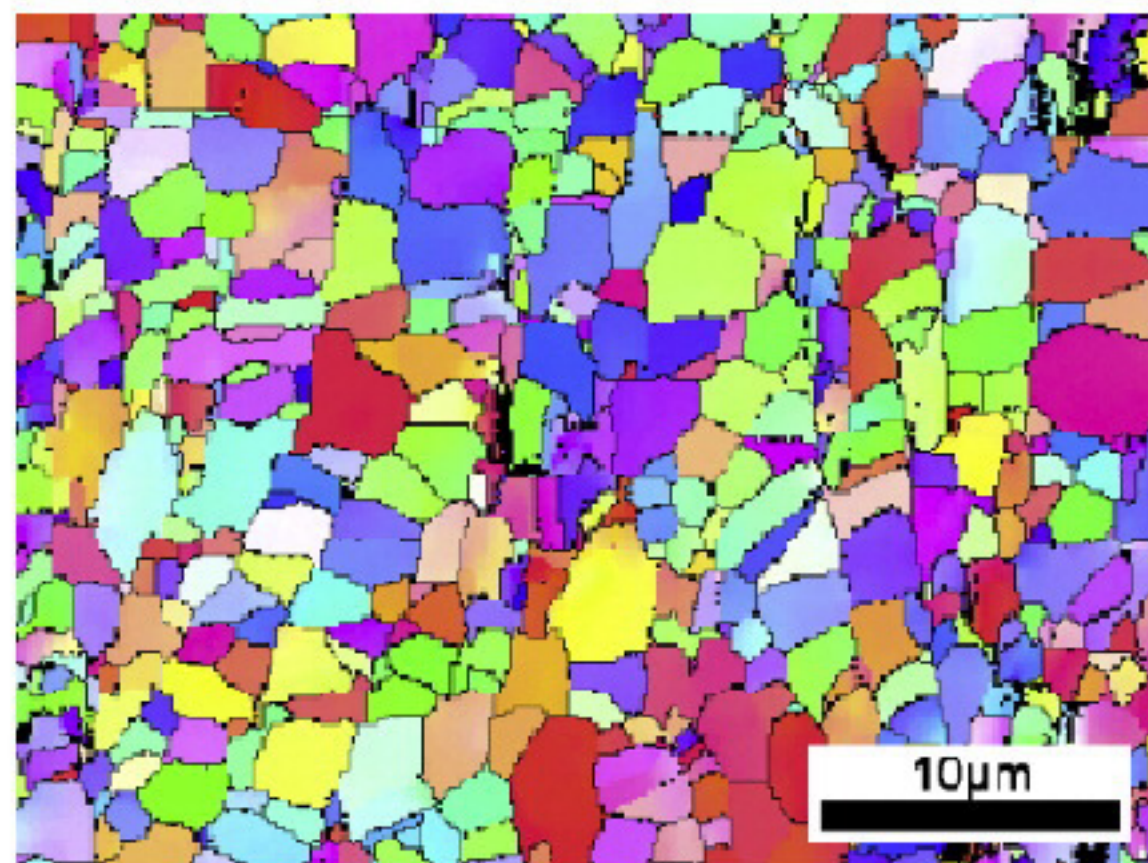
depend on grain uniformity (let alone H-P)



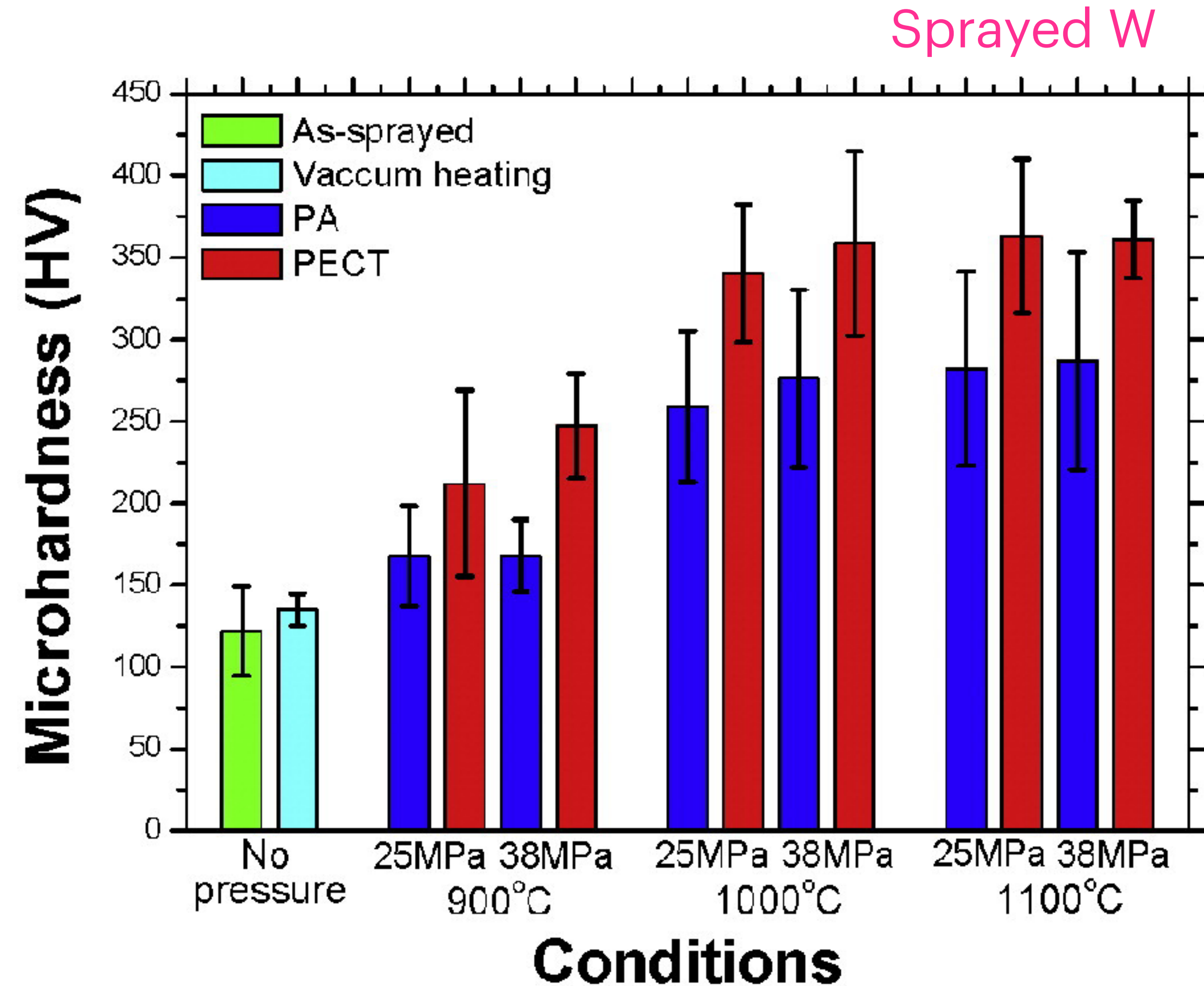
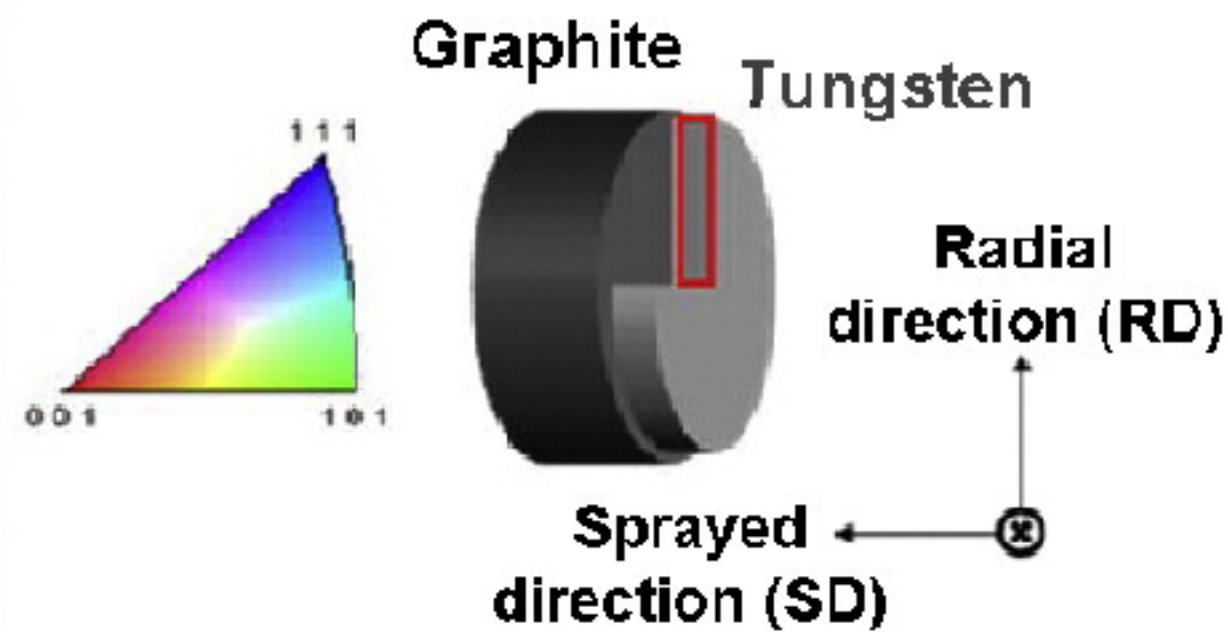
As-sprayed



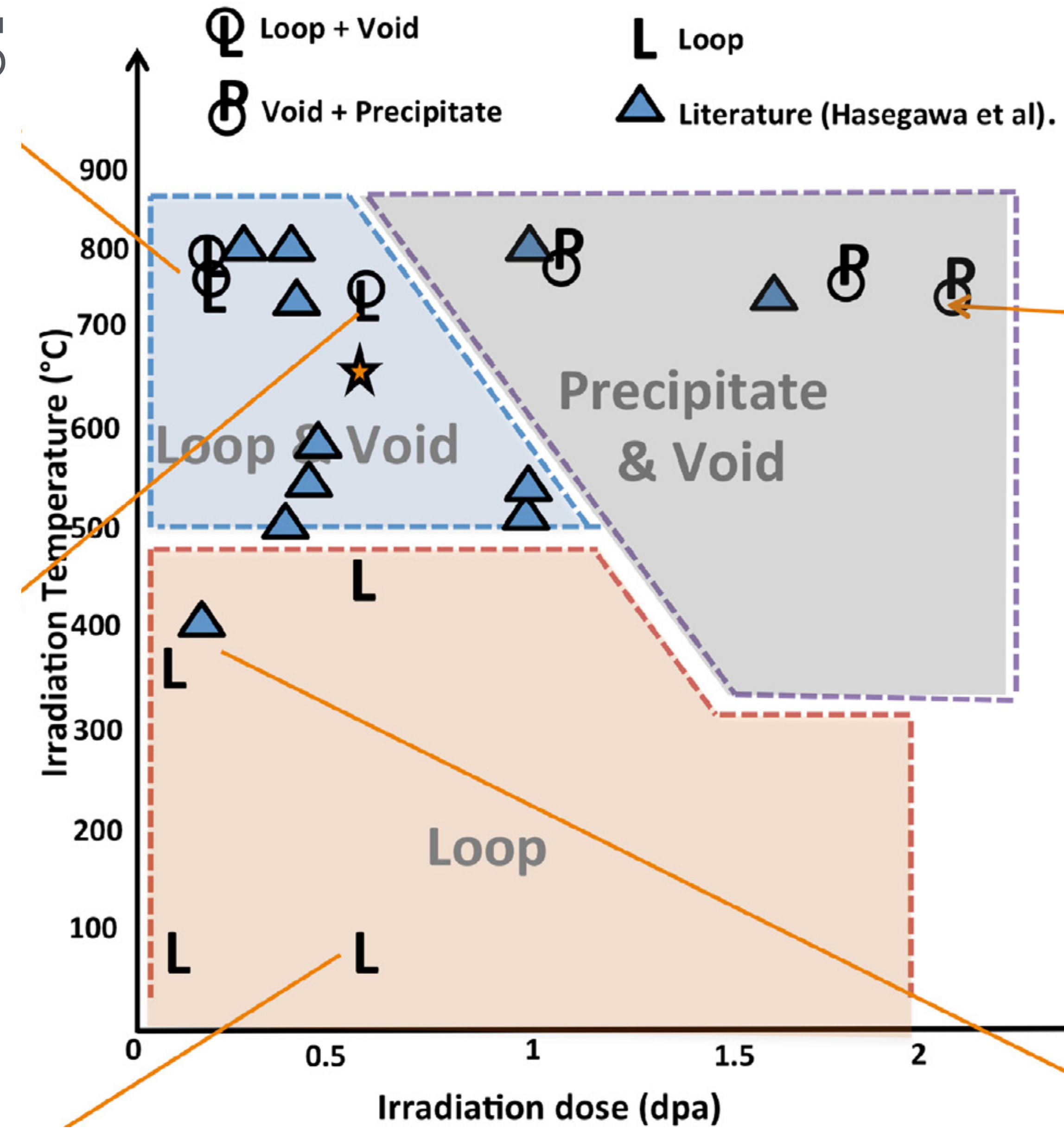
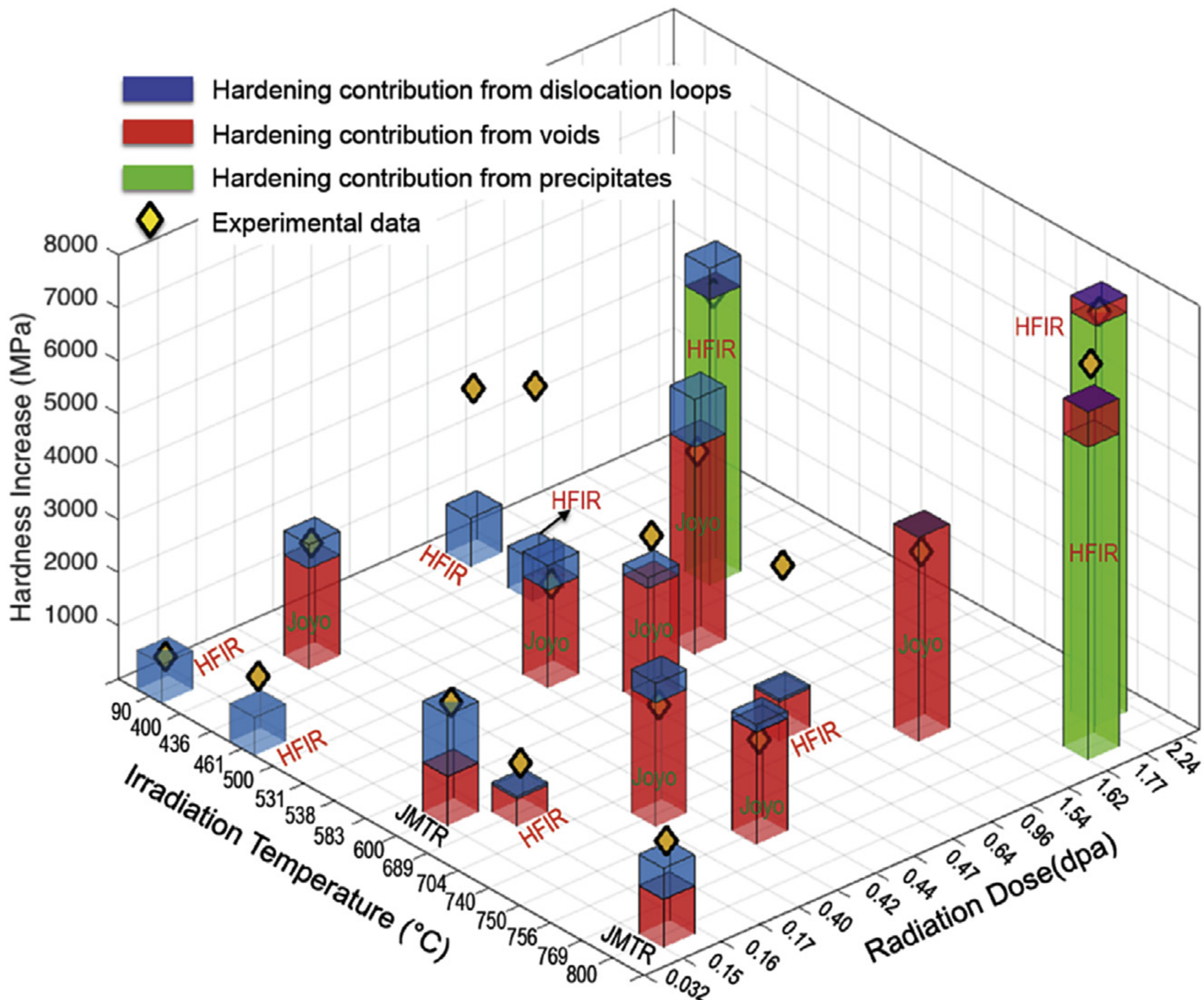
Press-annealing (PA)



Pulsed electric current treatment (PECT)



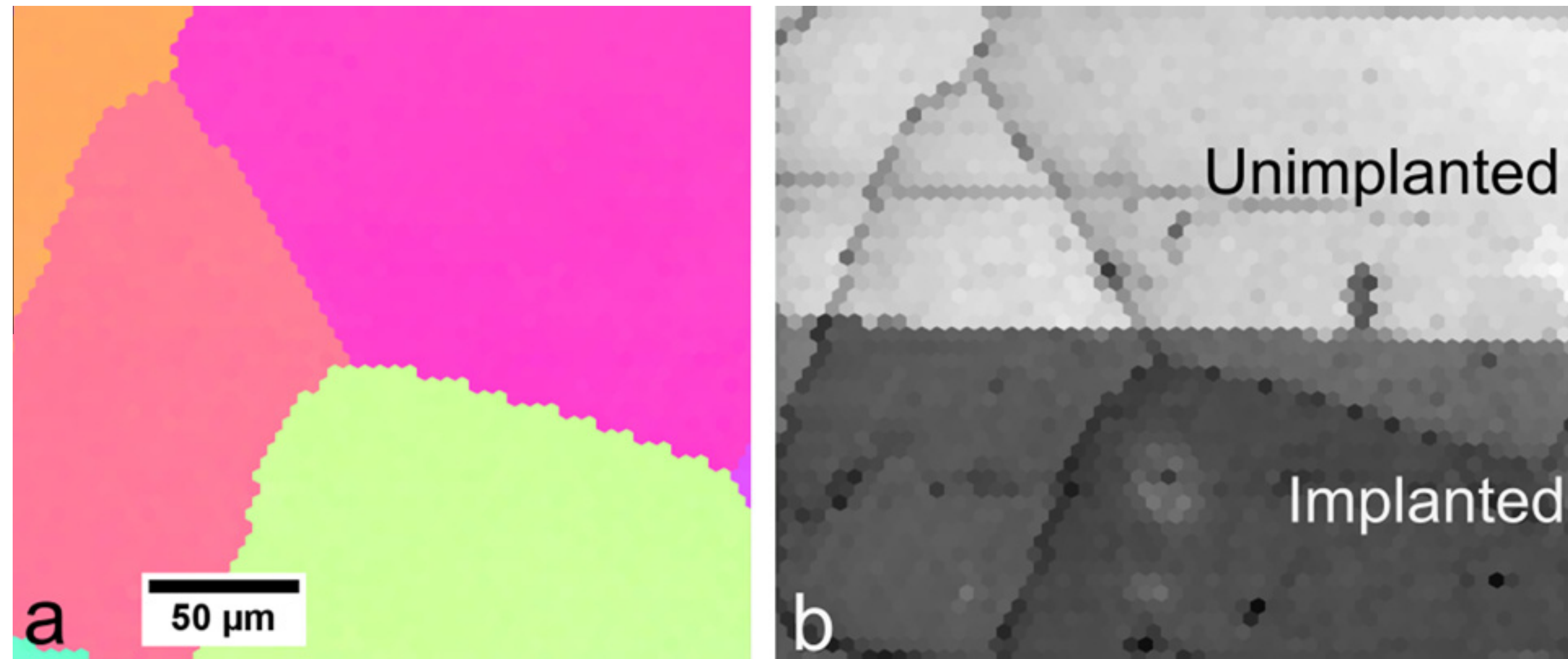
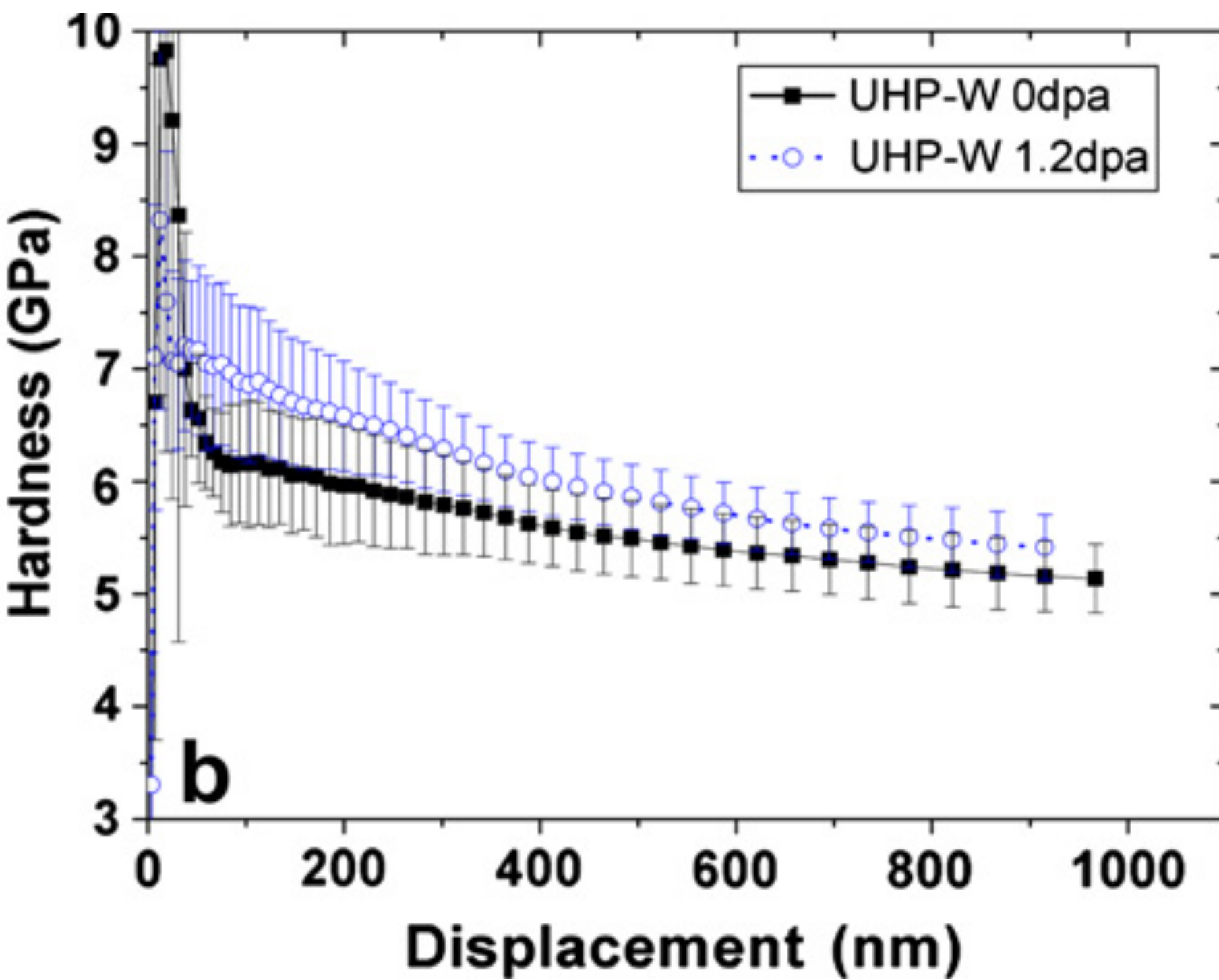
Mechanical properties of n-irradiated materials



Grain boundaries remain nearly unchanged

Microstructure has changed though

self-ion irradiated W



(a) Inverse pole figure and (b) image quality map produced using EBSD

Is GB transparent
to neutron irradiation?

Is GB transparent
to neutron irradiation?

Likely!!

**Insufficient facts
always invite danger.**



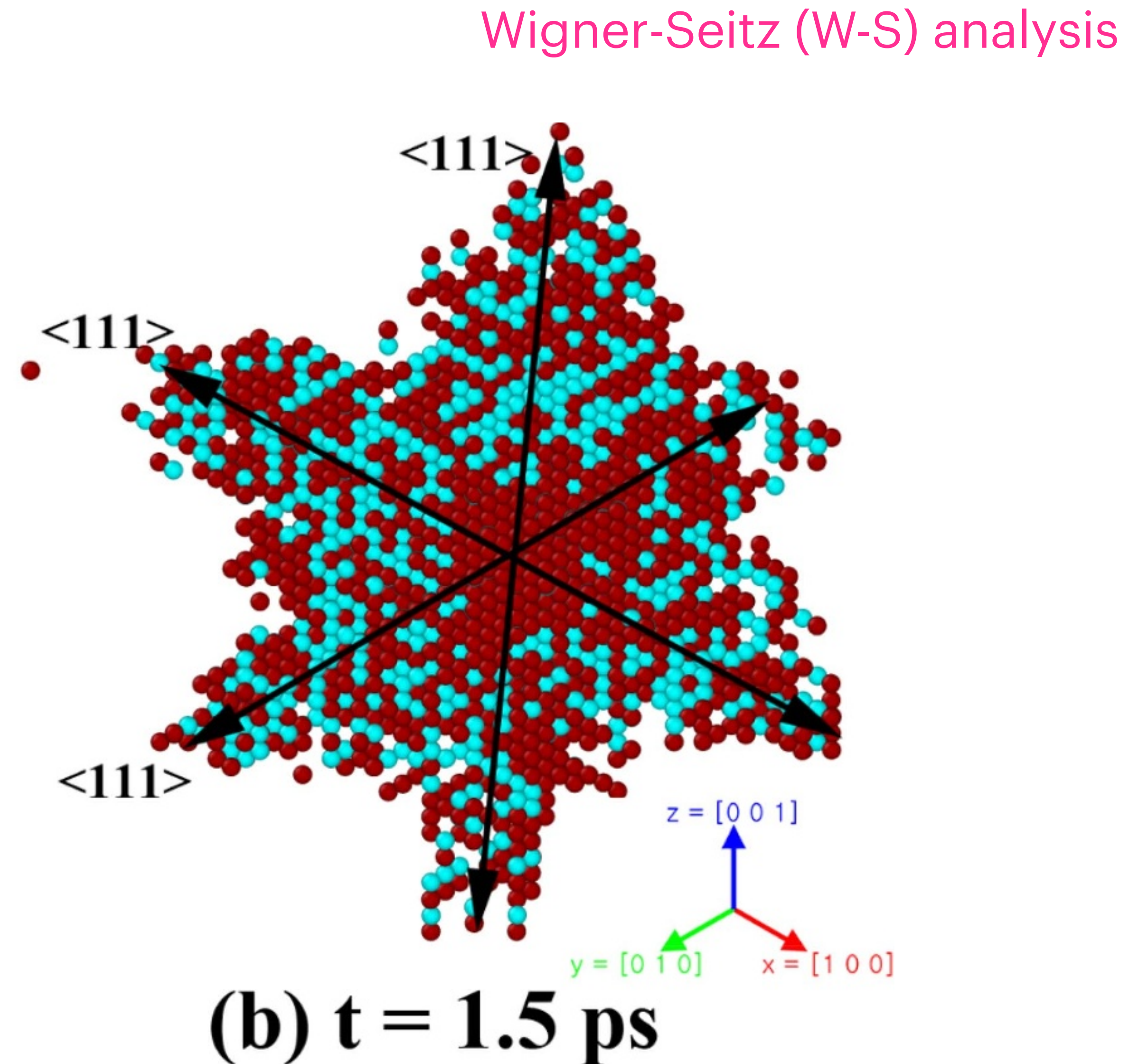
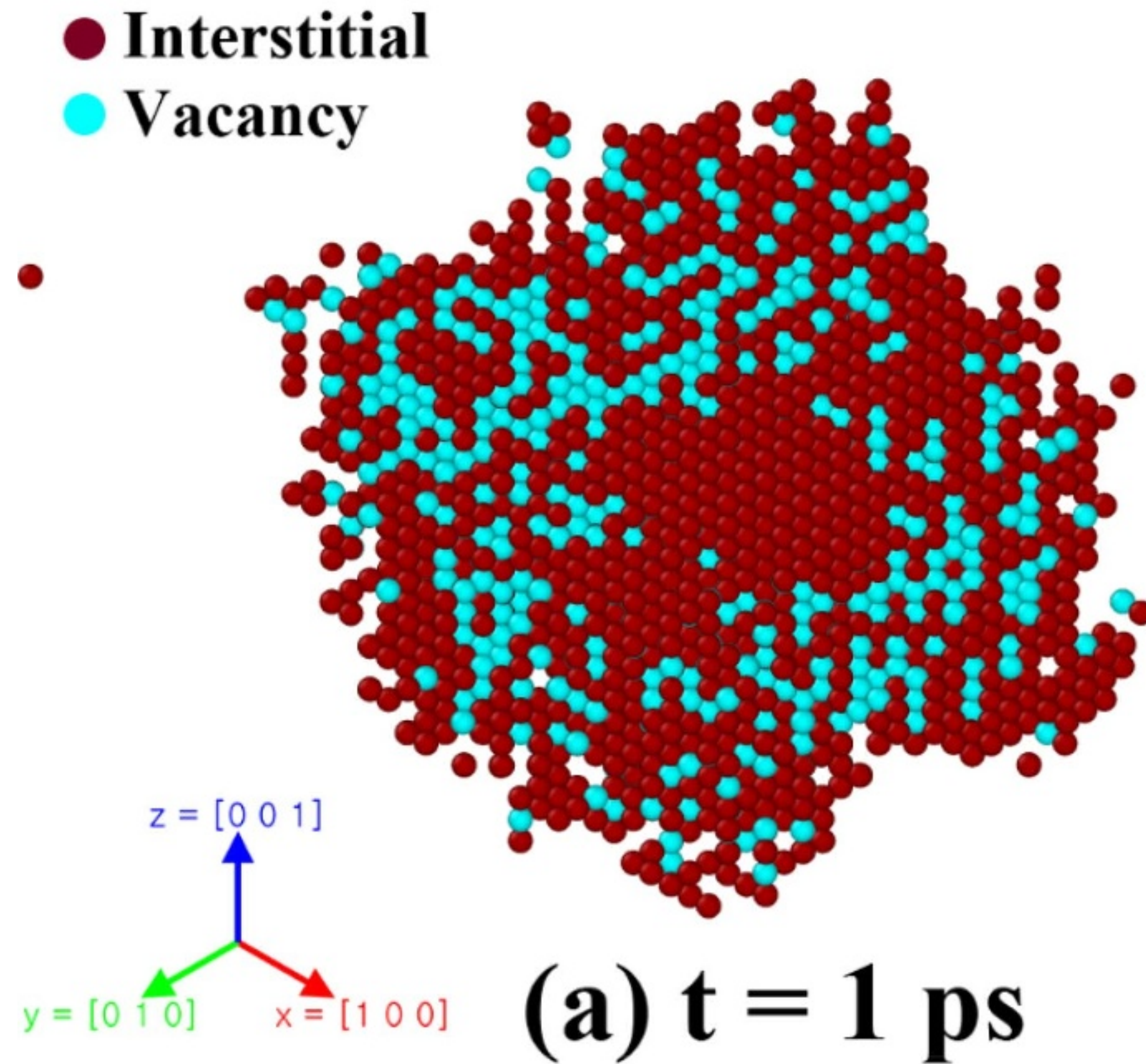
PKA simulations

Primary Knock-on Atoms to simulate n-irradiated cascades

- Cascades
 - In periodic bulk
 - Near surface
 - Near grain boundary
 - Damage detection methods
 - Preliminary results

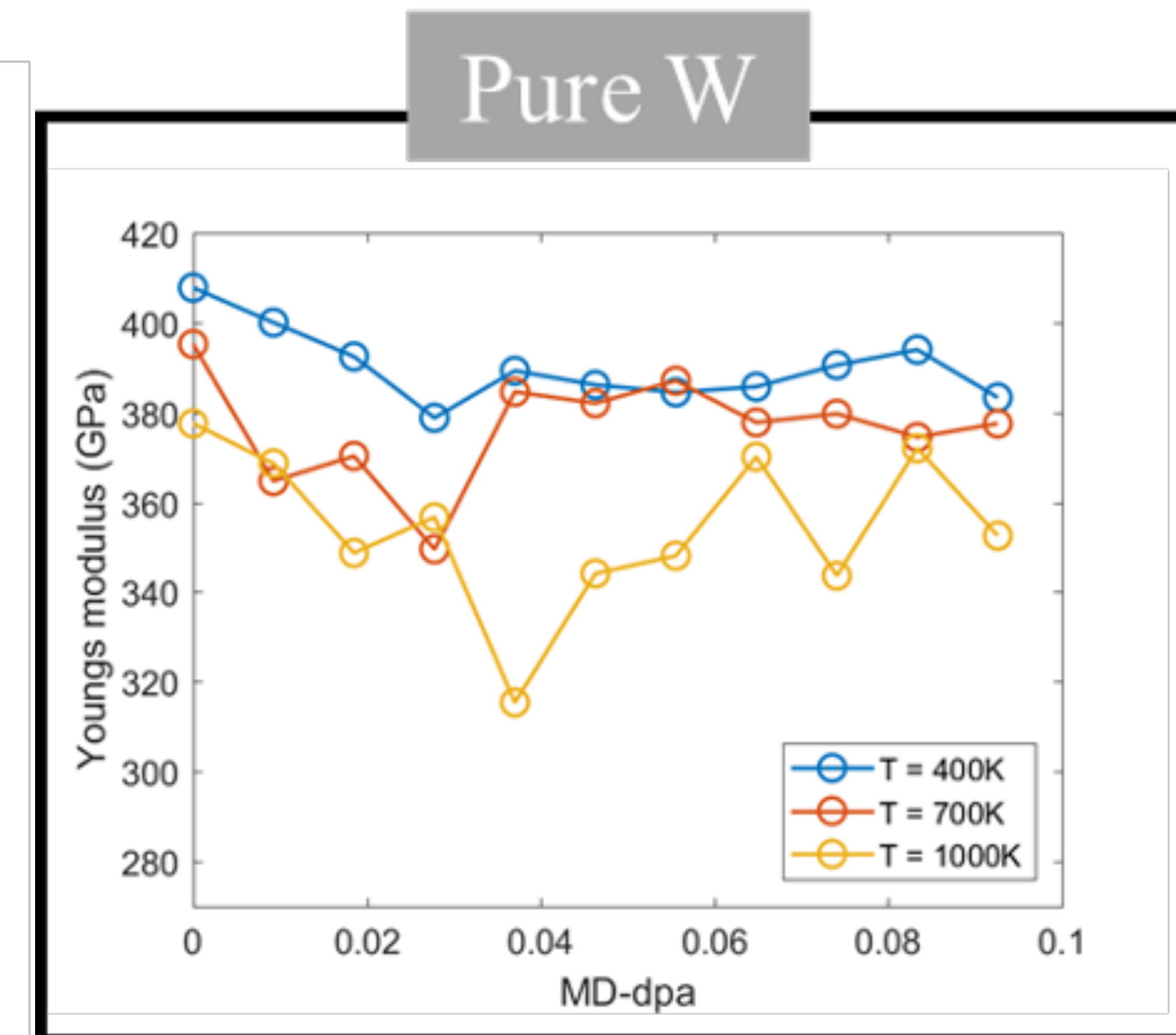
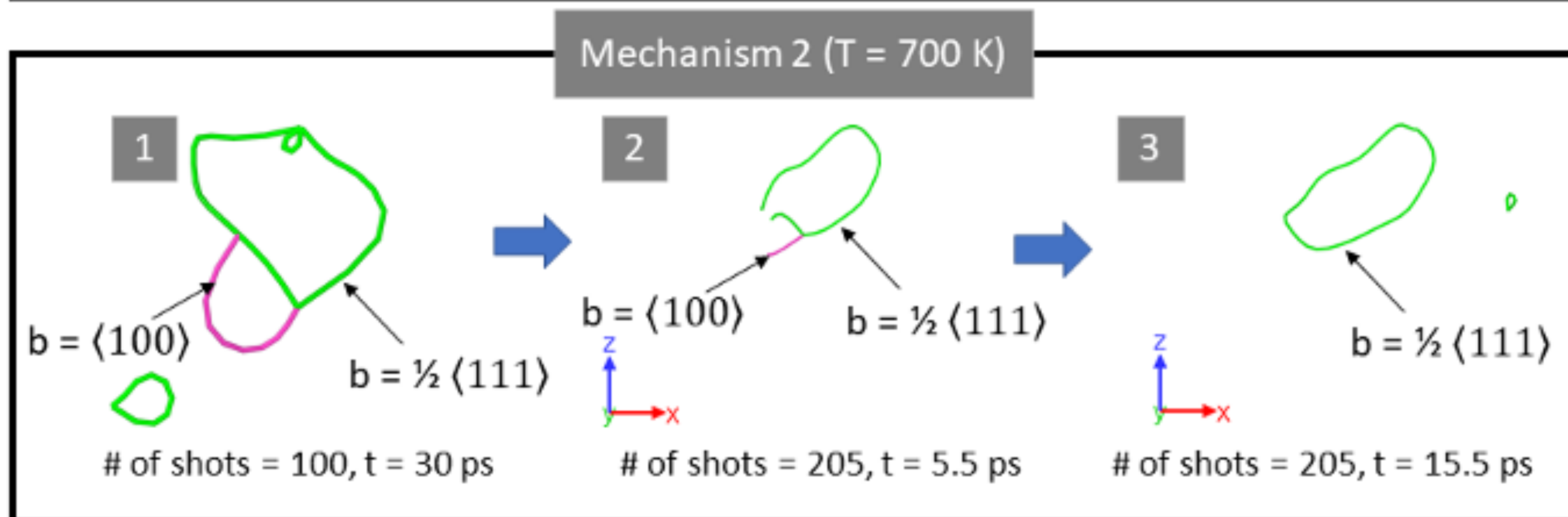
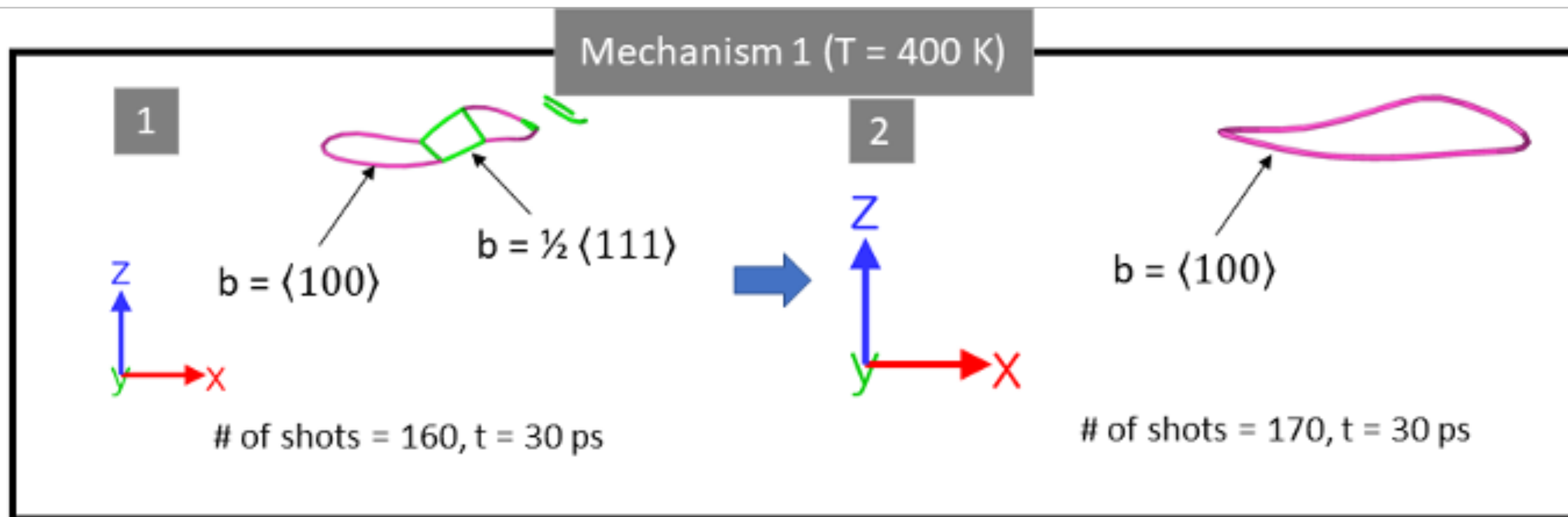
Irradiation damages

in single crystalline bulk



Total defect neutrality in periodic bulk

vacancies = # SIAs

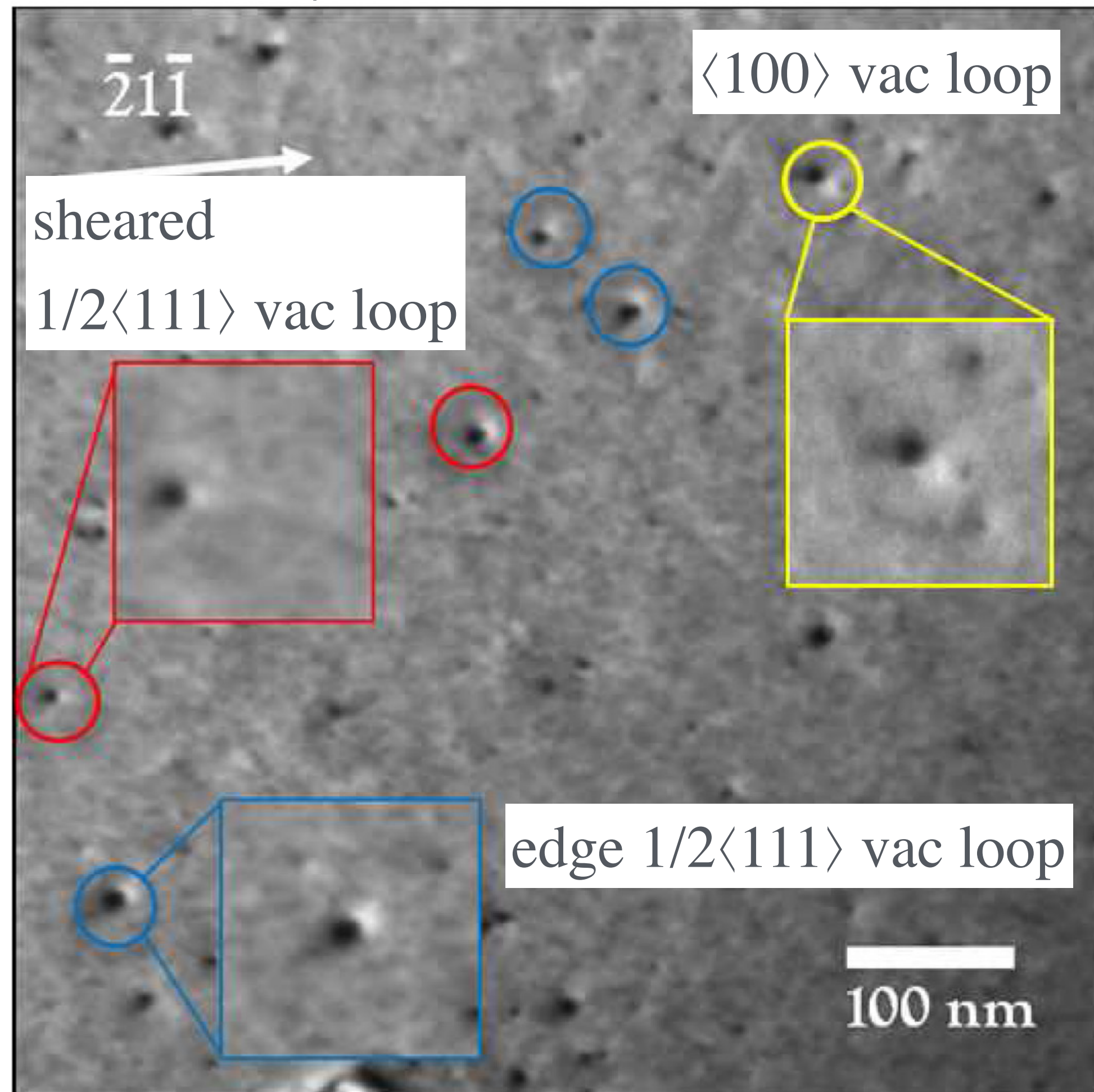


Young's modulus

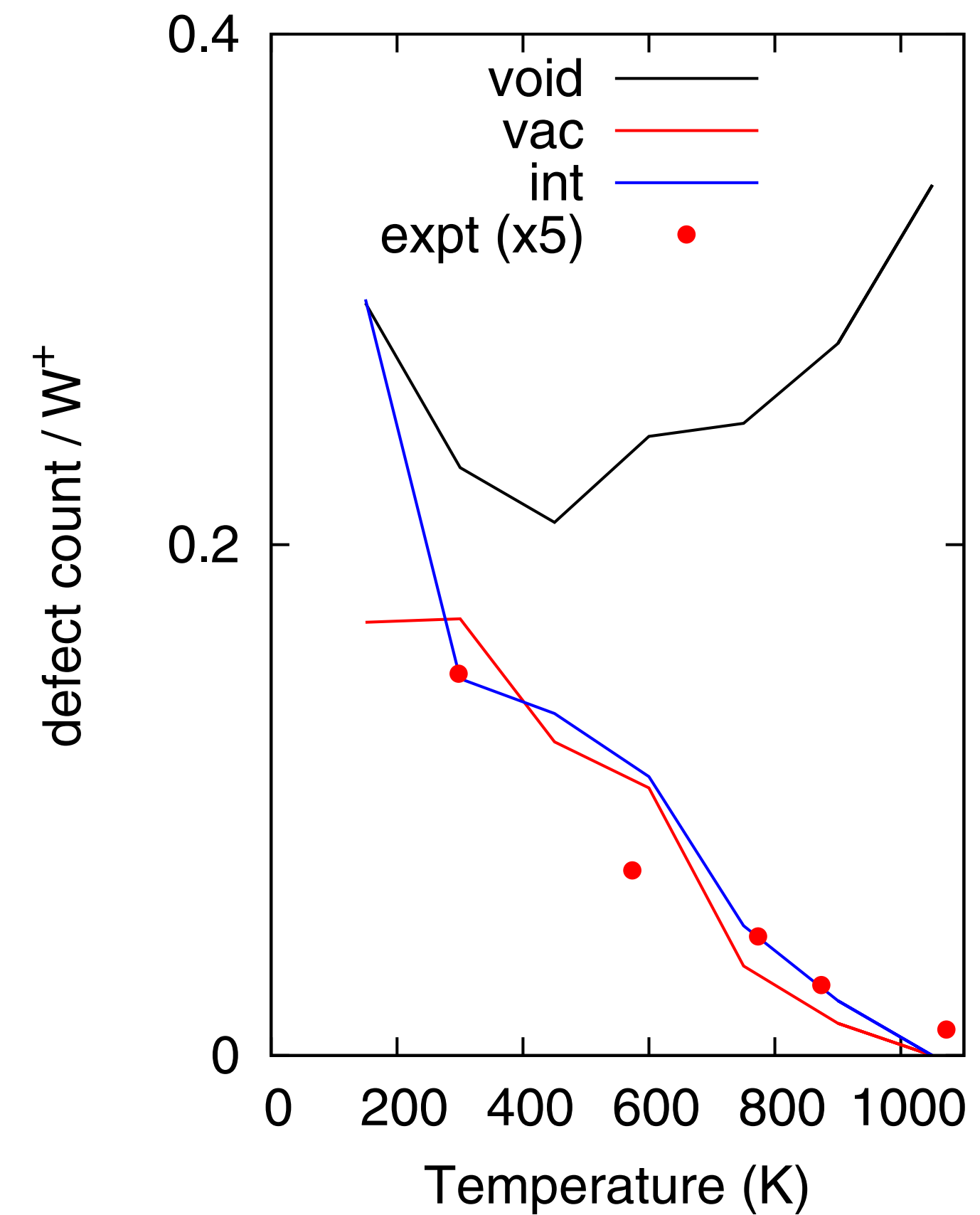
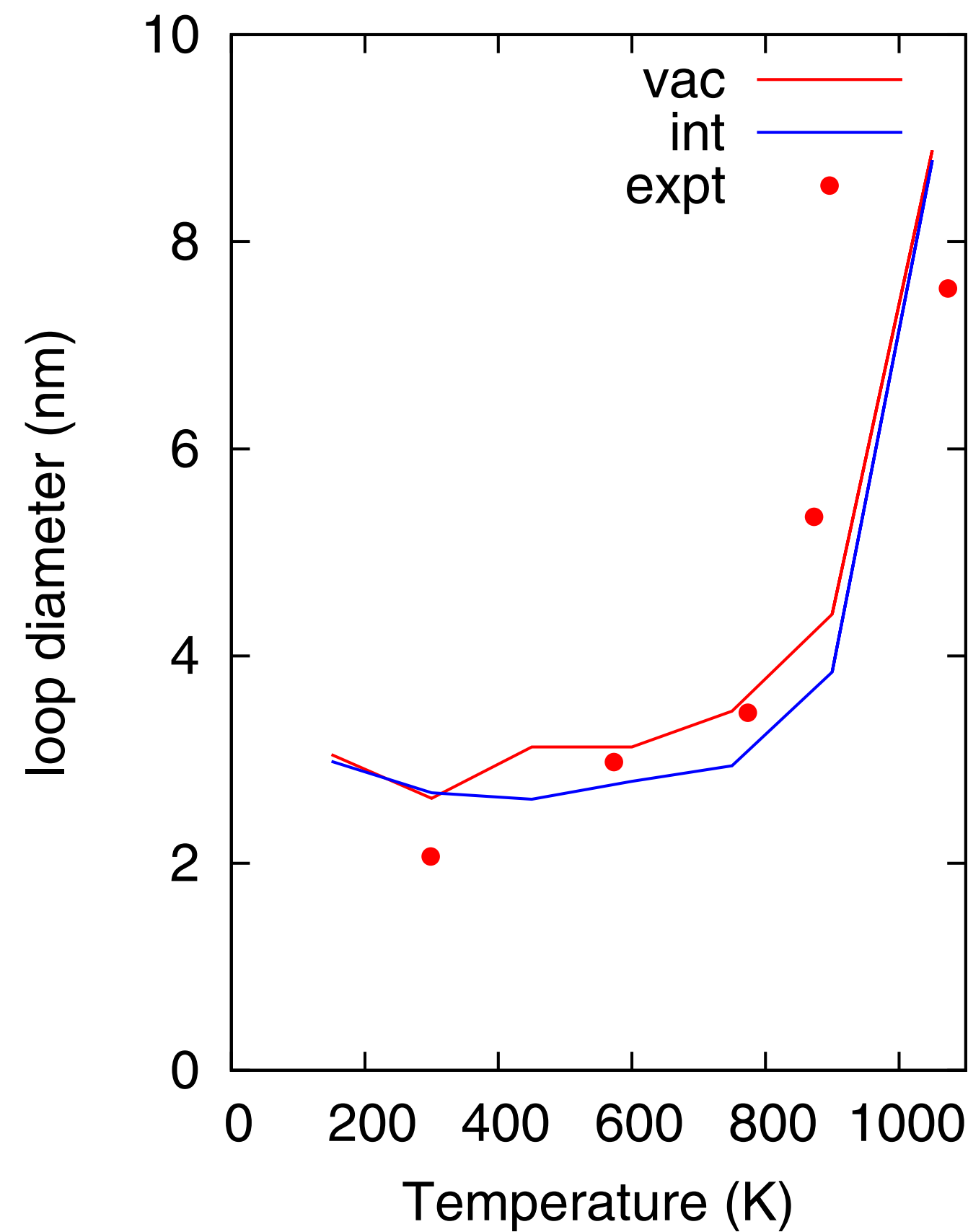
Irradiation damages

in W foils

~ 0.01 dpa w/ 150 keV W⁺ @ 300K

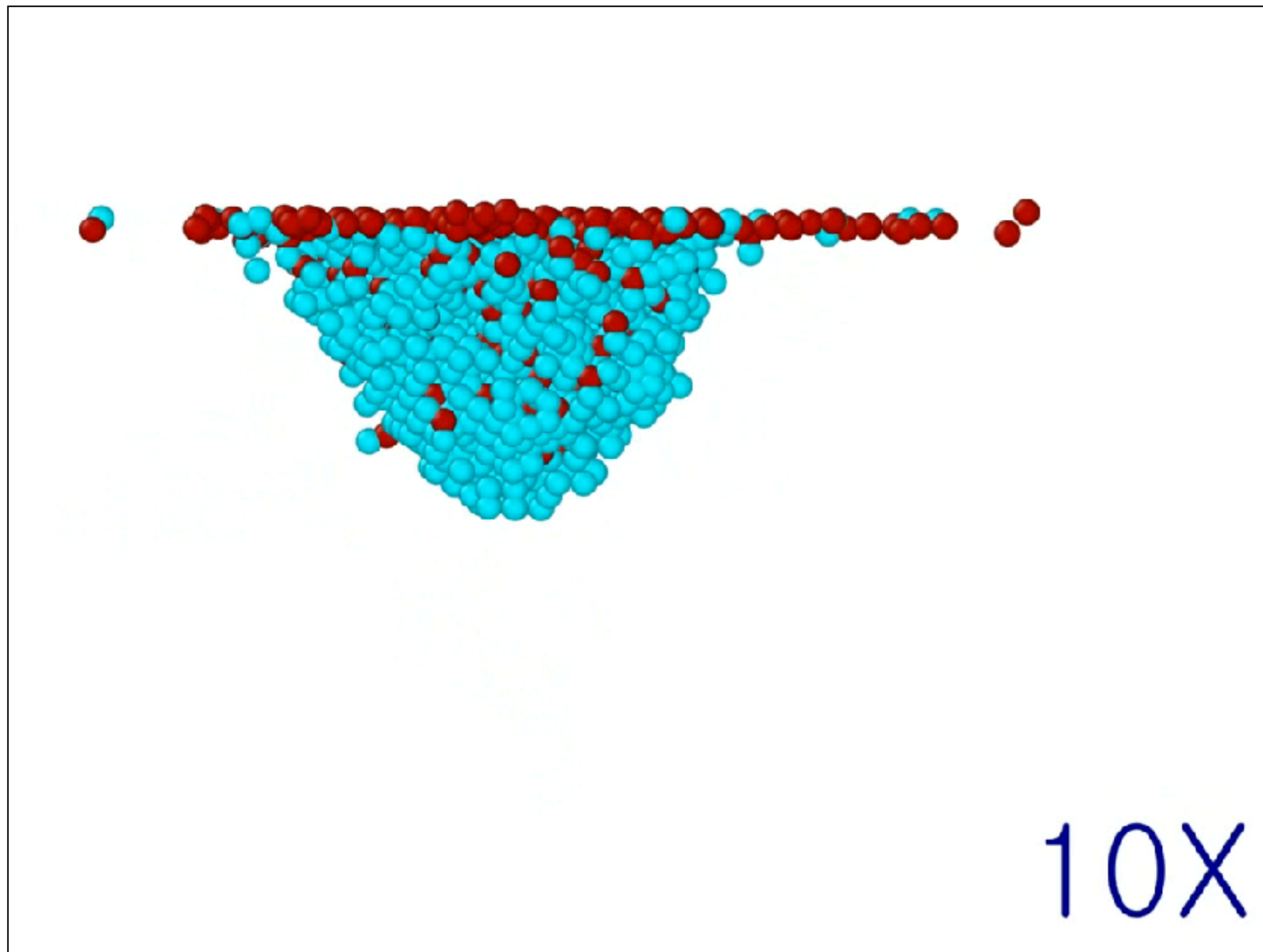


“the largest loops to be predominantly of prismatic $1/2\langle 111 \rangle$ type and of vacancy character”

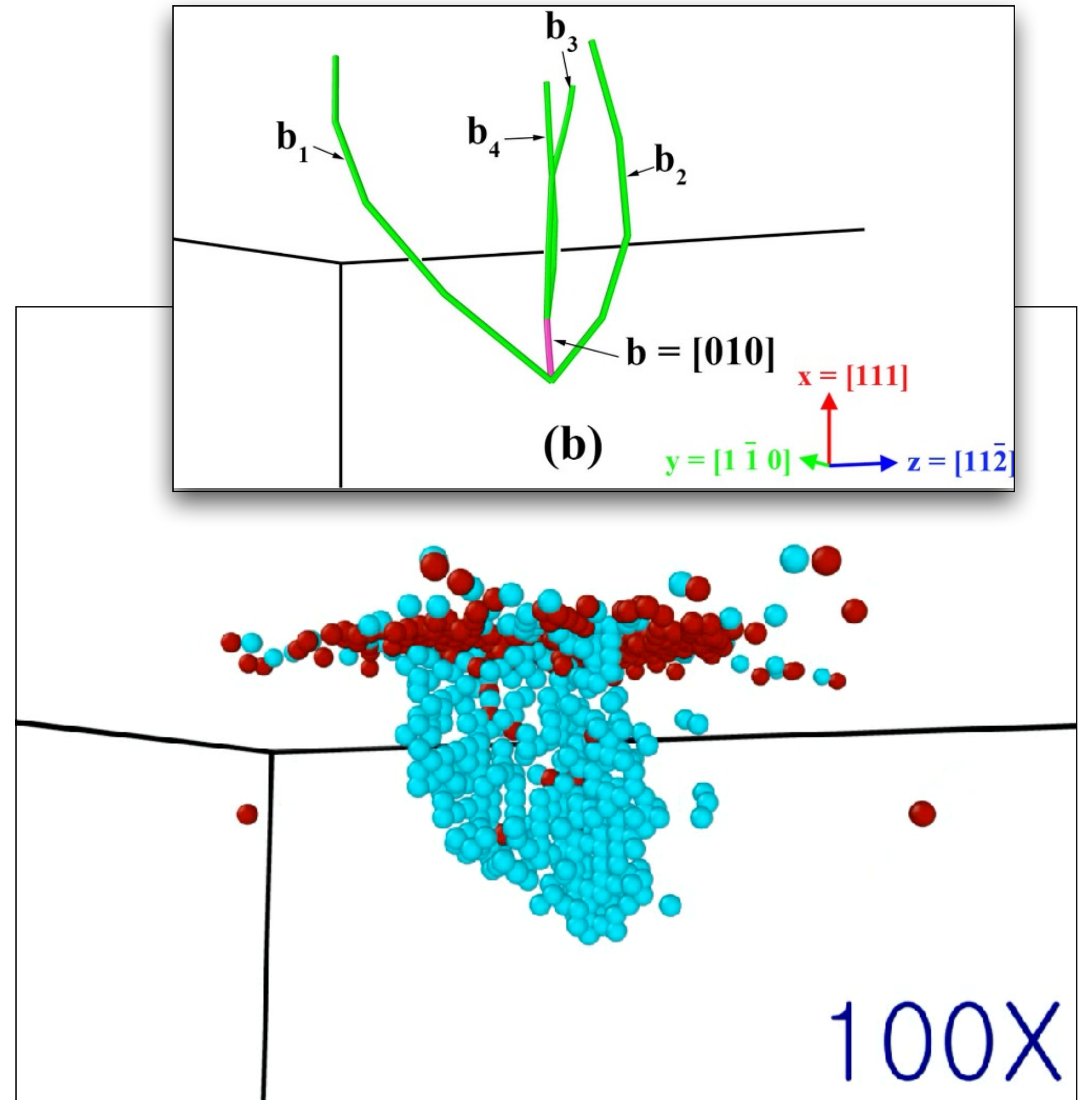


Irradiation damages

near surface



Mobile (vacancy) dislocation loop



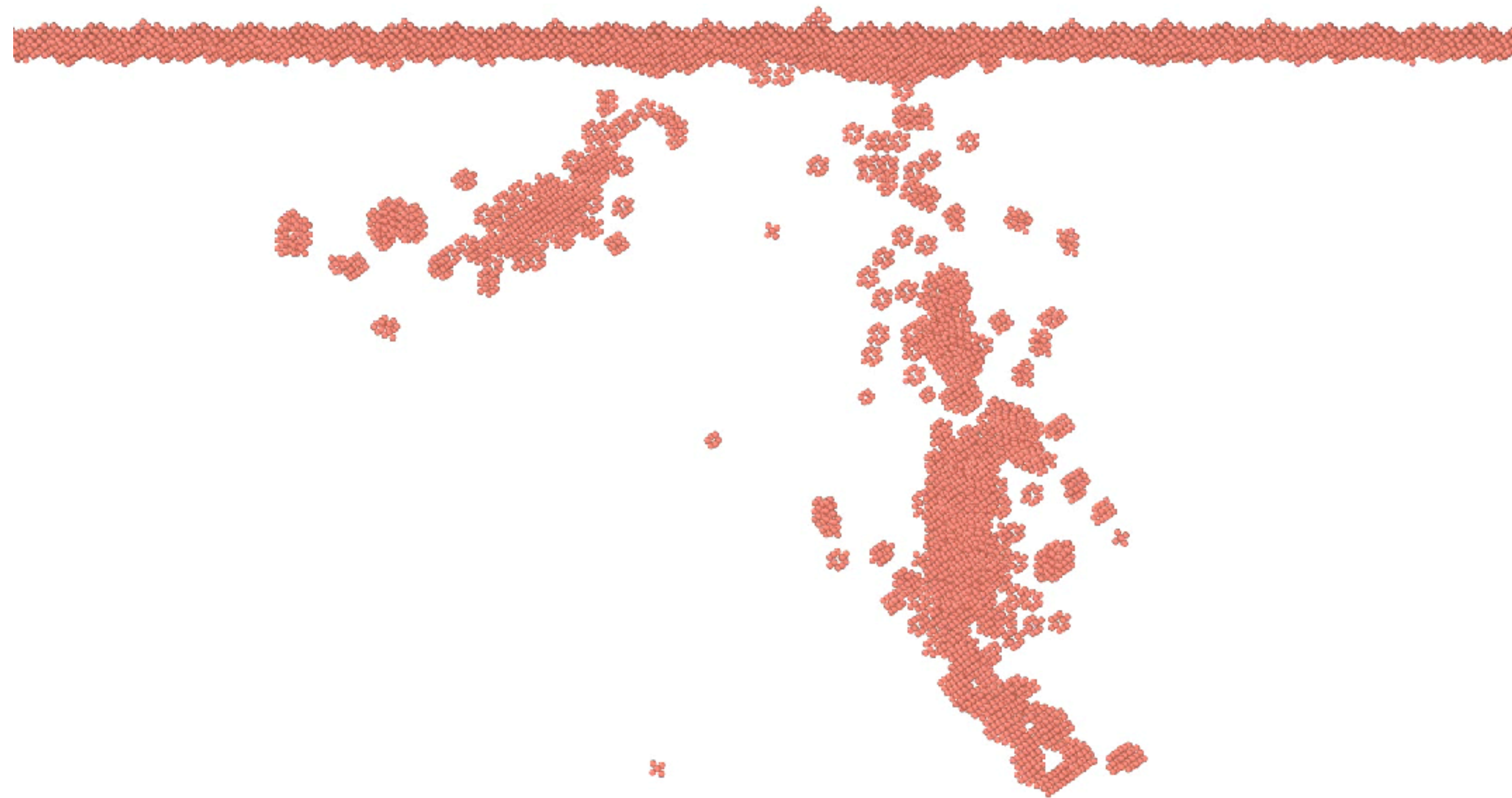
Immobile (vacancy) dislocation network

Surface ~ a defect sink w/ infinite capacity

Irradiation damages

near grain boundary (GB)

Grain boundary ~ a defect sink w/ **finite** capacity



Common Neighbor Analysis (**CNA**) showing non-BCC atoms

GBs absorb defects

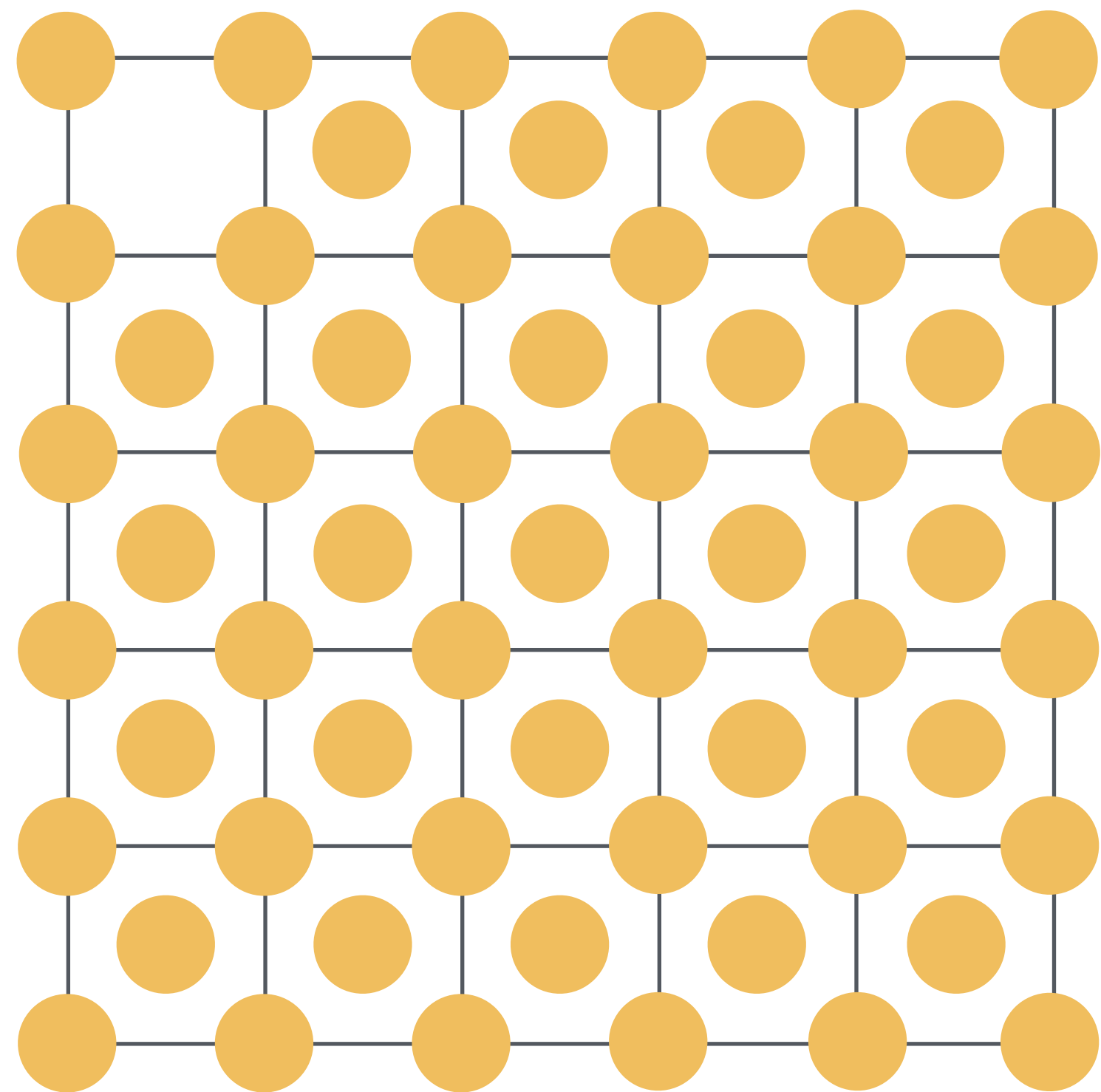
Then what?

- Possible scenarios
 - Diffusion inside GBs
 - Dislocations
 - **Crystal growth**
 - May be in the form of GB motion for small-area boundaries

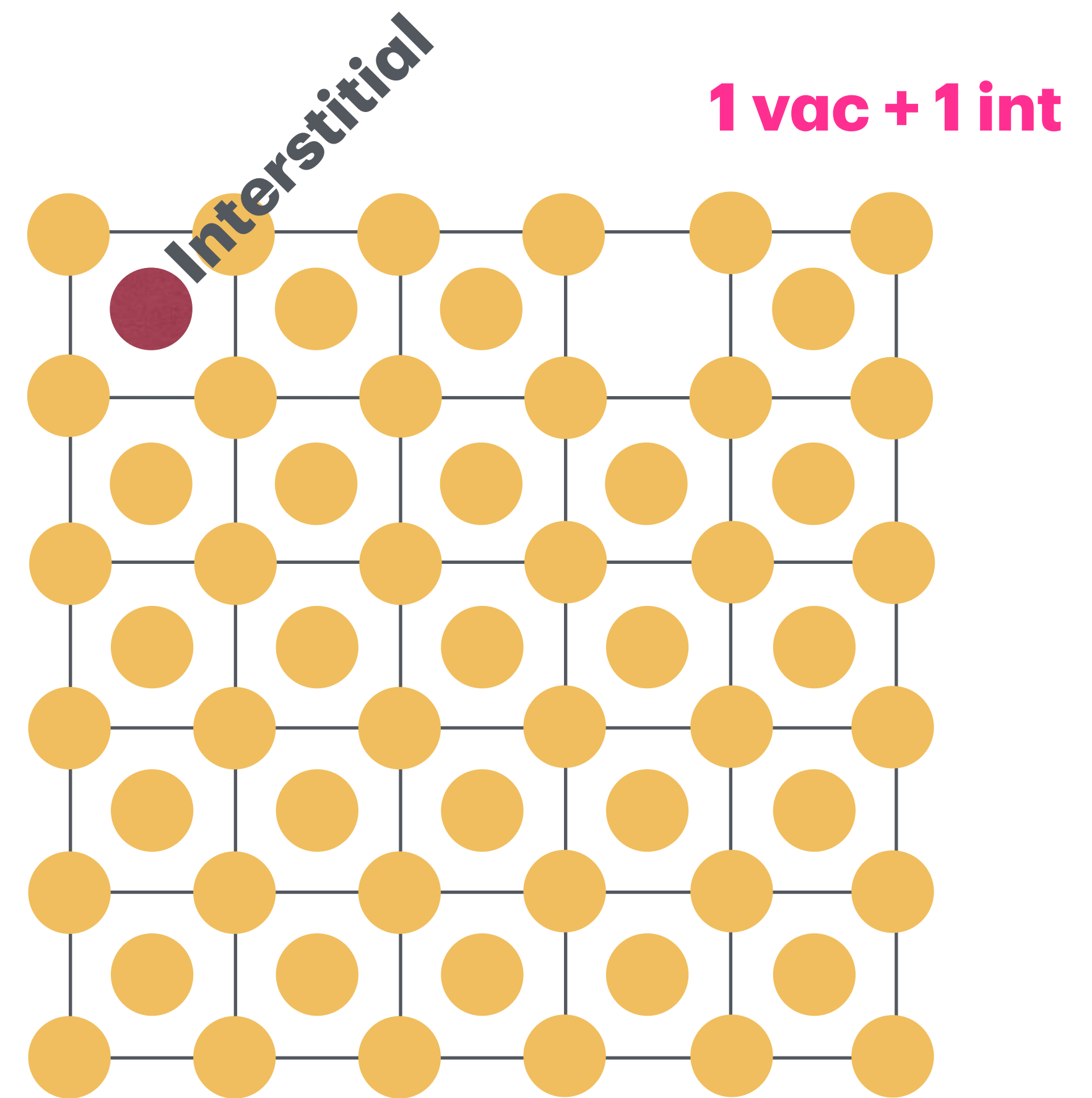
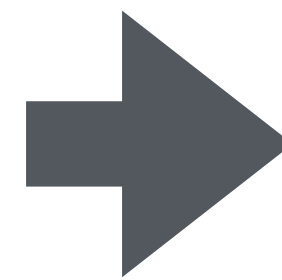
We need defect counts

Wigner-Seitz as implemented in popular tools

Based on initial atomic positions



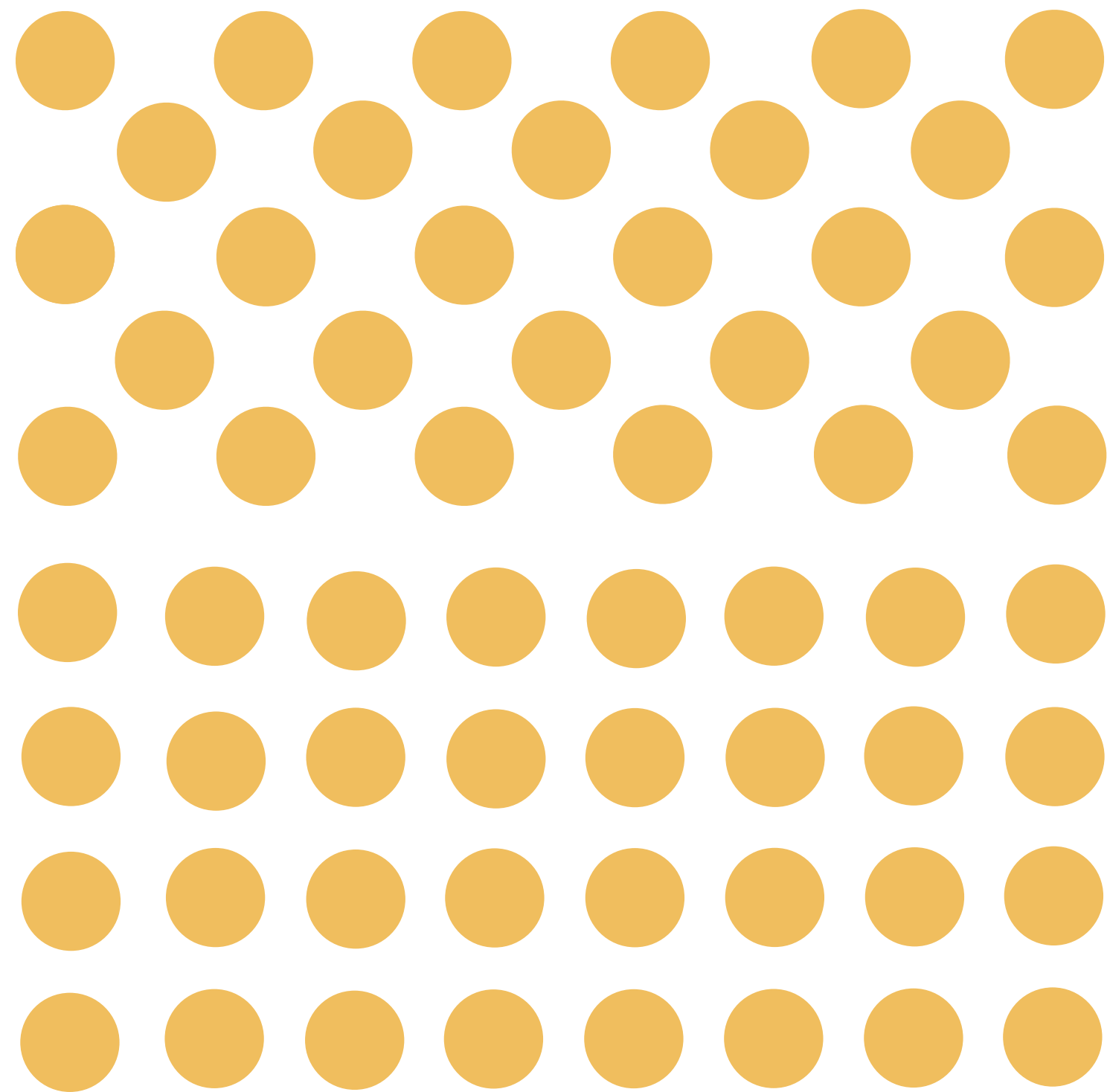
Initial state



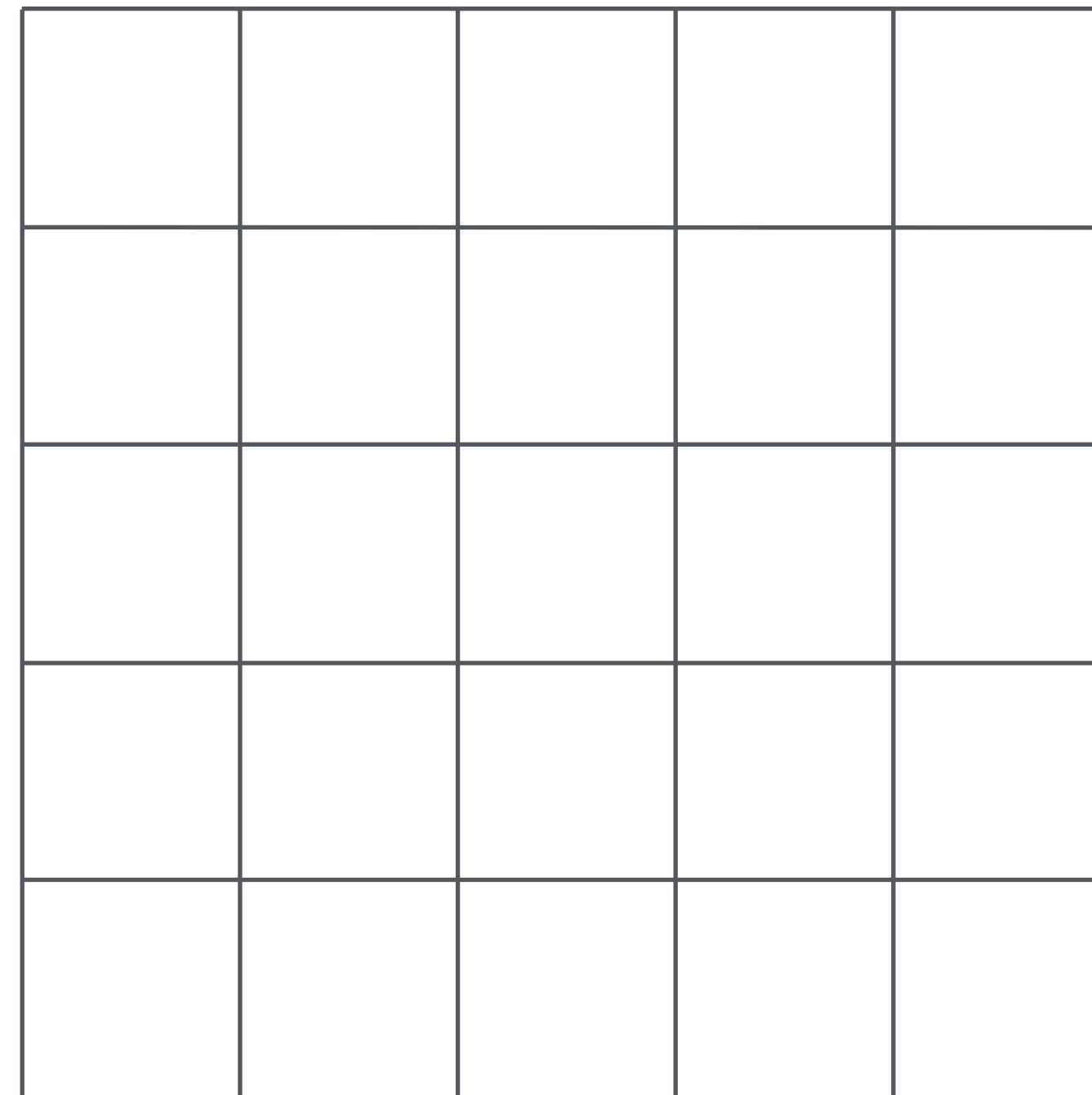
Final state

W-SO

with a fictitious perfect lattice

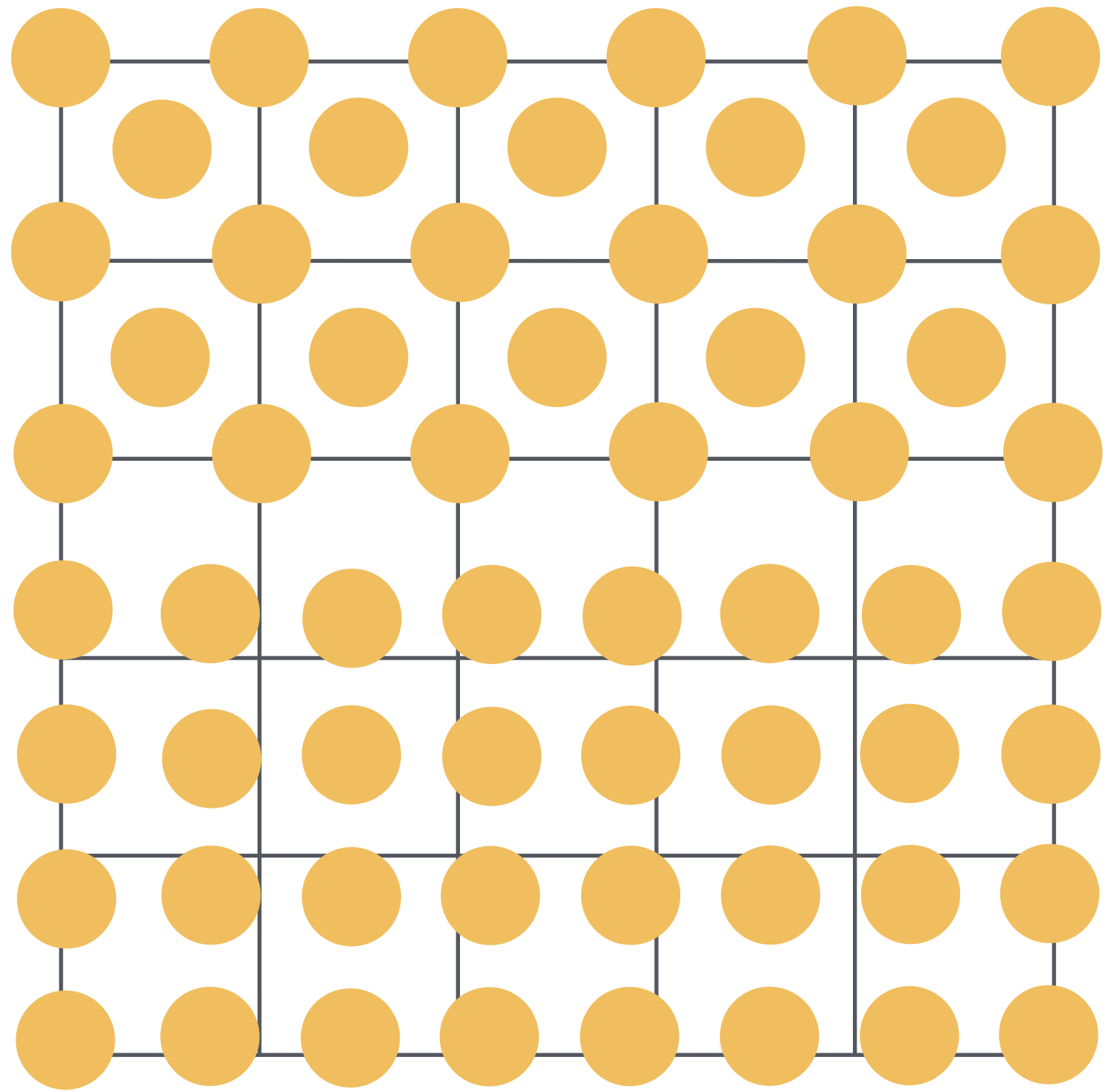


Volume-based



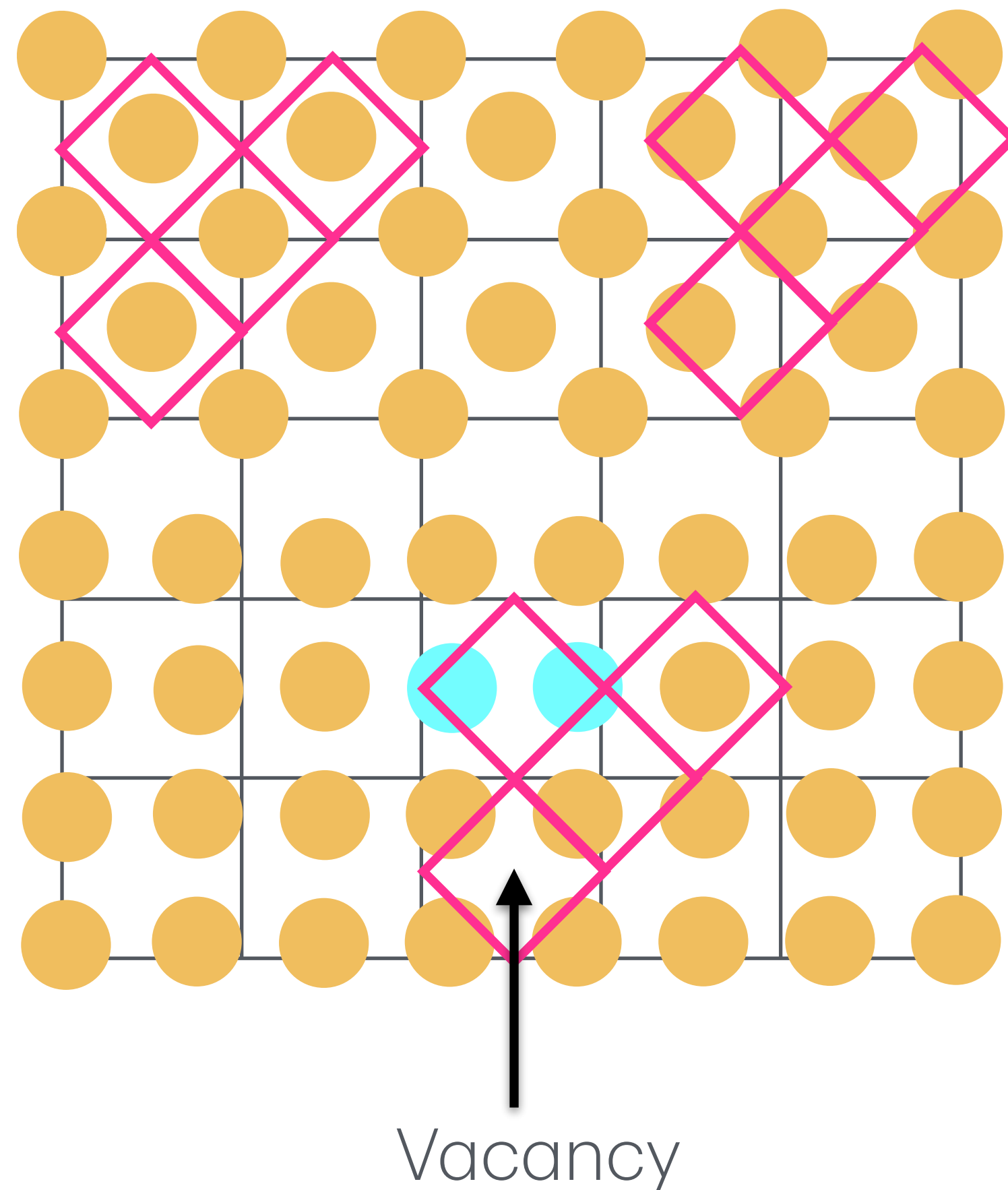
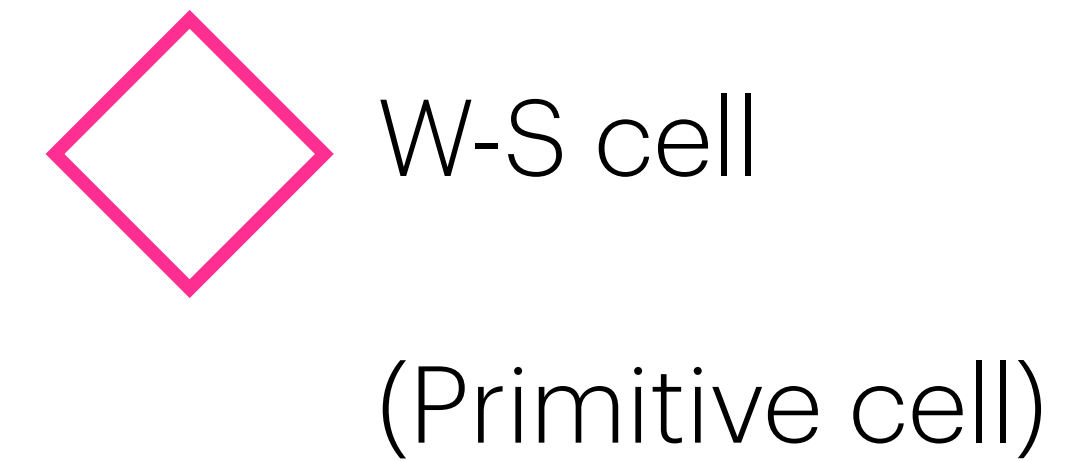
W-SO

with a fictitious perfect lattice



W-SO

fails in polycrystalline structures



Translation of reference lattice **hardly** matters

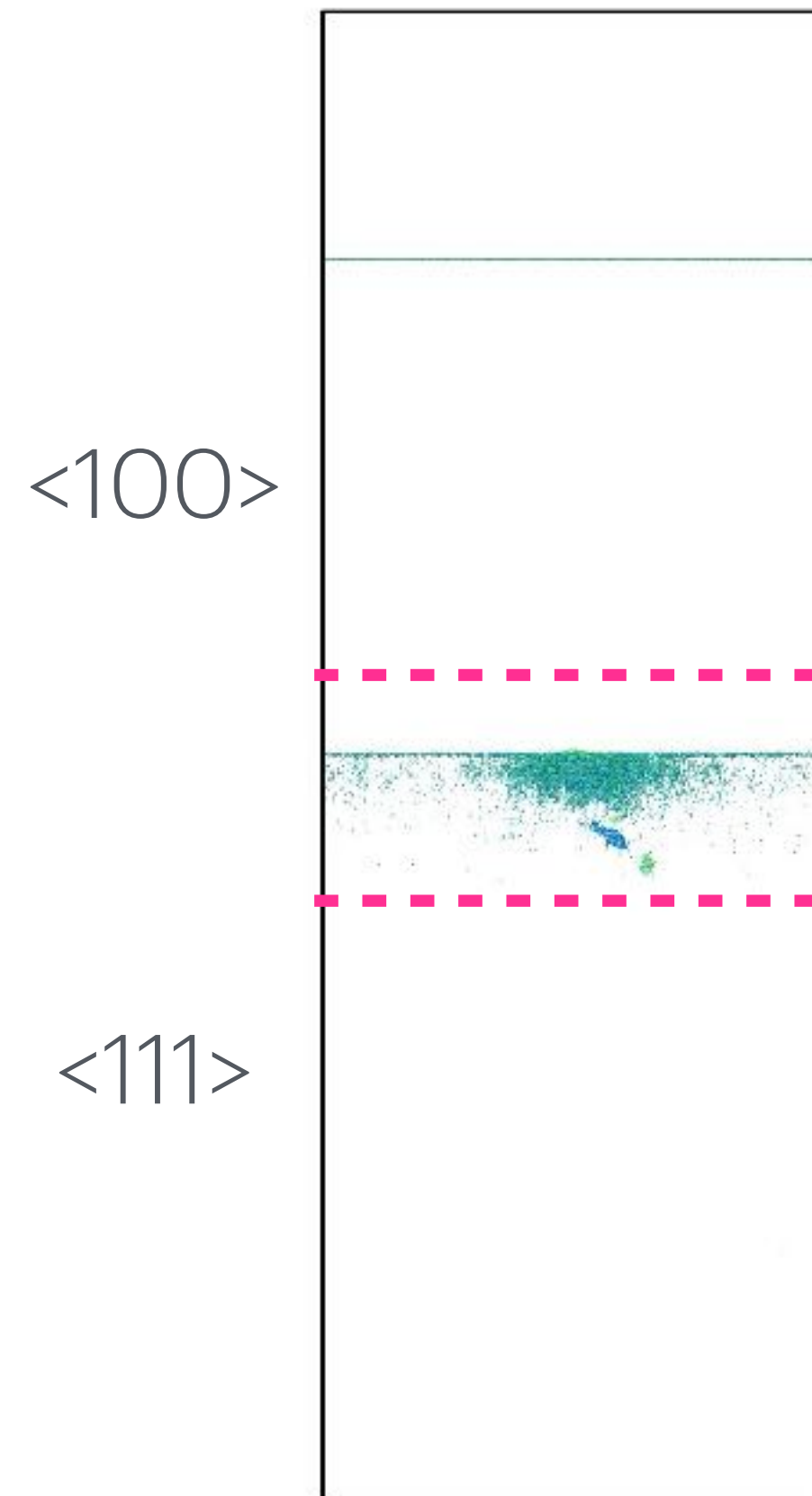
Rotation matters!!

● Interstitials

Interstitial-vacancy pair FALSE detection

W-S

fails even for a clean GB

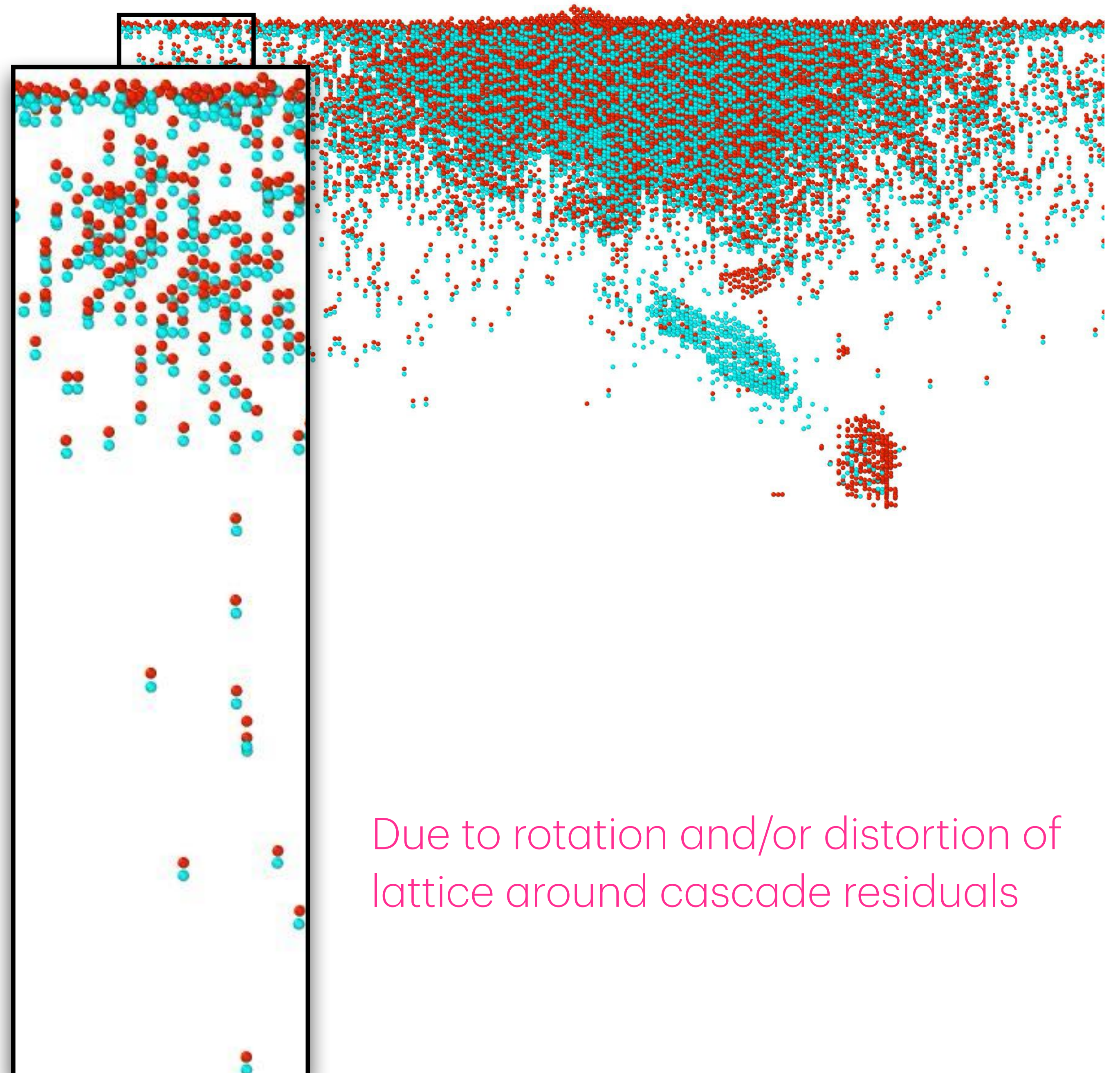


100 nm x 100 nm x 300 nm

PKA Energy = 300 keV @ 823K

PKA location = 1nm from GB

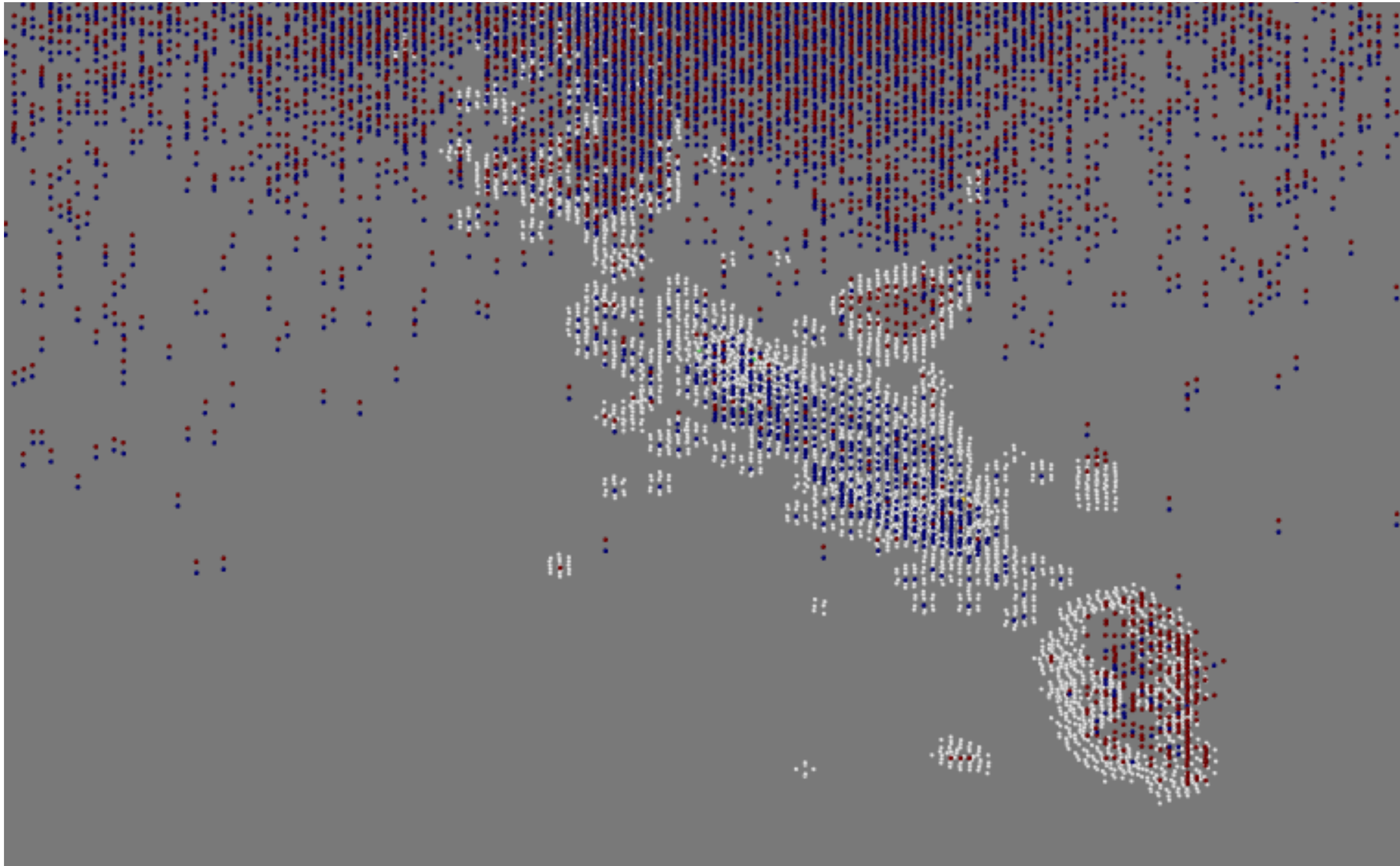
■ Vacancy ■ Interstitial



Due to rotation and/or distortion of lattice around cascade residuals

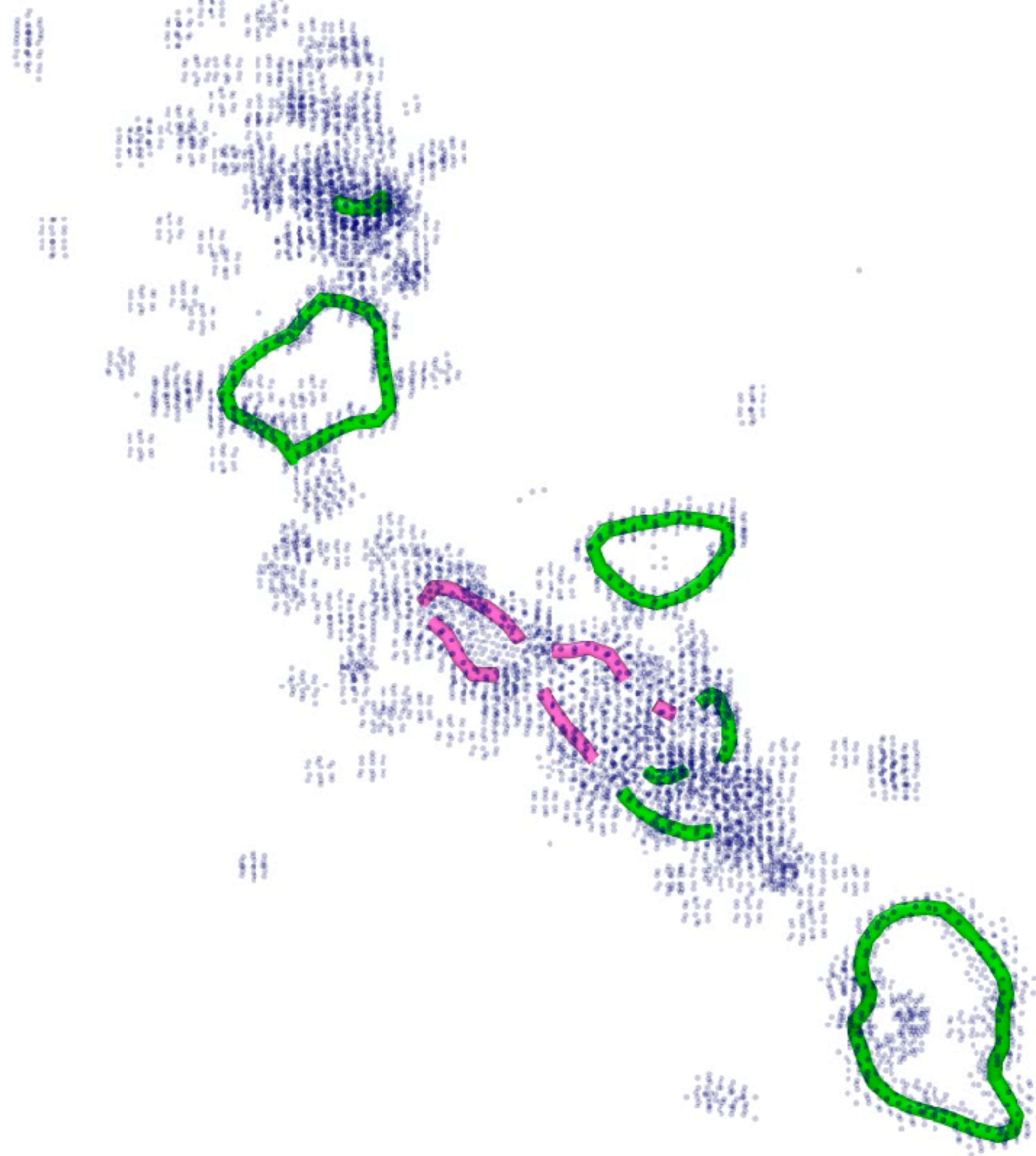
CNA+WS

■ Vacancy ■ Interstitial



Interstitial loops

CNA+DXA



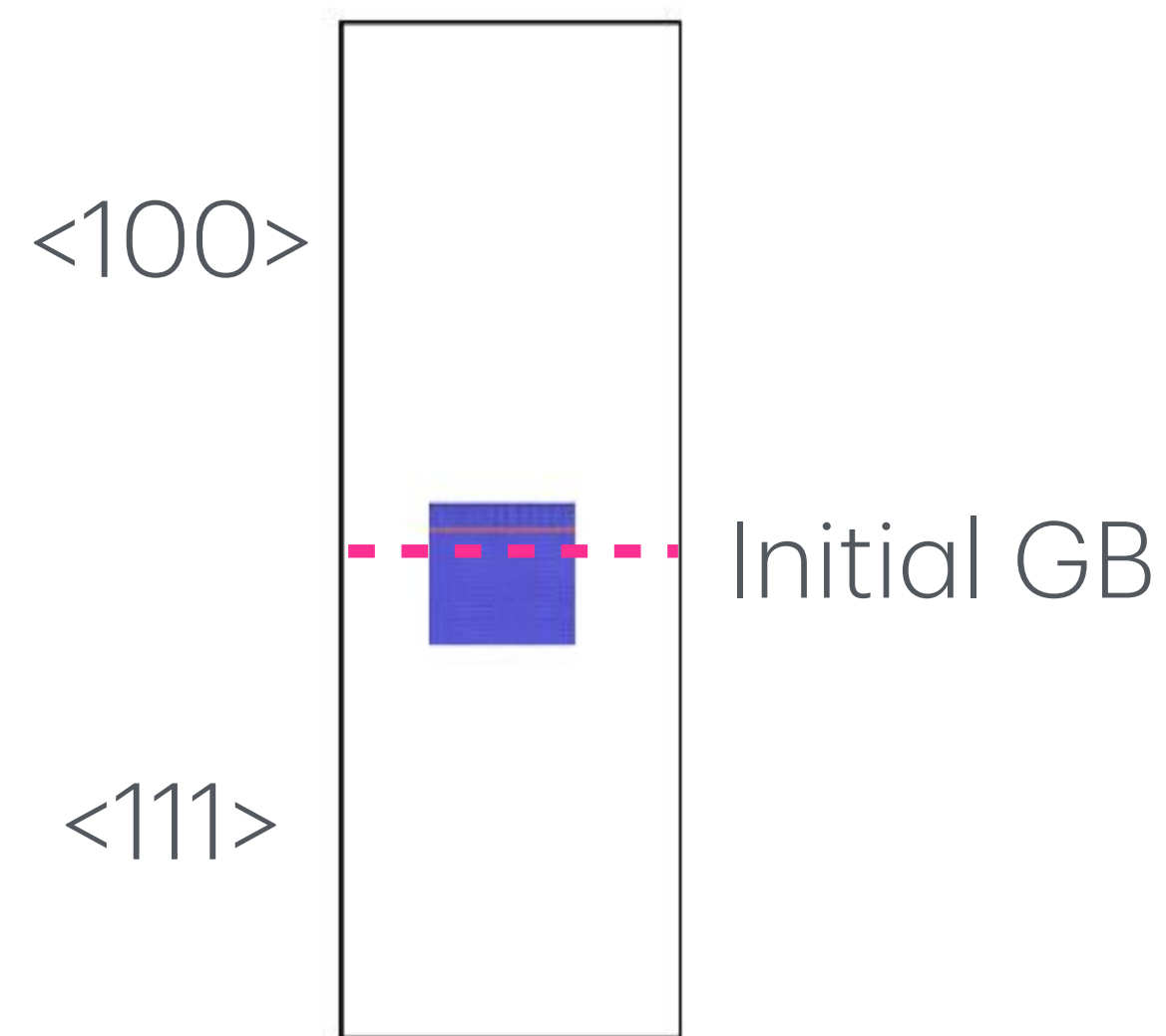
dislocation loop

Other $1/2\langle 111 \rangle$ $\langle 100 \rangle$ $\langle 110 \rangle$



How to separate defects

in GB from those in bulk w/ defect type info?



bcc $\langle 100 \rangle / \langle 111 \rangle$

100 nm x 100 nm x 300 nm

PKA Energy = 300 keV @ 823K

0 ps

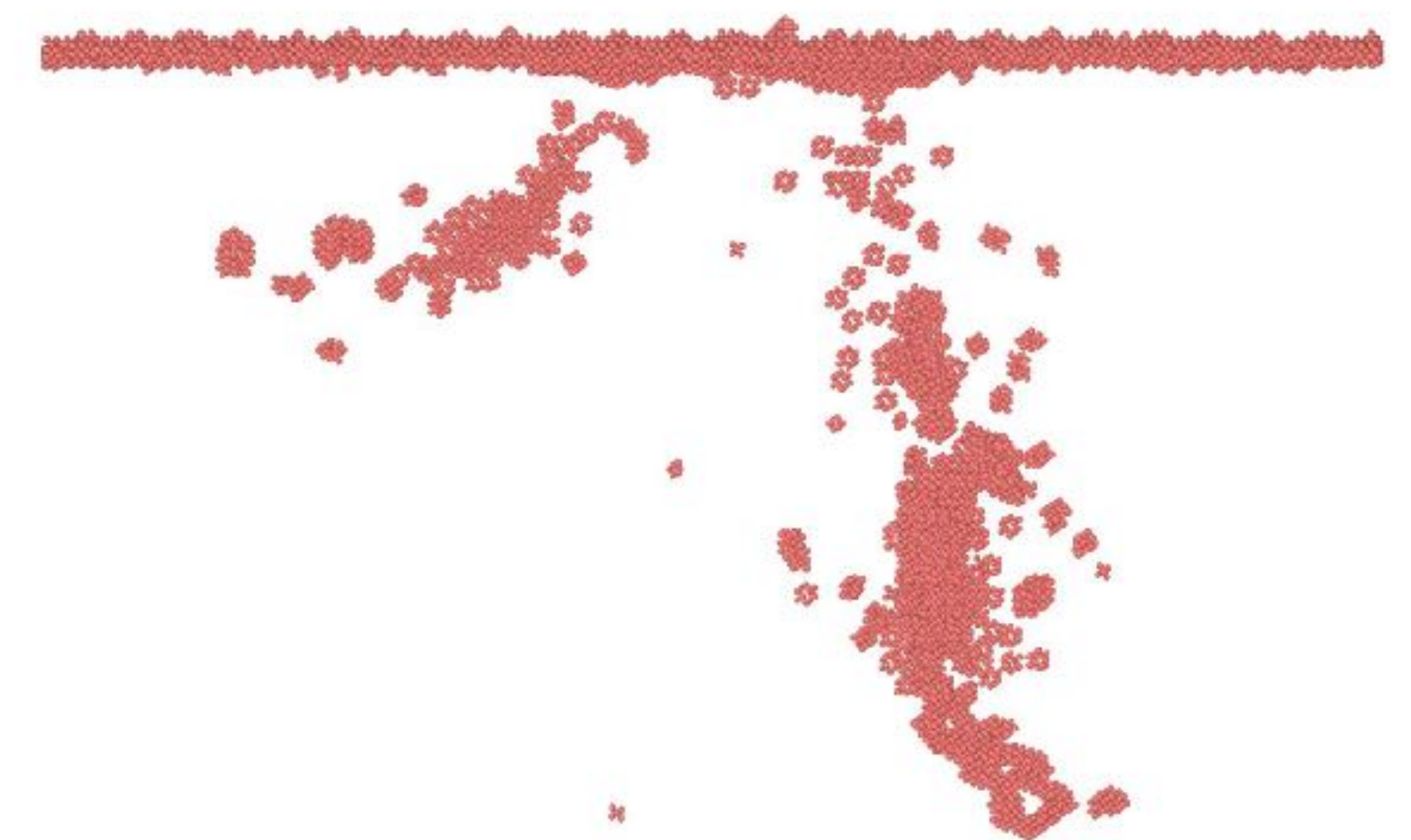
defect count = 65,798



CNA only tells you either crystalline or other

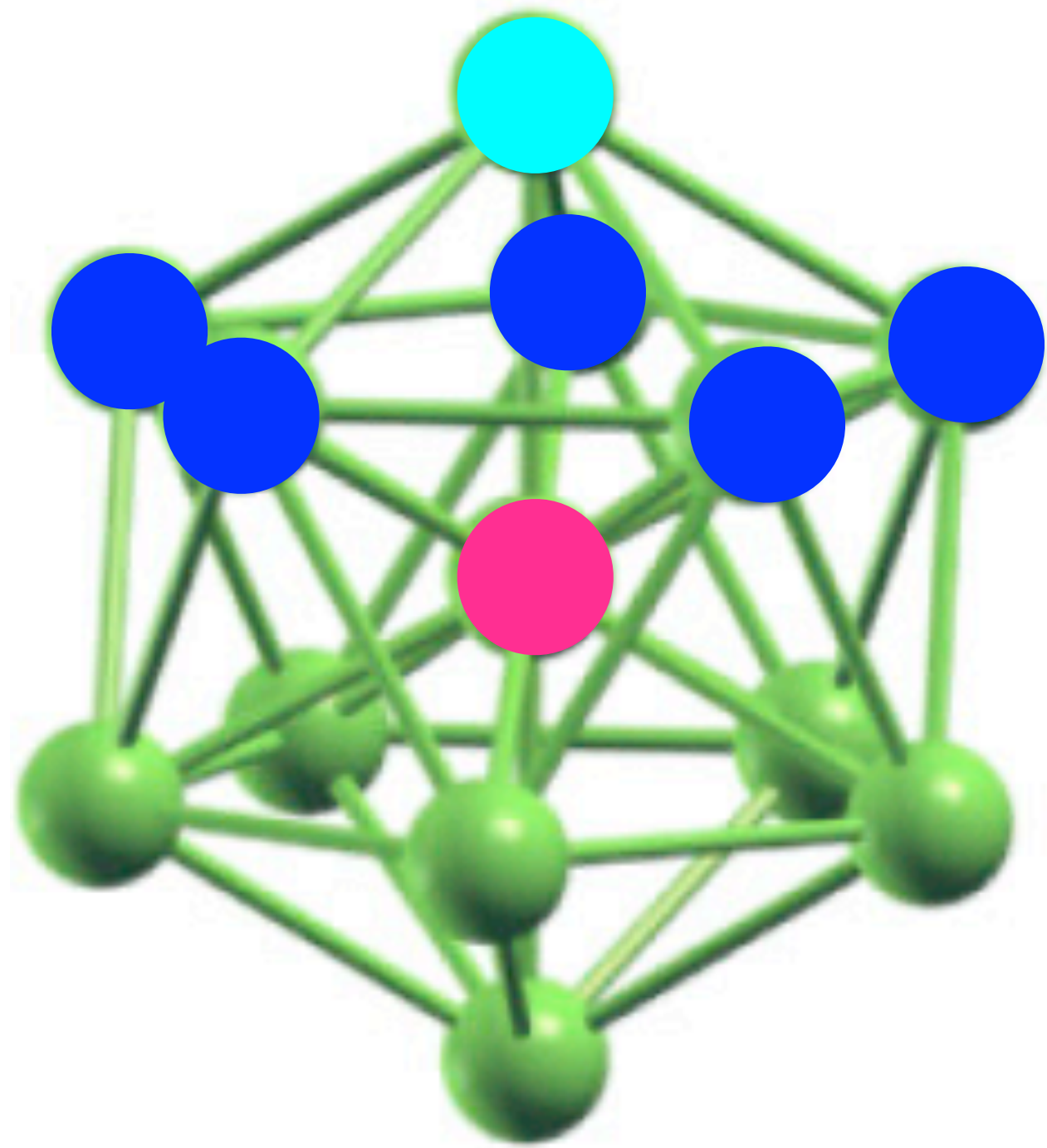
36.13 ps

defect count = 70,170



Deep inside

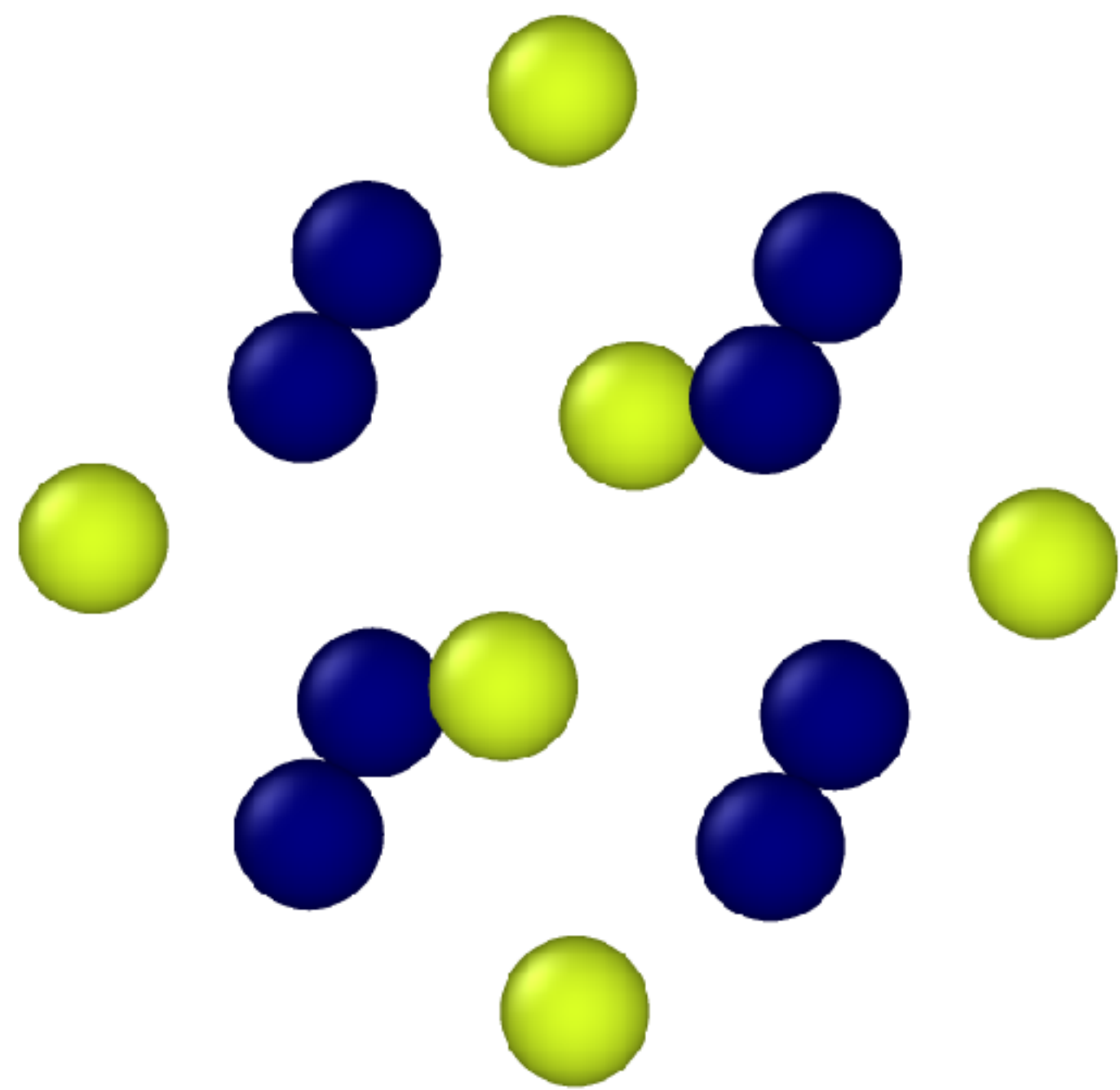
CNA



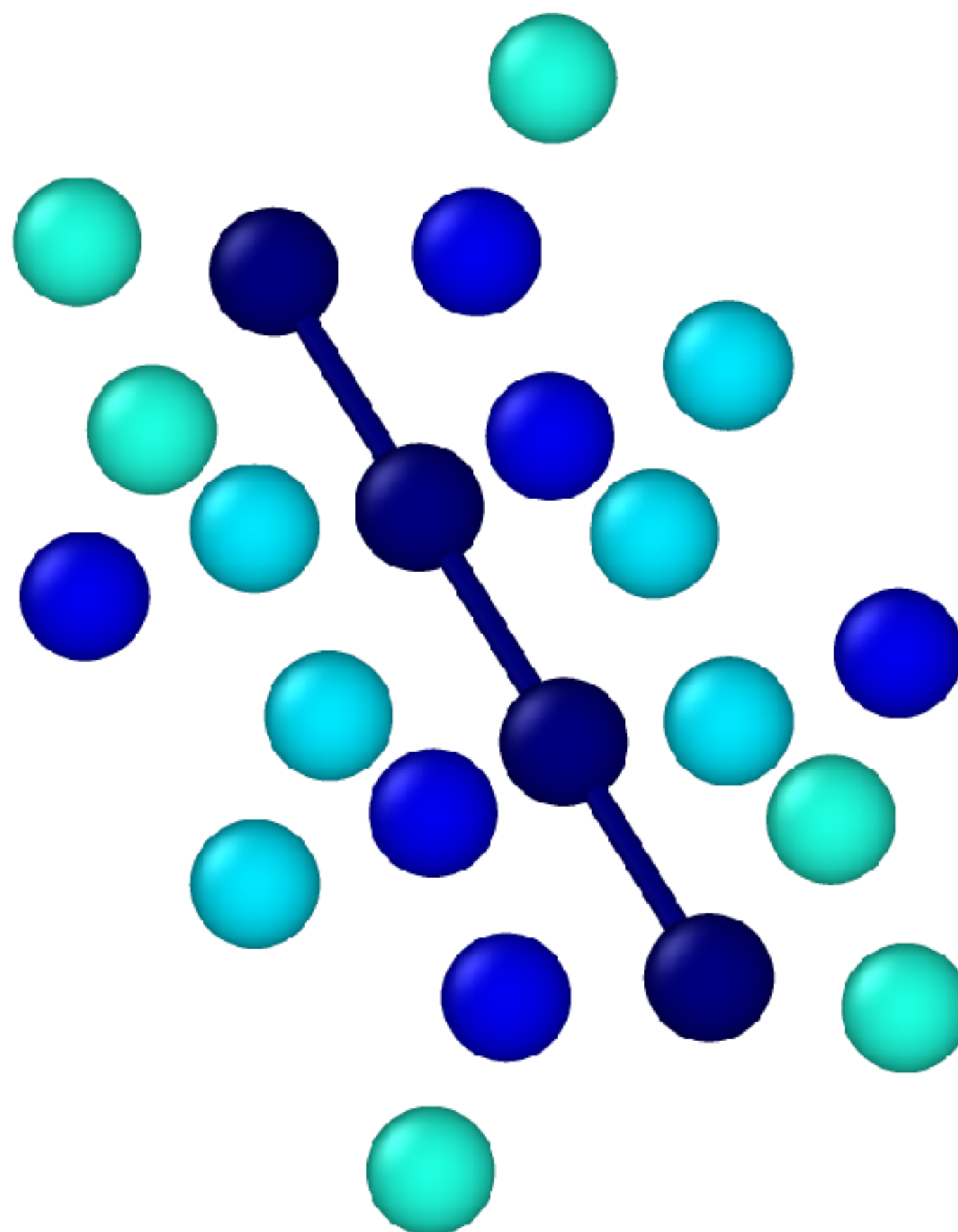
- atom of interest
- probe neighbor
- common neighbors

Bonded pairs	Stable liquid (2000 K)	Supercooled liquid (1700 K)	Ordered structure			
			bcc	fcc	hcp	ISRO
1101	0.04	0.02				
1201	0.12	0.08				
1211	0.06	0.04				
1301	0.07	0.06				
1311	0.22	0.19				
1321	0.09	0.09				0.71
1411	0.04	0.04				
1421	0.07	0.08		1.00	0.50	
1422	0.09	0.13			0.50	
1431	0.11	0.14				
1441	0.00	0.01	0.43			
1531	0.01	0.01				
1532	0.01	0.01				
1541	0.05	0.07				
1551	0.02	0.03				0.29
1661	0.00	0.01	0.57			

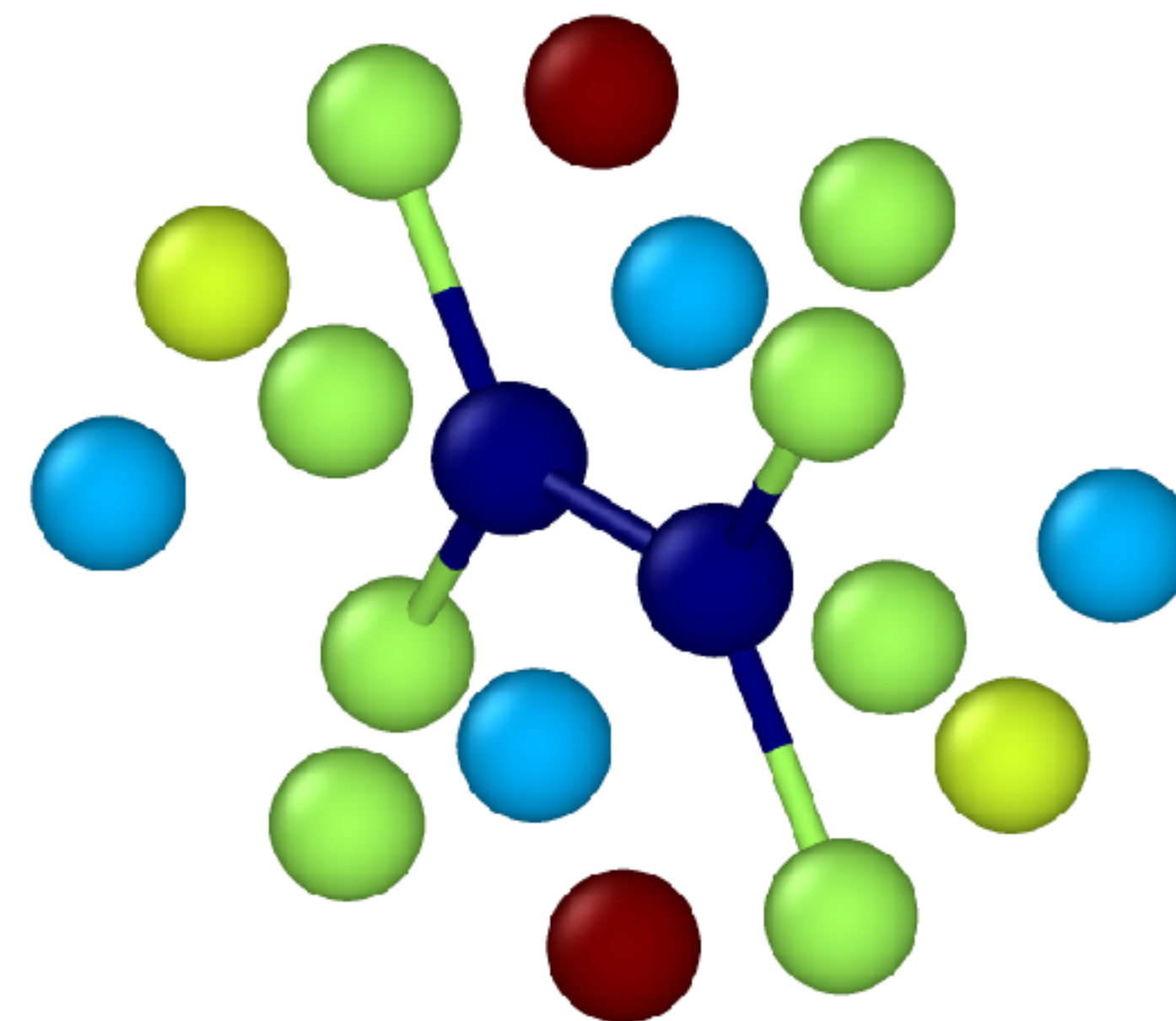
CNA goes beyond non-bcc



vacancy



$\langle 111 \rangle$ DB

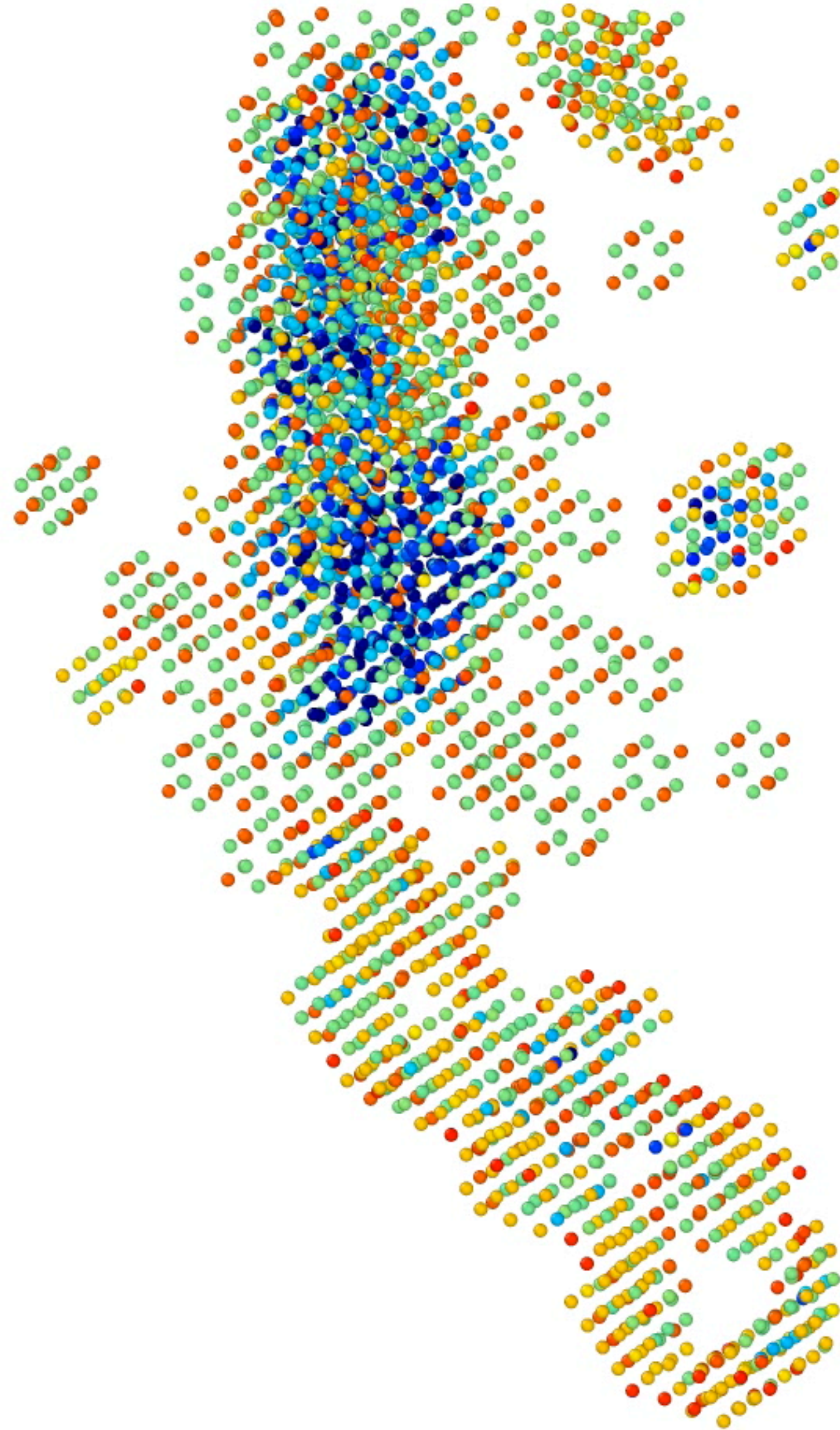


$\langle 110 \rangle$ DB



CNA

with bond types



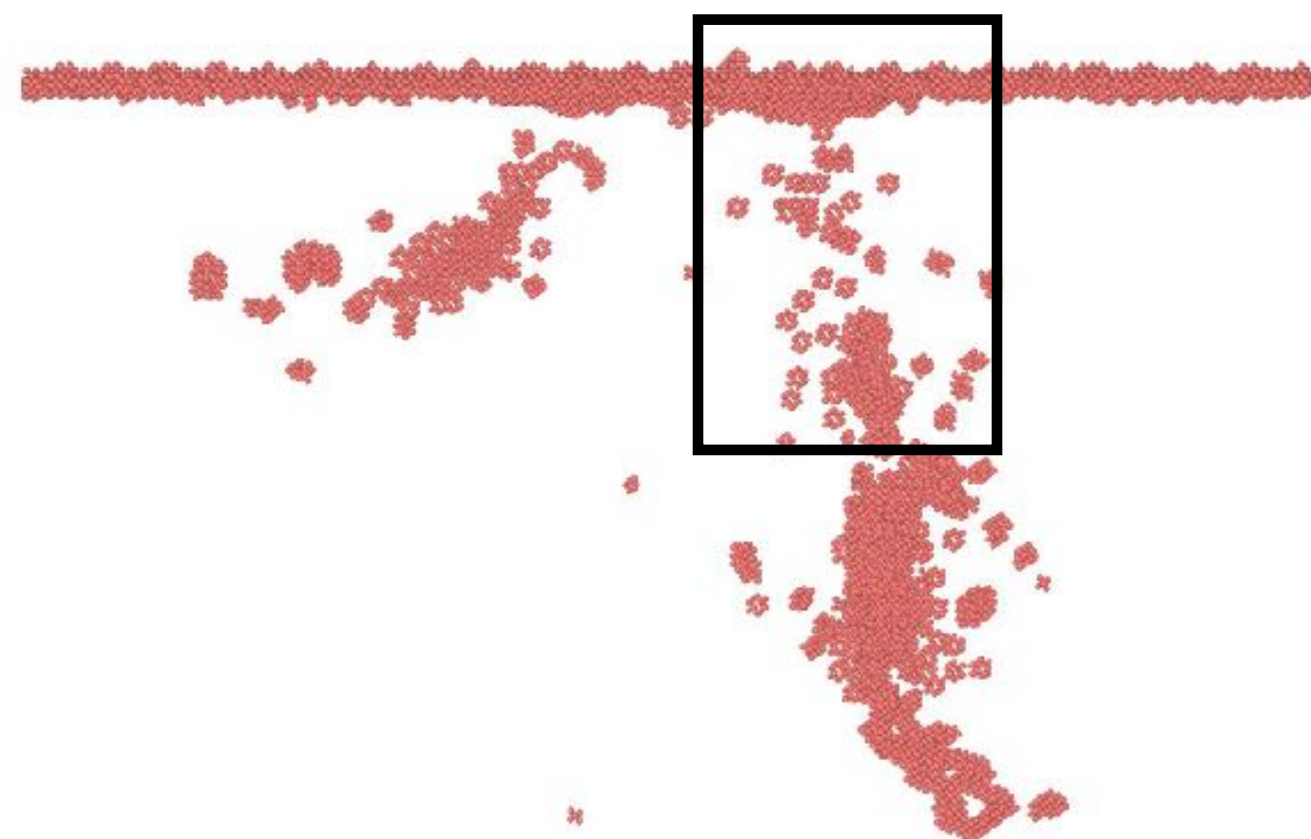
Bond Type

0

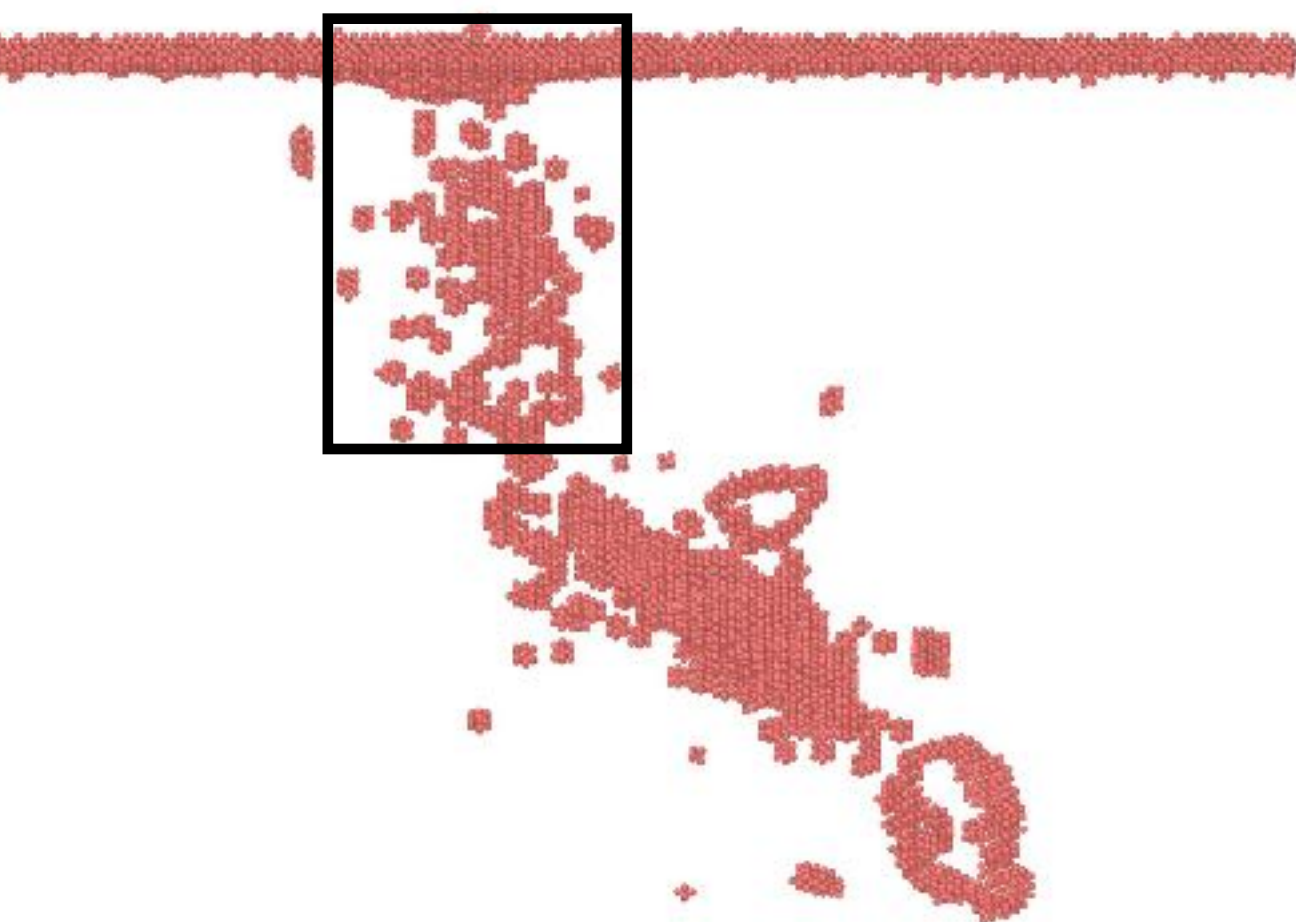


68

Crystal growth from GB



Front view

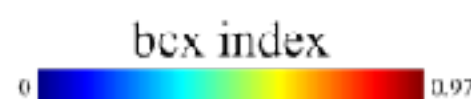
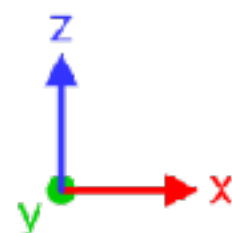
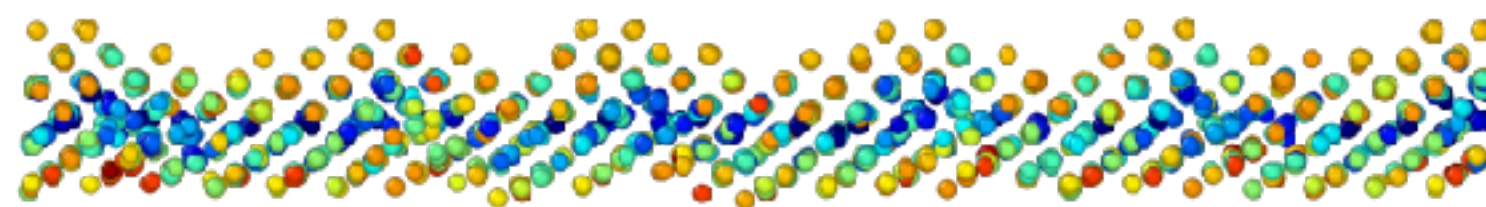


Side view

0 ps

GB atoms : 3,771

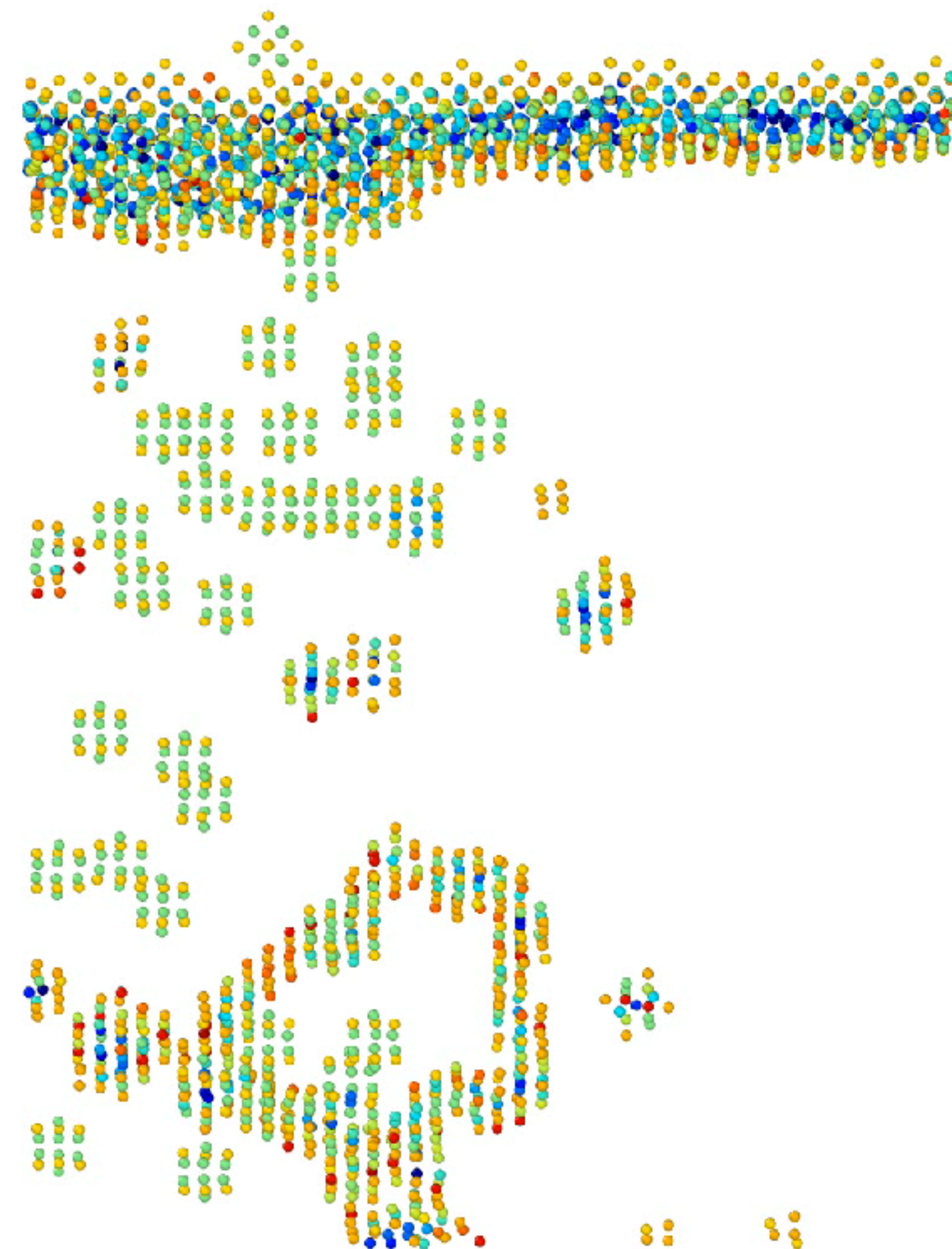
Bulk defects : 0



36.13 ps

GB atoms : 3,686

Bulk defects : 1,118



bcc index

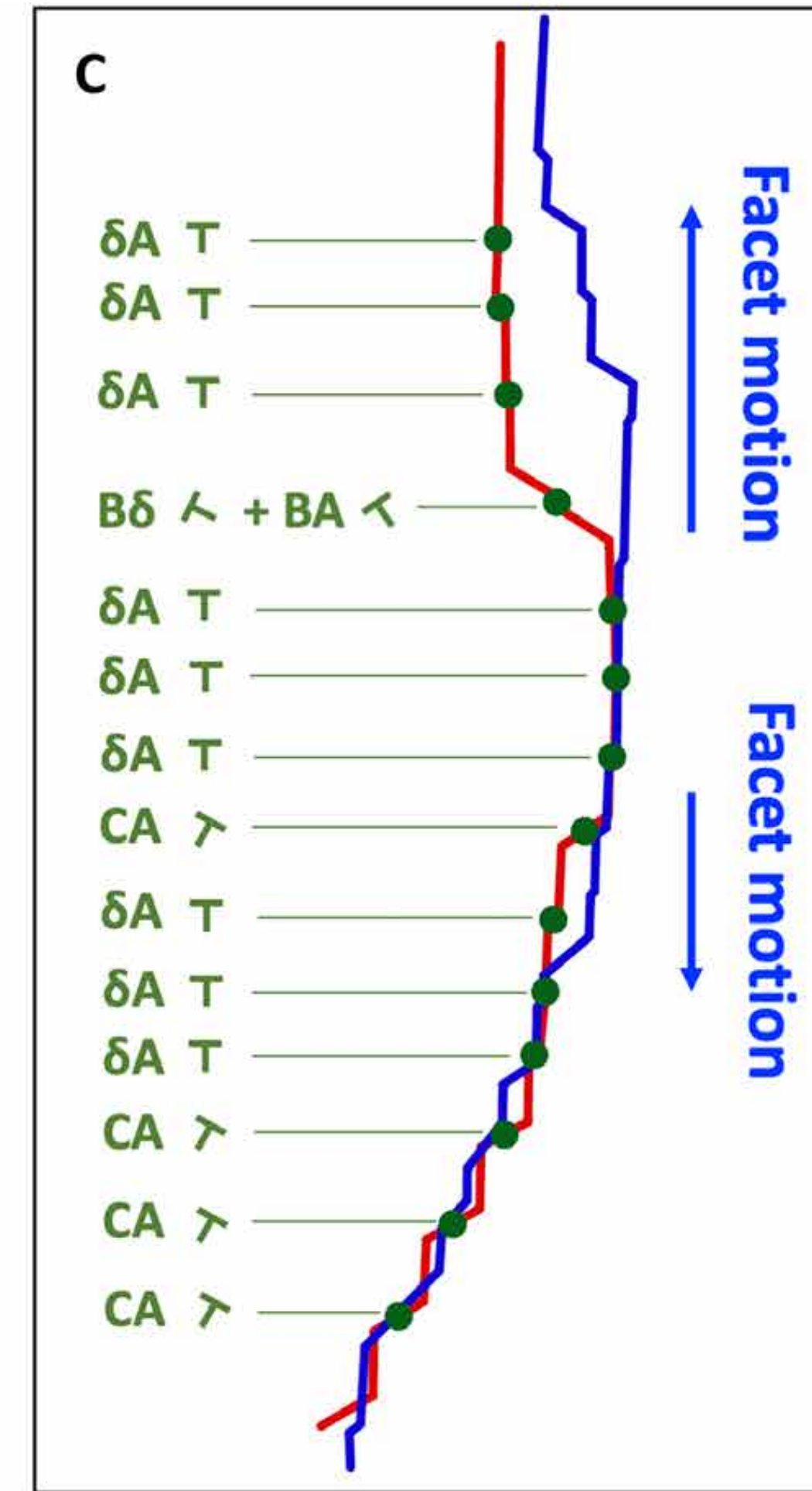
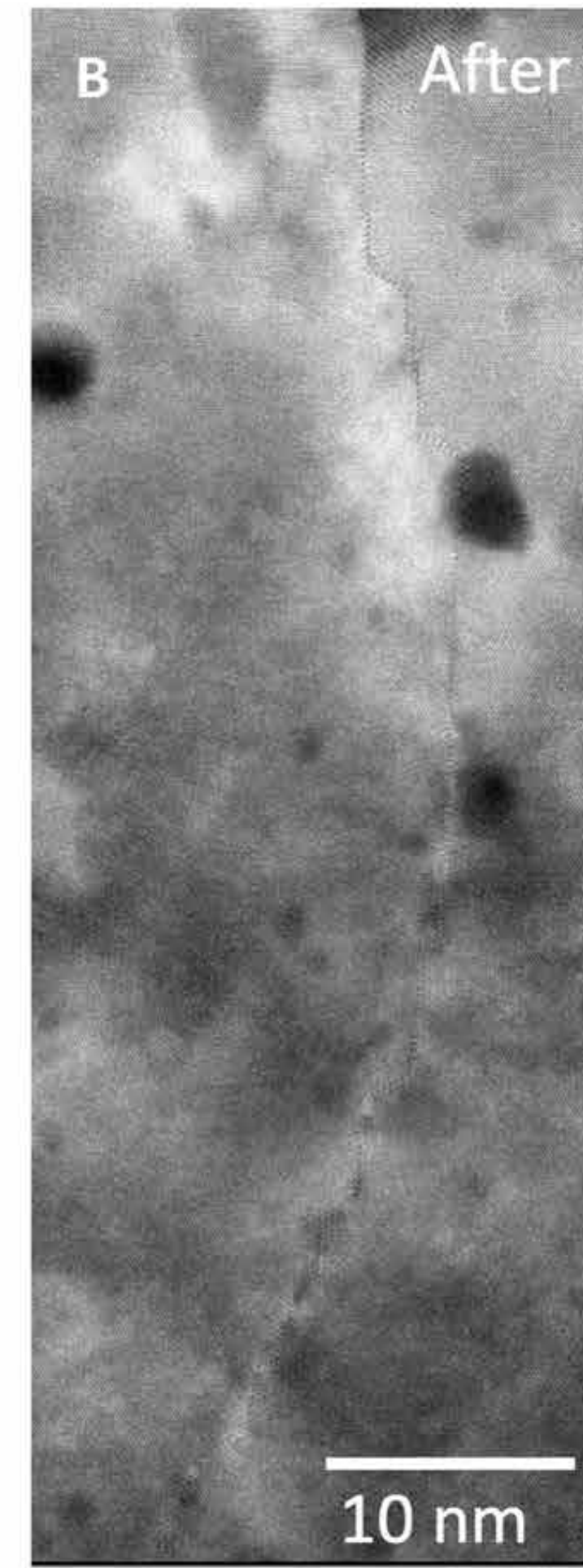
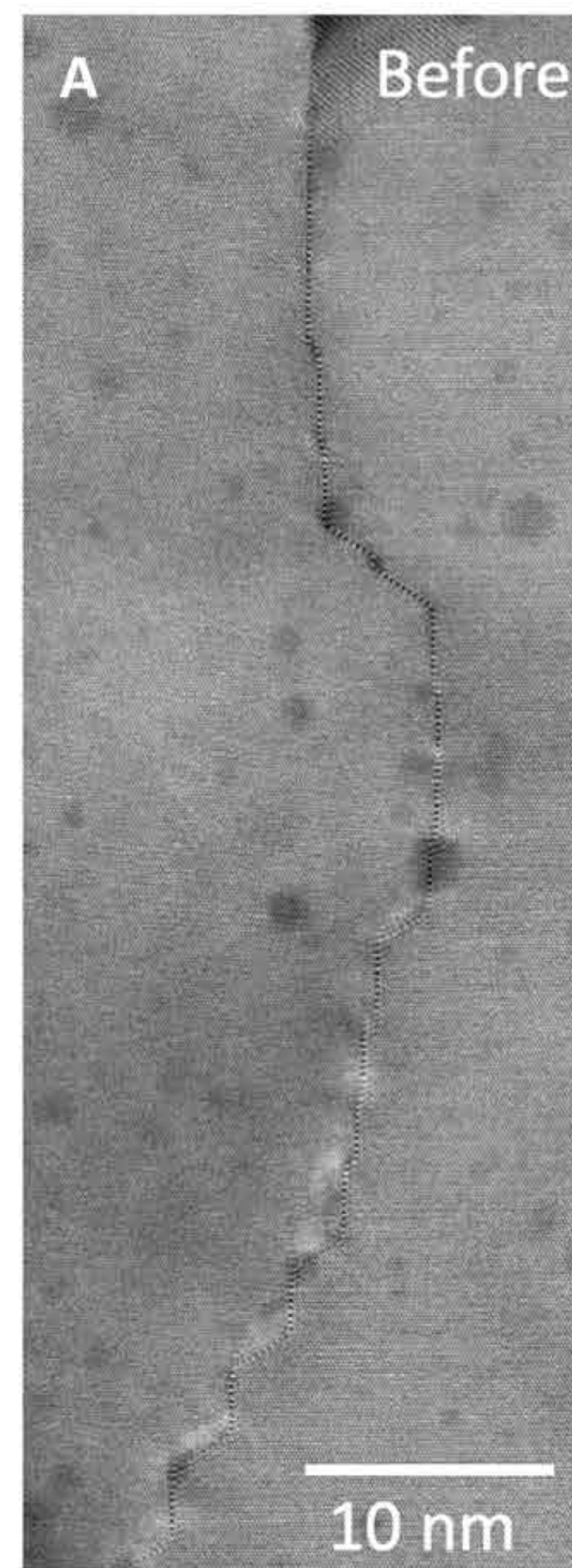
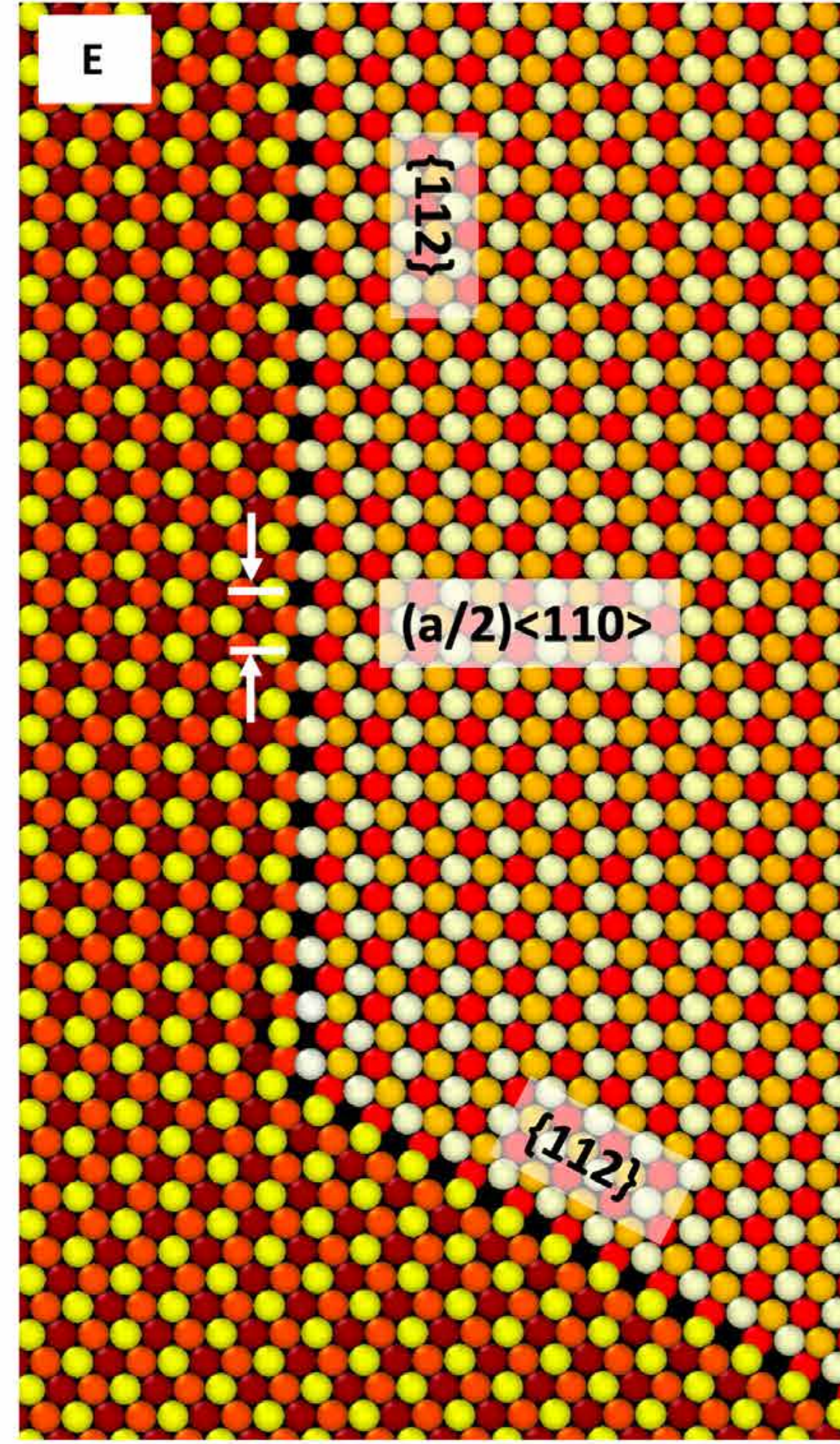
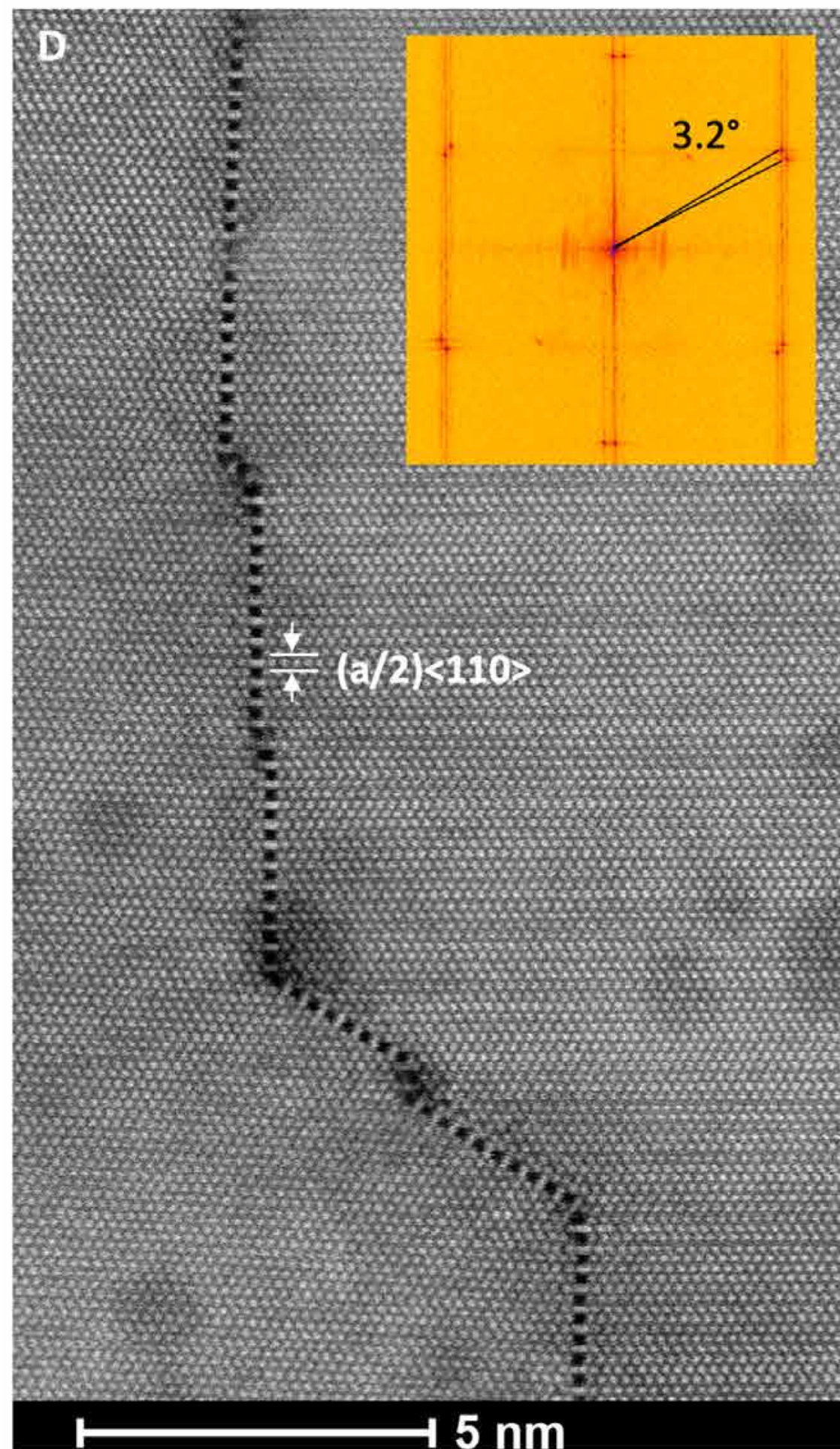


CNA w/ bond type results

Irradiation damages

near GB

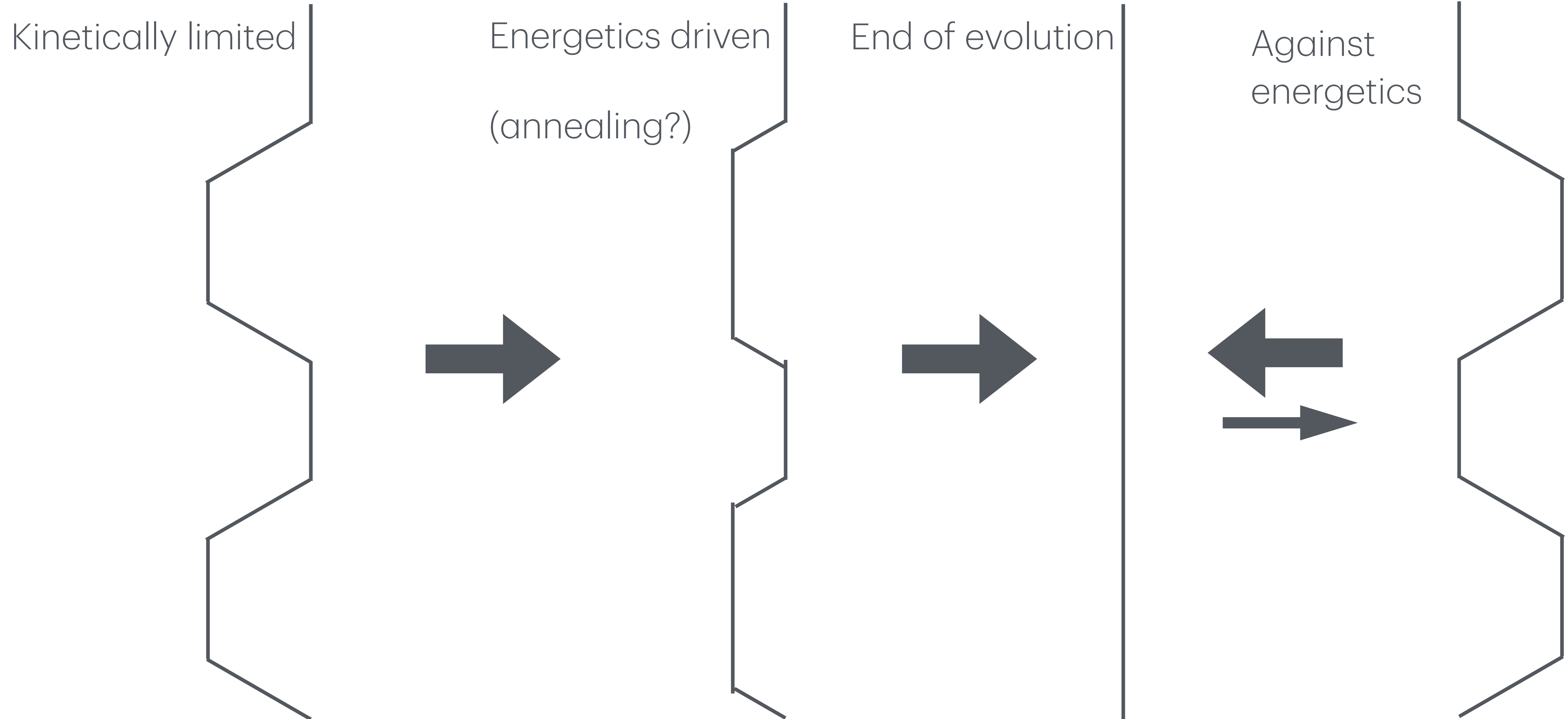
~ 1 dpa w/ 2.8 MeV Au⁴⁺



Pt $\Sigma 3$ {112} GBs

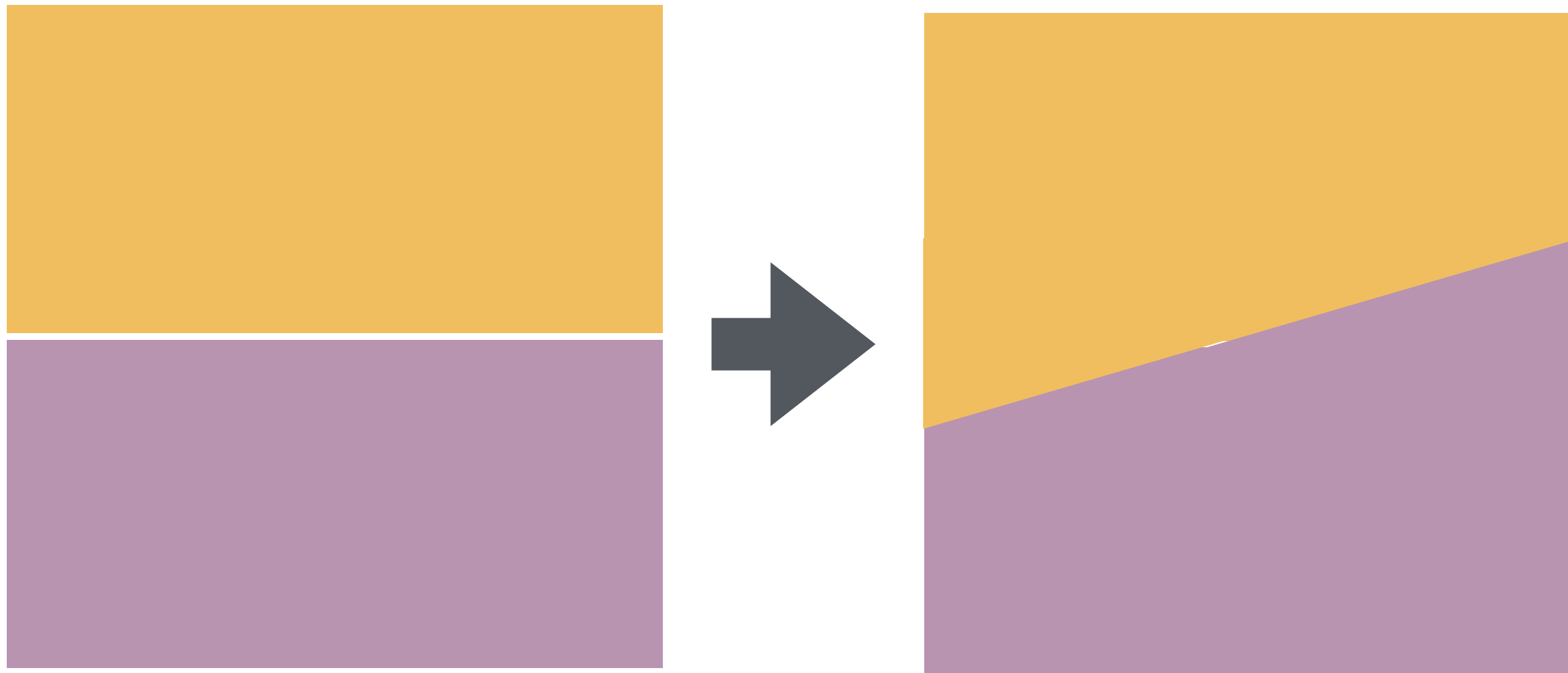
End of evolution scenario?

Moving Vs Flattening?



GB energetics

to find optimum



- Nucleation
 - Crystal growth
 - Dislocations
- Energetics
 - GB facet/surface energies
 - GB area

Is GB transparent
to neutron irradiation?



Likely... as far as GB degradation goes



GB is a defect sink, and the source of other defects



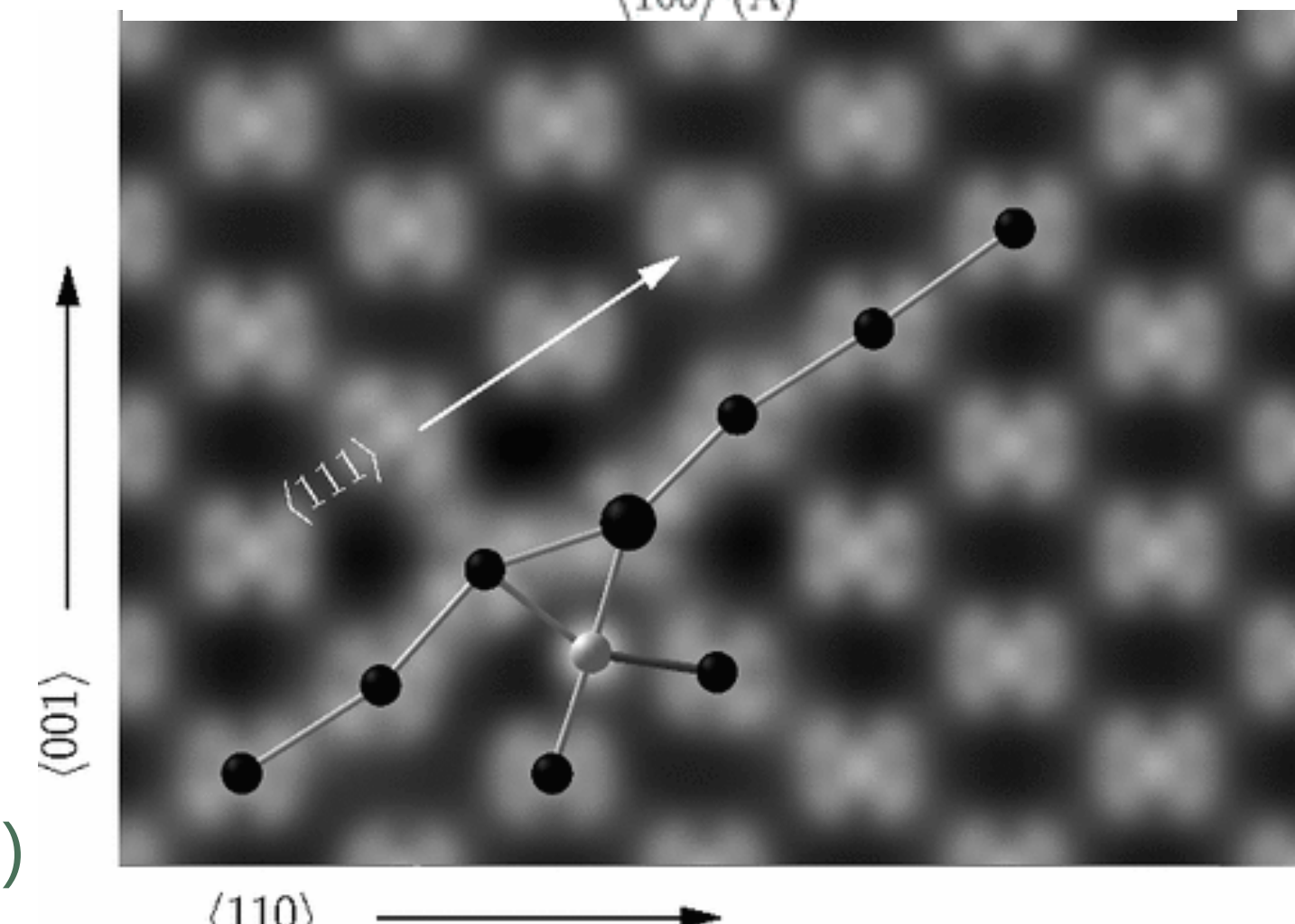
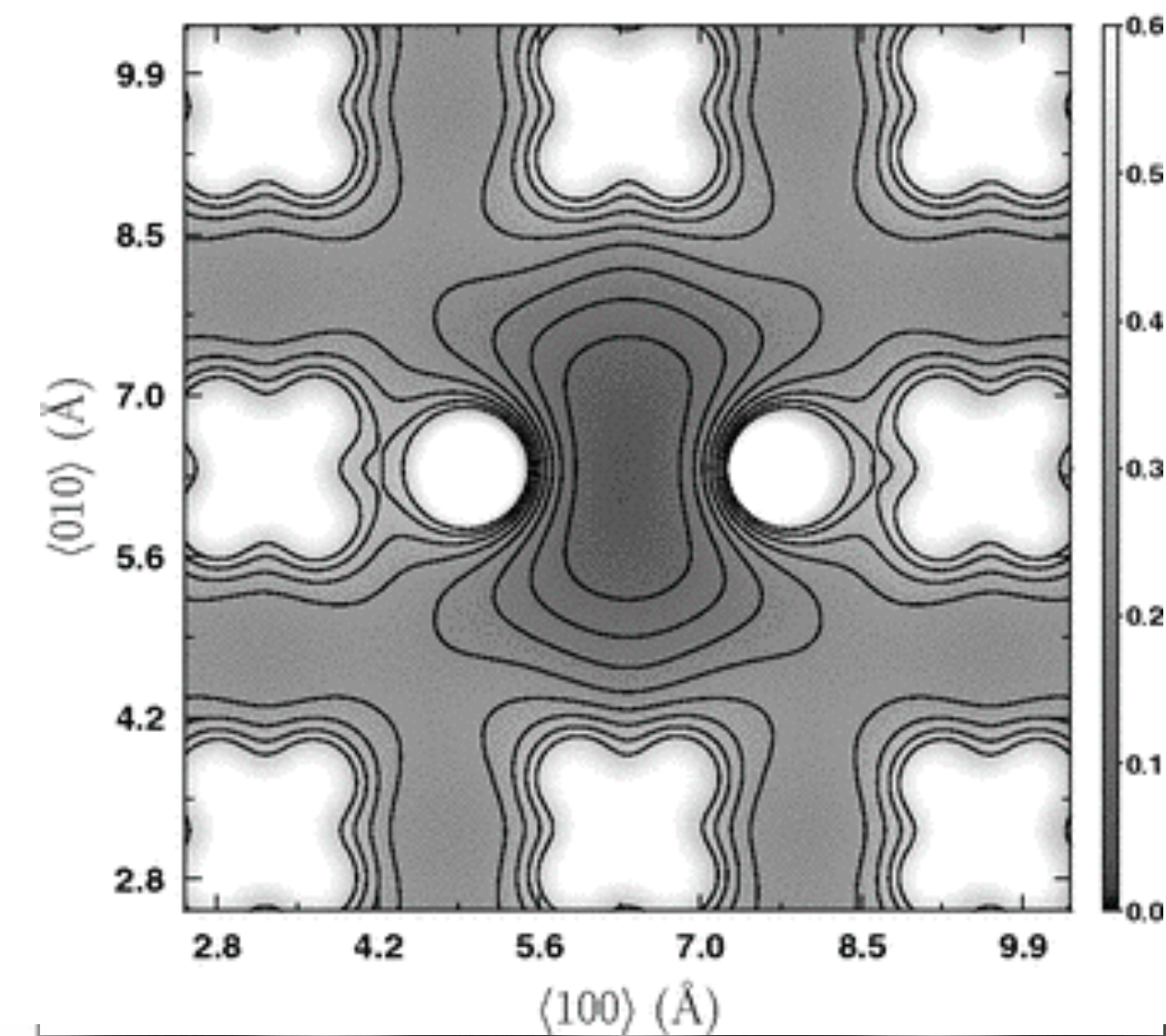
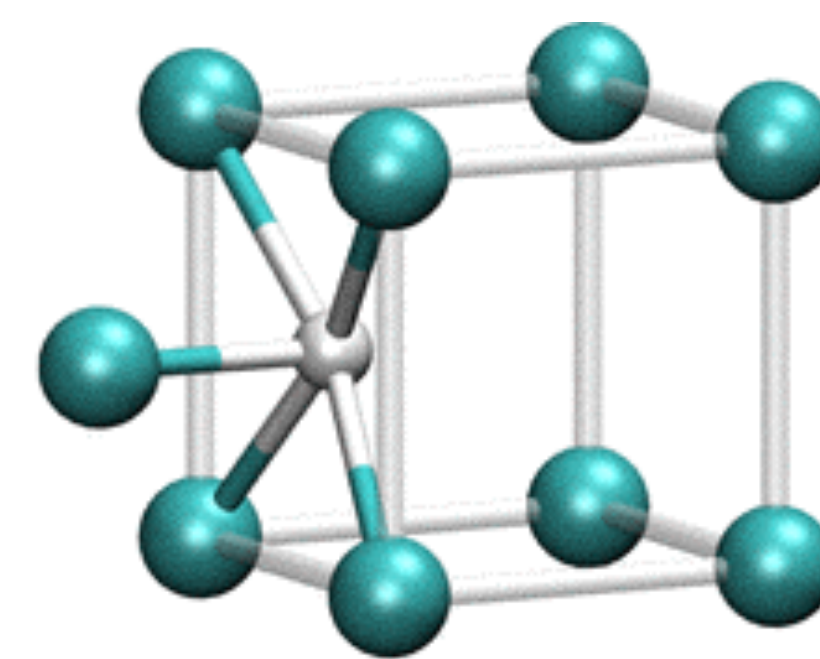
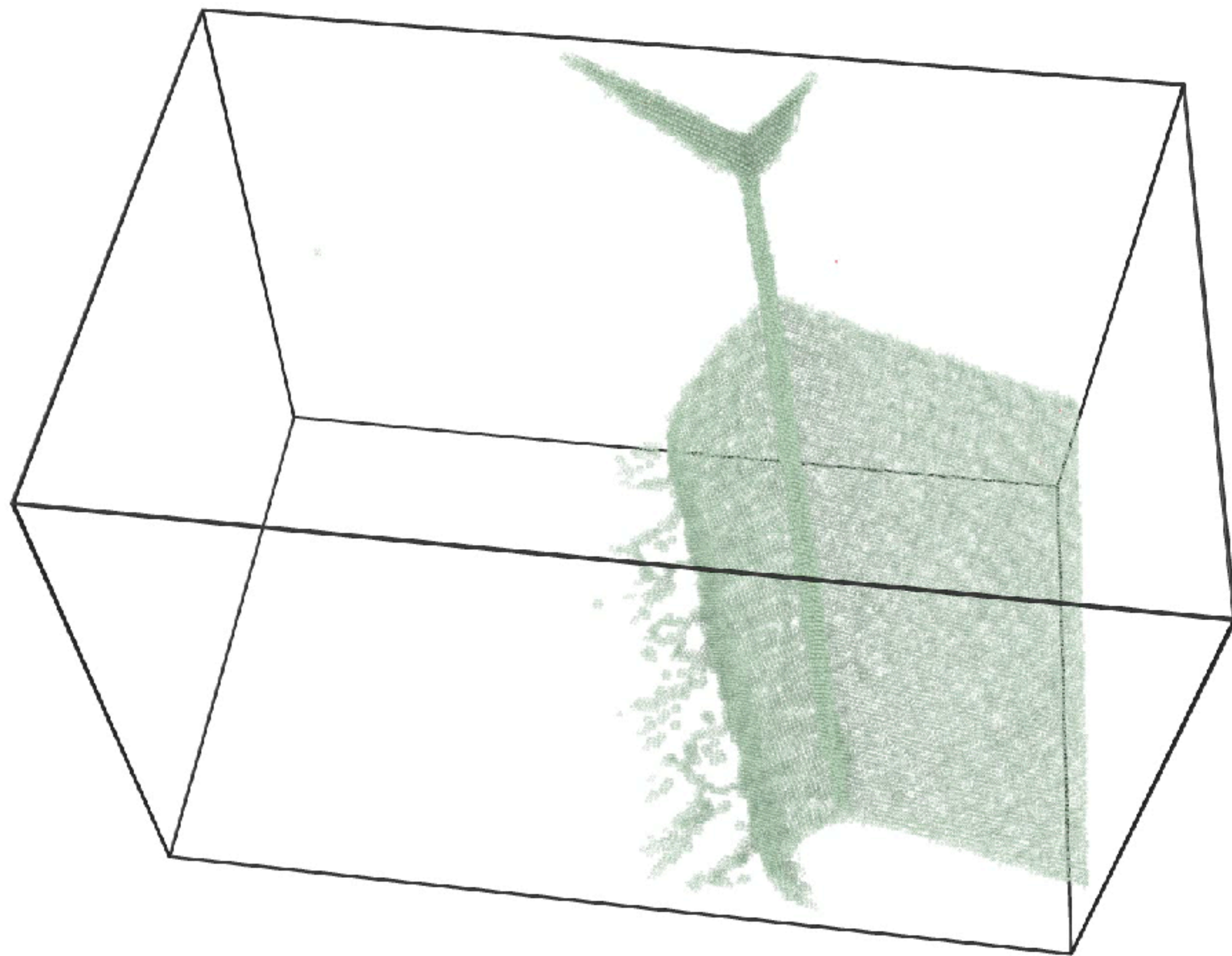
It will be interesting to see how GBs do
in the presence of transmutation and hydrogen

Remarks

During irradiation

- CNA seems promising for detections of various defects including GBs
 - **Fine-tuning in progress**
- Irradiation may stabilize the microstructure?
 - GBs may realign to reduce energy penalty
 - GB energy reduction = **grain growth?**

Future outlook



Thank you!